

1 **Supporting Information**

2

3 **Phosphorus-doped Porous Hollow Carbon Nanorods for High-performance**
4 **Sodium Based Dual-ion Battery**

5

6 Xiaoyan Wang, Shaofeng Wang, Kaixiang Shen, Shenggong He, Xianhua Hou*
7 Fuming Chen*

8

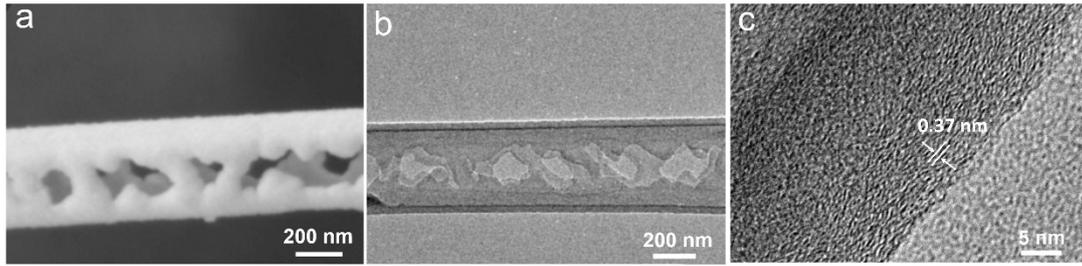
9 *Guangdong Provincial Key Laboratory of Quantum Engineering and Quantum*
10 *Materials, Guangdong Engineering Technology Research Center of Efficient Green*
11 *Energy and Environment Protection Materials, Engineering Research Center of*
12 *MTEES (Ministry of Education), School of Physics and Telecommunication*
13 *Engineering, South China Normal University, Guangzhou 510006, China*

14 * Corresponding author: fmchen@m.scnu.edu.cn (F.C.); houxianhua@m.scnu.edu.cn
15 (X. H.)

16

17

18



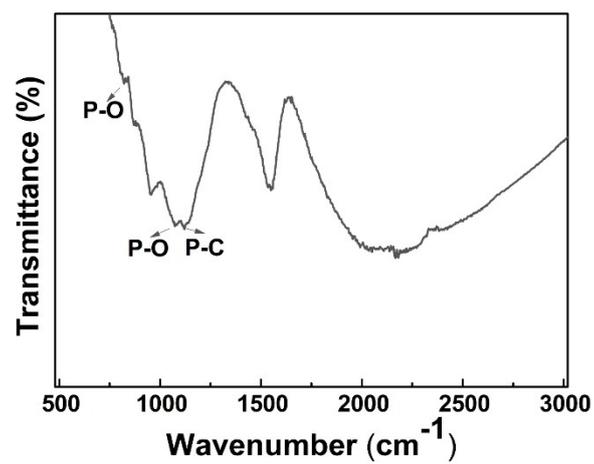
19

20

Figure S1. (a, b) SEM and TEM of signal HCNs. (c) HRTEM of P-HCNs.

21

22



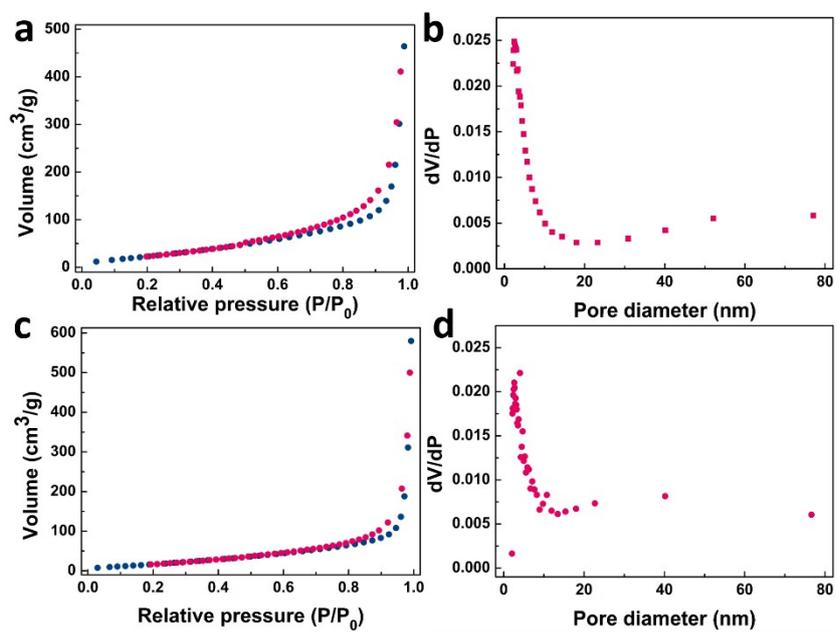
23

24 **Figure S2.** FTIR curve of P-HCNs, 1120.50 cm⁻¹ is assigned to P-C swing and C=C stretching,
25 and the peak located at 1063.8 cm⁻¹ is accounted for coupling of P-O symmetric stretching and P-
26 C.

27

28

29



30

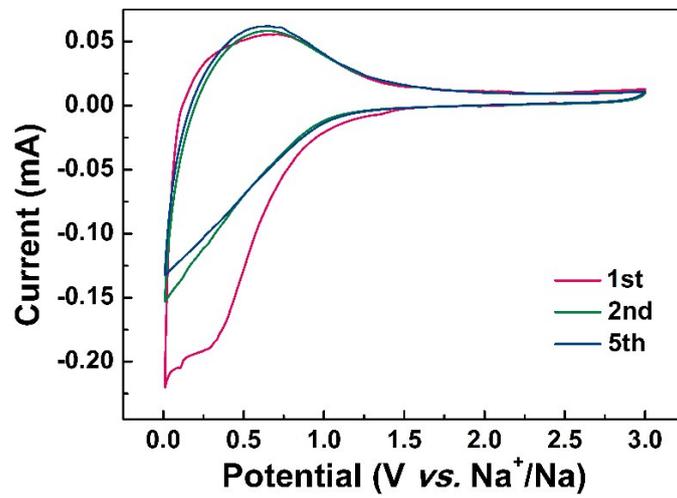
31 **Figure S3.** Nitrogen adsorption/desorption isotherms and the pore diameter of (a, b) HCNs and (c,

32 d) P-HCNs.

33

34

35

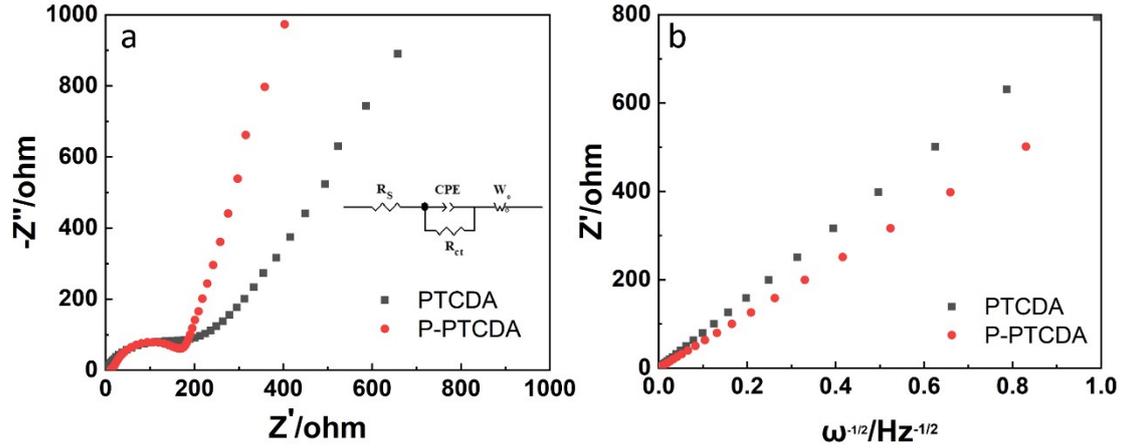


36

37 **Figure S4.** CV cures of HCNs in the potential range of 0.01-3.0 V at the scan rate of 0.2 mVs⁻¹

38

39



40

41 **Figure S5** (a) Nyquist plots of the NIBs based on P-HCNs anode. (b) Warburg impedance, Z' as a
 42 function of $\omega^{-1/2}$.

43

44 Diffusion coefficient value for Na-ion in case of HCN and P-HCN was calculated
 45 by the following equation:[1]

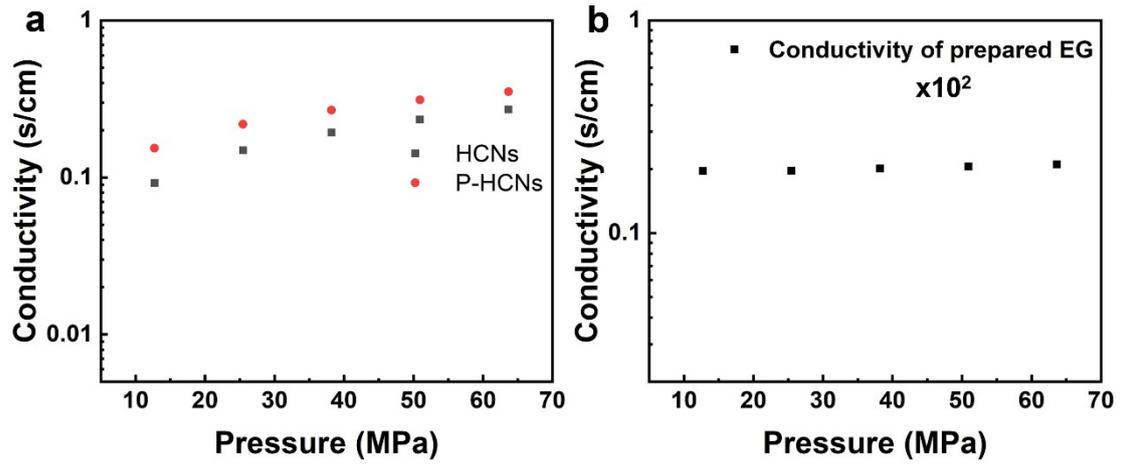
46

$$47 \quad D = \frac{R^2 T^2}{2 \sigma^2 n^4 F^4 A^2 C^2}$$

48 R is the gas constant, T is the Kelvin of thermodynamic temperature (293K), σ is the
 49 Warburg coefficient, n is electron transfer number ($n=1$), F is the Faraday's number
 50 ($96485.33289 \pm 0.00059 \text{C/mol}$), A is electrode area and C is the concentration of sodium
 51 in the electrode. The 2032 type of half-cell was assembled for EIS test. The electrode
 52 area is $\pi (0.6 \text{ cm})^2$. When the battery is full discharged, sodium and carbon can form
 53 NaC_6 compounds, which likes Li inserting to carbon. The mass of active material is 0.8

54 mg, so $C = \frac{0.8 \times 10^{-3}}{12} \times \frac{1}{6} \times \frac{1}{\pi 0.6^2}$. Because the thickness of the electrode is less than 20
 55 μm , so the volume of the electrode is approximately calculated equal to $\pi (0.6 \text{ cm})^2$.

56 Finally, the diffusion coefficient value for Na-ion in case of HCN and P-HCN are about
 57 6.7×10^{-16} and $11.8 \times 10^{-16} \text{ cm}^2/\text{S}$, respectively

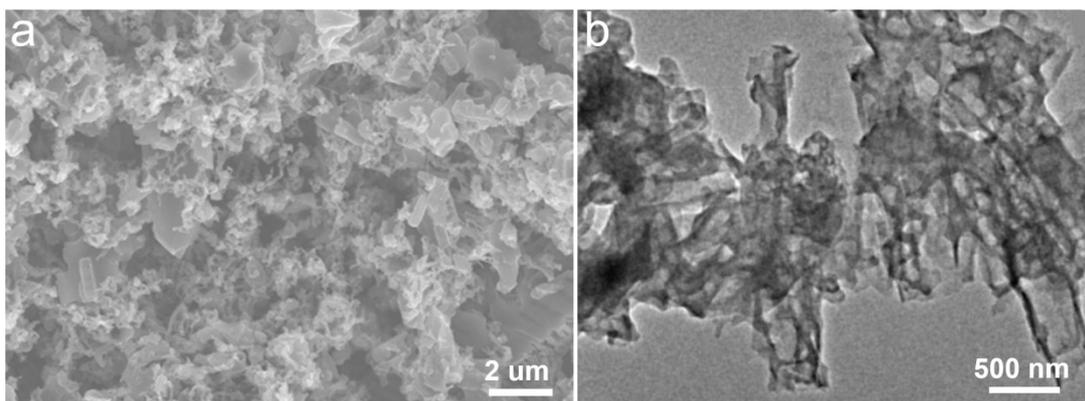


59

60 **Figure S6** (a) Pressure-Conductivity graph of HCNs and P-HCNs. (b) Pressure-Conductivity graph

61 of EG.

62

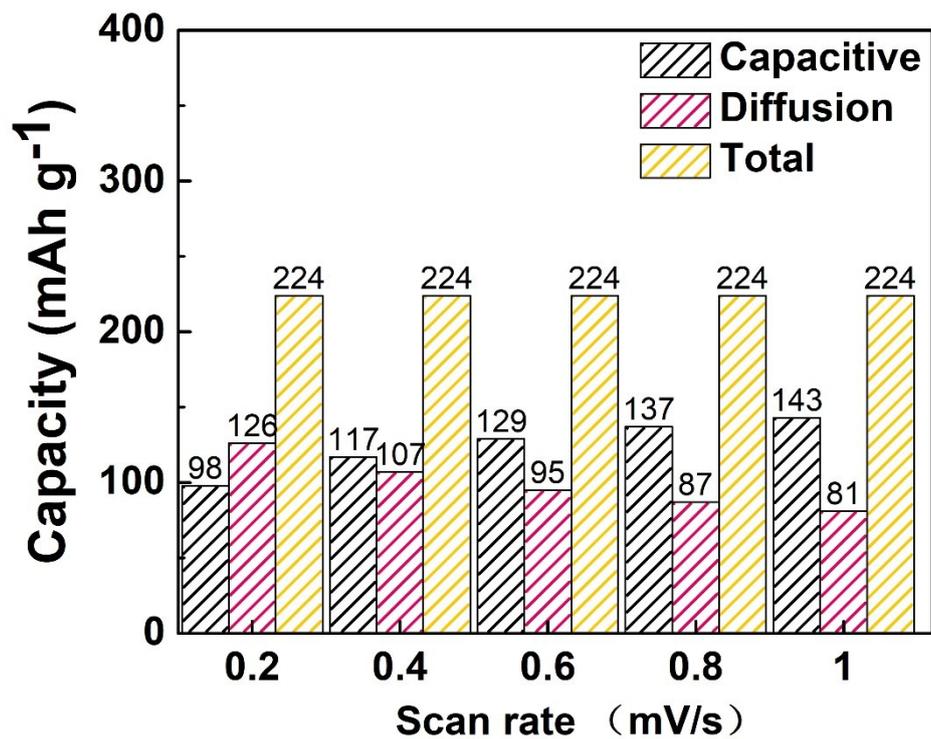


63

64 **Figure S7.** (a) SEM, (b) TEM image of P-HCNs for half-cell after 500 cycles at the current density

65 of 500 mA g⁻¹.

66



67

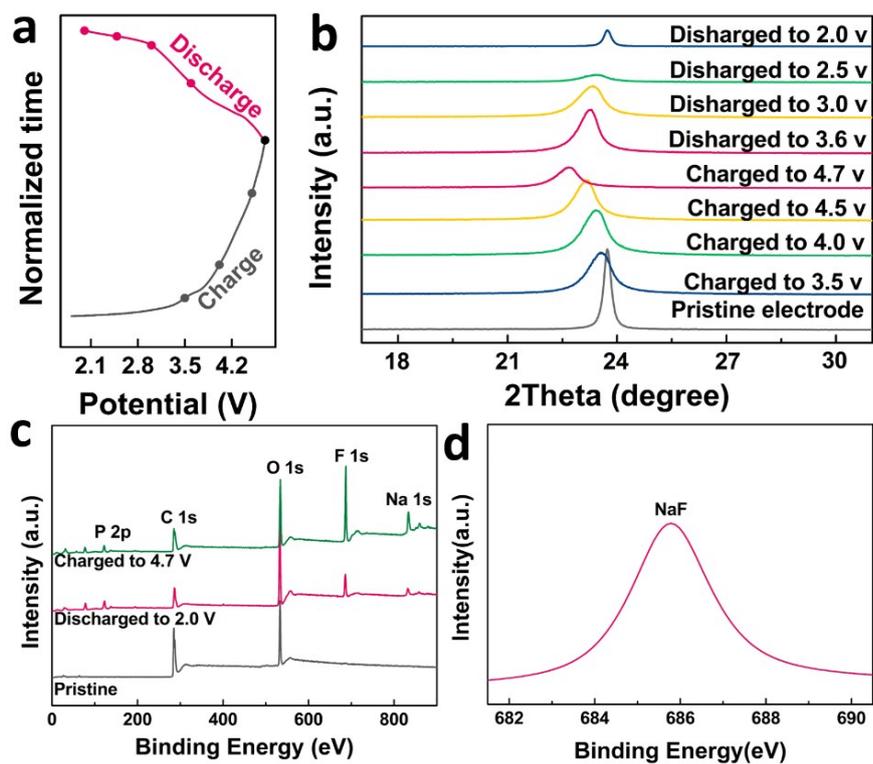
68 **Figure S8** Capacitive, diffusion-control and total capacities for varied scan rates of P-

69 HCNs electrode.

70

71 Note S1

72 To better testify that PF_6^- has involved in the insertion and desorption processes for
73 NDIBs, the Ex-situ XRD patterns of EG cathode at different charge/discharge states
74 (Figure S9a) are carried out in Figure S9b. There is only one characteristic graphite
75 (002) peak at $2\theta = 23.8^\circ$. Once charged from 3.5-4.7 V, the (002) peak shifts to a lower
76 angle, indicating a gradually increased layer spacing during the PF_6^- interposed into the
77 EG cathode. While discharged from 4.7 V to 2.0 V, the (002) peak recovers approached
78 to 23.8° , which means the PF_6^- de-intercalation from EG. Different from the initial
79 stage, the (002) peak of EG becomes broad and decline, meaning reduced crystallinity.
80 It is because of the formation of SEI in charge process. Ex-situ XPS spectra in Figure
81 S9c are performed to further study the PF_6^- reacted with EG. In the pristine state, it only
82 exhibits two peaks around 285.0 eV and 534.0 eV, assigning to C 1s, and O 1s. When
83 charged to 4.7 V, three new peaks appear around 192.7 eV and 823.0 eV, meaning P
84 2p, F 1s and Na 1s, respectively. The increased Na and F contents in fully charged state
85 at 4.7 V suggests that PF_6^- has inserted into the EG cathode. When discharged to 2 V,
86 the intensity of F 1s and Na 1s is reduced obviously, demonstrating that PF_6^- release
87 from EG to the electrolyte. Figure S9d is the high-resolution F1s XPS of P-HCNs, it
88 reflects that the F element for EG is attributed to the formation of NaF in the EG
89 electrode of NDIBs after the initial cycles.



90

91 **Figure S9** (a) Normalized voltage-time curves of full cell. (b) The ex-situ XRD patterns of the EG

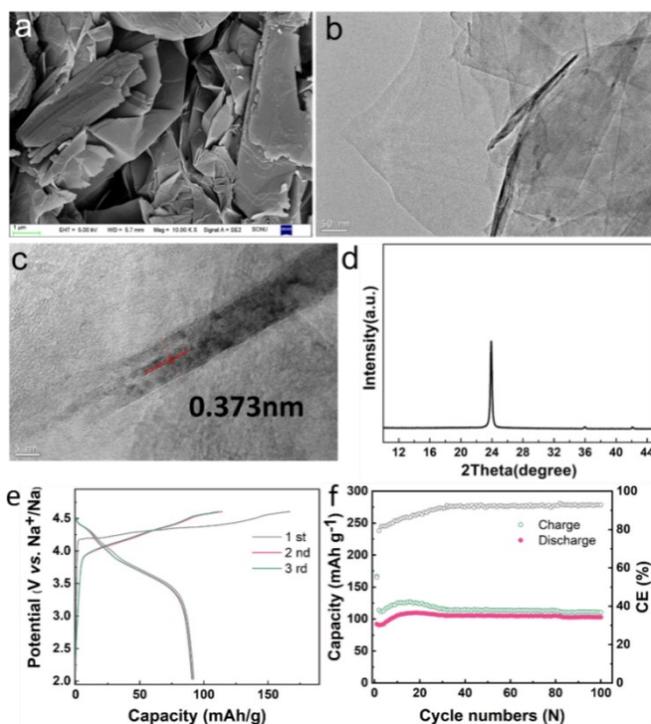
92 at different states during initial charge/discharge process. (c) The full XPS spectra of the EG

93 electrodes at different state after one cycle. (d) The high-resolution F 1s XPS of the EG.

94

95 Note S2

96 The morphology of EG was investigated by SEM and TEM in Figure S10a-10c.
97 The layer spacing is 0.373 nm as shown in Figure S10c, which is larger than graphite
98 (0.3354 nm). The XRD pattern of EG (Figure S10d) shows only one peak at 23.8°,
99 assigned to the (002) plane. Figure S10e displays the charge/discharge curves of EG
100 cathode within the potential window of 2.0-4.7 V. The first CE is 55.0%. After 100
101 cycles, the reversible capacity of 105 mAh g⁻¹ can be still obtained for the EG electrode
102 as demonstrated in Figure S10f.



103

104 **Figure S10** (a) SEM image of as-prepared EG, (b) TEM image of as-prepared EG, (c) HR-TEM
105 image of as-prepared EG, (d) XRD patterns of prepared EG. (e) The charge/discharge curves of EG
106 cathode in half-cell within the potential windows of 2.0-4.7 V, (f) The cycling performance of EG
107 at 100 mA g⁻¹.

108

109

Table S1 The atomic content of C, O and P in P-HCNs analyzed by EDS.

Materials	Element Content
P-HCN	by atomic (%)
C	88.4
O	10.2
P	1.4

110

111 Note S3: Energy density and power density calculation of P-HCNs//EG NDIBs.

112 The power and energy densities were calculated by the following formula[2]:

113
$$E = \int_{t1}^{t2} IVdt = \frac{1}{2}C(V_{max} + V_{min})(V_{max} - V_{min}) = \frac{V_{max} + V_{min}}{2} \times \frac{It}{m} \times \frac{1}{3.6} \quad (1)$$

114
$$P = \frac{3600 \times E}{t} \quad (2)$$

115 where V_{max} and V_{min} are the initial and final discharge voltages (V), I is the discharge
116 current (A), t is the discharge time (s), m is the total mass of the anode and cathode
117 active material (mg), E is the energy density (Wh kg^{-1}), and P is the power density (W
118 kg^{-1}). For example, when the discharge current density is maintained at 0.2 A g^{-1} for
119 57.75×60 s, and the total mass of anode and cathode active materials is 3.12 mg, the
120 energy and power densities can be calculated as below:

121
$$E = \frac{2.00509 + 4.698}{2} \times \frac{0.2 \times 0.8 \times 57.75 \times 60}{3.12} \times \frac{1}{3.6} = 165 \text{ Wh } kg^{-1} \quad (3)$$

122
$$p = \frac{3600 \times E}{t} = \frac{3600 \times 165}{57.75 \times 60} = 171 \text{ W kg}^{-1} \quad (4)$$

123 Calculated from the same method as above, the energy and power densities at different
124 current density could be obtained, which are showed in Table S2.

125

126 **Table S2.** A summary of the performance of reported devices

Anode//Cathode	Potential range (V)	E (Wh/kg)	P (W/kg)	Current density (A g⁻¹)
CNT//V ₂ O ₅ ^[3]	1.8-4.0	40	210	-
TiO ₂ -Graphene//AC ^[4]	1-3.8	64.2	56.3	3.35
Graphene//GMP ^[5]	0-1.8	42.3	950	10
Nb ₂ O ₅ @C/rGO//AC ^[6]	1-4.3	76	80	1
C//AC ^[7]	0-4.0	112	67	5
TNW//CNT ^[8]	0-2.8	12.5	300-320	10C
400-KON-Ti ₃ C ₂ ^[9]	0-1.6	27.4	--	1
	0-1.6	17.1	4	--
HAT-CNF//Carbon ^[10]	0.5-4.0	95	190	--
	0.5-4	65	1800	--
Na ₃ V ₂ (PO ₄) ₃ @C//AC ^[11]	0-3.0	118	95	--
	0-3.0	60	850	--
This work	2-4.7	76	891	1
	2-4.7	69	1821	2
	2-4.7	103	906	0.8
	2-4.7	143	580	0.5
	2-4.7	165	171	0.2

127

128

129 Reference

- 130 [1] H. Liu, Q. Cao, L.J. Fu, C. Li, Y.P. Wu, H.Q. Wu, Doping effects of zinc on
131 LiFePO_4 cathode material for lithium ion batteries, *Electrochemistry Communications*,
132 8(10) 1553-1557.
- 133 [2] X. Ren, Z. Ren, Q. Li, W. Wen, X. Li, Y. Chen, L. Xie, L. Zhang, D. Zhu, B. Gao,
134 Tailored Plum Pudding-Like $\text{Co}_2\text{P}/\text{Sn}$ Encapsulated with Carbon Nanobox Shell as
135 Superior Anode Materials for High-Performance Sodium-Ion Capacitors, *Advanced*
136 *Energy Materials*, (2019).
- 137 [3] Z. Chen, V. Augustyn, J. Wen, Y. Zhang, M. Shen, B. Dunn, Y. Lu, High-
138 performance supercapacitors based on intertwined CNT/ V_2O_5 nanowire
139 nanocomposites, *Advanced Materials*, 23 (2011) 791-795.
- 140 [4] Z. Le, F. Liu, P. Nie, X. Li, X. Liu, Z. Bian, G. Chen, H.B. Wu, Y. Lu,
141 Pseudocapacitive sodium storage in mesoporous single-crystal-like TiO_2 -graphene
142 nanocomposite enables high-performance sodium-ion capacitors, *ACS nano*, 11 (2017)
143 2952-2960.
- 144 [5] J. Fu, J. Yun, S. Wu, L. Li, L. Yu, K.H. Kim, Architecturally Robust Graphene-
145 Encapsulated MXene $\text{Ti}_2\text{CT}_x@$ Polyaniline Composite for High-Performance Pouch-
146 Type Asymmetric Supercapacitor, *ACS applied materials & interfaces*, 10 (2018)
147 34212-34221.
- 148 [6] E. Lim, C. Jo, M.S. Kim, M.H. Kim, J. Chun, H. Kim, J. Park, K.C. Roh, K. Kang,
149 S. Yoon, Hybrid Supercapacitors: High-Performance Sodium-Ion Hybrid
150 Supercapacitor Based on $\text{Nb}_2\text{O}_5@$ Carbon Core-Shell Nanoparticles and Reduced
151 Graphene Oxide Nanocomposites (*Adv. Funct. Mater.* 21/2016), *Advanced Functional*
152 *Materials*, 26 (2016) 3553-3553.
- 153 [7] H. Wang, D. Mitlin, J. Ding, Z. Li, K. Cui, Excellent energy-power characteristics
154 from a hybrid sodium ion capacitor based on identical carbon nanosheets in both
155 electrodes, *Journal of Materials Chemistry A*, 4 (2016) 5149-5158.
- 156 [8] Q. Wang, Z. Wen, J. Li, A hybrid supercapacitor fabricated with a carbon nanotube
157 cathode and a TiO_2 -B nanowire anode, *Advanced Functional Materials*, 16 (2006)
158 2141-2146.

159 [9] J. Li, X. Yuan, C. Lin, Y. Yang, L. Xu, X. Du, J. Xie, J. Lin, J. Sun, Achieving high
160 pseudocapacitance of 2D titanium carbide (MXene) by cation intercalation and surface
161 modification, *Advanced Energy Materials*, 7 (2017) 1602725.

162 [10] R. Yan, E. Josef, H. Huang, K. Leus, M. Niederberger, J.P. Hofmann, R. Walczak,
163 M. Antonietti, M. Oschatz, Understanding the Charge Storage Mechanism to Achieve
164 High Capacity and Fast Ion Storage in Sodium-Ion Capacitor Anodes by Using
165 Electrospun Nitrogen-Doped Carbon Fibers, *Advanced Functional Materials*, (2019)
166 1902858.

167 [11] K. Kaliyappan, Going Beyond Lithium Hybrid Capacitors: Proposing a New High-
168 Performing Sodium Hybrid Capacitor System for Next-Generation Hybrid Vehicles
169 Made with Bio-Inspired Activated Carbon, *Advanced Energy Materials*, 6 (2016).