

Supporting information

Redox engineering of strontium titanate-based thermoelectrics

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Table S1

Binding energies and relative areas of Ti 2p and Mo 3d core-levels for Sr(Ti,Mo)O_{3±δ} ceramics obtained by curve fitting of XPS spectra.

Composition	Ti 2p 3/2		Ti 2p 1/2		Mo 3d 5/2		Mo 3d 3/2	
	BE, eV ^a	Area, % ^b	BE, eV ^a	Area, % ^b	BE, eV ^a	Area, % ^b	BE, eV ^a	Area, % ^b
Sr _{1.00} Ti _{0.94} Mo _{0.06} O _{3±δ}	457.2 (1.10)	3.0	463.4 (2.00)	1.5	227.8 (0.80)	2.6	230.9 (0.96)	1.8
	458.6 (1.10)	63.3	464.4 (2.00)	32.2	230.5 (1.28)	14.4	233.6 (1.40)	10.0
					231.9 (1.28)	19.4	235.0 (1.40)	13.3
					232.8 (1.28)	22.8	236.0 (1.40)	15.7
Sr _{0.97} Ti _{0.94} Mo _{0.06} O _{3±δ}	457.0 (1.13)	2.1	463.1 (2.01)	1.1	227.8 (0.66)	24.2	230.9 (0.86)	16.9
	458.5 (1.13)	64.2	464.3 (2.01)	32.6	230.5 (1.22)	3.6	233.6 (1.44)	2.6
					231.8 (1.22)	11.1	235.0 (1.44)	7.8
					232.8 (1.22)	19.9	235.9 (1.44)	13.9
Sr _{1.00} Ti _{0.97} Mo _{0.03} O _{3±δ}	457.2 (1.08)	1.9	463.3 (1.97)	1.0	227.8 (0.50)	2.9	230.9 (0.80)	2.1
	458.6 (1.08)	64.4	464.4 (1.97)	32.7	230.7 (1.17)	11.4	233.8 (1.13)	7.8
					232.1 (1.17)	28.7	235.2 (1.13)	19.7
					233.1 (1.17)	16.2	236.2 (1.13)	11.2

^aValues between brackets refer to the FWHM (full-width at half maximum) of bands;

^bArea of each component relative to the total core-level peak area.

Table S2

Relative fractions of titanium and molybdenum oxidation states (in %) calculated from the XPS high resolution spectra.

Composition	[Ti ³⁺]/[Ti] _{tot}	[Mo ⁶⁺]/[Mo] _{tot}	[Mo ⁵⁺]/[Mo] _{tot}	[Mo ⁴⁺]/[Mo] _{tot}	[Mo ⁰]/[Mo] _{tot}
SrTi _{0.97} Mo _{0.03} O ₃	2.9	27.4	48.4	19.2	5.0
SrTi _{0.94} Mo _{0.06} O ₃	4.5	38.6	32.7	24.4	4.3
Sr _{0.97} Ti _{0.94} Mo _{0.06} O ₃	3.2	33.8	18.9	6.2	41.1

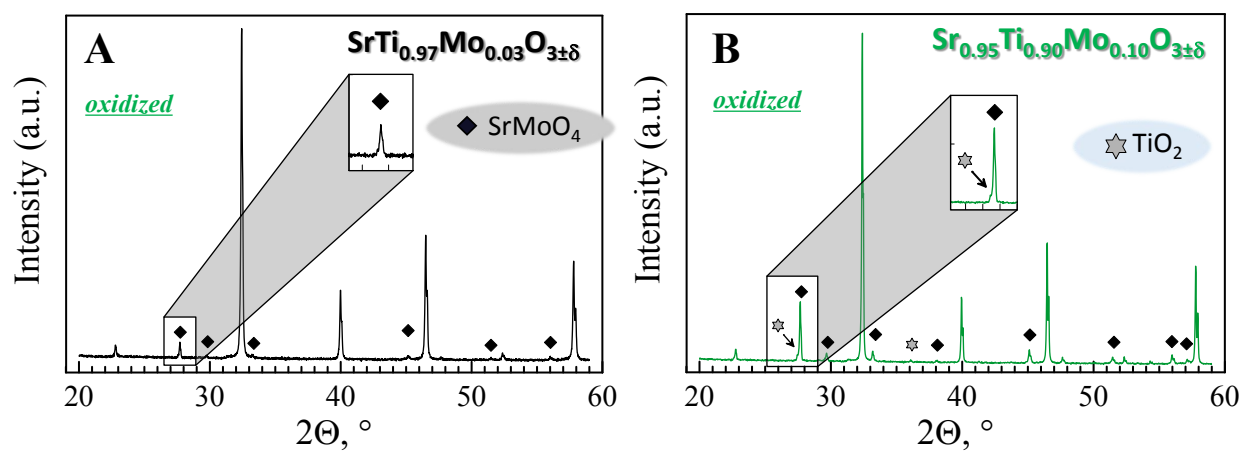


Figure S1. Room-temperature XRD patterns of the oxidized $\text{SrTi}_{0.97}\text{Mo}_{0.03}\text{O}_{3\pm\delta}$ (A) and $\text{Sr}_{0.95}\text{Ti}_{0.90}\text{Mo}_{0.10}\text{O}_{3\pm\delta}$ (B).

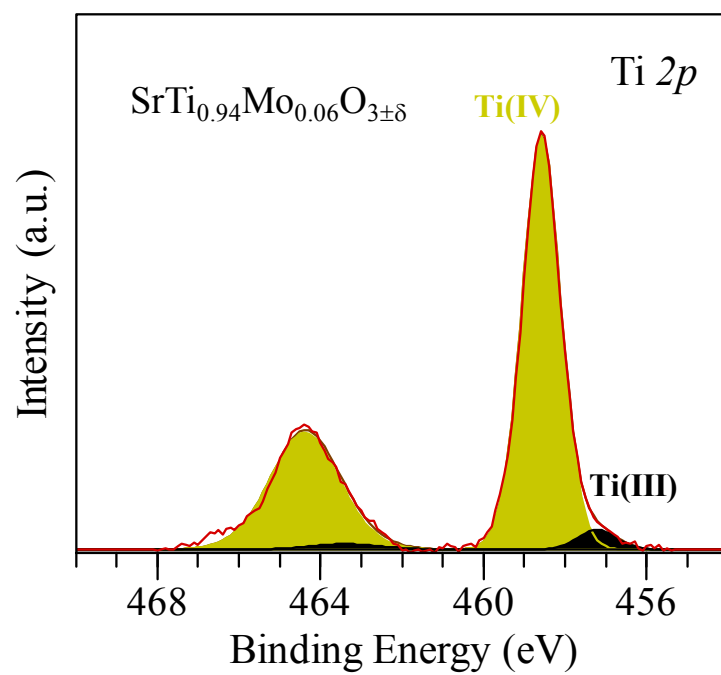


Figure S2. High-resolution XPS spectrum in Ti $2p$ region with the corresponding fit for $\text{SrTi}_{0.94}\text{Mo}_{0.06}\text{O}_{3\pm\delta}$ sample.

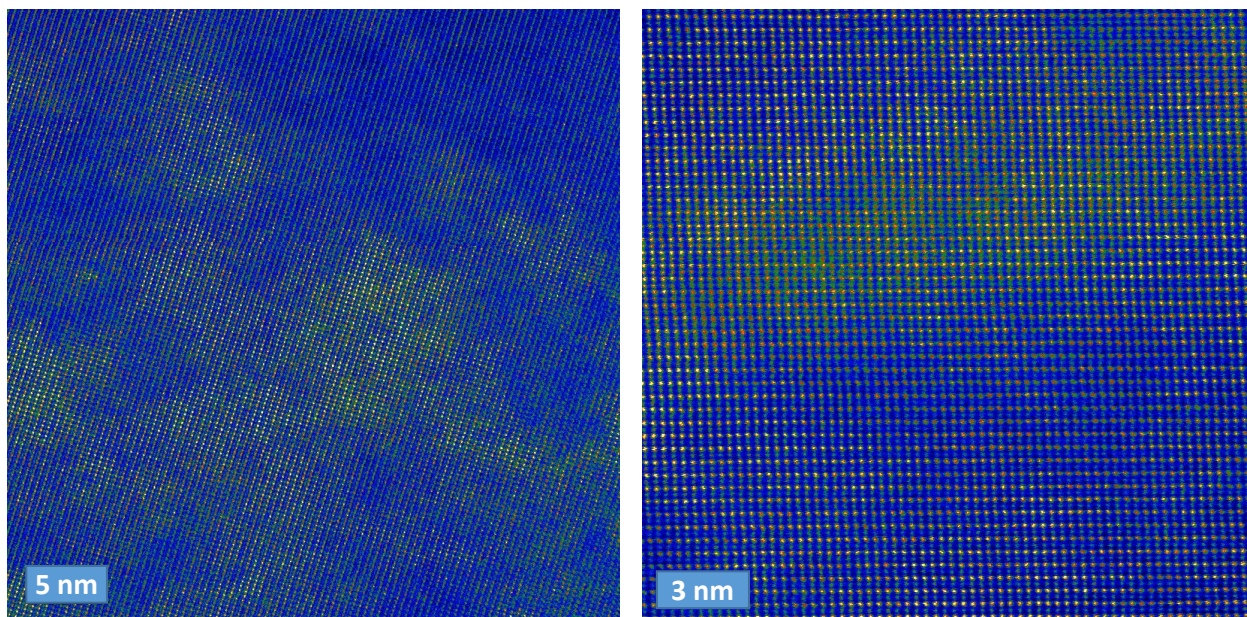


Figure S3. Atomic-resolution elemental maps (blue – lighter, red – heavier) for $\text{Sr}_{0.97}\text{Ti}_{0.94}\text{Mo}_{0.06}\text{O}_{3\pm\delta}$ showing nano-islands composed of SrMoO_3 - inclusions.

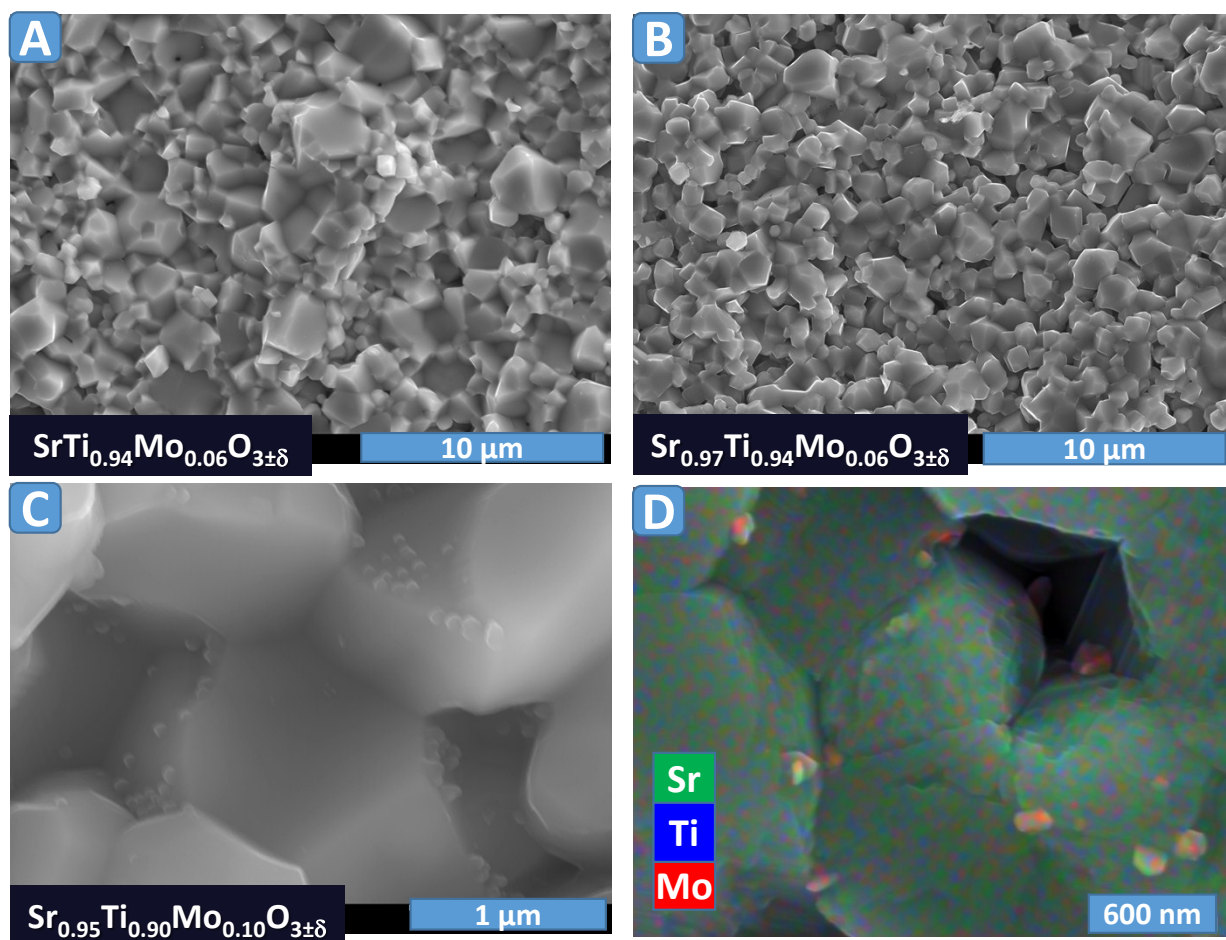


Figure S4. Representative SEM micrographs of $\text{Sr}(\text{Ti},\text{Mo})\text{O}_{3\pm\delta}$ fractured ceramic samples (A-C) and EDS mapping results for $\text{Sr}_{0.95}\text{Ti}_{0.90}\text{Mo}_{0.10}\text{O}_{3\pm\delta}$ (D).

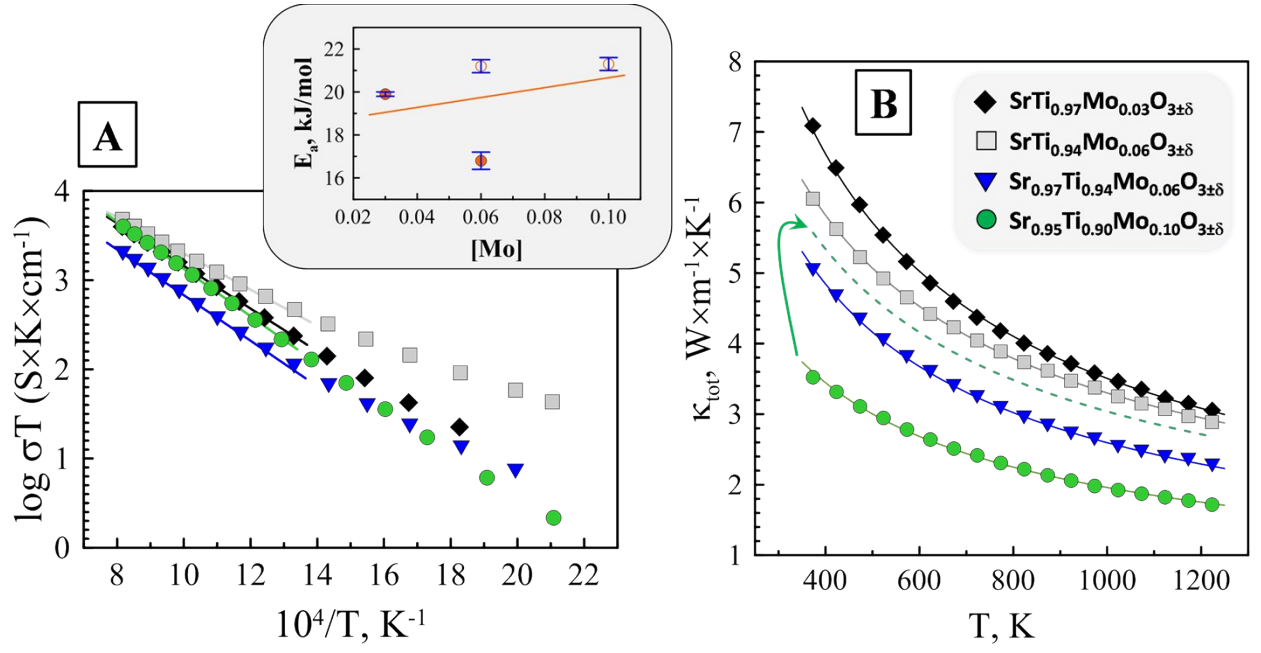


Figure S5. Temperature dependence of the electrical (A) and thermal conductivity (B) of the Sr(Ti,Mo)O_{3±δ} ceramics. The inset in A illustrates the effect of composition on the activation energies of electronic transport at 800-1223 K. The thermal conductivity of Sr_{0.95}Ti_{0.90}Mo_{0.10}O_{3±δ} corrected for porosity effects is shown by a green dashed line.

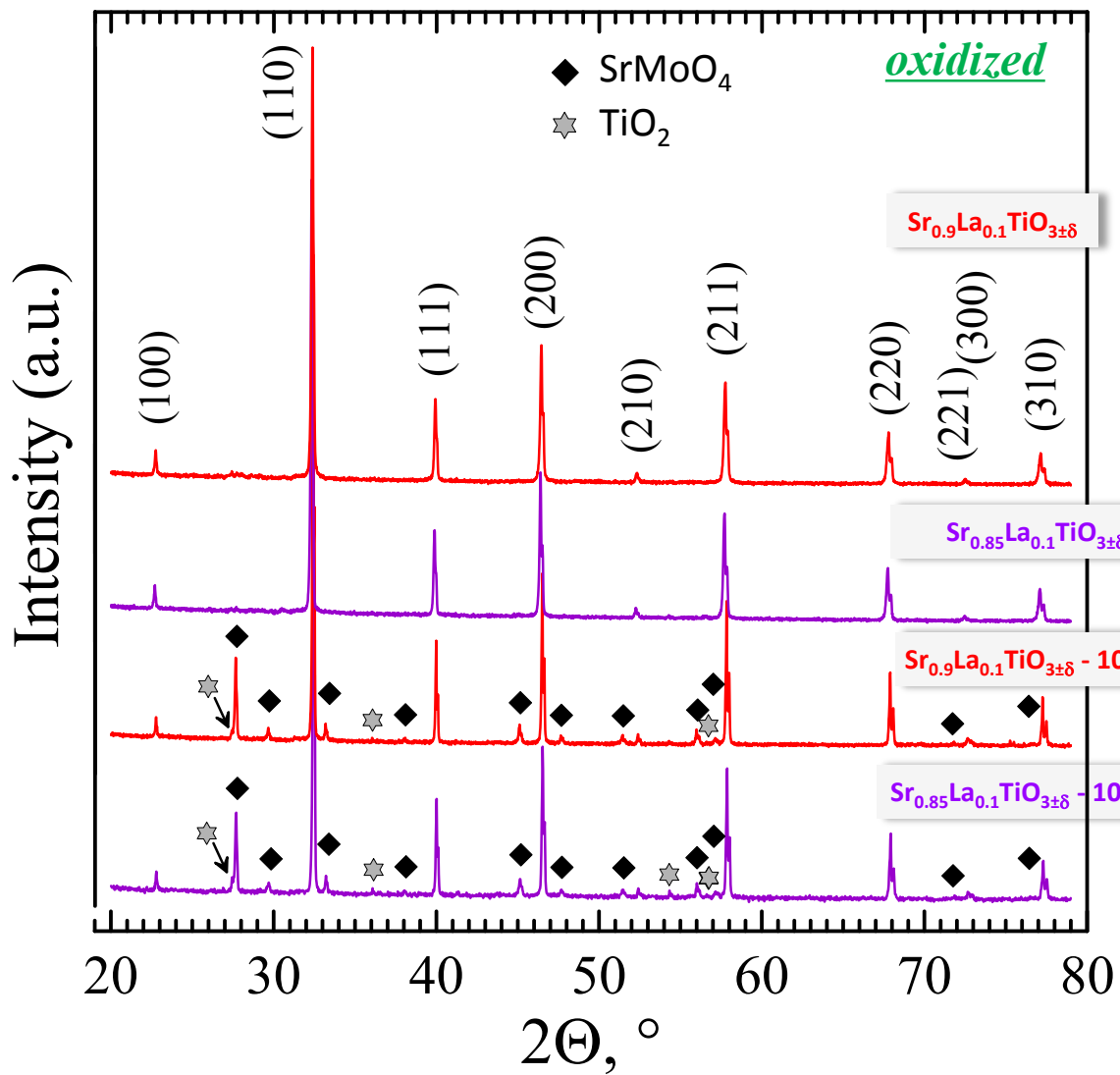


Figure S6. Room-temperature XRD patterns of the oxidized $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$, $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$, $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ and $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ samples.

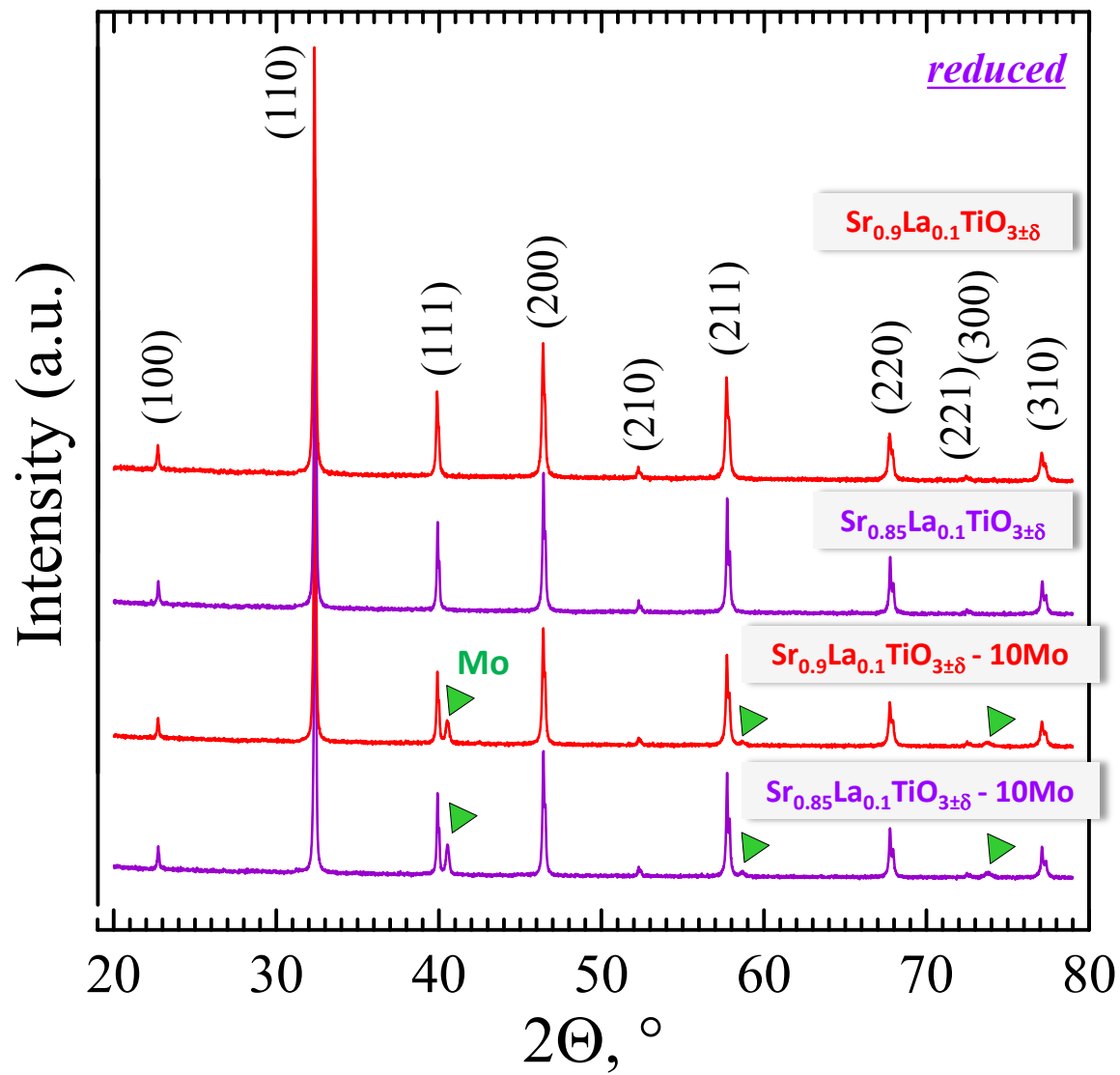


Figure S7. Room-temperature XRD patterns of the reduced $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$, $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$, $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ and $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ samples.

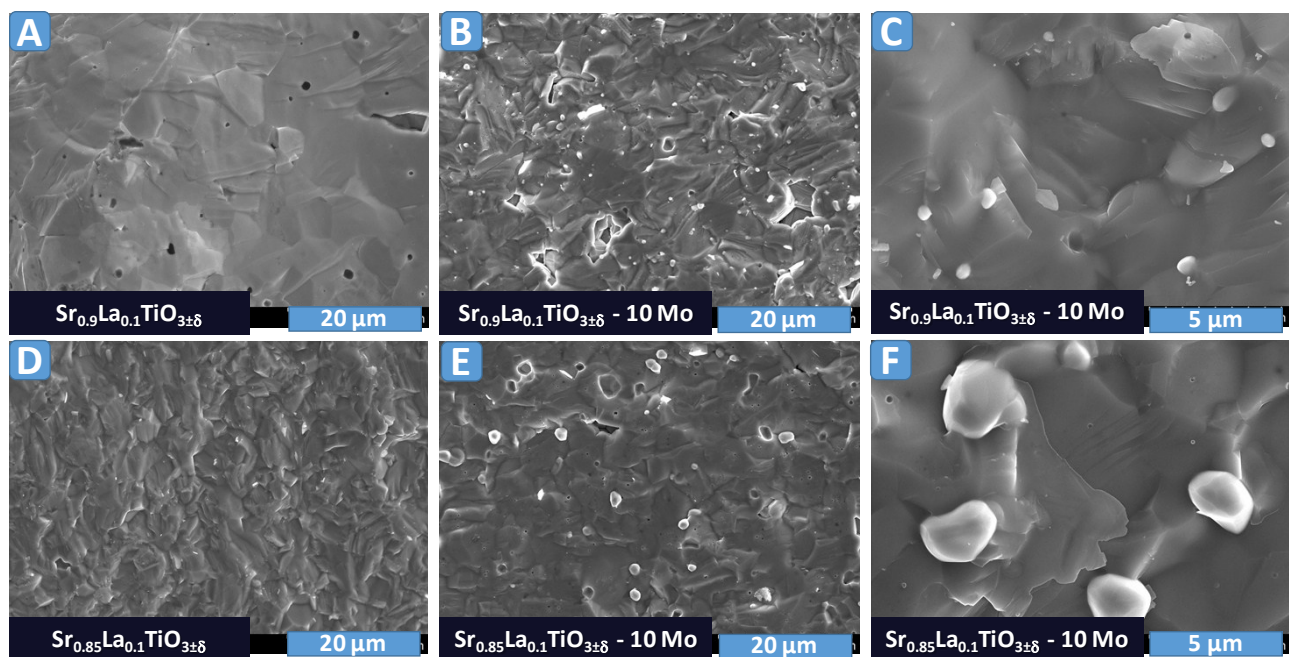


Figure S8. Representative SEM micrographs of the reduced fractured $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$ (A), $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ (B,C), $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$ (D) and $\text{Sr}_{0.85}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ (E,F) samples.

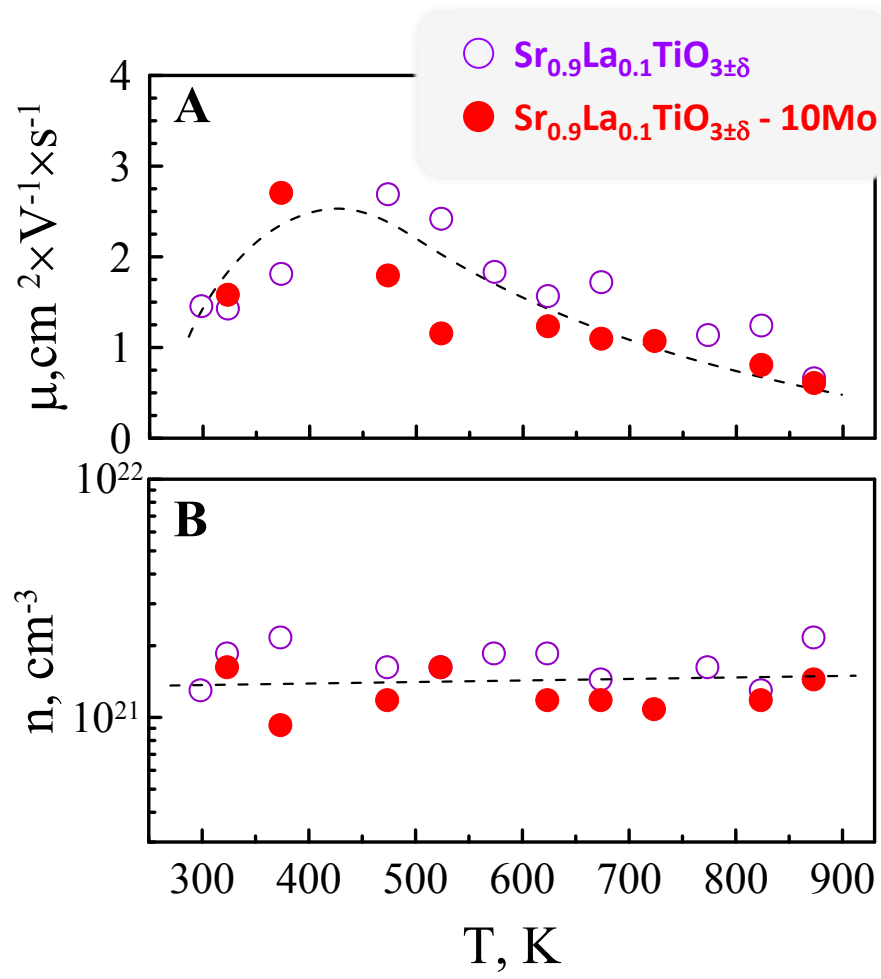


Figure S9. Temperature dependence of the measured charge carrier mobility (A) and charge carrier concentration (B) for reduced $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta}$ and $\text{Sr}_{0.9}\text{La}_{0.1}\text{TiO}_{3\pm\delta} - 10\text{Mo}$ ceramic samples.