Supporting Information for

Graphene-MoS₂ Vertically Anchored on MXene-Derived Accordion-like TiO₂/C

Skeleton: An Ultrastable HER Catalyst

Jian Li, Zi Wen, Zhen Xin Hui, Zhi Wen Chen, Chun Cheng Yang,* Qing Jiang*

Key Laboratory of Automobile Materials (Jilin University), Ministry of Education, and School of Materials Science and Engineering, Jilin University, Changchun 130022, China

^{*} Corresponding authors. Tel.: +86-431-85095371; Fax: +86-431-85095876; E-mails: ccyang@jlu.edu.cn (C. C. Yang); jiangq@jlu.edu.cn (Q. Jiang).

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Materials and methods

Synthesis of T_2C MXene: The Ti₂AlC powder was sieved below 25 µm with a 500mesh sieve before use. For preparing 2D layered Ti₂C MXene, 3 g Ti₂AlC was added into 40 ml 10% HF with mechanical stirring for 12 h. The obtained powder was washed with deionized water and ethanol for several times and dried at 60 °C in a vacuum oven overnight.

Synthesis of $rGO-MoS_2/Acc-TiO_2/C$: Graphene oxide (GO) was synthesized using a modified Hummers' method.^{1,2} For the *in-situ* synthesis of rGO-MoS₂/Acc-TiO₂/C, 40 mg ammonium molybdate tetrahydrate, 0.6 g thiourea and 25 mg Ti₂C MXene were dissolved in 10 ml ultrapure water to form a uniform solution. 0.7 ml GO (5 mg ml⁻¹) was then added into the solution. The mixture solution was transferred into a Teflonlined stainless-steel autoclave and heated to 160 °C for 24 h. The resulting powder was cleaned by centrifugation with deionized water, and dried at 60 °C in a vacuum oven overnight.

Synthesis of $MoS_2/Acc-TiO_2/C$, $rGO-MoS_2$ and MoS_2 : $MoS_2/Acc-TiO_2/C$ and $rGO-MoS_2$ were prepared with the same method as $rGO-MoS_2/Acc-TiO_2/C$ except for the addition of GO and Ti₂C MXene, respectively, before the hydrothermal reaction. MoS_2 was synthesized using the same means as $MoS_2/Acc-TiO_2/C$ without the addition of Ti₂C MXene when preparing the precursor solution.

Synthesis of TiO_2/C : For the fabrication of TiO_2/C , 25 mg as-prepared Ti_2C was dissolved into 10 ml ultrapure water to form a mixture solution, which was then hydrothermally heated to 160 °C for 24 h.

Material Characterization: X-ray diffraction (XRD) patterns were obtained using a D/max2500pc diffractometer with Cu K α radiation ($\lambda = 0.15406$ nm). Raman analysis was performed on a micro-Raman spectrometer (Renishaw) through a 532-nm wavelength laser source. Specific area and pore size distribution were recorded by nitrogen (N₂) adsorption-desorption isotherms with a Micromeritics ASAP 2020 analyzer. A field-emission scanning electron microscope (FESEM, JSM-6700F, JEOL, 15 keV) and a transmission electron microscope (TEM, JEM-2100F, JEOL, 200 keV) were employed to investigate the surface morphology and microstructure. X-ray photoelectron spectroscopy (XPS) measurement was carried out through a monochromatic Al-K_{α} (1486.6 eV) source on an ESCALAB 250Xi spectrometer (Thermo Fisher Scientific, US).

Electrochemical Measurements: The electrochemical measurements were carried out on an Ivium-n-Stat electrochemical workstation in 0.5 M H₂SO₄ at room temperature. A standard three-electrode system was used with a catalyst covered glassy carbon rotating disk (5 mm in diameter) electrode (RDE, Pine Research Instrumentation) as the working electrode, a graphite rod as the counter electrode, and a saturated calomel electrode (Hg/Hg₂Cl₂, SCE) as the reference electrode. To prepare the catalyst ink, 2.5 mg catalyst powders were dispersed into 0.5 ml mixture of waterisopropanol solution (4:1, v/v) containing 50 µl 5 wt% Nafion solution through sonication for 30 min. 15 µl as-prepared catalyst ink was then spread on the surface of a polished glassy carbon electrode (GCE) and dried at room temperature. Linear sweep voltammetry (LSV) was carried out with a rotation rate of 2025 rpm at a scan rate of 5 mV s⁻¹. Electrochemical impedance test was measured in the frequency range from 100 kHz to 0.1 Hz at an overpotential of 250 mV vs. the reversible hydrogen evolution (RHE). Cyclic voltammetry (CV) measurements were conducted under an overpotential range from -200 to 100 mV (vs. RHE) for pristine MoS₂, MoS₂/Acc- TiO_2/C , rGO-MoS₂ and rGO-MoS₂/Acc-TiO₂/C at a sweep rate of 50 mV s⁻¹ and -50 to 100 mV (vs. RHE) for Pt/C at a sweep rate of 100 mV s⁻¹, respectively.

Chronoamperometry was carried out at a current density of -10 mA cm⁻² for 200,000 s. All the measured potentials were converted to the potentials versus RHE by $E_{\text{RHE}} = E_{\text{SCE}} + 0.059 \text{ pH} + E_{SCE}^{0}$, where E_{RHE} , E_{SCE} and E_{SCE}^{0} denote the reversible hydrogen evolution potential, the measured potential and the saturated calomel electrode potential, respectively. N₂ gas flow was carried out through the electrolyte for 30 min to purify the solution before the electrochemical measurements. The Faradic efficiency of hydrogen production was measured at -0.5 V (*vs.* RHE) with a gas chromatograph (GC-2014). CO₂ was introduced into the electrolyte at a rate of 40 ml min⁻¹ as a carrier gas and the H₂ quantification was detected every 10 min through a thermal conductivity detector (TCD).

Computational Details: For catalyst design in this research, we carried out the spin-unrestricted DFT calculations in the Dmol³ code.^{3,4} A generalized gradient approximation (GGA) with Perdew-Burke-Ernzerhof correlation (PBE) was utilized to describe the effects of exchange-correlation.⁵ Core electrons are substituted by a single effective potential through DFT semi-core pseudo potentials (DSPPs) considering the relativistic effects.⁶ Furthermore, the basis set was double numerical plus polarization (DNP) with high quality for the orbital cutoff of 4.9 Å. To get geometrically optimized structures, the convergence criteria were 2.0×10^{-3} Ha Å⁻¹ for the energy gradient, 5.0×10^{-3} Ha Å⁻¹ for the displacement, and 1.0×10^{-5} Ha for the energy change. In this simulation, the actual spacing of k-point sampling was set around 0.03 Å⁻¹ for three directions and three-dimensional periodic boundary conditions were taken.

For the calculation of transition states, we applied the climbing image nudges elastic band method (CI-NEB) as implemented in the Vienna ab-initio simulation package (VASP).⁷ The energy cutoff for the plane-wave basis was set at 400 eV. The convergences of energy and interatomic forces were 10⁻⁶ eV and 0.05 eV Å⁻¹,

respectively. Among the Heyrovsky and Tafel reactions mechanisms, the proton was simulated by two H_2O molecules with one H^+ ($H_5O_2^+$), which has been proved to be successful by previous works.^{8,Ref. 27 of the text}

In this work, we considered both vertical and parallel MoS_2 nanosheets on TiO_2 substrate. The corresponding adhesion energy value (W_{ad}) between MoS_2 and TiO_2 was calculated by

$$W_{\rm ad} = \frac{E_{\rm MoS_2} + E_{\rm TiO_2} - E_{\rm MoS_2 - TiO_2}}{S}$$
(1)

where ${}^{E_{MoS_2}}$, ${}^{E_{TiO_2}}$, ${}^{E_{MoS_2}-TiO_2}$ denoted the total energy of MoS₂, TiO₂, and MoS₂-TiO₂ composites, respectively, while *S* represented the interfacial area between MoS₂ and TiO₂.⁹

Supplementary Figures



Fig. S1 XRD patterns of Ti_2C , MoS_2 , TiO_2/C , $MoS_2/Acc-TiO_2/C$ and $rGO-MoS_2/Acc-TiO_2/C$.



Fig. S2 Raman analysis of comparative samples. (a) Raman spectrum of pristine MoS_2 . (b) Raman spectrum of bare TiO_2/C . The inset is the local spectrum from 1200 to 1800 cm⁻¹.



Fig. S3 The specific surface area and pore size distribution analysis of comparative samples. N_2 adsorption/desorption isotherms and pore size distribution (the inset) of (a) bare MoS₂ and (b) Ti₂C MXene.



Fig. S4 Morphology characterization of comparative samples. SEM images of (a) Ti_2C MXene, (b) bare TiO_2/C , (c) pristine MoS_2 , and (d) rGO-MoS_2.



Fig. S5 Atomic structure of MoS_2/TiO_2 . (a) Top view and (b) side view of atomic structures of MoS_2/TiO_2 , where MoS_2 grows parallel to TiO_2 .



Fig. S6 Identification of the MoS_2 thickness of rGO- $MoS_2/Acc-TiO_2/C$. (a) TEM and (b) HRTEM images of rGO- $MoS_2/Acc-TiO_2/C$.



Fig. S7 XPS analysis of bare MoS_2 . (a) XPS survey spectrum of bare MoS_2 . (b) and (c) High-resolution XPS spectra of Mo 3d and S 2p, respectively.



Fig. S8 XPS analysis of bare Ti_2O/C . (a) XPS survey spectrum of Ti_2O/C . (b)-(d) High-resolution XPS spectra of Ti 2p, C 1s and O 1s, respectively.



Fig. S9 XPS analysis of Ti_2C MXene. (a) XPS survey spectrum of Ti_2C MXene. (b)-(d) High-resolution XPS spectra of Ti 2p, C 1s and O 1s, respectively.



Fig. S10 XPS analysis of $MoS_2/Acc-TiO_2/C$. (a) XPS survey spectrum of $MoS_2/Acc-TiO_2/C$. (b)-(f) High-resolution XPS spectra of Mo 3d, S 2p, Ti 2p, O 1s and C 1s of $MoS_2/Acc-TiO_2/C$, respectively.



Fig. S11 HER performances of the substrate. (a) Polarization curves and (b) Tafel plots of Ti_2C and TiO_2/C .



Fig. S12 Polarization curve of rGO-MoS₂/Acc-TiO₂/C at a potential range from -1.6 to 0 V vs. the standard hydrogen electrode (SHE).



Fig. S13 Polarization curves of bare MoS₂, MoS₂/Acc-TiO₂/C, rGO-MoS₂, rGO-MoS₂/Acc-TiO₂/C and commercial Pt/C without *IR*-compensation.



Fig. S14 Reaction paths of (a) Volmer, (b) Heyrovsky, and (c) Tafel reactions of HER on MoS_2 edges.

Catalyst	Specific surface area (m ² g ⁻¹)	Pore size distribution (nm)	η at 10 mA cm ⁻² (mV)
Ti ₂ C	2.4	mainly 10 ~ 25	
MoS_2	25.0	mainly $7 \sim 25$	289
rGO-MoS ₂ /Acc-TiO ₂ /C	45.3	mainly ~ 4	207

 $\label{eq:sigma} \begin{array}{l} \textbf{Table S1}. \ Summary \ of the specific surface area, pore size distribution and catalytic activity of Ti_2C, \\ MoS_2 \ and \ rGO-MoS_2/Acc-TiO_2/C. \end{array}$

Catalyst	The valence of Ti	The ratio of O-Ti bond to O-	The existence	The ratio of S_2^{2-} to S-containing	η at 10 mA cm ⁻²
		containing bonds	of Mo ⁵⁺	bonds	(mV)
MoS ₂			does not exist	16.2%	289
MoS ₂ /Acc- TiO ₂ /C	Ti ⁴⁺	64.1%	exists	27.2%	239
rGO-MoS ₂ /Acc- TiO ₂ /C	Ti ⁴⁺	51.3%	exists	24.8%	207
rGO-MoS ₂ /Acc- TiO ₂ /C after the stability test	Ti ⁴⁺	8.9%	exists	17.1%	218
Ti ₂ O/C	Ti ⁴⁺	51.0%			
Ti ₂ C	90.5% Ti ³⁺ ; 5.7% Ti ²⁺ ; 3.8% Ti ⁴⁺	14.2%			

Table S2. Summary of the bonding forms of different elements and the corresponding catalytic activity of MoS₂, MoS₂/Acc-TiO₂/C, rGO-MoS₂/Acc-TiO₂/C, rGO-MoS₂/Acc-TiO₂/C after the stability test, Ti₂O/C and Ti₂C.

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