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# **Centric-to-Acentric Structure Transformation Induced by Stereochemically Active Lone Pair: A New Insight for Design of IR Nonlinear Optical Materials**

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## **1. Experimental Section**

### **1.1 Materials and Instruments**

All reagents were obtained from commercial sources and used without further purification. La ingot (2N) was purchased from Hohhot Jinrui Rare Earth Co., Ltd., Cu shot (5N), Sb lump (5N), and S powder (5N) were purchased from Alfa-Aesar.

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Note that the external oxide layer on the surface of La ingot required to be thoroughly scraped before using it as reagent. Powder X-ray diffraction (PXRD) analysis was carried out in a Rigaku Mini-Flex II powder diffractometer (Cu- $K_{\alpha}$ ,  $\lambda = 1.5418 \text{ \AA}$ ). The semi-quantitative energy dispersive X-ray (EDX, Oxford INCA) spectra were measured with a field emission scanning electron microscope (FESEM, JSM6700F). Solid-state UV–Vis–NIR diffuse reflectance spectrum was recorded on a Perkin-Elmer Lambda 950 UV–Vis spectrophotometer with  $\text{BaSO}_4$  as the standard.

### **1.2 Synthesis**

The compound  $\text{La}_2\text{CuSbS}_5$  was prepared from the elements (elemental ratio La : Cu : Sb : S = 2 : 1 : 1 : 5) in a  $\text{BaCl}_2/\text{CsBr}$  (1 : 1) flux at 1273 K. The starting reactants were loaded in a glove-box into carbon-coated silica tubes that were subsequently evacuated to  $10^{-3}$  Pa and sealed. The tubes were heated at 1273 K for 72 h, and then cooled at  $3 \text{ K h}^{-1}$  to 673 K, and later the computer-controlled furnace was shut off. The dark-red flake crystals of  $\text{La}_2\text{CuSbS}_5$  (see Figure S1a) are easily isolated from the products by washing with distilled water and dried by ethanol. The composition La : Cu : Sb : S of approximately 2 : 1 : 1 : 5 was established by EDX with the use of a JSM-6700F SEM (see Figure S1b). The homogeneous  $\text{La}_2\text{CuSbS}_5$  was obtained as analyzed by the PXRD patterns shown in Figure S1c. Title compounds are stable in air more than three months.

### **1.3 Single crystal X-ray Crystallography**

The single-crystal diffraction data collections was taken on a Mercury CCD equipped with graphite-monochromated Mo- $K_{\alpha}$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ) at room-temperature.

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The absorption correction was done<sup>1</sup> and the structure was solved by the direct methods and refined by the full-matrix least-square fitting on  $F^2$  using the *SHELX-2014* software.<sup>2</sup> All of the non-hydrogen atoms were refined with anisotropic thermal parameters and the coordinates were standardized using *STRUCTURE TIDY*.<sup>3</sup> The refinement details are summarized in Table S1. The positional coordinates and isotropic equivalent thermal parameters are given in Table S2, and the selected bond distances are listed in Table S3.

### **2. Computational details**

The DFT calculations have been performed using the *Vienna ab initio simulation package* (VASP)<sup>4-6</sup> with the Perdew-Burke-Ernzerhof (PBE)<sup>7</sup> exchange correlation functional. The projected augmented wave (PAW)<sup>8</sup> potentials with the valence states 3s and 3p for S, 4s and 3d for Cu, 5s and 5p for Sb, 6s and 5d for La, respectively, have been used. A  $\Gamma$ -centered  $3 \times 5 \times 5$  Monkhorst-Pack grid for the Brillouin zone sampling<sup>9</sup> and a cutoff energy of 500 eV for the plane wave expansion were found to get convergent lattice parameters. Both the cell and atomic relaxations were carried out until the residual forces are below 0.02 eV/Å. A Monkhorst-Pack k-point mesh<sup>6</sup> of  $5 \times 9 \times 9$  was used for the calculation of the linear and nonlinear optical properties. To better describe the exchange–correlation effects of localized electrons, a hybrid functionals (HSE06)<sup>10</sup> was used for the electronic structure calculations. The spin-polarization was considered in all calculations.

The imaginary part of the dielectric function due to direct inter-band transitions is given by the expression:

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$$\varepsilon_2(\hbar\omega) = \frac{2e^2\pi}{\Omega\varepsilon_0} \sum_{k,v,c} \left| \langle \psi_k^c | u \cdot r | \psi_k^v \rangle \right|^2 \delta(E_k^c - E_k^v - E) \quad \dots\dots\dots (1)$$

Where  $\Omega$ ,  $\omega$ ,  $u$ ,  $v$  and  $c$  are the unit-cell volume, photon frequencies, the vector defining the polarization of the incident electric field, valence and conduction bands, respectively. The real part of the dielectric function is obtained from  $\varepsilon_2$  by a Kramers-Kronig transformation:

$$\varepsilon_1(\omega) = 1 + \left(\frac{2}{\pi}\right) \int_0^{+\infty} d\omega' \frac{\omega'^2 \varepsilon_2(\omega')}{\omega'^2 - \omega^2} \quad \dots\dots\dots (2)$$

The refractive index  $n(\omega)$  can be obtained based on  $\varepsilon_1$  and  $\varepsilon_2$ .

In calculation of the static  $\chi^{(2)}$  coefficients, the so-called length-gauge formalism derived by Aversa and Sipe<sup>11</sup> and modified by Rashkeev et al<sup>12</sup> is adopted, which has been proved to be successful in calculating the second order susceptibility for semiconductors and insulators.<sup>13-16</sup> In the static case, the imaginary part of the static second-order optical susceptibility can be expressed as:

$$\begin{aligned} \chi^{abc} &= \frac{e^3}{\hbar^2\Omega} \sum_{nml,k} \frac{r_{nm}^a (r_{ml}^b r_{ln}^c + r_{ml}^c r_{ln}^b)}{2\omega_{nm}\omega_{ml}\omega_{ln}} [\omega_n f_{ml} + \omega_m f_{ln} + \omega_l f_{nm}] \\ &+ \frac{ie^3}{4\hbar^2\Omega} \sum_{nm,k} \frac{f_{nm}}{\omega_{mn}^2} [r_{nm}^a (r_{mn;c}^b + r_{mn;b}^c) + r_{nm}^b (r_{mn;c}^a + r_{mn;a}^c) + r_{nm}^c (r_{mn;b}^a + r_{mn;a}^b)] \\ &\dots\dots\dots(3) \end{aligned}$$

where  $r$  is the position operator,  $\hbar\omega_{nm} = \hbar\omega_n - \hbar\omega_m$  is the energy difference for the bands  $m$  and  $n$ ,  $f_{mn} = f_m - f_n$  is the difference of the Fermi distribution functions, subscripts  $a$ ,  $b$ , and  $c$  are Cartesian indices, and  $r_{mn;a}^b$  is the so-called generalized derivative of the coordinate operator in  $k$  space,

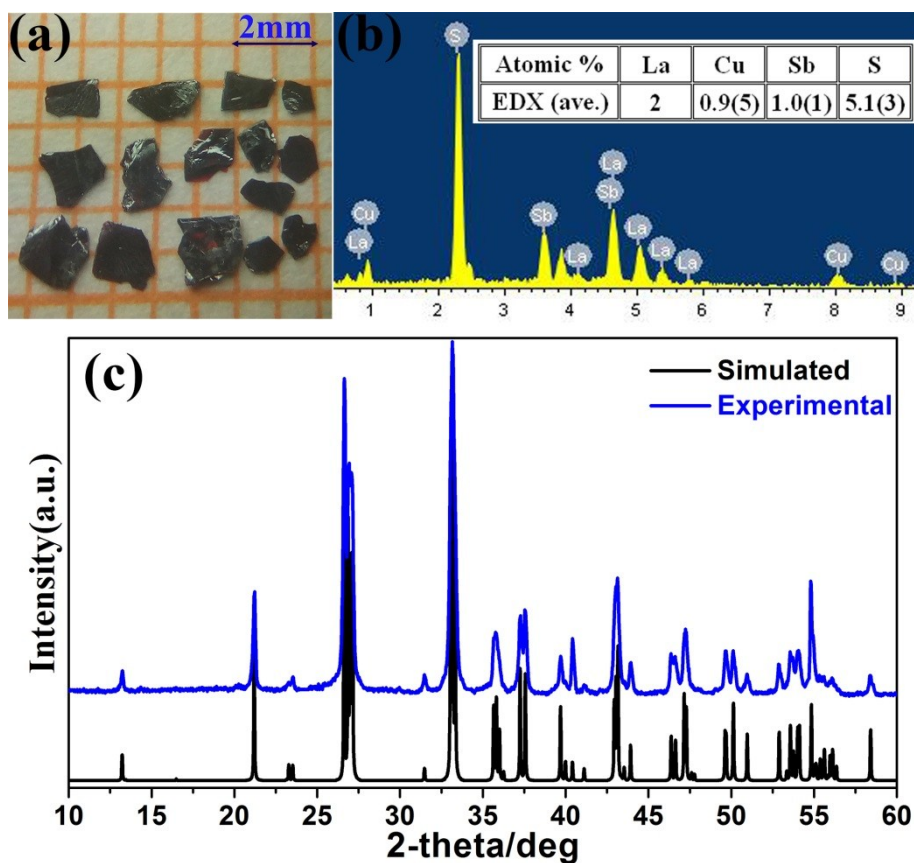
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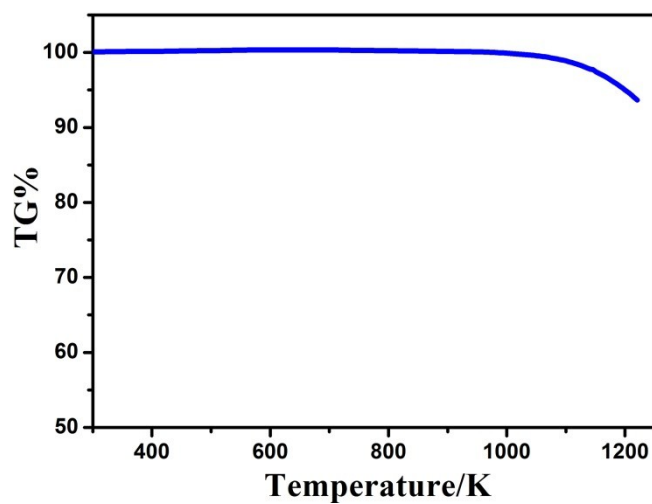
$$r_{nm;a}^b = \frac{r_{nm}^a \Delta_{mn}^b + r_{nm}^b \Delta_{mn}^a}{\omega_{nm}} + \frac{i}{\omega_{nm}} \times \sum_l (\omega_{lm} r_{nl}^a r_{lm}^b - \omega_{nl} r_{nl}^b r_{lm}^a) \dots\dots\dots (4)$$

where  $\Delta_{nm}^a = (p_{nm}^a - p_{mm}^a) / m$  is the difference between the electronic velocities at the bands  $n$  and  $m$ .

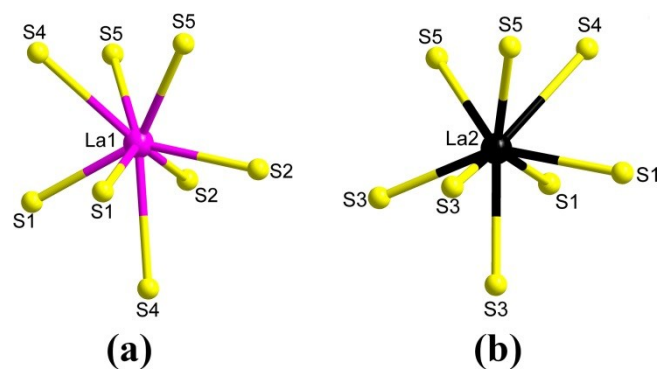
As the nonlinear optical coefficients is sensitive to the momentum matrix, much finer k-point grid and large amount of empty bands are required to obtain a convergent  $\chi^{(2)}$  coefficient. The  $\chi^{(2)}$  coefficients here were calculated from PBE wavefunctions with a  $5 \times 9 \times 9$  k-point grid and about 260 empty bands. A scissor operator has been added to correct the conduction band energy (corrected to the experimental gap), which has been proved to be reliable in predicting the second order susceptibility for semiconductors and insulators.<sup>17-19</sup>



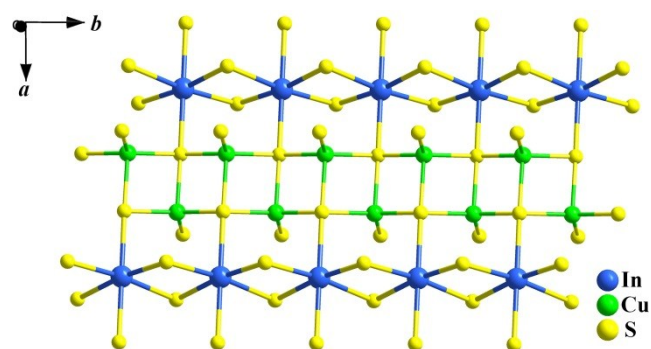
**Figure S1.** The characterizations of  $\text{La}_2\text{CuSbS}_5$ : (a) the photograph of crystals, (b) EDX results; and (c) PXR patterns.



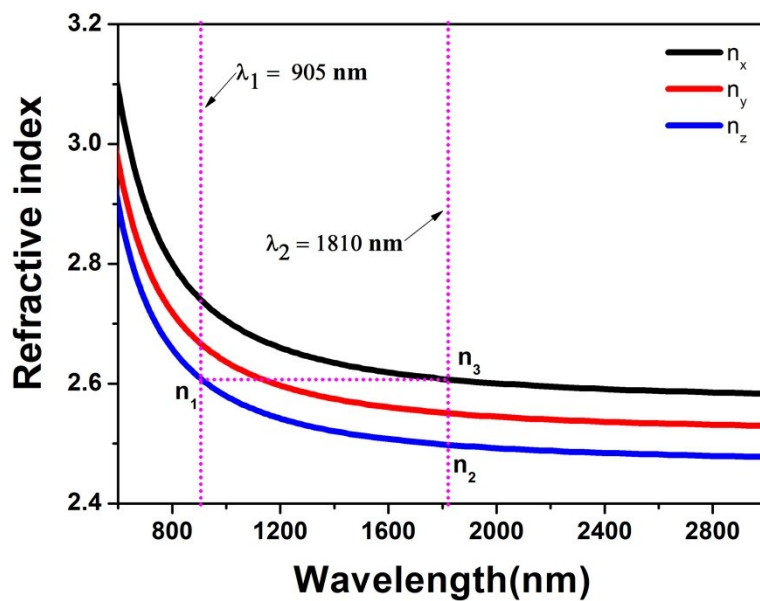
**Figure S2.** The TG curve of  $\text{La}_2\text{CuSbS}_5$ .



**Figure S3.** The local coordination environment of (a) La1 and (b) La2 atoms in  $\text{La}_2\text{CuInS}_5$ .



**Figure S4.** The 1D  $\{[\text{CuSbS}_5]^{6-}\}_n$  chain in  $\text{La}_2\text{CuInS}_5$ .



**Figure S5.** Phase-matching capabilities for La<sub>2</sub>CuSbS<sub>5</sub> at 1810 nm. The minimum refractive index at 905 nm ( $n_1$ ) located between the minimum ( $n_2$ ) and maximum refractive index ( $n_3$ ) at 1810 nm indicates the fulfillment of the requirement of the phase-matching condition.



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**Table S1.** Crystallographic data and refinement details for La<sub>2</sub>CuSbS<sub>5</sub>.

Empirical formula	La <sub>2</sub> CuSbS <sub>5</sub>
Formula weight	623.41
Temperature(K)	293(2)
Crystal system	Orthorhombic
Space group	<i>Ima2</i> (No.46)
<i>a</i> (Å)	13.382(6)
<i>b</i> (Å)	7.558(3)
<i>c</i> (Å)	7.635(3)
<i>V</i> (Å <sup>3</sup> )	772.1(5)
<i>Z</i>	4
<i>D<sub>c</sub></i> (g·cm <sup>-3</sup> )	5.36
$\mu$ (mm <sup>-1</sup> )	18.27
GOOF on <i>F</i> <sup>2</sup>	1.092
<i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> ( <i>I</i> > 2σ( <i>I</i> )) <sup>a</sup>	0.0439, 0.1145
<i>R</i> <sub>1</sub> , <i>wR</i> <sub>2</sub> (all data)	0.0444, 0.1147
Largest diff. peak and hole (e·Å <sup>-3</sup> )	3.527, -2.741

<sup>a</sup>:  $R_1 = \frac{\sum ||F_o| - |F_c||}{\sum |F_o|}$ ,  $wR_2 = [\frac{\sum w(F_o^2 - F_c^2)^2}{\sum w(F_o^2)^2}]^{1/2}$

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**Table S2.** Atomic coordinates and equivalent isotropic displacement parameters of  $\text{La}_2\text{CuSbS}_5$ .

Atom	Wyckff	$x$	$y$	$z$	$U_{\text{eq}}(\text{\AA})^a$
La1	$4a$	0	0	0.0011(2)	0.0115(4)
La2	$4b$	0.25	0.1341(2)	0.3864(3)	0.0117(4)
Cu	$4a$	0	0	0.5400(8)	0.054(2)
Sb	$4b$	0.25	0.6716(3)	0.2188(3)	0.016(5)
S1	$4b$	0.25	0.508(2)	0.502(2)	0.019(2)
S2	$8c$	0.1054(4)	0.3197(7)	0.1744(6)	0.011(2)
S3	$8c$	0.6037(4)	0.1644(7)	0.3295(7)	0.010(2)

$U_{\text{eq}}$  is defined as one third of the trace of the orthogonalized  $U_{ij}$  tensor.

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**Table S3.** Selected bond lengths ( $\text{\AA}$ ) of  $\text{La}_2\text{CuSbS}_5$ .

La1–S2 $\times 2$	3.095(5)	La2–S2 $\times 2$	2.886(5)
La1–S3 $\times 2$	3.123(5)	La2–S2 $\times 2$	2.950(5)
La1–S2 $\times 2$	3.172(5)	La2–S1	2.961(8)
La1–S3 $\times 2$	3.174(5)	La2–S3 $\times 2$	3.019(6)
La1–S1 $\times 2$	3.346(2)	La2–S1	3.130(2)
Cu–S2 $\times 2$	2.214(6)	Sb–S3 $\times 2$	2.467(6)
Cu–S3 $\times 2$	2.460(7)	Sb–S1	2.486(2)
		Sb–S1	2.934(9)

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**Table S4.** SHG intensities comparison of the well-known IR-NLO sulfides in recent

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years at the same measured wavelength ( $\lambda = 2050$  nm).

Compounds	SHG	LIDT	Ref.
SnGa <sub>4</sub> S <sub>7</sub>	1.3 × AGS	19 × AGS	20
Li <sub>2</sub> ZnSiS <sub>4</sub>	1.1 × AGS	10 × AGS	21
Na <sub>2</sub> ZnGe <sub>2</sub> S <sub>6</sub>	0.9 × AGS	6 × AGS	22
Li <sub>2</sub> BaSnS <sub>4</sub>	0.7 × AGS	6.5 × AGS	23
Li <sub>2</sub> BaGeS <sub>4</sub>	0.5 × AGS	11 × AGS	23
Ba <sub>6</sub> Zn <sub>7</sub> Ga <sub>2</sub> S <sub>16</sub>	0.5 × AGS	28 × AGS	24
BaAl <sub>4</sub> S <sub>7</sub>	0.5 × AGS	10 × AGS	25
Li <sub>2</sub> MnGeS <sub>4</sub>	0.5 × AGS	40 × AGS	26
Na <sub>2</sub> BaSnS <sub>4</sub>	0.5 × AGS	5 × AGS	27
Na <sub>2</sub> BaGeS <sub>4</sub>	0.3 × AGS	8 × AGS	27
<b>La<sub>2</sub>CuSbS<sub>5</sub></b>	<b>0.5 × AGS</b>	<b>6.7 × AGS</b>	<b>This work</b>
AgGaS <sub>2</sub> (AGS)	1.0 × AGS	1.0 × AGS	28

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