

# Theoretical Discovery of half-metallic Dirac dispersion in experimentally synthesized two dimensional metal semiquinoid frameworks

Table S1. Energies for M-SF under different magnetic states (ferromagnetic (FM), antiferromagnetic (AFM), and nonferromagnetic (NM)). The bold energies are relatively lower in each configurations.

Energy (eV)	FM	AFM	NM
Ti-SF	<b>-281.027</b>	-280.944	-280.024
V-SF	-280.038	-280.230	<b>-280.329</b>
Cr-SF	-278.894	<b>-279.144</b>	-277.726
Fe-SF	<b>-273.192</b>	-272.759	-272.203

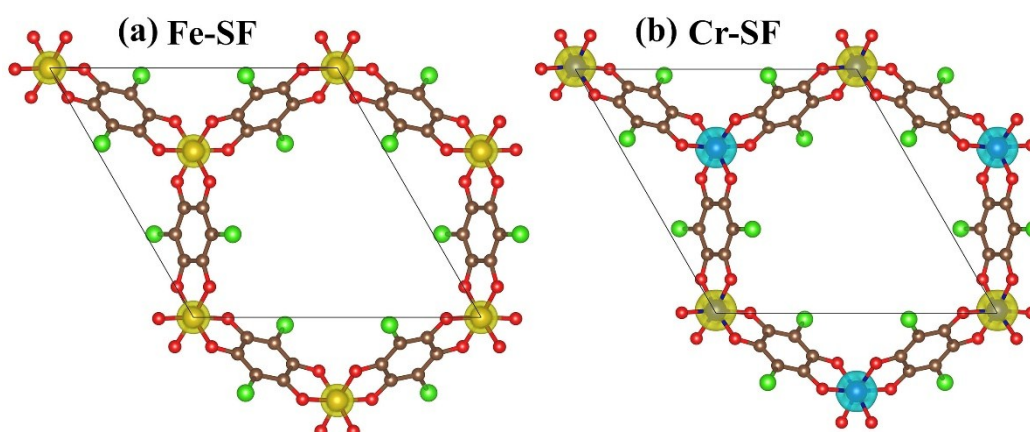
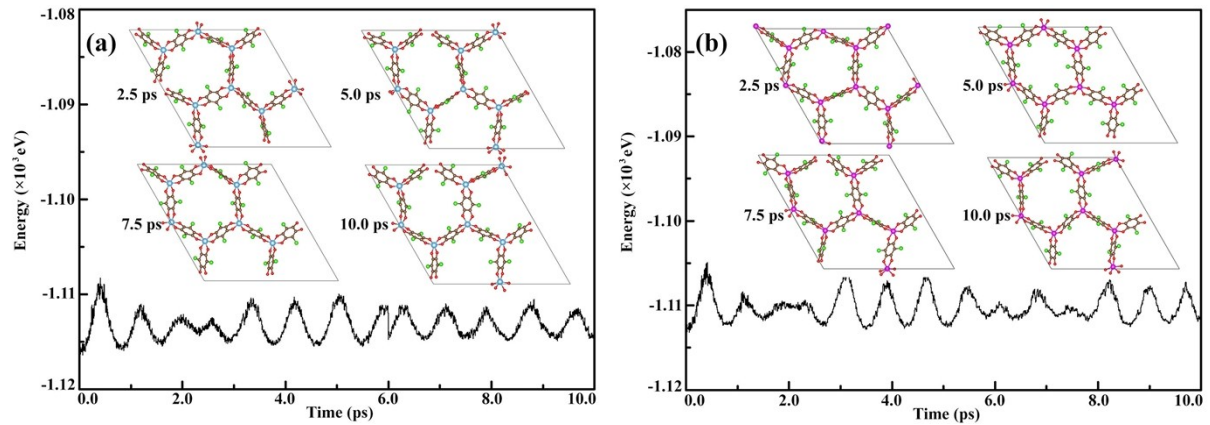


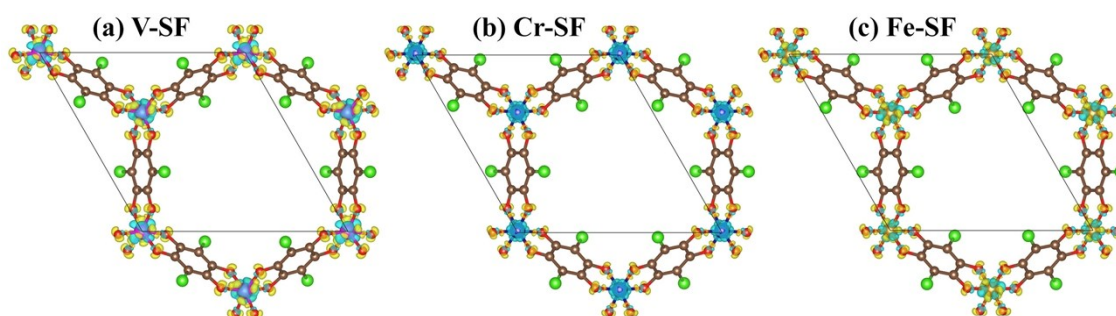
FIG. S1. The spin density of (a) Fe-SF (ferromagnetic) and (d) Cr-SF (antiferromagnetic) with isovalue of  $0.02 \text{ e}/\text{\AA}^3$ .

## Thermodynamic stability

The thermodynamic stability was evaluated through Ab initio molecular dynamics (AIMD) simulations in a  $2 \times 2 \times 1$  supercell at room temperature (300K), based on the Nosé-Hoover method. The whole simulations last 10 picoseconds (ps) with a time step of 1 femtosecond (fs).



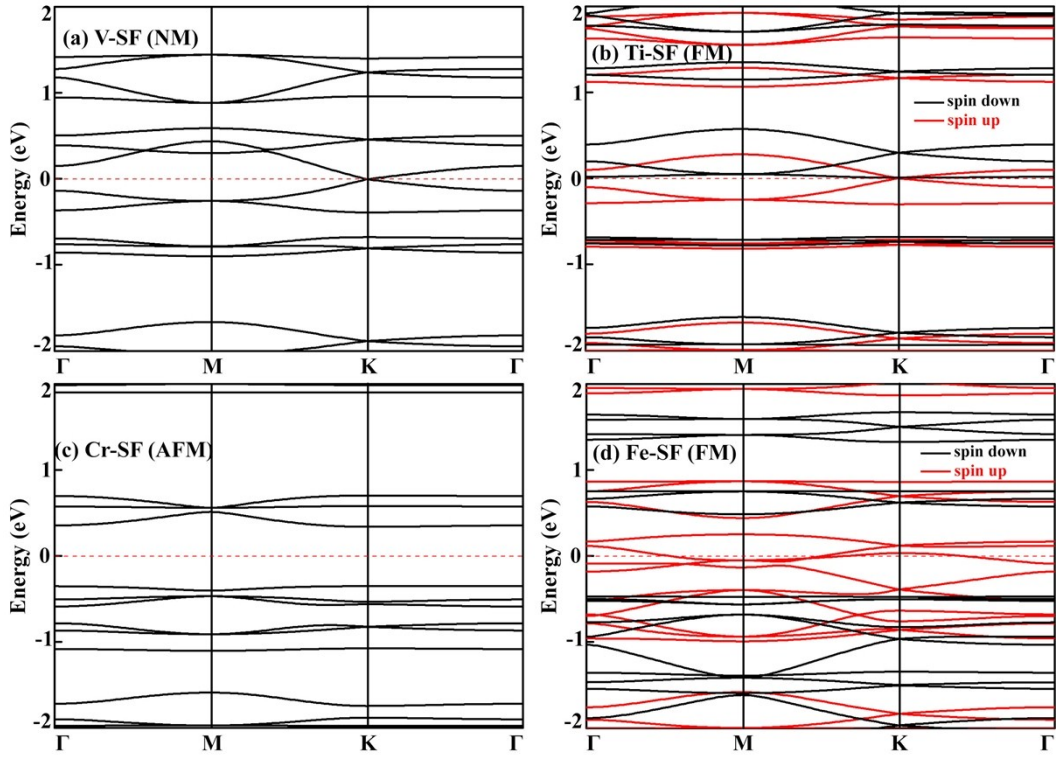
**FIG. S2.** Total potential energies of (a) Ti-SF, and (b) V-SF fluctuate during the AIMD simulation at 300K. The top view of Ti- and V-SF during the whole simulation are inserted in the pictures.



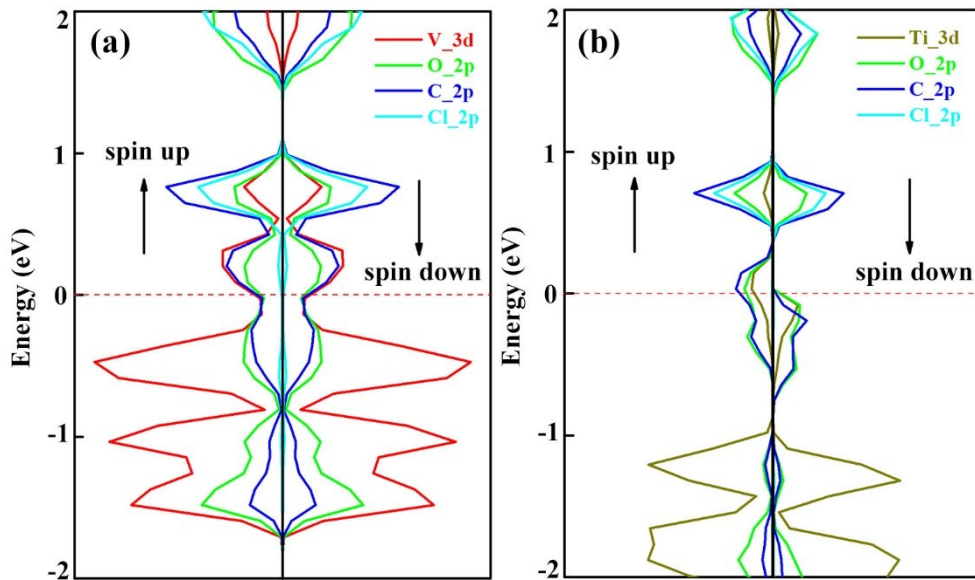
**FIG. S3.** Charge density difference for (a) V, (b) Cr, and (c) Fe-SF with respect to the metal atoms and ligands. The yellow and cyan region represent the electron accumulation and depletion area with an isovalue of  $0.02 \text{ e}/\text{\AA}^3$ .

**Table S2.** Bader charge analysis of M-SF.

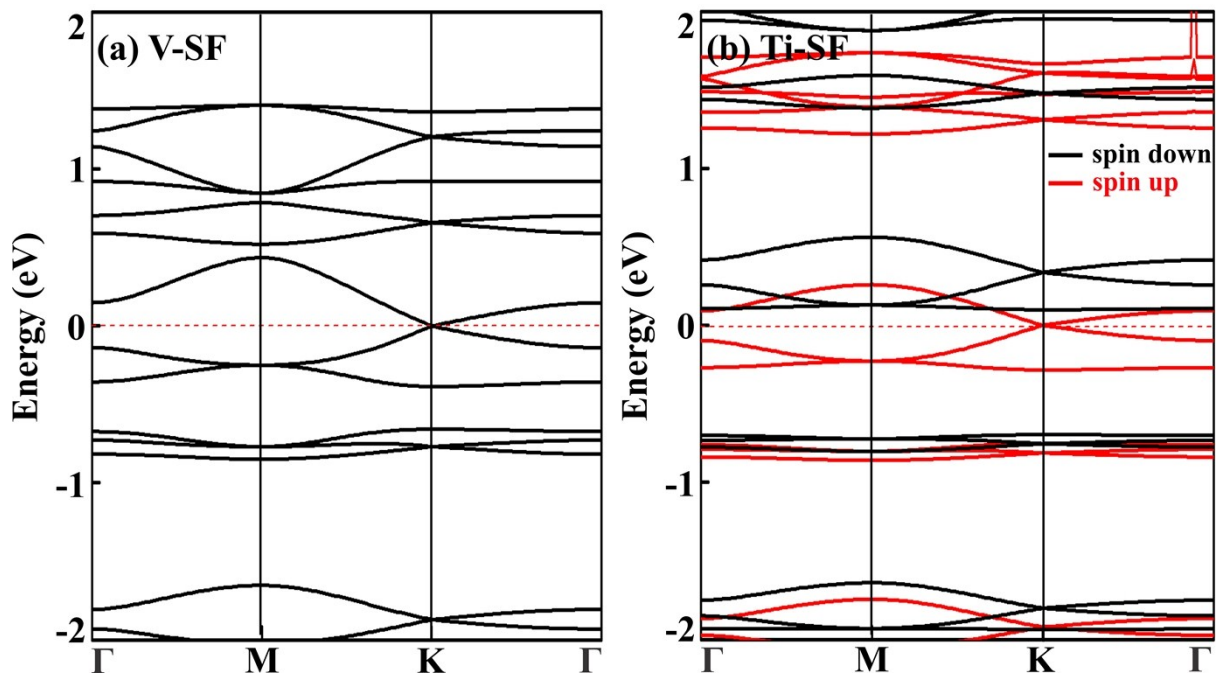
	Average atomic charge (a.u.)		
	metal atom	O atom	C atom
Ti-SF	+1.90	-0.91	+0.05
V-SF	+2.14	-0.95	+0.44
Cr-SF	+1.81	-0.92	+0.42
Fe-SF	+1.55	-0.89	+0.39



**FIG. S4.** Band structures of M-SF at magnetic ground states calculated by GGA\_PBE functional.



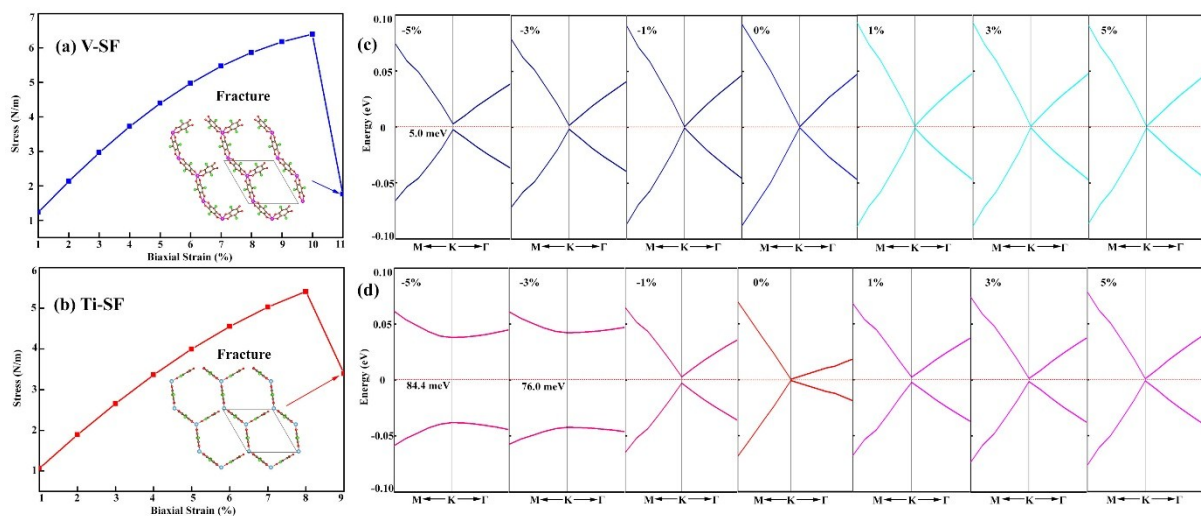
**FIG. S5.** Density of states of (a) V-SF and (b) Ti-SF. Fermi level (red dashed line) has been set to zero.



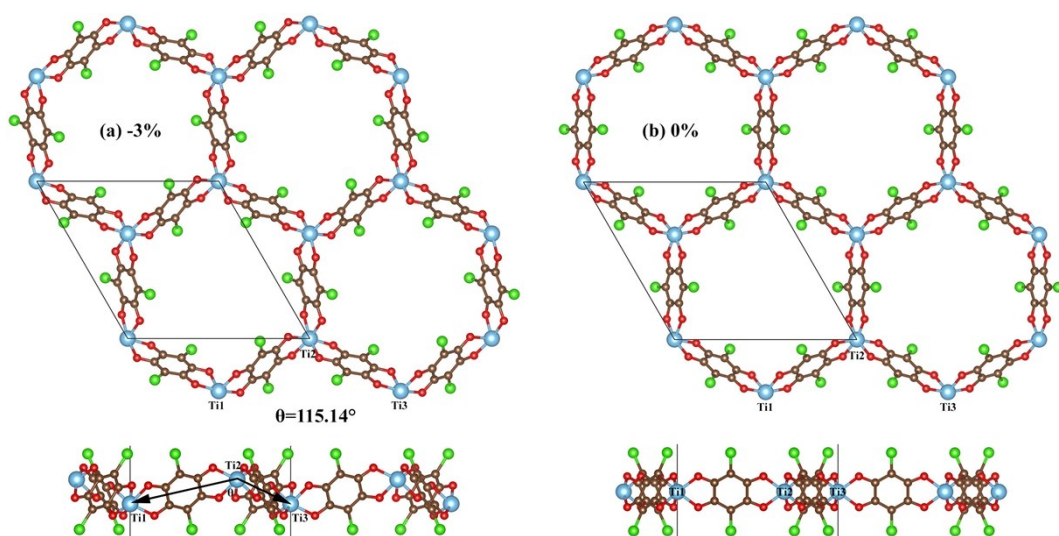
**FIG. S6.** Band structures of V-SF and Ti-SF at magnetic ground states calculated by GGA+U method, the U values are set to 2.0 eV for V, and 2.3 eV for Ti 3d orbitals<sup>1</sup>.

## Strain effect.

Generally speaking, the mechanical strength of M-SF is not very well, the critical breaking strain of V-SF and Ti-SF are 10% and 8%, respectively (Fig. S7). Therefore, we only employed a small biaxial strain (from -5% to 5%) in our band calculations. For V-SF, strains show little impact on Dirac points, only a bandgap of 5.0 meV opens under -5% compression. Whereas, for Ti-SF, small bandgaps appear under tension and -1% compression. It is noteworthy that the Dirac cones will be broken when the compression reaching -3% in Ti-SF, due to the rotation of organic ligands (Fig. S8). The fermi velocity of the conical bands under strains show on significant change in comparison with the full relaxed structures.



**FIG. S7.** Strains in the (a) V-SF and (b) Ti-SF subjected to biaxial strain. Band structures of (c) V-SF and (d) Ti-SF near K point under finite equal-biaxial strain. HSE functional is employed in band calculations. The fermi level (red dashed line) has been set to zero.



**FIG. S8.** The optimized structures of Ti-SF under the strain of (a) -3% and (b) 0%.

## The calculation of exchange interaction ( $J$ )

For Ti-SF, the centre metal atoms (Ti) show ferromagnetic coupling with its neighbouring atoms. Here, to be simple, we regard the Ti atoms as the magnetic centre (Fig. S9), so that, Ti-SF will show similar magnetic structure with  $\text{CrI}_3$ .<sup>2</sup> According to Heisenberg model, the spin Hamiltonian can be written as:

$$H = -J \sum_{\langle i,j \rangle} \sigma_i \sigma_j$$

where  $\sigma$  is +1 or -1 at each magnetic site. Thus, the energies for the magnetic configuration FM and AFM are derived as following equations:

$$E_{FM} = E_0 - \frac{3}{2}NJ|\sigma|^2$$

$$E_{AFM} = E_0 + \frac{3}{2}NJ|\sigma|^2$$

therefore,

$$J = \frac{E_{AFM} - E_{FM}}{3N|\sigma|^2}$$

where  $N$  is the number of the magnetic atom in the unit cell.  $E_{AFM}$  and  $E_{FM}$  represent the energies under AFM and FM magnetic order, respectively. Besides, on the basis of our DFT calculation,  $T_c$  for  $\text{CrI}_3$  is 46 K, which is very close to the experimental value<sup>3</sup>.

### Tight binding model.

Due to the strong  $\pi$ -d conjugation and local charge density near fermi level, here we treated the central metals as the vertexes of a honeycomb sublattice, like the Carbon atoms of Graphene. Considering the hopping between nearest neighbors (site A and B), the TB

Hamiltonian  $H(\vec{k}) = \begin{pmatrix} \varepsilon & tf^* \\ tf & \varepsilon \end{pmatrix}$  leads to two energy eigenvalues

$$E_{1,2} = \varepsilon \pm t\sqrt{f(\vec{k}) \times f^*(\vec{k})}$$

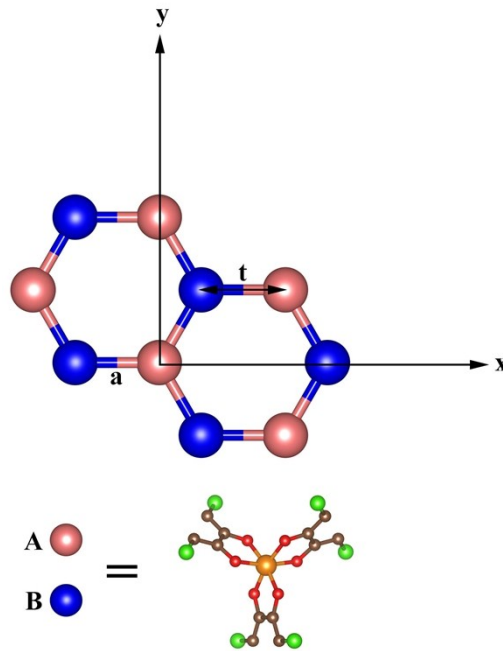
with

$$f(\vec{k}) \times f^*(\vec{k}) = 3 + 4\cos\left(\frac{\sqrt{3}}{2}ak_y\right)\cos\left(\frac{3}{2}ak_x\right) + 2\cos\left(\frac{\sqrt{3}}{2}ak_y\right)$$

and

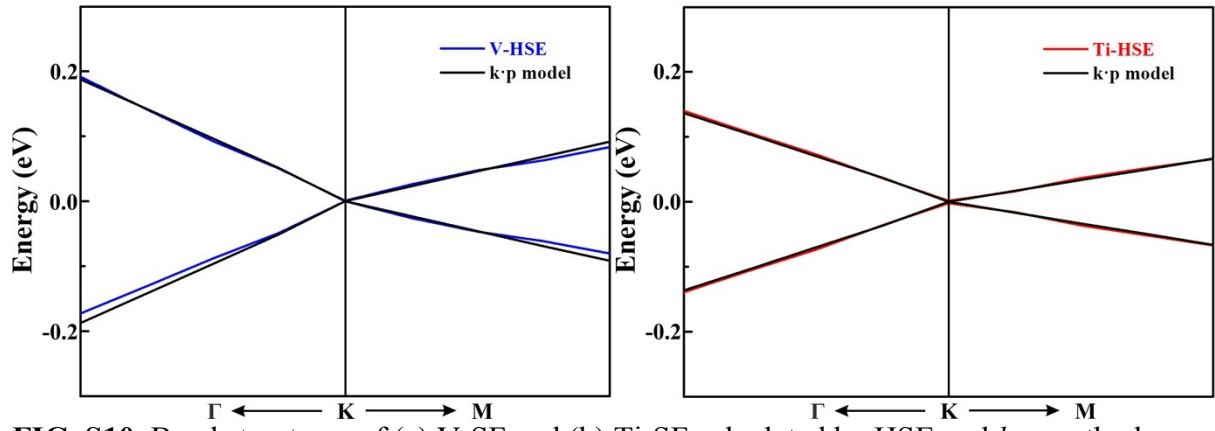
$$t = \langle \varphi_A | H | \varphi_B \rangle$$

Then we fit our HSE band structures of V-SF and Ti-SF in the first BZ with our TB model.



**FIG. S9.** Tight binding model for M-SF. Pink and blue balls represent different sites in the unit cell.

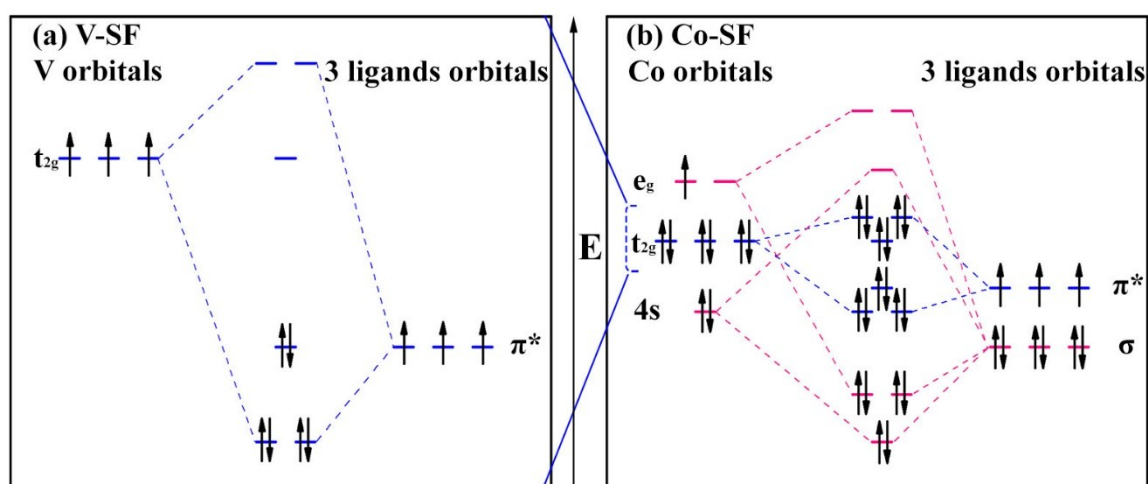




**FIG. S10.** Band structures of (a) V-SF and (b) Ti-SF calculated by HSE and  $k \cdot p$  methods.

## Molecular orbital model.

Inspired by the previous works<sup>4-6</sup>, herein, we perfectly explained the magnetism of M-SF with a local  $D_3$  symmetry at the metal-center, and successfully predicted the magnetism of unsynthesized metal-semiquinoid frameworks. First, we considered that the bonding  $\pi$  orbitals of organic ligands are full-filled, because the outmost orbitals of the ligand atoms are all over half occupied (non-metal atoms). Therefore, the orbitals with  $\pi$  features ought to be  $\pi^*$  orbitals of ligands<sup>4</sup>. Besides, as mentioned above, the highest valence band and lowest conduction band of M-SF ligands both show  $\pi^*$  orbital feature, so that, we considered that these orbitals are half occupied. Previous works gave a simple combination of  $\pi^*$  and  $t_{2g}$  orbitals<sup>5</sup> (Fig. S11(a)). This molecular orbital model only can be used when the metal atoms have no more than 6 3d electrons. In order to predict the magnetism of all 3d transition metals, the combination of ligand  $\sigma$  and metal 4s and  $e_g$  orbitals should be involved. As an example, we exhibited molecular orbital model of V-SF and Co-SF. The previous model shows no single electrons in V-SF molecular orbitals, indicating its nonmagnetic properties, which also can be illustrated by our molecular model (Fig. S11(b)). However, through considering the interaction between ligand  $\sigma$  and metal 4s and  $e_g$  orbitals, our molecular orbital model can predict the magnetic state of metal-semiquinoid frameworks with all 3d transition metal center. Here we predicted that M-SF (M=Sc, V, Co, Cu) are all nonmagnetic, which can be confirmed by our DFT calculations (Table S3).



**FIG. S11.** Proposed molecular orbital model of (a) V-SF and (b) Co-SF with local  $D_3$  symmetry at the metal center.

**Table S3.** Energies of M-SF (here, M represents all 3d transition metals) under different magnetic states (ferromagnetic (FM), antiferromagnetic (AFM), and nonferromagnetic (NM)). The magnetic moments ( $M_{M-SF}$ ) under ferromagnetic state of them are shown in the table. The bold energies are relatively lower in each configurations.

	Energy (eV)			$ M_{M-SF} $ ( $\mu_B$ )
	FM	AFM	NM	
Sc-SF	-279.206	-279.729	<b>-279.731</b>	<b>0.0</b>
Ti-SF	<b>-281.027</b>	-280.944	-280.024	2.0
V-SF	-280.038	-280.230	<b>-280.329</b>	<b>0.0</b>
Cr-SF	-278.894	<b>-279.144</b>	-277.726	6.0
Mn-SF	-275.997	<b>-276.245</b>	-275.625	4.0
Fe-SF	<b>-273.192</b>	-272.759	-272.203	2.0
Co-SF	-269.509	-270.134	<b>-270.135</b>	<b>0.0</b>
Ni-SF	<b>-264.306</b>	-264.211	-264.304	2.0
Cu-SF	-259.241	-259.240	<b>-259.454</b>	<b>0.0</b>
Zn-SF	<b>-257.814</b>	-257.720	-256.484	2.0

## Reference

1. S. Lutfalla, V. Shapovalov and A. T. Bell, *Journal of chemical theory and computation*, 2011, **7**, 2218-2223.
2. Y. Zhao, L. Lin, Q. Zhou, Y. Li, S. Yuan, Q. Chen, S. Dong and J. Wang, *Nano letters*, 2018, **18**, 2943-2949.
3. C. Gong, L. Li, Z. Li, H. Ji, A. Stern, Y. Xia, T. Cao, W. Bao, C. Wang and Y. Wang, *Nature*, 2017, **546**, 265.
4. H. H. Downs, R. M. Buchanan and C. G. Pierpont, *Inorganic Chemistry*, 1979, **18**, 1736-1740.
5. M. E. Ziebel, L. E. Darago and J. R. Long, *Journal of the American Chemical Society*, 2018, **140**, 3040-3051.
6. L. Zhang, Z. Wang, B. Huang, B. Cui, Z. Wang, S. Du, H.-J. Gao and F. Liu, *Nano letters*, 2016, **16**, 2072-2075.