

Tunable 3D light trapping architectures based on self-assembled SnSe₂ nanoplate arrays for ultrasensitive SERS detection

Weiwei Li,^{1,2#} Lei Xiong,^{1,3#} Nianci Li,^{1,4} Shuo Pang,^{1,2} Guoliang Xu,^{1,2} Chenghan Yi,^{1,2}
Zhixun Wang,⁵ Guoqiang Gu,⁶ Kaiwei Li,⁷ Weimin Li,¹ Lei Wei,⁴ Guangyuan Li,^{1*}
Chunlei Yang,^{1*} and Ming Chen^{1*}

¹*Shenzhen Institutes of Advanced Technology, Chinese Academy of Sciences,
Shenzhen 518055, People's Republic of China*

²*Department of Nano Science and Technology Institute, University of Science and
Technology of China, Suzhou 215123, People's Republic of China*

³*School of Information Science and Engineering, Yunnan University, Yunnan 650500,
People's Republic of China*

⁴*School of Computer and Control Engineering, University of Chinese Academy of
Sciences, Beijing 100049, People's Republic of China*

⁵*School of Electrical and Electronic Engineering, Nanyang Technological University,
50 Nanyang Avenue, 639798, Singapore*

⁶*Department of Electrical and Electronic Engineering, Southern University of Science
and Technology, Shenzhen, 518055, People's Republic of China*

⁷*Guangdong Provincial Key Laboratory of Optical Fiber Sensing and
Communications, Institute of Photonics Technology, Jinan University, Guangzhou,
510632, People's Republic of China*

**email: gy.li@siat.ac.cn; cl.yang@siat.ac.cn; ming.chen2@siat.ac.cn*

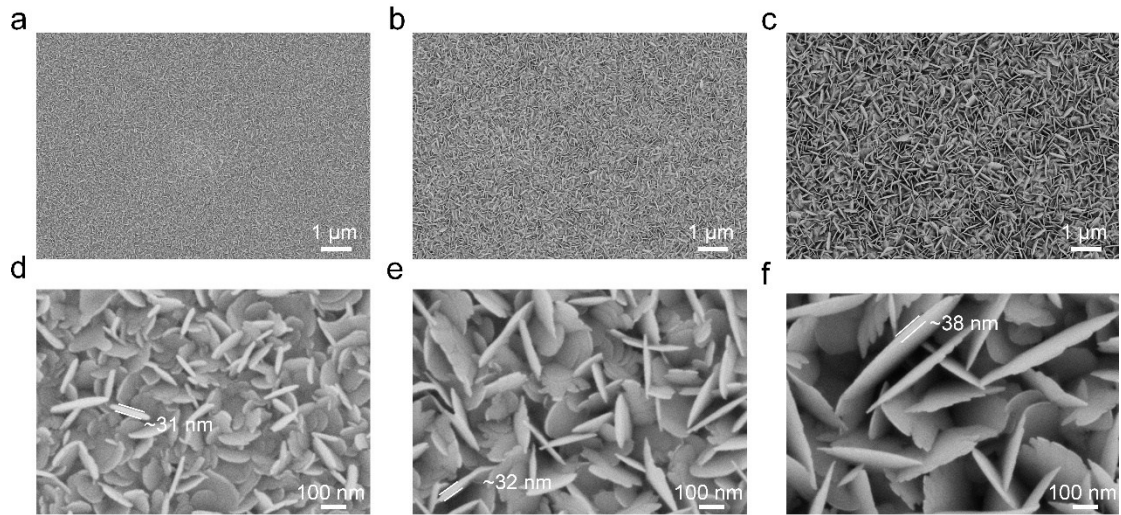


Figure S1. (a-c) Representative top-view SEM images of the resulting SnSe₂ NPAs, corresponds to a certain deposition time (5, 10 and 20 min, respectively). (d-f) Representative top-view SEM images of the resulting SnSe₂ NPAs, corresponds to a certain deposition time (5, 10 and 20 min, respectively). The thickness of the SnSe₂ NPAs is about 30~40 nm (> 10 layers, bulk).

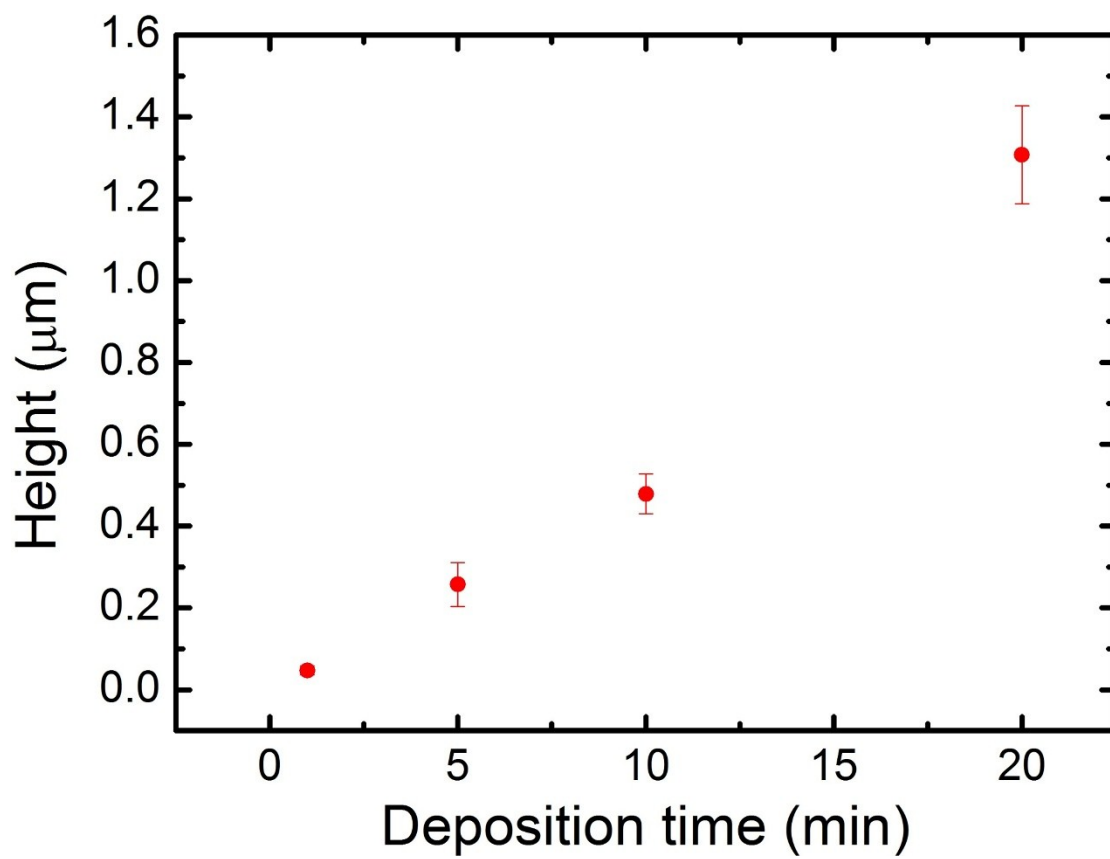


Figure S2. Height of SnSe₂ NPs as a function of deposition time.

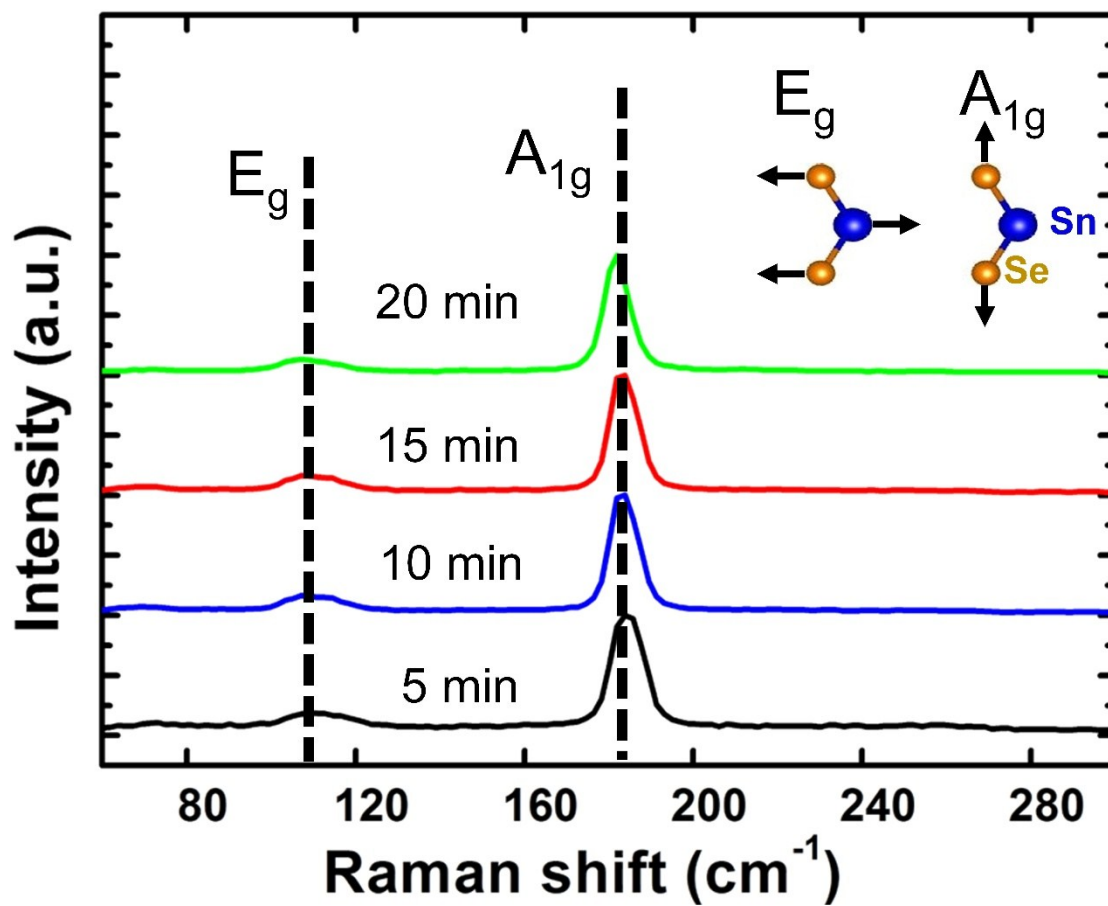


Figure S3. Raman spectra of SnSe₂ NPs with different deposition time.

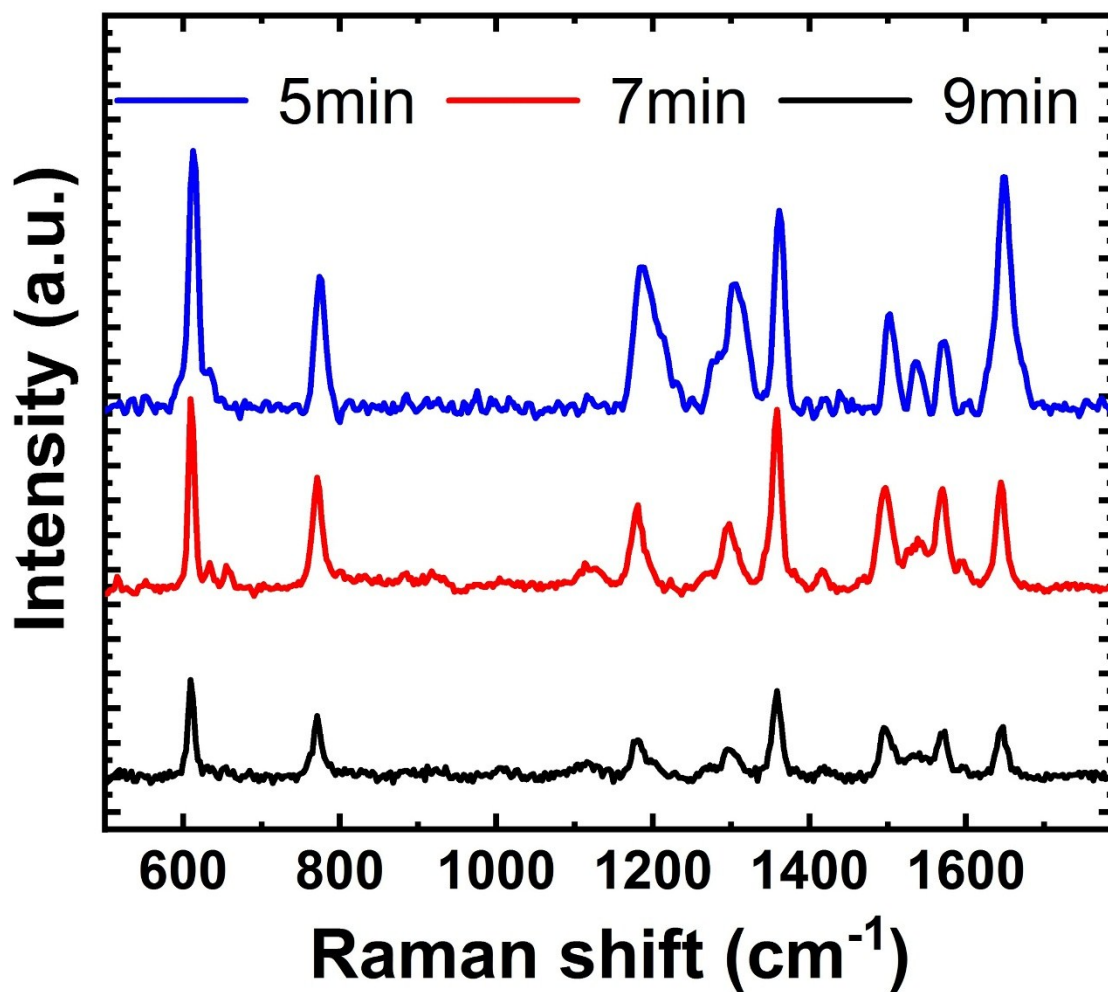


Figure S4. SERS spectra of R6G (10⁻⁶ M) adsorbed on SnSe₂ NPAs with different deposition time.

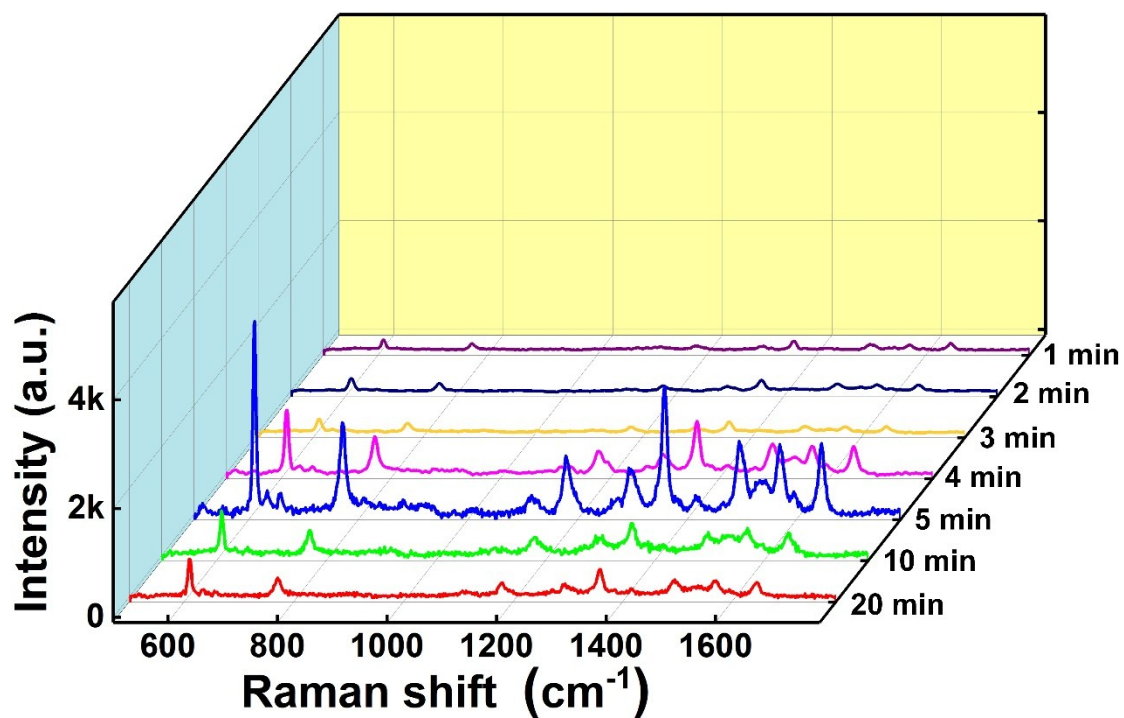


Figure S5. SERS spectra of R6G (10⁻⁶ M) adsorbed on SnSe₂ NPAs (another batch of samples) with different deposition time. The result indicates the highly reproductive feature, compared with the results shown in Figure 3a.

Charge-transfer process

The valence band (VB) and the conduction band (CB) of SnSe₂ were calculated by the following equations (S1) and (S2) [1]

$$E_c = -(\chi(A)^a * \chi(B)^b * \chi(C)^c)^{1/(a+b+c)} + 1/2E_g \quad (S1)$$

$$E_v = E_c + E_g \quad (S2)$$

where E_v and E_c are the valence band and conduction band edge, respectively. χ is the electronegativity of the semiconductor, which is geometric mean of the electronegativity of the constituent atoms. For SnSe₂, the value of bandgap E_g was calculated to be 1.2 eV by the following equation (S3)[2]

$$E_g = 1240 / \lambda_{abs} \quad (S3)$$

where λ_{abs} is 1028 nm, as required from the photoluminescence spectra shown Figure S6a. Thus, the E_v and E_c of SnSe₂ was -4.7 eV and -3.5 eV, respectively. Figure S6b depicts the charge-transfer process at the interface between R6G probe molecule and SnSe₂. The valence band (VB) (-4.7 eV) and conduction band (CB) (-3.5 eV) of SnSe₂ are located between the HOMO (-5.70 eV) and LUMO (-3.40 eV). The vibronic coupling between SnSe₂ (CB state |S> and VB state |S'>) and R6G molecule (excited state |K> and ground state |I>) leads to the charge transfer process. Considering of the excitation energy and R6G molecule, when a laser with wavelength of 532 nm was used, the following transfer processes can occur according to the energy level relationship [3] : (1) (T_{SS}), (2) (T_{IK}), (3) (T_{SK}), (4) (T_{IS}), (5) (T_{IS'}) and (6) (T_{SK}), which are shown in Figure S6b. Attributing to enormous enhancement of the polarizability of the R6G molecule, recombination of electrons and holes is increasing, which enhances the peak intensity of R6G molecules.

In additional, for the two dimensional SnSe₂ nanomaterials, the large specific surface area will also benefit to the easier combination of surface atoms (Sn, Se) with adsorbed R6G molecules, in turn leading to stronger photoinduced charge-transfer process [4].

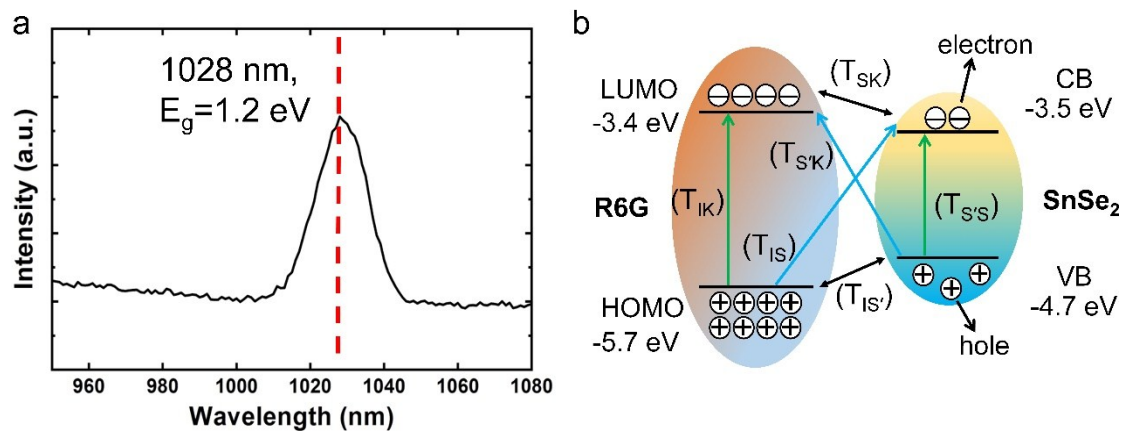


Figure S6. (a) Photoluminescence spectra of SnSe₂ NPAs. (b) Schematic illustration of the photoinduced charge-transfer process between SnSe₂ NPAs and R6G under the excitation of 532 nm.

Calculation of enhancement factor (EF)

The Raman enhancement factor (EF) of as-fabricated SnSe₂ NPAs SERS substrate was calculated by the following equation (S4) [5]:

$$EF = (I_{SERS} / N_{SERS}) / (I_{NR} / N_{NR}) \quad (S4)$$

In equation S4, I_{SERS} and I_{NR} are the Raman intensity of normal Raman signal collected from probe molecule adsorbed on SnSe₂ NPAs (1cm×1cm) and normal Raman signal collected from probe molecule (0.1 M) drop on rectangular glass (1cm×1cm), respectively. N_{SERS} and N_{NR} are the amounts of molecules involved in the SERS experiments and estimated for 100 μL of the aqueous solution of probe molecule (0.1 M) dropped on a glass which produced a deposition area about 1cm² (1cm×1cm), respectively. Here we use R6G probe molecules and 611 cm⁻¹ peak to calculate the EF .

The amounts of the detected molecule N_{SERS} can be calculated by:

$$N_{SERS} = CV_{solution}S_{spot}N_A / S_{sub} \quad (S5)$$

where C is the concentration of probe molecule, $V_{solution}$ is the volume of used probe molecules drop, S_{spot} and S_{sub} are the area of laser spot and the area of rectangular glass substrate (1cm²), respectively, N_A is Avogadro constant (6.22×10^{23}). The laser spot diameter d ($d = 1.22\lambda_{laser} / N_A$, λ_{laser} is the incident wavelength 532 nm, the numerical aperture of the objective lens $N_A = 0.5$) is about 1.3 μm. Thus, the laser spot area $S_{spot} = \pi(d/2)^2$ is about 1.33 μm².

The Raman 611 cm⁻¹ intensity of R6G/SnSe₂ NPAs (1×10^{-10} M) is 35.6 counts (Figure S7b) with 10 s acquisition time, and that of bulk R6G is 562.5 counts with 1 s acquisition time (Figure S7a). With normalization in case of acquisition time, the Raman intensity ratio is estimated to be $I_{SERS} / I_{NR} = (35.6/10)/(562.5) = 6.33 \times 10^{-3}$. Therefore, according to the equation (S4) and (S5), the EF of as-fabricated SnSe₂ NPAs SERS substrate was calculated to be $EF = (6.33 \times 10^{-3})(10^{-1} / 10^{-10}) = 6.33 \times 10^6$.

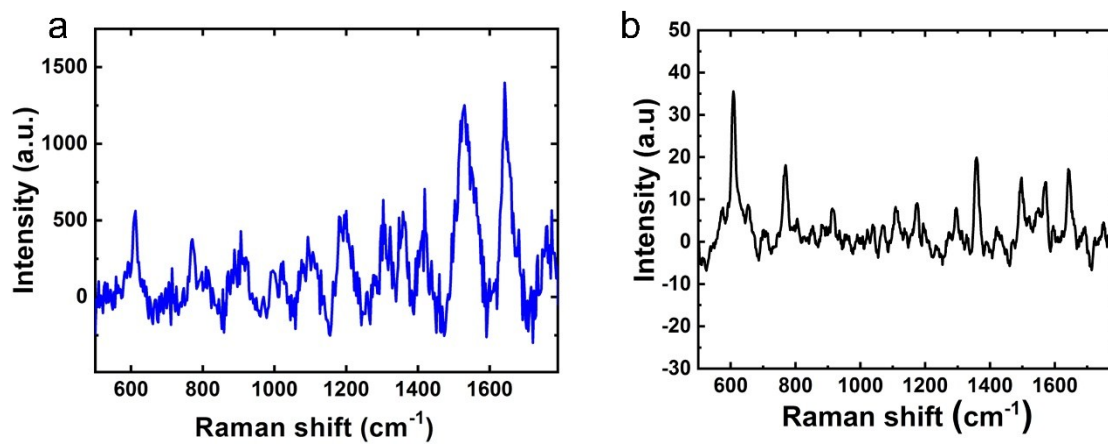


Figure S7. (a) SERS spectra of R6G (0.1 M) adsorbed on glass substrate (1 s acquisition time). (b) SERS spectra of R6G (1×10^{-10} M) adsorbed on glass substrate (10 s acquisition time).

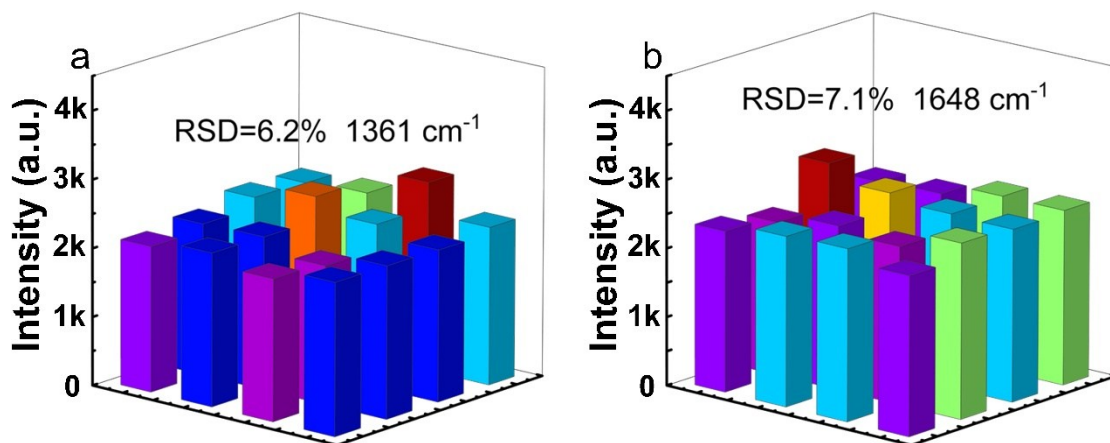


Figure S8. The Raman intensities of R6G at peak of 1361 cm⁻¹ and 1648 cm⁻¹, acquired from SERS spectra shown in Figure 4a.

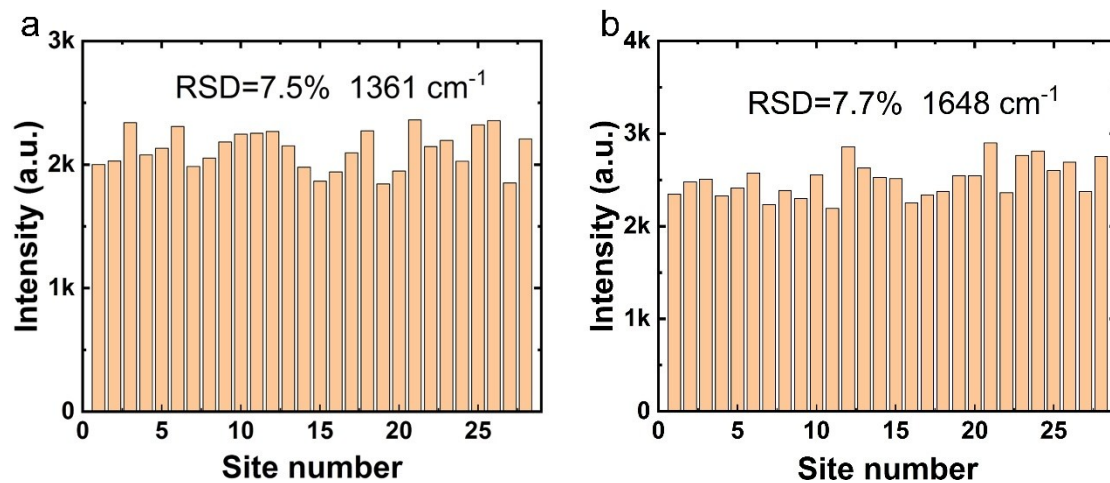


Figure S9. The Raman intensities of R6G at peak of 1361 cm⁻¹ and 1648 cm⁻¹, acquired from SERS spectra shown in Figure 4c.

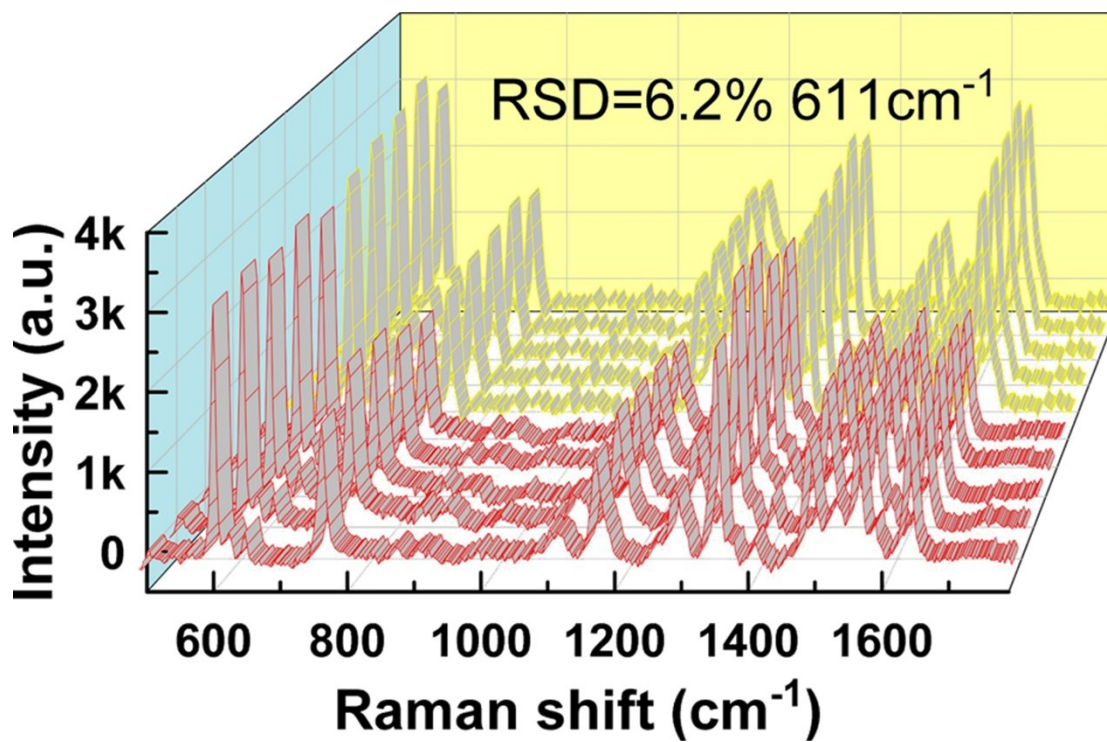


Figure S10. Sample to sample Raman intensity variations. 5 SERS spectra (yellow line) and 5 SERS spectra (red lines) are collected from different samples. A small sample-to-sample RSD of 6.2% was achieved.

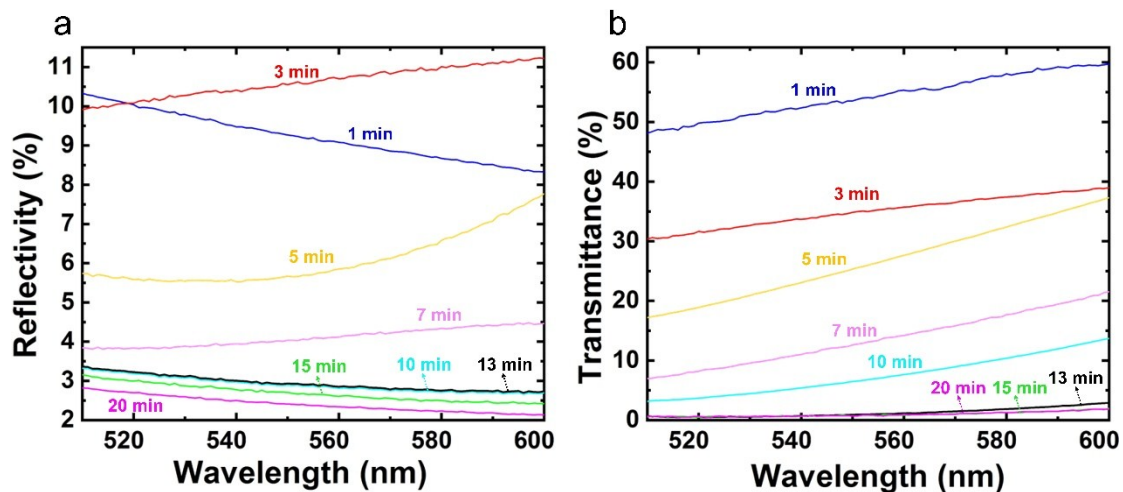


Figure S11. (a) Measured reflectivity of SnSe₂ NPAs with different deposition time. (b) Measured transmittance of SnSe₂ NPAs with different deposition time.

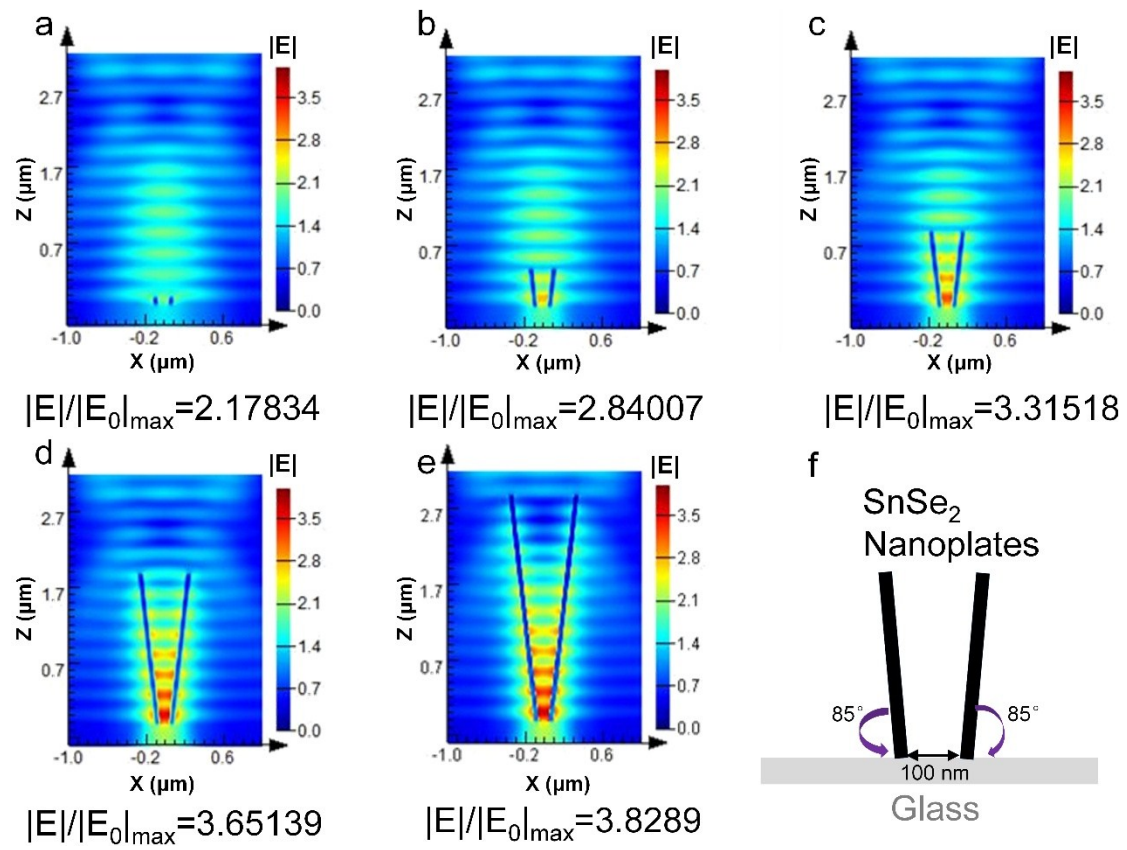


Figure S12. (a-e) Calculated electric field ($|E|/|E_0|$) distributions of SnSe₂ NPAs with different heights. (f) Schematic diagram of the modeled structure (symmetric structure, with 85° angle for both nanoplates).

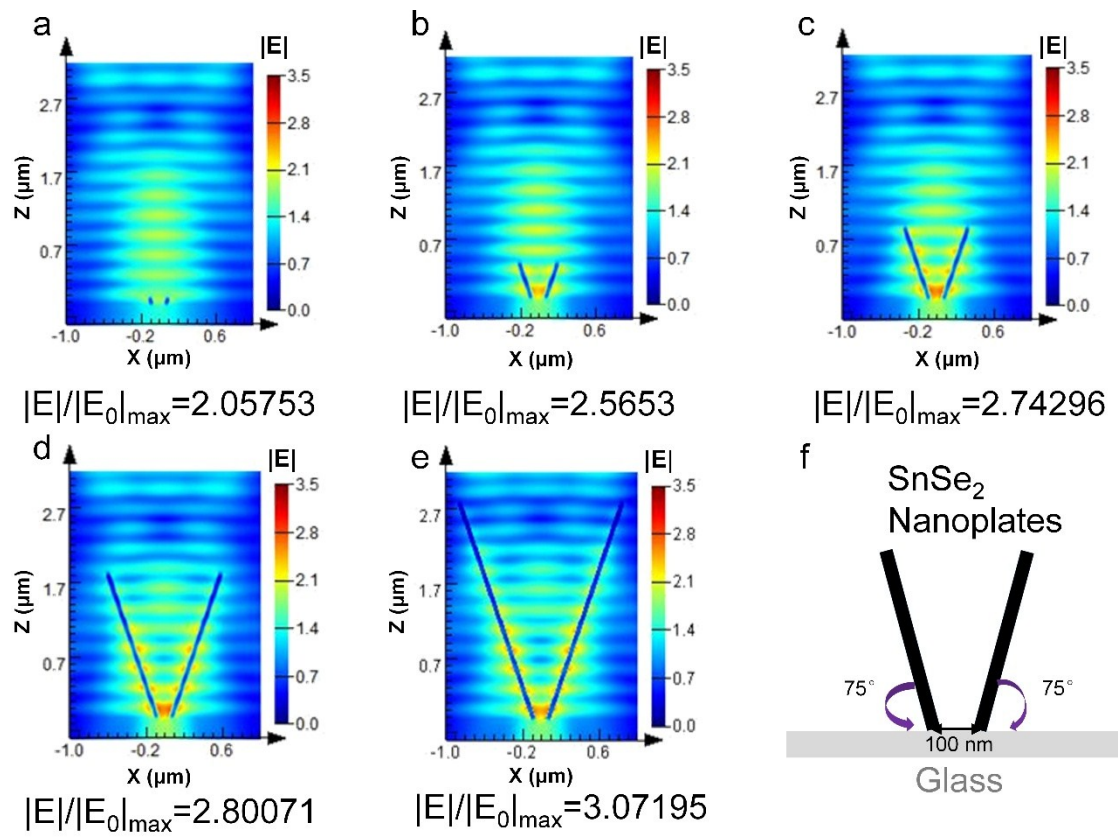


Figure S13. (a-e) Calculated electric field ($|E|/|E_0|$) distributions of SnSe₂ NPAs with different heights. (f) Schematic diagram of the modeled structure (symmetric structure, with 75° angle for both nanoplates).

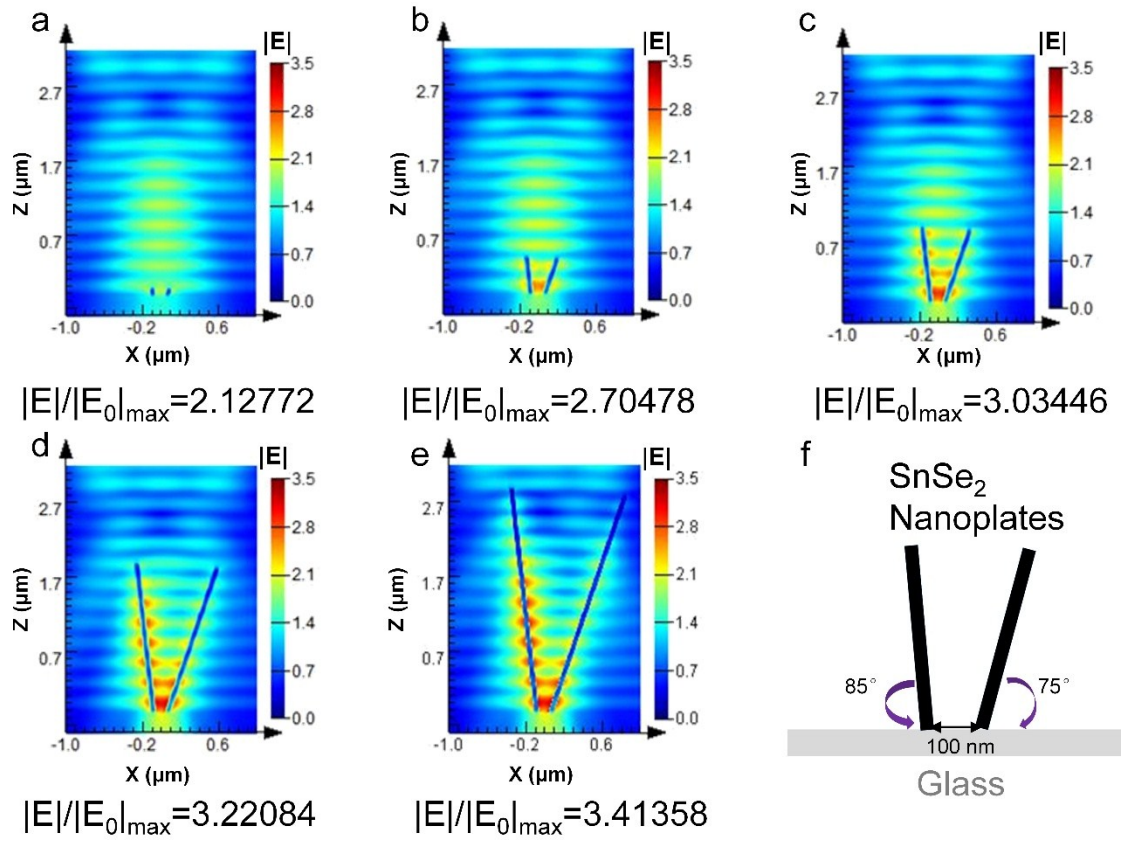


Figure S14. (a-e) Calculated electric field ($|E|/|E_0|$) distributions of SnSe₂ NPAs with different heights. (f) Schematic diagram of the modeled structure (asymmetric structure, with 85° angle for the left nanoplate and 75° angle for the right nanoplate).

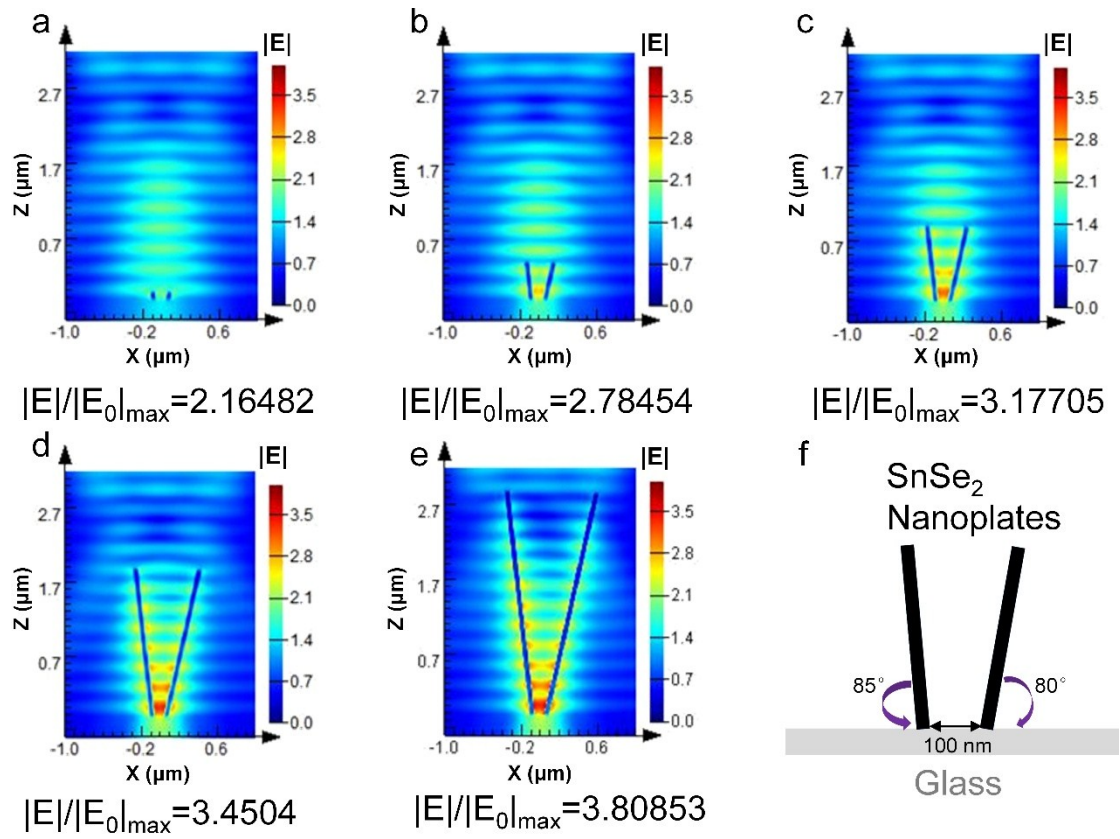


Figure S15. (a-e) Calculated electric field ($|E|/|E_0|$) distributions of SnSe₂ NPAs with different heights. (f) Schematic diagram of the modeled structure (asymmetric structure, with 85° angle for the left nanoplate and 80° angle for the right nanoplate).

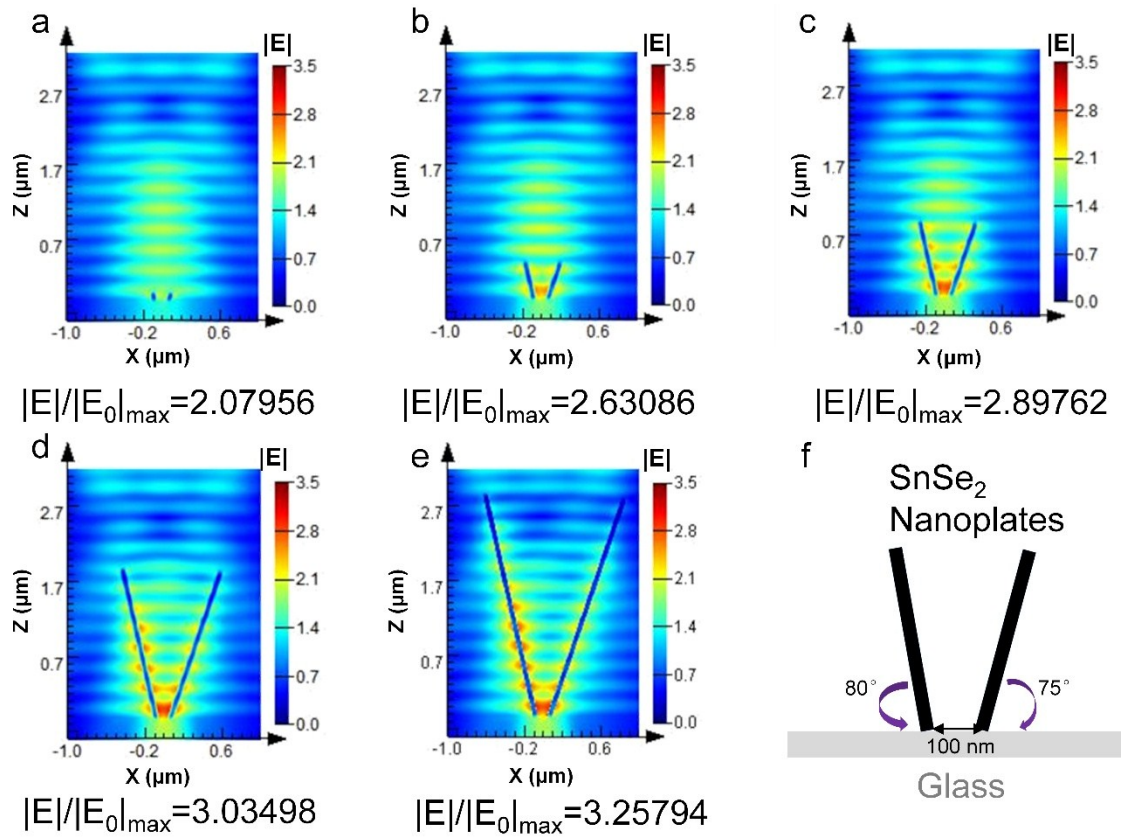


Figure S16. (a-e) Calculated electric field ($|E|/|E_0|$) distributions of SnSe₂ NPAs with different heights. (f) Schematic diagram of the modeled structure (asymmetric structure, with 80° angle for the left nanoplate and 75° angle for the right nanoplate).

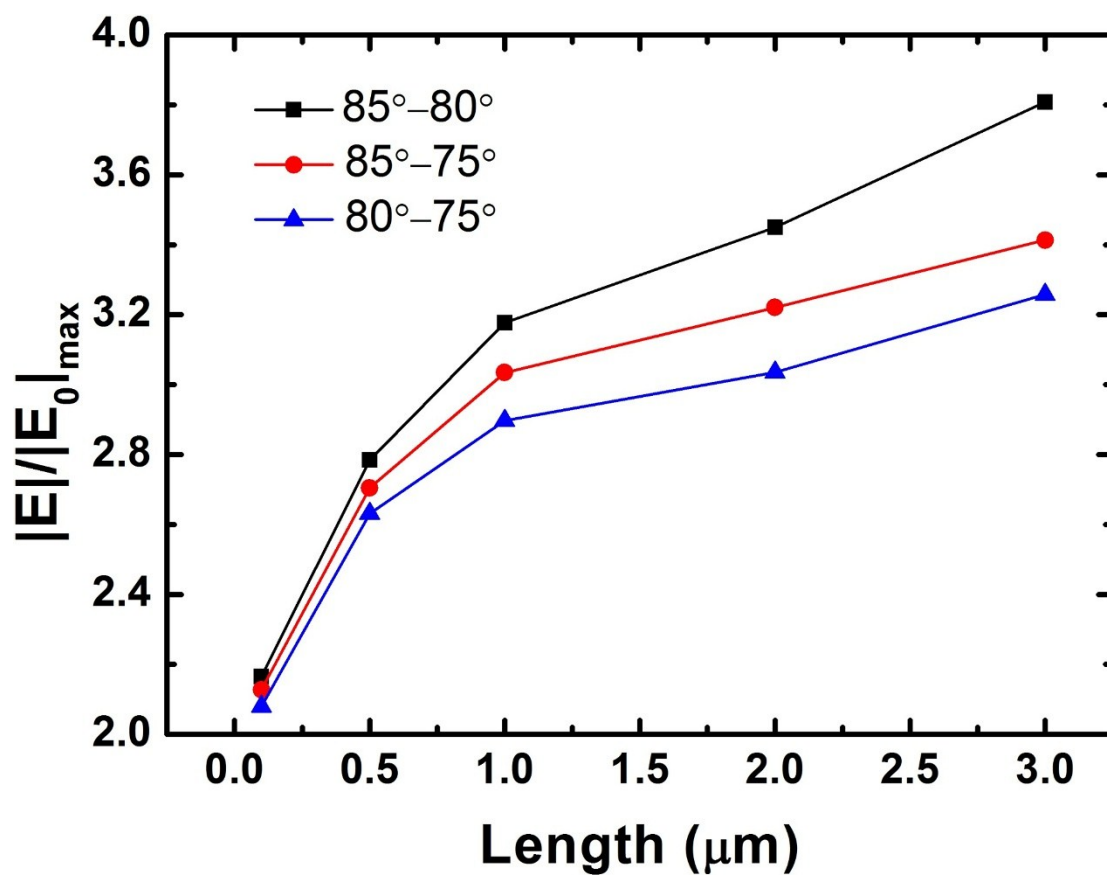


Figure S17. The calculated maximum intensity $|E|/|E_0|$ of SnSe₂ NPAs with different heights and angles (asymmetric structure)

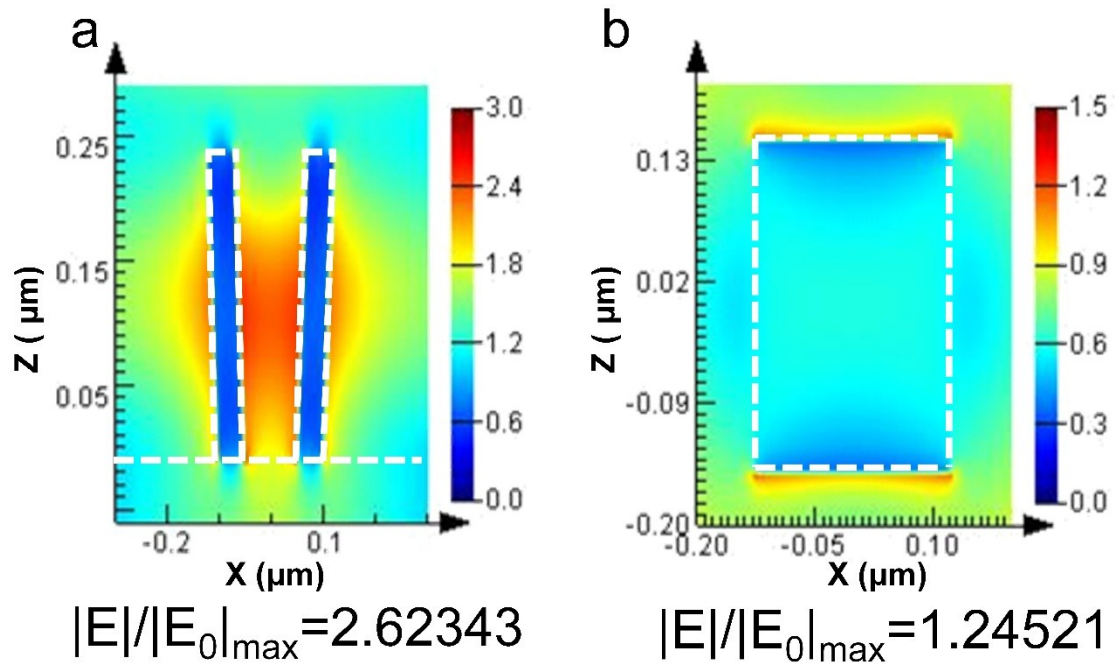


Figure S18. (a-b) Calculated electric field ($|E|/|E_0|$) distributions for vertical-structured SnSe₂ NPAs ($t=5$ min, height= $0.25 \mu\text{m}$, width= $0.3 \mu\text{m}$, thickness= 35 nm) and tiled SnSe₂ nanoplate ($t=1$ min, thickness= 35 nm , length= $0.25 \mu\text{m}$, width= $0.3 \mu\text{m}$), respectively.

Enhanced SERS performance *via* inserting high reflective layer-Distributed Bragg Reflectors (DBR)

We use Ta₂O₅/SiO₂ DBR as the high reflective layer. The n, k value of Ta₂O₅ are 2.1469 and 0.00081579 for 532 nm (laser wavelength), respectively. The n value of SiO₂ is 1.46 for 532 nm.

Then the thickness of the Ta₂O₅ and SiO₂ layer can be obtained by the following equation[6]:

$$L = \frac{\lambda_{Bragg}}{4n} \quad (S6)$$

where L is the thickness of the layer, λ_{Bragg} is the Bragg central wavelength (532 nm). The

calculated thickness of of the Ta₂O₅ and SiO₂ layer are 62 nm and 91 nm, respectively. The

reflectivity R of the DBR can then be obtained by the following equation[6]:

$$R = \left(\frac{1 - (n_{Ta_2O_5} / n_{SiO_2})^{2m}}{1 + (n_{Ta_2O_5} / n_{SiO_2})^{2m}} \right)^2 \quad (S7)$$

where m is the pairs of Ta₂O₅/SiO₂. The result is shown in Figure S18a, the reflectivity of the

Ta₂O₅/SiO₂ DBR (10 pairs) is 0.998 (close to 100%).

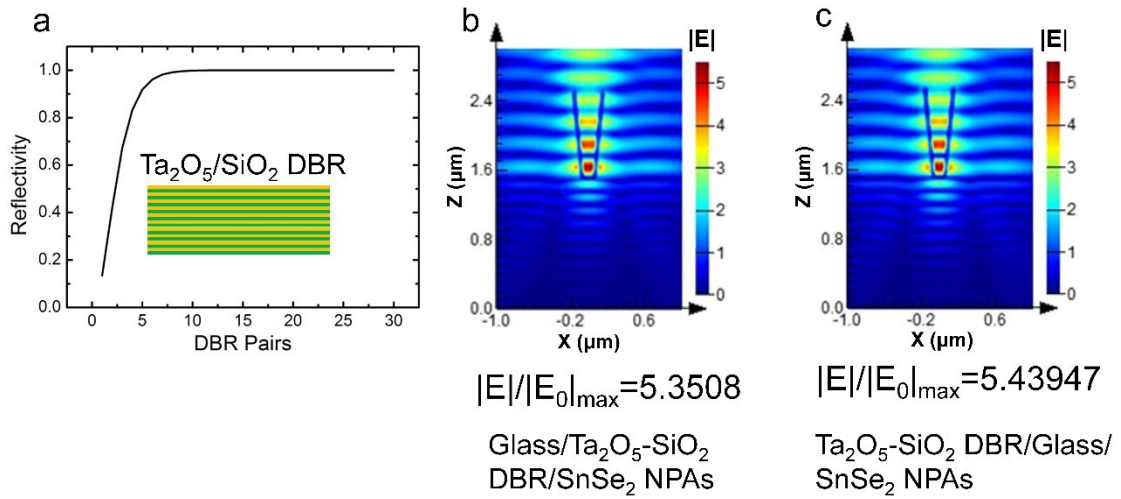


Figure S19. (a) Calculated reflectivity of Ta₂O₅/SiO₂ DBR as a function of DBR pairs. (b-c) The electric field ($|E|/|E_0|$) distributions of Glass/Ta₂O₅-SiO₂ DBR/SnSe₂ NPAs scheme and Ta₂O₅-SiO₂ DBR/Glass/SnSe₂ NPAs scheme (10 pairs DBR symmetric structure, length of SnSe₂ NPAs=1 μm , angle=85°, thickness of Glass=50 nm). Compared to the results shown in Figure S12c, the maximum intensity $|E|/|E_0|$ increases from 3.32 to 5.35 (5.44), which can promote the coupling between the incident light and SnSe₂ NPAs, in turn enhance the SERS performance.

Table S1: Summary of semiconductors and noble metals SERS activity

Material	Analyte	EF	LOD(M)	Laser	Ref.
SnSe ₂ NPAs	R6G	6.33x10 ⁶	1x10 ⁻¹²	532nm	This work
SnSe ₂ Nanoflakes (isolated flakes)	R6G	-	1x10 ⁻⁸	532nm	[3]
Cu ₂ O Concave Sphere (single particle)	CV and R6G	2.8x10 ⁵ (CV), 2.5x10 ⁵ (R6G),	2x10 ⁻⁸	647nm	[7]
TiO ₂ microarrays	Methylene Blue	2x10 ⁴	6x10 ⁻⁶	532nm	[8]
SiO ₂ @TiO ₂ core-shell spheres	Methylene Blue	-	1x10 ⁻⁵	633nm	[9]
Au@SiO ₂	R6G	1x10 ⁶	1x10 ⁻⁹	785nm	[10]
Cu ₂ O Nanospheres	4-MBA	1x10 ⁵	1x10 ⁻³	488nm	[11]
Bulk MoS ₂	R6G	1.9x10 ⁶	1x10 ⁻⁸	532nm	[12]
3D-stacked Ag NWs	Methylene Blue	-	1.5x10 ⁻⁹	632.8nm	[13]
AuNPs@MoS ₂	R6G	8.2x10 ⁵	1x10 ⁻⁶	633nm	[14]
3D TiO ₂ nanofibers	CV	1.3x10 ⁶	-	532nm	[15]

References:

1. P. Y. Ma, H. J. Yu, Y. Yu, W. M. Wang, H. Wang, J. Y. Zhang, Z. Y. Fu, *Phys. Chem. Chem. Phys.* **2016**, *18*, 3638–3643
2. J. Tauc, R. Grigorovici, A. Vancu, *Phys. Status Solidi* **1966**, *15*, 627
3. Y. Zhang, Y. Shi, M. Wu, K. Zhang, B. Man, M. Liu, *Nanomaterials* **2018**, *8*, 515.
4. C. Muehlethaler, C. R. Consideine, V. Menon, W. C. Lin, Y. H. Lee, J. R. Lombardi, *ACS Photonics* **2016**, *3*, 1164.
5. J. Huang, D. Ma, F. Chen, M. Bai, K. Xu and Y. Zhao, *Anal. Chem.* **2015**, *87*, 10527.
6. C. Wilmsen, H. Temkin, L. A. Coldren, “Vertical cavity surface emitting lasers: design, fabrication, characterization and applications [M],” *Cambridge Studies in Modern Optics*. **2001**, 24, pp. 73.
7. X. X. Li, Y. Shang, J. Lin, A. R. Li, X. T. Wang, B. Li, L. Guo, *Adv. Funct. Mater.* **2018**, *28*, 1801868.
8. D. Qi, L. Lu, L. Wang, J. Zhang, *J. Am. Chem. Soc.* **2014**, *136*, 9886-9889.
9. I. Alessandri, *J. Am. Chem. Soc.* **2013**, *135*, 5541-5544.
10. Y. Shin, J. Song, D. Kim, T. Kang, *Adv. Mater.* **2015**, *27*, 4344-4350
11. L. Jiang, T. You, P. Yin, Y. Shang, D. Zhang, L. Guo, S. Yang, *Nanoscale* **2013**, *5*, 2784-2789.
12. D. Yan, W. Qiu, X. Chen, L. Liu, Y. Lai, Z. Meng, J. Song, Y. Liu, X. Liu, and D. Zhan, *J. Phys. Chem. C* **2018**, *122*, 14467–14473
13. S. G. Park, C. Mun, M. Lee, T. Y. Jeon, H. S. Shim, Y. J. Lee, J. D. Kwon, C. S. Kim, D. H. Kim, *Adv. Mater.* **2015**, *27*, 4290-4295.
14. S. Su, C. Zhang, L. Yuwen, J. Chao, X. Zuo, X. Liu, C. Song, C. Fan, L. Wang, *ACS Appl. Mater. Interfaces* **2014**, *6*, 18735.
15. D. Maznichenko, K. Venkatakrishnan, B. Tan, *J. Phys. Chem. C* **2012**, *117*, 578.