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# **Supplementary Material:**

# Negative and near-zero Poisson's ratio of interlayer space in 2D Graphene/MoS<sub>2</sub> and Graphene/*h*-BN heterostructures

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# 1. Interlayer binding energies and equilibrium distance results

**Table 1.** Interlayer binding energies  $(E_{bind})$  and equilibrium distances (d) for bilayer graphene, bilayer *h*-BN, bilayer MoS<sub>2</sub>, G/MoS<sub>2</sub> and G/*h*-BN heterostructures.





	Stacking mode C		Stacking mode D		Stacking mode E	
G/h-BN heterostructure	Top view d		Top view d		• Carbon • Boron • Nitrogen	
	$E_{bind}$ (meV/Å <sup>2</sup> )	d (Å)	$E_{bind}$ (meV/Å <sup>2</sup> )	d (Å)	$E_{bind}$ (meV/Å <sup>2</sup> )	<i>d</i> (Å)
D2	17.43	3.366	18.41	3.349	23.26	3.129
OptB88	21.32	3.443	21.93	3.404	25.11	3.327
MBD	16.18	3.577	16.75	3.445	19.61	3.345

#### 2. Out-of-plane Poisson's ratio results



**Figure S1.** The strain generated in the z direction  $({}^{\varepsilon_z})$  versus the strain applied in the x direction ( ${}^{\varepsilon_x}$ ) data of uniaxial deformation of the (**a**, **b**, **c**) bilayer graphene, (**d**, **e**, **f**) bilayer *h*-BN, (**g**, **h**, **i**) bilayer MoS<sub>2</sub>, (**j**, **k**, **l**) G/MoS<sub>2</sub> and (**m**, **n**, **o**) G/*h*-BN heterostructures, calculated by density functional theory with different van der Waals corrections.

**Table S2.** Computational results of out-of-plane Poisson's ratio  $(v_{13})$  for the bilayer graphene,

bilayer *h*-BN, bilayer MoS<sub>2</sub>, G/MoS<sub>2</sub> and G/*h*-BN heterostructures by density functional theory

		Out-of-plane Poisson's ratio $(v_{13})$				
Bilayer		D2	optB88	MBD		
Graphene		-0.097	-0.061	-0.089		
<i>h</i> -BN		-0.010	-0.00023	-0.0025		
	$d_{Top}$	0.247	0.256	0.250		
MoS <sub>2</sub>	$d_{Mo}$	0.240	0.258	0.247		
	d	0.228	0.264	0.243		
heterostructure						
G/h-BN		-0.033	0.0005	-0.002		
G/MoS <sub>2</sub>	$d_{Top}$	0.080	0.099	0.089		
	$d_S$	0.295	0.297	0.296		
	d	-0.120	-0.090	-0.107		

with different van der Waals corrections.

d represents interfacial layer equilibrium distance.  $d_S$  represents the distance between S and S atoms.  $d_{Mo}$  represents the distance between Mo atoms in upper MoS<sub>2</sub> monolayer and Mo atoms in lower MoS<sub>2</sub> monolayer and  $d_{Top}$  represents the distance between top S atoms in upper MoS<sub>2</sub> monolayer and S (C) atoms in lower MoS<sub>2</sub> (Graphene) monolayer (Fig. 1).

#### 3. Out-of-plane Stiffness results



**Figure S2.** Total energy of the (**a**, **b**, **c**) bilayer graphene, (**d**, **e**, **f**) bilayer *h*-BN, (**g**, **h**, **i**) bilayer MoS<sub>2</sub>, (**j**, **k**, **l**) G/MoS<sub>2</sub> and (**m**, **n**, **o**) G/*h*-BN heterostructures with the strain applied in the z direction ( $\mathcal{E}_z$ ) calculated by density functional theory with different van der Waals corrections.

**Table S3.** Computational results of out-of-plane stiffness for the bilayer graphene, bilayer h-BN, bilayer MoS<sub>2</sub>, G/MoS<sub>2</sub> and G/h-BN heterostructures by density functional theory with different van der Waals corrections.

	Out-of-plane Stiffness (Gpa)						
	D2		optB88		MBD		
Бпауег	C <sub>33</sub>	C <sub>333</sub>	C <sub>33</sub>	C <sub>333</sub>	C <sub>33</sub>	C <sub>333</sub>	
Graphene	106	-939	71	-586	54	-463	
<i>h</i> -BN	109 -463		67	-463	56	-463	
MoS <sub>2</sub>	28	-202	34	-202	30	-210	
heterostructure							
G/h-BN	121	-1040	70	-571	61	-508	
G/MoS <sub>2</sub>	56	-500	57	-475	56	-489	

#### 4. In-plane Poisson's ratio results



**Figure S3.** The strain generated in the y direction  $({}^{\mathcal{E}}y)$  versus the strain applied in the x direction ( ${}^{\mathcal{E}}x$ ) data of uniaxial deformation of the (**a**, **b**, **c**) bilayer graphene, (**d**, **e**, **f**) bilayer *h*-BN, (**g**, **h**, **i**) bilayer MoS<sub>2</sub>, (**j**, **k**, **l**) G/MoS<sub>2</sub> and (**m**, **n**, **o**) G/*h*-BN heterostructures, calculated by density functional theory with different van der Waals corrections.

**Table S4.** Computational results of in-plane Poisson's ratio  $(v_{12})$  for the bilayer graphene, bilayer *h*-BN, bilayer MoS<sub>2</sub>, G/MoS<sub>2</sub> and G/*h*-BN heterostructures by density functional theory with

	In-plane Poisson's ratio $(v_{12})$					
Bilayer	D2	optB88	MBD			
Graphene	0.163	0.159	0.168			
<i>h</i> -BN	0.169	0.199	0.197			
MoS <sub>2</sub> 0.249		0.250	0.250			
heterostructure						
G/h-BN	0.204	0.176	0.186			
G/MoS <sub>2</sub>	0.171	0.177	0.174			

different van der Waals corrections.

#### 5. In-plane Stiffness results



**Figure S4.** Total energy of the (**a**, **b**, **c**) bilayer graphene, (**d**, **e**, **f**) bilayer *h*-BN, (**g**, **h**, **i**) bilayer MoS<sub>2</sub>, (**j**, **k**, **l**) G/MoS<sub>2</sub> and (**m**, **n**, **o**) G/*h*-BN heterostructures with the strain applied in the x direction ( ${}^{\varepsilon}x$ ) calculated by density functional theory with different van der Waals corrections.

Table S5. Computational results of in-plane stiffness for the bilayer graphene, bilayer h-BN,

bilayer MoS<sub>2</sub>, G/MoS<sub>2</sub> and G/h-BN heterostructures by density functional theory with different

	In-plane Stiffness (Nm <sup>-1</sup> )						
Dilawa	D2		optB88		MBD		
Bilayer	Y <sub>11</sub>	Y <sub>111</sub>	Y <sub>11</sub>	Y <sub>111</sub>	Y <sub>11</sub>	Y <sub>111</sub>	
Graphene	673	-3034	679	-4124	683	-3471	
<i>h</i> -BN	551	-2671	554	-2900	538	-1101	
MoS <sub>2</sub>	244	-1548	247	-2157	256	-1808	
heterostructure							
G/h-BN	596	-1385	625	-4352	615	-2974	
G/MoS <sub>2</sub>	465	-2969	468	-3053	465	-3004	

van der Waals corrections.

By analyzing total energy (see Figure 1), the stiffness (Young's Modulus) SOEC and TOEC of the G/MoS<sub>2</sub> and G/*h*-BN heterostructures in the in-plane direction ( $Y_{11}$  and  $Y_{111}$ ) were derived (see Methods section). The calculated  $Y_{11}$  of the G/MoS<sub>2</sub> and G/*h*-BN heterostructures are almost equal to the sum of the stiffness (Young's Modulus) of the two layers of the materials. The  $Y_{11}$  of the twodimensional Graphene, MoS<sub>2</sub>, and *h*-BN are 340 Nm<sup>-1</sup>, 109 Nm<sup>-1</sup> and 238 Nm<sup>-1</sup>, respectively<sup>1-3</sup>. This result reveals the reason for the enhanced stability of the carbon-based heterostructures observed in the experiment<sup>4</sup>.

#### 6. The relationship between Poisson's ratio and stiffness

The stiffness tensor of a hexagonal crystal system can be written as follows:

$$C_{\alpha\beta} = \begin{pmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{C_{11} - C_{12}}{2} \\ \end{array} \right)_{-}$$

The compliance tensor can be obtained by taking inverse of the stiffness tensor:

$$S_{\alpha\beta} = C_{\alpha\beta}^{-1} = \begin{pmatrix} S_{11} & S_{12} & S_{13} & 0 & 0 & 0 \\ S_{12} & S_{11} & S_{13} & 0 & 0 & 0 \\ S_{13} & S_{13} & S_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & S_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & S_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & 2(S_{11} - S_{12}) \end{pmatrix}_{.}$$

The in-plane Poisson's ratio  $\mathbf{v}_{12}$  and the out-of-plane Poisson's ratio  $\mathbf{v}_{13}$ :

$$\begin{cases} v_{12} = -\frac{S_{12}}{S_{11}} \\ v_{13} = -\frac{S_{13}}{S_{11}} \end{cases}$$

After simplification,

$$\begin{cases} v_{12} = \frac{C_{13}^{2} - C_{12}C_{33}}{C_{13}^{2} - C_{11}C_{33}} \\ v_{13} = \frac{C_{13}(C_{12} - C_{11})}{C_{13}^{2} - C_{11}C_{33}} \end{cases}$$

Therefore, we can verify the accuracy of the results by the relationship between Poisson's ratio and stiffness tensor.

#### 7. the relationship between $\theta$ and $v_{13}$

When i fixed in the x direction and j varying in the y-z plane by an angle of  $\theta$ , the transformation

from the stress  $\sigma_{\beta}$  (unprimed) to the stress  $\sigma_{\beta'}$  in arbitrary system (primed) is described by:

$ \begin{matrix} \sigma_{1'} \\ \sigma_{2'} \\ \sigma_{3'} \\ \sigma_{4'} \\ \sigma_{5'} \\ \sigma_{6'} \end{matrix} $	$ \begin{smallmatrix} 1 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\ 0 \\$	$0 \\ \cos^2 \theta \\ \sin^2 \theta \\ -\sin \theta \cos \theta \\ 0 \\ 0 \\ 0$	$0 \\ \sin^2 \theta \\ \cos^2 \theta \\ \sin \theta \cos \theta \\ 0 \\ 0$	$0$ $2\sin\theta\cos\theta$ $-2\sin\theta\cos\theta$ $\cos^{2}\theta - \sin^{2}\theta$ $0$ $0$	$\begin{array}{c} 0\\ 0\\ 0\\ cos\theta\\ sin\theta \end{array}$	$\begin{array}{c} 0\\ 0\\ 0\\ -\sin\theta\\ \cos\theta \end{array}$	$ \begin{matrix} \sigma_1 \\ \sigma_2 \\ \sigma_3 \\ \sigma_4 \\ \sigma_5 \\ \sigma_6 \end{matrix} $
$\breve{\sigma}_{\beta'}$	=		7	σ <sub>β'</sub> β			$\overleftarrow{\sigma}_{\beta}$

Note that we use contracted notations  $(11 \rightarrow 1, 22 \rightarrow 2, 33 \rightarrow 3, 13 \rightarrow 4, 23 \rightarrow 5, 12 \rightarrow 3, C_{ijkl} \rightarrow C_{\alpha\beta}$ 

 $C_{ijklmn} \rightarrow C_{\alpha\beta\gamma}$ ) for the tensor indices. The transformation from the strain  $\varepsilon_{\alpha}$  (unprimed) to the strain  $\varepsilon_{\alpha'}$  in arbitrary system (primed) is described by:



The compliance coefficients  $S_{\alpha'\beta'}$  are defined as the proportionality constants between stress  $\sigma_{\beta'}$ and strain by the Hooke's law:

$$\varepsilon_{\alpha'} = S_{\alpha'\beta'}\sigma_{\beta'}$$

It can be expressed by the coordinate transformation equation:

$$T_{\varepsilon_{\alpha'\alpha}}\varepsilon_{\alpha} = S_{\alpha'\beta'}T_{\sigma_{\beta'\beta}}\sigma_{\beta}$$

We obtain

$$\varepsilon_{\alpha} = \frac{T_{\varepsilon_{\alpha'\alpha}}^{-1} S_{\alpha'\beta'} T_{\sigma_{\beta'\beta}}}{S_{\alpha\beta}} \sigma_{\beta},$$

and

 $S_{\alpha'\beta'} = T_{\varepsilon_{\alpha'\alpha}} S_{\alpha\beta} T_{\sigma_{\beta'\beta}}^{-1}$ 

The out-of-plane Poisson's ratio  $v_{13}$ :

$$v_{13} = -\frac{S_{13}}{S_{11}}$$

After simplification,

$$v_{13}(\theta) = -\frac{(C_{11} - C_{12})C_{13}\cos^2\theta - (C_{13}^2 - C_{12}C_{13})\sin^2\theta}{C_{13}^2 - C_{11}C_{13}}.$$

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