## **Supplementary Information**

# Promoting Li<sub>2</sub>O<sub>2</sub> oxidation by La<sub>1.7</sub>Ca<sub>0.3</sub>Ni<sub>0.75</sub>Cu<sub>0.25</sub>O<sub>4</sub> layered perovskite in lithium-oxygen batteries

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# **1. Materials preparation**

For preparation of an La<sub>1.7</sub>Ca<sub>0.3</sub>Ni<sub>0.75</sub>Cu<sub>0.25</sub>O<sub>4</sub> (LCNC) material, desired amounts of La(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O (99.99%, Aldrich), Ca(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O (> 99.0 %, Aldrich), Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (99.999 %, Aldrich) and Cu(NO<sub>3</sub>)<sub>2</sub>·xH<sub>2</sub>O (99.999 %, Aldrich) were dissolved in a mixture of deionized water and ethylene glycol (EG, HOCH<sub>2</sub>CH<sub>2</sub>OH, 99.8%, Aldrich). Citric acid (HOC(COOH)(CH<sub>2</sub>COOH)<sub>2</sub>,  $\geq$  99.5%, Aldrich) was added to the solution under stirring conditions and then heated to 70°C. The molar ratios of citric acid to metal were maintained at 3:1. The resulting solution was heated to 150°C to form a viscous gel. The gel was dried at 250°C and then calcined at 900°C for 5 h to remove any remaining organic materials. The sample was finally pulverized by grinding into fine powders.

## 2. Materials characterization

## (a) X-ray diffraction (XRD) analysis

XRD patterns were recorded with an automated Rigaku diffractometer (2500 D/MAX, Rigaku) using Cu K<sub> $\alpha$ </sub> radiation. The measurements were conducted over the scanning angle range of 20 – 80° using a step width of 0.025° and a holding time of 5 s.

## (b) Neutron powder diffraction (NPD) analysis

NPD data were collected at room temperature for all the compounds using the highresolution powder diffractometer (HRPD) at Hanaro Center of the Korea Atomic Energy Research Institute. NPD data were collected over scattering angles ranging from 0° to 160° using 1.8348 Å neutron. 3 g of each sample was contained in a vanadium can. The collection time was 4 h per pattern. Diffraction data were analyzed by the Rietveld method, using the General Structure Analysis System (GSAS), with the use of its internal tables for scattering lengths. The line shape of the diffraction peaks was generated by a pseudo-Voigt function and the background refined to a 1st degree polynomial. In the final run the following parameters were refined: background coefficients, zero point, half width, pseudo-Voigt and asymmetry parameters for the peak shape, scale factor and unit-cell parameters.

## (c) Surface area and FTIR measurements

Brunauer-Emmett-Teller (BET) surface area was determined from  $N_2$  sorption isotherms by using a BEL-SORP mini system. FTIR measurements were carried out on a Nicolet 6700 spectrometer (ThermoFisher Scientific) in transmission mode with a KBr pellet.

## 3. Electrochemical experiments

# (a) Charging experiments with Li<sub>2</sub>O<sub>2</sub>-packed electrodes

Electrochemical cells were constructed based on a Swagelok design and composed of a Li metal anode, an electrolyte (1 M LiTFSI in TEGDME) impregnated into a glass fiber separator and a porous cathode. Cathodes artificially packed with  $Li_2O_2$  were prepared by casting a mixture of Ketjen black (KB), catalyst, ground  $Li_2O_2$  (Alfa Aesar) and PVdF-co-HFP (Kynar 2801) in N-Methyl pyrrolidinone (NMP) onto Ni mesh current collector, followed by vacuum drying at 70°C for 24 h. The composite electrode consisted of 50 wt.% KB, 15 wt.% catalyst, 15 wt.% Li<sub>2</sub>O<sub>2</sub> and 20 wt.% binder. Galvanostatic charging tests were performed at a current density of 67 mA g<sup>-1</sup> (based on the mass of  $Li_2O_2$ ) using a Maccor Series 4000, and potentiostatic current transient measurements were conducted by jumping the potential from an open circuit potential to 4.1 V vs. Li/Li<sup>+</sup>.

## (b) Evaluation of $Li-O_2$ battery performances

 $Li-O_2$  batteries and cathodes were prepared under the same conditions as described above, except no  $Li_2O_2$  was initially included in the cathode. Galvanostatic discharge-charge profiles were measured at a current density of 40 mA g<sup>-1</sup> (based on the mass of carbon). For cycling experiments, batteries were discharged and charged at 80 mA g<sup>-1</sup>.

# 4. NPD refinement results

**Table S1** Refined structural parameters for  $La_{1.7}Ca_{0.3}Ni_{0.75}Cu_{0.25}O_4$  obtained from the Rietveld refinement using NPD data at room temperature. Here, *g* and *U*<sub>iso</sub> represent the occupation and isotropic thermal parameters, respectively. The numbers in parentheses are the estimated standard deviations of the last significant figure.

atom	cito				~	$100 + II / \lambda^2$
atom	site	X	У	Z.	8	$100 * U_{iso}/A$
La	4 <i>e</i>	0	0	0.3621(2)	0.85 <sup>a)</sup>	$0.49(2)^{b)}$
Ca	4 <i>e</i>	0	0	0.3621(2)	0.15	0.49(2)
Ni	2a	0	0	0	0.75 <sup>a)</sup>	0.58(2) <sup>b)</sup>
Cu	2a	0	0	0	0.25	0.58(2)
01	4 <i>c</i>	0	1/2	0	1.0	0.85(1)
02	4 <i>e</i>	0	0	0.1764(2)	1.0	1.73(2)
	S	pace group: I	<b>!</b> 4/m m m (	No. 139) and Z	= 2	
$a = b = 3.8173(6)$ Å, $c = 12.7752(2)$ , $\alpha = \beta = \gamma = 90^{\circ}$						
	$R_{ m w}$	p = 7.39 %, <i>R</i>	$R_{\rm p} = 5.30$ %	, $S (= R_{wp}/R_e) =$	2.82	
a) Constraints on isotronia thermal factors: $U_{i}$ (La) = $U_{i}$ (Ca) $U_{i}$ (Ni) = $U_{i}$ (Cu)						

a) Constraints on isotropic thermal factors:  $U_{iso}$  (La) =  $U_{iso}$ (Ca),  $U_{iso}$  (Ni) =  $U_{iso}$ (Cu).

b) Constraints on occupation factors: g(La) + g(Ca) = 1, g(Ni) + g(Cu) = 1.