

Electronic Supplementary Information

Microwave assisted aqueous synthesis of core-shell $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS}$ quantum dots for high performance sensitized solar cells

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Synthesis of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS}$ quantum dots

In the present work, a two-step microwave assisted method is employed to realize the rapid and homogeneous growth of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS}$ QDs. First, the $\text{CdSe}_x\text{Te}_{1-x}$ core QDs were synthesized as below. CdCl_2 (1.25 mmol) was dissolved to 150 ml de-ionized water in 250 mL three necked flask. Then MPA (0.230 mL) and tri-sodium citrate dehydrate (50 mg) were added into the flask in sequence under vigorous magnetic stirring. After the pH of premixed solution was tuned to 9.0, Na_2TeO_3 (0.1 mmol), Na_2SeSO_3 (0.02 mmol) and NaBH_4 (100 mg) were added successively. The reaction vessel was transferred to the microwave oven equipped with a reflux condenser and heated to 95 °C for 10 min. $\text{CdSe}_x\text{Te}_{1-x}$ QDs was taken when the temperature cooled down to lower than 50 °C. Next, the $\text{CdSe}_x\text{Te}_{1-x}$ QDs with thin shell CdS ($\text{CdSe}_x\text{Te}_{1-x}/\text{CdS}$ -I) were obtained by simply adding the 30 mL premixed precursor solution containing CdCl_2 (1.25 mmol) and MPA (0.185 ml) without prior purification. Specifically, the pH of solution was adjusted to 12.2 and then heated at 100 °C with microwave irradiation for another 20 min. For purification, alcohol was added, and then the suspension was centrifuged and the supernatant was removed.

Preparation of Na_2SeSO_3 solution

Anhydrous sodium sulfite (0.336 g) and selenium powder (0.0843 g) were added in 100 ml one-neck flask followed by 20 mL deionized water, which was heated to 80 °C in an oil bath for 4 h. After cooling down to room temperature, the clear portion of the solution was filtered and stored in the refrigerator at 4 °C for use.

Fabrication of TiO₂ photoanodes

The double layered TiO₂ photoanodes were fabricated on well-cleaned FTO glass via doctor blade technique. A 11 μm -thickness transparent layer of 20 nm sized anatase TiO₂ particles was first deposited on FTO substrates, followed by sintering at 450 °C for 30 min. and 6 μm -thickness light scattering layer of 300 nm sized rutile TiO₂ particles was further coated, subsequently annealed at 450 °C for 30 min. Finally, the obtained TiO₂ electrodes were treated with an aqueous solution of 40 mM TiCl₄ at 70 °C for 40 min, followed by a sintering process at 500 °C for 30 min. The as-prepared TiO₂ electrodes were used for further sensitization with various QDs.

Fabrication of QDs sensitized TiO₂ photoanodes

To demonstrate the viability of CdSe_xTe_{1-x}/CdS-I QDs as sensitizer in QSCs, core-shell CdSe_xTe_{1-x}/CdS-I/CdS-II QDs sensitized TiO₂ was made to test their photoactivity. First, the bare TiO₂ electrodes were directly immersed into presynthesized CdSe_xTe_{1-x}/CdS-I QDs aqueous solution with pH~12 for 24 h to ensure maximized surface coverage. The overcoating of thick CdS-II shell had been achieved via in situ successive ionic layer adsorption and reaction (SILAR) method according to the literature procedure.¹ Briefly, the CdSe_xTe_{1-x}/CdS-I QDs modified TiO₂ electrodes were first dipped into a methanol solution containing Cd(CH₃COO)₂·2H₂O (0.05 M) for 1 min, and then subject it into a Na₂S·9H₂O (0.05 M) water/methanol solution (1:1 by volume) for 1 min in sequence. The electrodes were thoroughly rinsed with distilled water to remove the excess precursor after each dipping step. This procedure was repeated for 6 cycles to form core-shell CdSe_xTe_{1-x}/CdS-I/CdS-II sensitized electrodes. For comparison, CdSe_xTe_{1-x}/CdS-I and CdS-II QDs sensitized TiO₂ electrodes were also made.

Preparation of Cu₂S counter electrodes

Porous Cu₂S was prepared by immersing brass foil in 37% HCl solution at 70 °C for 5 min , followed by rinsing with deionized water and dring with nitrogen flow, and subsequently dipping it into polysulfide solution for 10 min for further use.

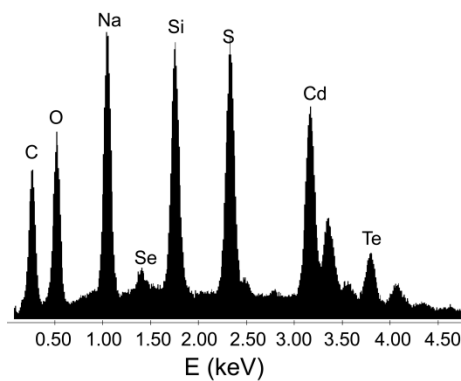


Fig. S1. EDX spectrum of $\text{CdSe}_x\text{Te}_{1-x}$, the corresponding atomic ratio of Te and Se is 1.03:0.18.

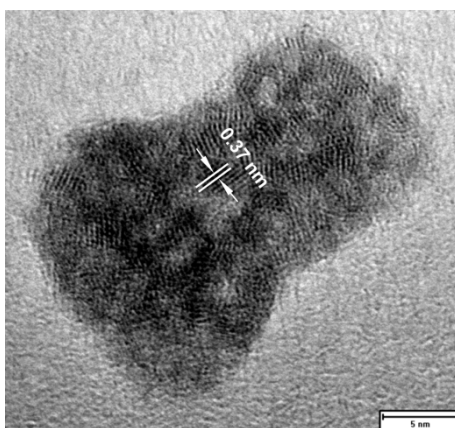


Fig. S2. HRTEM image of $\text{CdSe}_x\text{Te}_{1-x}$ QDs with an average size of 2.5 nm.

Stability of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ and $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ based QSCs

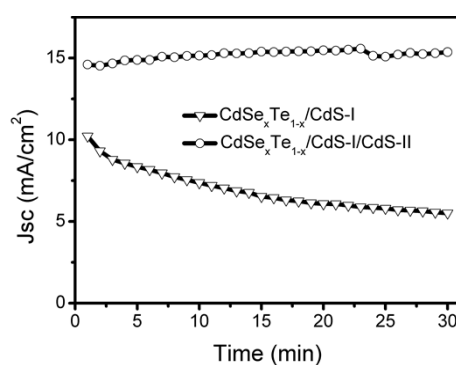


Fig. S3 Photocurrent stability of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ and $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ under continuous illumination of $100 \text{ mW}/\text{cm}^2$.

Stability of the devices is evaluated by measuring J_{sc} with respect to time under continuous illumination of $100 \text{ mW}/\text{cm}^2$. As shown in Fig. S3, for $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ based QSCs,

a steady photocurrent is observed. While for the $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ based QSCs, the photocurrent decreases quickly after 30 min irradiation. Strong corrosion of the polysulfide electrolyte on the $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ likely leads to the reduction in photocurrent during long period operation. Therefore, it can be concluded that the in situ successive ionic layer adsorption and reaction growth of thick CdS shell on the $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ is a promising strategy to enhance the photostability of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ QDs.

The influence of CdS-II cycles on the performance of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ based QSCs

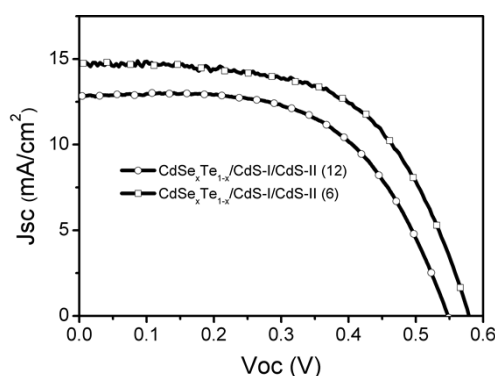


Fig. S4 The influence of CdS-II cycles on the performance of $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ based QSCs

Table S1 Device parameters of different CdS-II cycles on $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ based QSCs

Sample	J_{SC} (mA/cm ²)	V_{OC} (V)	FF	η (%)
$\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ (6)	14.79	0.578	0.59	5.04
$\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ (12)	12.83	0.548	0.59	4.15

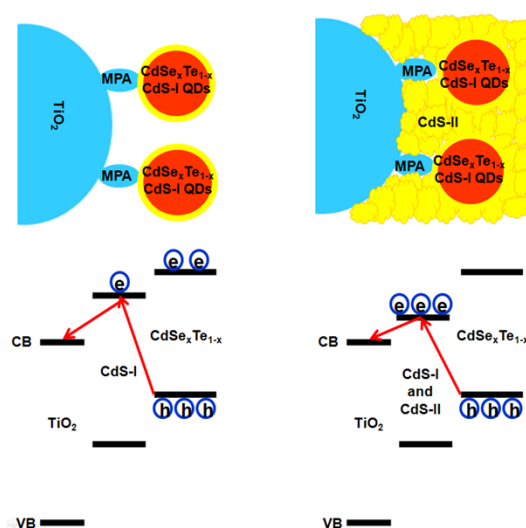


Fig. S5 Schematic illustrating of the energy level alignment for $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ and $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ quantum dot sensitized TiO_2 electrodes

As is well known, the effective bandgap of type II core-shell heterostructured QDs can be manipulated by selecting appropriate semiconductor materials and varying the core size and shell thickness. The effective bandgap determined by the valence band of core and conductive band of shell. To understand the physical processes underlying the broad light absorption, schematic illustrating of the energy level alignment for $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}$ and $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ quantum dot sensitized TiO_2 electrodes (Fig. S5) were added. Due to the quantum confinement effect, the conduction band of $\text{CdSe}_x\text{Te}_{1-x}$ (core) is up-shifted and a type-II structure with a staggered band offset is formed with the increasing shell thickness. The coupling of $\text{CdSe}_x\text{Te}_{1-x}$ with CdS shell results in stronger wave function interfacial interaction which leads to spatially indirect energy gap with the significant redshift of the absorption edge of this core-shell QDs, in which the electronic alignment throughout the core-shell structure causes spatial separation of the electrons and holes between the $\text{CdSe}_x\text{Te}_{1-x}$ (holes) and CdS (electrons). This spatial separation of charge carriers enables fast electron transfer from the QD sensitizer to the TiO_2 , which simultaneously suppresses the charge-recombination process because the shell acts as a tunneling barrier for the hole localized inside the core. Therefore, the obtained $\text{CdSe}_x\text{Te}_{1-x}/\text{CdS-I}/\text{CdS-II}$ core/thick shell NCs has a narrower indirect band gap than both the direct bandgap of $\text{CdSe}_x\text{Te}_{1-x}$ and CdS. In our work, the resultant narrow bandgap can be easily controlled by varying the SILAR cycles of CdS shell to form thick shell with lower conductive band position, which caused the significant redshift of light absorption range. In addition, with the introduction of in situ SILAR deposition CdS-II layer, much surface state on aqueous QDs and TiO_2 can be passivated to retard the internal interface recombination.

Table S2 Reproducibility of solar cell performance

Sample	J_{SC} (mA/cm ²)	V_{OC} (V)	FF	η (%)
CdSe _x Te _{1-x} /CdS-I	9.01	0.495	0.58	2.59
	10.23	0.502	0.53	2.72
	10.29	0.457	0.47	2.21
	Average value	9.84	0.485	2.53
CdS-II(6)	7.57	0.515	0.55	2.14
	7.92	0.509	0.52	2.10
	8.37	0.501	0.47	1.97
	Average value	7.95	0.508	2.06
CdSe _x Te _{1-x} /CdS-I/CdS-II(6)	14.49	0.575	0.58	4.83
	14.63	0.582	0.58	4.94
	14.79	0.578	0.59	5.04
	Average value	14.64	0.578	4.91

Reference

- (1) V. Gonzalez-Pedro, C. Sima, G. Marzari, P. P. Boix, S. Gimenez, Q. Shen, T. Dittrich and I. Mora-Sero, *Phys. Chem. Chem. Phys.*, 2013, **15**, 13835.