# Insights Into an Intriguing Gas Sorption Mechanism in a Polar Metal–Organic Framework with Open-Metal Sites and Narrow Channels

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# [Cu(Me-4py-trz-ia)] System Cell



Figure S1. (a) The *a* axis view, (b) *b* axis view, and (c) shifted *c* axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)]. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.



Figure S2. (a) A view of the ca.  $45^{\circ}$  angle between the a and b axes and (b) the shifted c axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)]. A singular location of binding sites 1 and 2 are also shown. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.



Figure S3. (a) A view of the *ca.*  $45^{\circ}$  angle between the *a* and *b* axes of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)]. The magenta circles indicate the carboxylate group functionality. The region between the two carboxylate oxygen atoms can only be accessed after prior interaction with the adjacent methyl groups. (b) The shifted *c* axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)]. The magenta circles indicate the methyl group functionality. The sorbate molecule must come across the region between the two adjacent methyl groups before it can sorb into the interior of the MOF. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.

#### **Simulation Parameters**

Repulsion/dispersion interactions were modeled using the Lennard-Jones 12–6 potential<sup>1</sup> and these parameters ( $\epsilon$  and  $\sigma$ ) were taken from the Optimized Potentials for Liquid Simulations – All Atom (OPLS–AA) force field<sup>2</sup> for all aromatic C and H atoms and the Universal Force Field (UFF)<sup>3</sup> for all other atoms.

The atomic point partial charges for the atoms in [Cu(Me-4py-trz-ia)] were determined from electronic structure calculations on several fragments large enough to mimic the chemical environment of the MOF. Examination of the unit cell for [Cu(Me-4py-trz-ia)] revealed 36 atoms in chemically distinct atomic environments (Figure S4). Indeed, there are two chemically distinct  $Cu^{2+}$  ions and each atom on the Me-4py-trz-ia<sup>2-</sup> linker is chemically distinguishable. This served as the basis for selecting appropriate fragments for charge-fitting calculations. The addition of hydrogen atoms, where appropriate, was required for the chemical termination of fragment boundaries. The fragments that were selected for the charge-fitting calculations for [Cu(Me-4py-trz-ia)] in this study are shown in Figure S5.

All calculations on each fragment were performed using the NWChem *ab initio* simulation software.<sup>4</sup> All light atoms were treated with the 6-31G<sup>\*</sup> basis set, while the LANL2DZ<sup>5-7</sup> effective core potential basis set was used to treat the inner electrons of the  $Cu^{2+}$  ions. The former has been shown to produce overpolarized charges that are appropriate for condensed phase simulations.<sup>8</sup> The partial charges were determined through a least-squared fit approach<sup>9,10</sup> to the electrostatic potential surface of each fragment. For each chemically distinct atom, the partial charges were averaged between the fragments. Afterwards, the partial charges were adjusted such that the total charge of the framework was equal to zero. The average partial charges for all chemically distinct atoms for each fragment are shown in Table S1. Note, atoms that are buried or located on the edges of the fragments were not included in the averaging. The final partial charges for each chemically distinct atom for [Cu(Me-4py-trz-ia)] can be found in Table S2.

The atomic point polarizabilities for all light atoms used a set of rigorously parametrized empirical values that were shown to be highly transferable.<sup>11–23</sup> These atoms were assigned the exponential polarizabilities (C = 1.28860 Å<sup>3</sup>, H = 0.41380 Å<sup>3</sup>, O = 0.85200 Å<sup>3</sup>, N = 0.97157 Å<sup>3</sup>) and associated damping parameter ( $\lambda = 2.1304$ ) provided by the work of van Duijnen et al.<sup>24</sup> An atomic point polarizability value of 2.19630 Å<sup>3</sup> was used to parameterize the Cu<sup>2+</sup> ions; this parameter was determined in previous work.<sup>17</sup>



Figure S4. Chemically distinct atoms in [Cu(Me-4py-trz-ia)] as referred to in Tables S1 and S2. Numerical labeling correspond to chemically distinct atoms evaluated. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.

Figure S5. Fragments of [Cu(Me-4py-trz-ia)] that were selected for gas phase charge fitting. Numerical labeling corresponds to chemically distinct atoms evaluated. Atom colors: C = cyan, H = white, O = red, Cu = tan.







Atom	Label	Frag 1	Frag 2	Frag 3	Frag 4	Frag 5	Frag 6	Frag 7	Frag 8	Frag 9
Cu	1	0.0808	0.0970	-	-	-	0.0050	-0.0011	0.0165	0.0382
Cu	2	-	-	0.4432	0.4317	0.4716	0.4727	0.4262	0.4311	0.5439
0	3	-	-	-	-	-	-0.5063	-0.5018	-0.5513	-0.5691
0	4	-	-	-	-	-	-0.6381	-0.6232	-0.6003	-0.6471
0	5	-	-	-	-	-	-0.7061	-0.6881	-0.7046	-0.7621
0	6	-	-	-	-	-	-0.7440	-0.7503	-0.7058	-0.7597
С	7	-	-	-	-	-	0.8259	0.8308	0.7877	0.8852
С	8	-	-	-	-	-	0.8363	0.8444	0.8092	0.8926
С	9	-	-	-	-	-	-0.1294	-0.1535	-0.0711	-0.1232
С	10	-	-	-	-	-	-0.0360	-0.0272	0.0426	0.0196
С	11	-	-	-	-	-	-0.1162	-0.1233	-0.1119	-0.1642
Н	12	-	-	-	-	-	0.1504	0.1578	0.1690	0.1370
С	13	-	-	-	-	-	-	-0.1858	-0.1966	-
С	14	-	-	-	-	-	-	-0.2192	-0.2494	-
Н	15	-	-	-	-	-	-	0.1072	0.1134	-
Н	16	-	-	-	-	-	-	0.1620	0.1717	-
С	17	-	0.1517	-	-	-	-	-	0.1659	0.2131
Ν	18	-	-	-	-0.1737	-0.2045	-	-	-	-0.1111
С	19	-	-	0.3564	0.2797	0.3117	-	-	-	0.3043
С	20	-	-	-0.0996	-0.1444	-0.1458	-	-	-	-0.1167
Н	21	-	-	0.1333	0.1662	0.1656	-	-	-	0.1516
Н	22	-	-	0.0600	0.0461	0.0421	-	-	-	0.0074
Н	23	-	-	0.0335	0.0433	0.0440	-	-	-	0.0420
Ν	24	-	-	0.0790	0.0691	-0.0047	-	-	-	-0.0064
Ν	25	-	-	-0.4163	-0.3963	-0.4018	-	-	-	-0.3997
С	26	-	-	-	0.2720	0.2971	-	-	-	0.3269
С	27	-0.0233	-0.0025	-	-	-	-	-	-	-0.0484
С	28	-0.2388	-0.2172	-	-	-	-	-	-	-0.2459
С	29	-0.3074	-0.3374	-	-	-	-	-	-	-0.3087
Н	30	0.1360	0.1228	-	-	-	-	-	-	0.1213
Н	31	0.1327	0.1520	-	-	-	-	-	-	0.1672
С	32	0.1900	0.1464	-	-	-	-	-	-	0.1633
С	33	0.1190	0.1458	-	-	-	-	-	-	0.1410
Н	34	0.1098	0.1107	-	-	-	-	-	-	0.1028
Н	35	0.1156	0.1137	-	-	-	-	-	-	0.1152
Ν	36	-0.0616	-0.0405	-	-	-	-	-	-	0.0002

**Table S1.** Partial charges for individual fragments of [Cu(Me-4py-trz-ia)] as listed in Figure S5 in units of  $e^-$ . Label of atoms correspond to Figure S4.

Atom	Label	$q~(e^-)$		
Cu	1	0.03940		
Cu	2	0.46006		
0	3	-0.49370		
0	4	-0.58188		
0	5	-0.66358		
0	6	-0.68652		
C	7	0.84563		
C	8	0.83240		
C	9	-0.00870		
C	10	-0.09042		
C	11	-0.11746		
Н	12	0.16478		
C	13	-0.21339		
C	14	-0.16610		
Н	15	0.12427		
Н	16	0.16703		
C	17	0.16613		
Ν	18	-0.15129		
C	19	0.31298		
C	20	-0.11746		
Н	21	0.15418		
H	22	0.04068		
Н	23	0.03889		
Ν	24	0.03425		
Ν	25	-0.37437		
C	26	0.29863		
C	27	-0.02293		
C	28	-0.21706		
C	29	-0.29488		
H	30	0.12670		
H	31	0.15062		
C	32	0.16655		
C	33	0.13525		
Н	34	0.10775		
H	35	0.11482		
Ν	36	-0.03153		

**Table S2**. The partial charges  $(e^-)$  for the chemically distinct atoms in [Cu(Me-4py-trz-ia)] as used in the simulations. Label of atoms correspond to Figure S4.

## Many-Body Polarization

Many-body polarization was explicitly included in the simulations by use of a Thole-Applequist type model.<sup>25–27</sup> An overview of the model used in this work is given here. Consider a static electric field applied to a system containing N atomic sites. The induced dipole,  $\vec{\mu}$ , at each site *i* is calculated *via* the following:

$$\vec{\mu}_i = \alpha_i^{\circ} \left( \vec{E}_i^{stat} + \vec{E}_i^{ind} \right) \tag{1}$$

where  $\alpha^{\circ}$  is a scalar atomic point polarizability,  $\vec{E}^{stat}$  is the electrostatic field vector due to the atomic point partial charges of the MOF atoms and the sorbate molecules, and  $\vec{E}^{ind}$  is the electrostatic field vector due to the atomic induced dipoles. Note,  $\vec{E}^{stat}$  is calculated using Wolf summation.<sup>28–30</sup> Rewriting equation 1 in terms of the dipole field tensor,  $\hat{\mathbf{T}}_{ij}$ , yields the following:

$$\vec{\mu}_i = \alpha_i^{\circ} \left( \vec{E}_i^{stat} - \sum_{j \neq i}^N \mathbf{\hat{T}}_{ij} \vec{\mu}_j \right)$$
(2)

In this work, the following dipole field tensor is used:

$$\hat{T}_{ij}^{\alpha\beta} = \nabla^{\alpha}\nabla^{\beta} \left(\frac{s(r)}{r_{ij}}\right) \tag{3}$$

where  $r_{ij}$  is the distance between sites *i* and *j* and the screening function  $s(r) = (\frac{\lambda r}{2} + 1)e^{-\lambda r}$  serves to describe the charge distributions of the interacting dipoles. The damping parameter  $\lambda$  is set to 2.1304, consistent with the work performed by Thole and others, eliminating the short range divergences that occur in the polarization model when dealing with point dipoles (known as the polarization catastrophe).<sup>24,26,31</sup>

Although matrix inversion may be used to find an exact solution to the self-consistent field problem (equation 2), iterative methods can produce a solution in a fraction of the computation time and are extremely stable when Gauss-Seidel relaxation is employed.<sup>30,32</sup> Initially, every induced dipole vector in the system is evaluated as the following:

$$\vec{\mu}_i = \alpha_i^\circ \vec{E}_i^{stat} \tag{4}$$

On each subsequent iteration  $\xi$ , the dipole field tensor is written as a sum of its lower triangular and strictly upper triangular components  $\hat{\mathbf{T}} \equiv \hat{\mathbf{T}}^{\mathbf{L}} + \hat{\mathbf{T}}^{\mathbf{U}}$ , and each dipole is updated sequentially *via* forward substitution:

$$\vec{\mu}_i^{\xi} = \alpha_i^{\circ} \left( \vec{E}_i^{stat} - \sum_{i>j} \mathbf{\hat{T}}_{ij}^L \vec{\mu}_j^{\xi+1} - \sum_{i(5)$$

Finally, the polarization energy for the MOF-sorbate system is calculated by the following based on the work of Palmo and Krimm<sup>33</sup> using the  $\xi^{\text{th}}$  iteration dipoles and the  $(\xi + 1)^{th}$  induced field:

$$U_{pol}^{\xi} = -\frac{1}{2} \sum_{i} \vec{\mu}_{i}^{\xi} \cdot \vec{E}_{i}^{stat} - \frac{1}{2} \sum_{i} \vec{\mu}_{i}^{\xi} \cdot \vec{E}_{i}^{ind,\xi+1}$$
(6)

#### Grand Canonical Monte Carlo

Simulations of H<sub>2</sub> and CO<sub>2</sub> sorption in [Cu(Me-4py-trz-ia)] were performed using grand canonical Monte Carlo (GCMC) on a  $2 \times 2 \times 2$  unit cell system of the MOF. This method entails constraining the chemical potential, volume, and temperature to be constant while allowing the particle number and other statistical mechanical quantities to fluctuate.<sup>34</sup> This was achieved by trial insertion and deletion of sorbate molecules based on a random number generator. Periodic boundary conditions were applied in order to approximate an infinitely extended crystal lattice. A spherical cutoff corresponding to half the shortest system cell dimension length was used. The average particle number was calculated by the following expression:<sup>35,36</sup>

$$\langle N \rangle = \frac{1}{\Xi} \sum_{N=0}^{\infty} e^{\beta \mu N} \left\{ \prod_{i=1}^{3N} \int_{-\infty}^{\infty} dx_i \right\} N e^{-\beta U(x_1, \dots x_{3N})}$$
(7)

where  $\Xi$  is the grand canonical partition function,  $\beta$  is the inverse temperature (equal to the quantity 1/kT, where k is the Boltzmann constant and T is the temperature),  $\mu$  is the chemical potential of the gas reservoir, and U is the total potential energy. The chemical potential for H<sub>2</sub> and CO<sub>2</sub> was determined for a range of temperatures through the BACK<sup>37</sup> and Peng-Robinson equation of state,<sup>38</sup> respectively. The total potential energy of the MOF–sorbate system is calculated by:

$$U = U_{rd} + U_{es} + U_{pol} \tag{8}$$

where  $U_{rd}$  is the repulsion/dispersion energy calculated by the Lennard-Jones potential,<sup>1</sup>  $U_{es}$  is the electrostatic energy calculated by Ewald summation,<sup>39</sup> and  $U_{pol}$  is the polarization energy which is calculated using equation 6.

For the simulations of  $H_2$  sorption at the temperatures considered in this work, quantum mechanical disperion effects were included semiclassically through the fourth-order Feynman-Hibbs correction according to the following equation:<sup>40</sup>

$$U_{FH} = \frac{\beta\hbar^2}{24\mu} \left( U'' + \frac{2}{r}U' \right) + \frac{\beta^2\hbar^4}{1152\mu^2} \left( \frac{15}{r^3}U' + \frac{4}{r}U''' + U'''' \right)$$
(9)

where  $\hbar$  is the reduced Planck's constant and the primes indicate differentiation with respect to pair separation r.

The MOF–sorbate system was also treated with long-range corrections to all terms of the potential due to the finite size of the simulation box. The long-range contribution to the Lennard-Jones potential was calculated by the following based on a previously reported method:<sup>41</sup>

$$U_{LJ}^{LRC} = \frac{16\pi}{3V} \sum_{i=0}^{N-1} \sum_{j=0}^{N-1} \epsilon_{ij} \left( \frac{\sigma_{ij}^9}{3R_c^9} - \frac{\sigma_{ij}^3}{R_c^3} \right)$$
(10)

where  $\epsilon_{ij}$  and  $\sigma_{ij}$  are Lorentz-Berthelot mixed Lennard-Jones parameters, V is the volume of the simulation box, and  $R_C$  is the cut-off distance, which is set to one-half the shortest simulation box length.

Long-range electrostatic interactions were handled by performing full Ewald summation. The long-range correction to the polarization energy was performed by replacing the static electric field with the shifted-field formula according to Wolf *et al.*,  $^{28,29}$  which is the following:

$$\vec{E}_{i}^{shift} = \sum_{j}^{N-1} q_{j} \left( \frac{1}{r_{ij}^{2}} - \frac{1}{R_{c}^{2}} \right) \hat{r}$$
(11)

where q is the atomic point partial charge and  $\hat{r}$  is the radial unit vector.

In GCMC simulation, the excess sorbate uptake, defined as the amount sorbed in the pore volume in excess of the bulk gas capacity in the same free space,<sup>42</sup> was determined from a calculation that utilized an experimental pore volume value<sup>43</sup> of 0.586 cm<sup>3</sup> g<sup>-1</sup> and bulk gas densities *via* the following expression:

$$R_{ex} = 1000 \frac{m(\langle N \rangle - V_p \rho_b)}{M} \tag{12}$$

where  $V_p$  is the pore volume,  $\rho_b$  is the bulk gas density, m is the molar mass of the sorbate, and M is the molar mass of the MOF.

The theoretical isosteric heats of adsorption,  $Q_{st}$ , were calculated based on the fluctuations of the particle number and total potential energy in the MOF–sorbate system through the following expression:<sup>44</sup>

$$Q_{st} = -\frac{\langle NU \rangle - \langle N \rangle \langle U \rangle}{\langle N^2 \rangle - \langle N \rangle^2} + kT$$
(13)

For all state points considered, the simulations consisted of  $2 \times 10^6$  Monte Carlo steps to guarantee equilibration, followed by an additional  $2 \times 10^6$  steps to sample the desired thermodynamic properties. The simulations utilized a correlation time of  $1 \times 10^4$  steps in order to guarantee uncorrelated configurations. All simulations were performed using the Massively Parallel Monte Carlo (MPMC) code,<sup>45</sup> which is currently available for download on Google Code.

#### Prediction of Pore Volume Through Helium Adsorption

The theoretical pore volume of [Cu(Me-4py-trz-ia)] was calculated using a previously reported molecular simulation procedure.<sup>46</sup> The relationship between the absolute amount of sorbed molecules,  $N_{abs}$ , and the amount of sorbed molecules in excess of the number of molecules that would occupy the free pore volume at bulk gas conditions,  $N_{ex}$ , is represented by the following equation:<sup>42</sup>

$$N_{ex} = N_{abs} - V_p \rho_b \tag{14}$$

where  $V_p$  is the pore volume and  $\rho_b$  is the bulk phase density. For helium at ambient temperature,  $N_{ex}$  is equal to 0 because it is a non-adsorbing (or weakly adsorbing) gas, and thus, equation 14 becomes the following:

$$V_p = \frac{N_{abs}}{\rho_b} \tag{15}$$

At low pressures, helium is assumed to exhibit ideal behavior and this yields equation 15 as:

$$V_p = \frac{N_{abs}kT}{P} \tag{16}$$

where k is the Boltzmann constant, T is the temperature, and P is the pressure. The adsorption second virial coefficient from experiment and simulation is calculated by:

$$B_{ex} = kT \lim_{P \to 0} \left(\frac{dN_{ex}}{dP}\right) \tag{17}$$

and

$$B_{abs} = kT \lim_{P \to 0} \left( \frac{dN_{abs}}{dP} \right) \tag{18}$$

Combining equations 14, 17, and 18 gives

$$B_{ex} = B_{abs} - V_p \tag{19}$$

For helium, both  $N_{ex}$  and  $B_{ex}$  are equal to zero, so equation 19 becomes:

$$V_p = B_{abs} \approx kTm \tag{20}$$

where *m* is the slope of the nearly linear helium adsorption isotherm as *P* approaches zero. Thus, GCMC simulations of helium sorption were performed in [Cu(Me-4py-trz-ia)] at 298 K and pressures up to 50.0 atm using a previously reported helium model ( $\epsilon = 10.22$  K;  $\sigma = 2.28$  Å).<sup>47</sup> The helium sorption isotherm in [Cu(Me-4py-trz-ia)] is shown in Figure S6. The slope (in units of cm<sup>3</sup> g<sup>-1</sup> atm<sup>-1</sup>) was determined from linear fitting of the isotherm (R<sup>2</sup> = 0.999) and it was used in equation 20 for the calculation of the theoretical pore volume. In this work, the pore volume was calculated to be 0.591 cm<sup>3</sup> g<sup>-1</sup>, which is very close to the experimental value of 0.586 cm<sup>3</sup> g<sup>-1</sup>.



Figure S6. The simulated He sorption isotherm at 298 K and up to 50.0 atm for [Cu(Me-4py-trz-ia)]. The linear fitting of the sorption isotherm is also shown.

## Sorbate Potentials

The H<sub>2</sub> potential employed in this work is a five–site polarizable model that includes partial charges and atomic point polarizabilities on the center-of-mass position and the location of the H atoms (H–H distance is 0.742 Å). These quantities represent the molecular quadrupole and polarizability tensor of H<sub>2</sub>, respectively. In addition, this model contains Lennard-Jones parameters that are localized on the center-of-mass site and two phantom sites that extend 0.363 Å from the centerof-mass site along the  $C_{\infty}$  axis. This model is known to reproduce high density hydrogen in the bulk environment and is capable of describing the proper sorption behavior in highly charged/polar MOFs.<sup>48</sup>

The CO<sub>2</sub> potential used herein is a five-site polarizable model that contains partial charges and atomic point polarizabilities localized on the atomic locations of the carbon and oxygen atoms (C–O distance is 1.162 Å). Although the point polarizability values are representative of the polarizability tensor for CO<sub>2</sub>, the partial charges slightly overestimate the molecular quadrupole for the compound; the latter finding could attributed to the limitation of the potential form. Further, there are Lennard-Jones parameters that are located on the carbon atom and the off-site positions that extend 1.114 Å away from the carbon atom. This model is capable of reproducing bulk CO<sub>2</sub> data and describing sorption in MOFs where induced dipole effects are critical.<sup>49</sup> The parameters for both H<sub>2</sub> and CO<sub>2</sub> models can be found in Table S3.

Model	Atomic Site	$\epsilon(K)$	$\sigma(\text{\AA})$	$q~(e^-)$	$\alpha^{\circ}(\text{\AA}^3)$
	COM	12.76532	3.15528	-0.74640	0.69380
$H_2$	Н	0.00000	0.00000	0.37320	0.00044
	OS	2.16726	2.37031	0.00000	0.00000
	С	19.61757	3.30366	0.77134	1.22810
$CO_2$	О	0.00000	0.00000	-0.38567	0.73950
	OS	46.47457	2.99429	0.00000	0.00000

Table S3. Parameters used to characterize the polarizable  $H_2$  and  $CO_2$  potentials used in this work. COM refers to the center-of-mass position and OS refers to the off-site positions.

## H<sub>2</sub> Sorption Results

Figure S7 shows that the simulated low-pressure (up to 100 kPa) excess H<sub>2</sub> sorption isotherms in [Cu(Me-4py-trz-ia)] are slightly higher than experiment at 77 K and 87 K at all pressures considered, whereas it is in rather good agreement with experiment at 97 K. All experimental data points in this figure were estimated from reference 43. However, at all state points considered, the simulated uptakes are in very good agreement with the corresponding experimental uptakes to within joint uncertainties. The simulated H<sub>2</sub> sorption isotherm in [Cu(Me-4py-trz-ia)] at 77 K and higher pressures (up to 3.0 MPa) is also in good agreement with experiment (data points estimated from reference 43) for the pressure range considered (Figure S8). The maximum calculated excess H<sub>2</sub> uptake was approximately 20.0 mmol  $g^{-1}$ , which is consistent with experimental measurements.



Figure S7. Low-pressure (up to 1.0 atm) excess  $H_2$  sorption isotherms in [Cu(Me-4py-trz-ia)] at 77 K (solid), 87 K (dashed), and 97 K (dotted) for experiment (black) and simulation (red).



Figure S8. High-pressure (up to 30.0 atm) excess  $H_2$  sorption isotherms in [Cu(Me-4py-trz-ia)] at 77 K for experiment (black) and simulation (red). Note, the error bars at each state point for the simulated plot are small, especially at higher pressures, and thus, have been omitted for clarity.



Figure S9. Radial distribution functions, g(r), for H<sub>2</sub> molecules about the Cu1 ions (atom label 1 in Figure S4) (red) and the Cu2 ions (atom label 2 in Figure S4) (blue) in [Cu(Me-4py-trz-ia)] at 77 K and pressures of 0.01 atm (solid), 0.40 atm (dashed), and 1.0 atm (dotted).



Figure S10. Normalized  $H_2$  dipole distribution at 77 K and various pressures in [Cu(Me-4py-trz-ia)]. The region from 0.065 to 0.20 Debye corresponds to sorption to site 1. The region from 0.00 to 0.065 Debye corresponds to sorption to site 2.



Figure S11. The shifted c axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)] showing the sites of H<sub>2</sub> sorption as a function of induced dipole magnitude according to Figure S10: (a) 0.065 to 0.20 Debye; (b) 0.00 to 0.065 Debye. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.



Figure S12. (a) A view of the *ca.*  $45^{\circ}$  angle between the *a* and *b* axes and (b) the shifted *c* axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)] showing the positions of the sorbed H<sub>2</sub> molecules about sites 1 and 2 from simulated annealing. The sorbate molecules are shown in orange. The distance between the two center-of-mass sites is 4.20 Å. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.

### CO<sub>2</sub> Sorption Results

All simulated excess CO<sub>2</sub> sorption isotherms in comparison with the corresponding experimental data (produced by using the parameters from reference 43) in [Cu(Me-4py-trz-ia)] are shown in Figure S13. Figure S13(a) shows that the simulated high pressure (up to 4.0 MPa) CO<sub>2</sub> sorption isotherm at 273 K are in excellent agreement with the experimental isotherm at all pressures considered at this temperature. It can be observed that the experimental isotherm reaches CO<sub>2</sub> saturation at approximately 1.0 MPa at this temperature with a value of about 12.0 mmol g<sup>-1</sup>. The simulated CO<sub>2</sub> sorption isotherm also saturates at approximately 12.0 mmol g<sup>-1</sup>, as demonstrated by the excellent agreement between experiment and simulation for all pressures considered at 273 K as shown in Figure S13(a). The simulated CO<sub>2</sub> sorption isotherm at 298 K is in excellent agreement with experiment at pressures up to 0.10 MPa (Figure S13(b)). However, at pressures of 0.20 MPa and above, the simulated uptakes are notably higher than experiment at this temperature. It can be observed in Figure S13(c) that, at 323 K, the simulated sorption isotherm undersorbs experiment at low pressures (up to 0.50 MPa) and oversorbs experiment at higher pressures (1.0 MPa and above). In general, the simulated CO<sub>2</sub> uptakes agree with the corresponding experimental measurements to within joint uncertainties at all state points considered and the conclusions presented in the manuscript are not materially affected by the differences.



Figure S13. High-pressure (up to 40.0 atm) excess  $CO_2$  sorption isotherms in [Cu(Me-4py-trz-ia)] at (a) 273 K, (b) 298 K, and (c) 323 K for experiment (black) and simulation (red). Note, the error bars at each state point for the simulated plots are small, especially at higher pressures, and thus, have been omitted for clarity.



Figure S14. Radial distribution functions, g(r), for CO<sub>2</sub> molecules about the Cu1 ions (atom label 1 in Figure S4) (red) and Cu2 ions (atom label 2 in Figure S4) (blue) in [Cu(Me-4py-trz-ia)] at 273 K and pressures of 0.05 atm (solid), 0.40 atm (dashed), and 1.0 atm (dotted).



Figure S15. Normalized CO<sub>2</sub> dipole distribution at 273 K and various pressures in [Cu(Me-4py-trz-ia)]. The region from 0.30 to 0.80 Debye corresponds to sorption to site 1. The region from 0.00 to 0.12 Debye corresponds to sorption to site 2. The region from 0.12 to 0.30 Debye corresponds to sorption to both sites 1 and 2.



Figure S16. The shifted c axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)] showing the sites of CO<sub>2</sub> sorption as a function of induced dipole magnitude according to Figure S15: (a) 0.30 to 0.80 Debye; (b) 0.00 to 0.12 Debye. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.



Figure S17. Molecular illustration of a sorbed  $CO_2$  molecule about (a) site 1 and (b) site 2 in [Cu(Me-4py-trz-ia)] as determined from simulation. Note, orientational constraints are not imposed on the  $CO_2$  molecule at the binding sites. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.



Figure S18. (a) A view of the *ca.*  $45^{\circ}$  angle between the *a* and *b* axes and (b) the shifted *c* axis view of the  $2 \times 2 \times 2$  unit cell system of [Cu(Me-4py-trz-ia)] showing the positions of the sorbed CO<sub>2</sub> molecules about sites 1 and 2 from simulated annealing. The distance between the two CO<sub>2</sub> carbon atoms is 4.29 Å. Atom colors: C = cyan, H = white, N = blue, O = red, Cu = tan.

#### **Molecular Dynamics Simulations**

Molecular dynamics (MD) simulations were performed in [Cu(Me-4py-trz-ia)] using an in-house modification of the Largescale Atomic/Molecular Massively Parallel Simulator (LAMMPS)  $code^{50}$  to include many-body polarization.<sup>51</sup> The same MOF parameters and sorbate models that were used in the Monte Carlo simulations were used for the MD simulations, including assuming the framework is rigid. The H<sub>2</sub> sorption simulation was performed with microcanonical (NVE) integration and a timestep of 2 fs. A single hydrogen molecule was simulated at a location between two adjacent methyl groups (site 2); this was obtained from the GCMC simulations. The hydrogen molecule was given a velocity large enough to overcome the potential energy well between the two methyl groups and explore the rest of the framework, including traveling to the region between two adjacent carboxylate oxygen atoms (site 1). A portion of the hydrogen molecule's trajectory was extracted around the time of a transition from site 2 to site 1 and single point energies at each timestep were calculated using the MPMC code.<sup>45</sup> The CO<sub>2</sub> sorption simulation was performed in a similar manner, with the exception that Nosé-Hoover style canonical (NVT) integration was used with the temperature set at 200 K and with a temperature damping time of 100 timesteps.

A plot of single-point energies as the H<sub>2</sub> molecule travels from site 2 to site 1 in [Cu(Me-4py-trz-ia)] is shown in Figure S19(a). The fluctuations in energy at the beginning can be attributed to librational effects from the starting position (site 2). Nevertheless, as the H<sub>2</sub> molecule escapes site 2 and makes its way over to site 1, it must overcome a barrier that is approximately 0.80 kJ mol<sup>-1</sup> in energy. This is a rather low value as it suggests that the hydrogen molecules can easily jump from site 2 to site 1 in this MOF at 77 K. A plot of similar type can be seen for CO<sub>2</sub> sorption in Figure S19(b). It was observed that the barrier height for the transition from site 2 to site 1 (measuring from the well of site 2 to the highest energy conformation) is approximately  $10.5 \text{ kJ mol}^{-1}$ , which is significant, but is clearly overcome in filling the sites under the conditions considered. Note that in the case of CO<sub>2</sub>, the transition between site 2 to site 1 is characterized by several energy peaks with shallow minima in between them. This is because the bulky methyl functionality hinders a smooth transition from site 2 to site 1 for the large CO<sub>2</sub> molecule.



Figure S19. A plot of single-point energies as the sorbate molecule travels from site 2 to site 1 in [Cu(Me-4py-trz-ia)] as determined with the aid of molecular dynamics simulations: (a)  $H_2$ ; (b)  $CO_2$ .

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