# Dopamine-Mo<sup>VI</sup> complexation-assisted large-scale aqueous synthesis of single-layer MoS<sub>2</sub>/carbon sandwich structure for ultrafast, long-life lithium-ion batteries

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# **Experimental Section**

# Materials

Dopamine hydrochloride (DOPA-HCl), thioacetamide (TAA), sodium molybdate dihydrate (Na<sub>2</sub>MoO<sub>4</sub>.2H<sub>2</sub>O), sodium tungstate dihydrate (Na<sub>2</sub>WO<sub>4</sub>.2H<sub>2</sub>O), ferric chloride hexahydrate (FeCl<sub>3</sub>.6H<sub>2</sub>O) and bulk MoS<sub>2</sub> powder (<  $2\mu$ m) were purchased from Sigma-Aldrich (USA) and used as received. Electrolyte (1M LiPF<sub>6</sub> in a mixture of ethylene carbonate (EC) and dimethyl carbonate (DMC) at a 1:1 volume ratio) and lithium foil were purchased from Charslton Technologies Pte Ltd (Singapore).

## Synthesis of SLMoS<sub>2</sub>/C composite

The SLMoS<sub>2</sub>/C composite was synthesized using hydrothermal method. Typically, 300 mg Na<sub>2</sub>MoO<sub>4</sub>, 900 mg DOPA-HCl and 900 mg TAA were dissolved in 20 ml deionized water, respectively, and then transferred to a 100 ml Teflon-lined autoclave. The suspension was kept at 200 °C for 16 hours. After cooling down, the black precipitates were collected by centrifugation and washed with DI water and ethanol for 3 times. The obtained sample was annealed at 700 °C for 3 hours in argon to eliminate the surface functional groups.

## Synthesis of Fe<sub>3</sub>O<sub>4</sub>/C and WO<sub>2</sub>/C composites

1 mmol Na<sub>2</sub>WO<sub>4</sub>.2H<sub>2</sub>O and FeCl<sub>3</sub>.6H<sub>2</sub>O were dissolved in 60 ml deionized water and then mixed with 4 mmol DOPA-HCl, respectively. The suspensions were transferred to a 100 ml Teflon-lined autoclave and kept at 160 °C for 16 hours directly. The obtained samples were annealed at 400 °C for 3 hours in argon.

### Characterization

The morphologies of the samples were studied using a field-emission scanning electron microscope (FESEM, JEOL JSM 7600) at an accelerating voltage of 5 kV and a transmission electron microscope (TEM, JEOL 2100) at 200 kV. X-ray photoelectron spectroscopy (XPS) measurements were conducted on a Kratos Analytical AXIS His spectrometer with a monochromatized Al Ka X-ray source (1486.6 eV photons). X-ray diffraction (XRD) patterns were recorded on a Bruker GADDS X-ray diffractometer. Fourier transform infrared spectroscopic (FTIR) measurements were performed using a Shimadzu FTIR IR Prestige-21 using KBr pellets. The compositions of the samples were determined by thermogravimetric analysis (TGA, TA Q500). All the samples were heated from room temperature to 600 °C in air. The UV absorption spectra were measured using a Shimadzu UV-3600 UV–vis–NIR spectrophotometer. Raman spectra were obtained by using WITec CRM200 confocal Raman microscopy system (WITec, Germany) with a laser wavelength of 532 nm.

### **Electrochemical measurements**

The electrochemical performances were evaluated with a standard CR2032 coin cell. A Celgard® 2600 membrane is used as the separator. The working electrode was composed of 70 wt% composites, 20 wt% Super P carbon black, and 10 wt% polyvinylidene fluoride. The typical loading of the anode in the electrode was 2 to 3 mg. The electrochemical tests were performed on a NEWARE BTS-5V10mA battery tester. The cells were cycled between

0.005~3.0 V vs Li<sup>+</sup>/Li. The specific capacity and current density were calculated based on the weight of the SLMoS<sub>2</sub>/C composite. The cyclic voltammetry (CV) was studied on a PGSTAT302N Autolab electrochemical workstation with a voltage window of 0.005~3.0 V and a scan rate of 0.1 mV/s. The electrochemical impedance spectroscopy (EIS) was measured in the frequency range of  $10^{-2}$  to  $10^{6}$  Hz with an AC voltage amplitude of 5 mV.



Fig. S1. Photo images of DOPA-HCl, Na<sub>2</sub>MoO<sub>4</sub> and DMC solutions.



Fig. S2. FTIR spectra of DOPA-HCl, Na<sub>2</sub>MoO<sub>4</sub> and DMC.



Fig. S3. XPS spectra of DOPA-HCl, Na<sub>2</sub>MoO<sub>4</sub>, DMC and SLMoS<sub>2</sub>/C.



Fig. S4. TEM images of the DMC.



Fig. S5. FTIR spectra of the DMC-HT and SLMoS<sub>2</sub>/C.



Fig. S6. TGA curves of (a) SLMoS<sub>2</sub>/C and (b) DMC-HT. The large weight loss between 200-450  $^{\circ}$ C is caused by the combustion of carbon and conversion of MoS<sub>2</sub> to MoO<sub>3</sub> in air.



Fig. S7. (a) SEM and (b) TEM images of DMC-HT.



Fig. S8. TEM images of (a)  $Fe_3O_4/C$  and (b)  $WO_2/C$  composites after annealing.



Fig. S9 XRD patterns of Fe<sub>3</sub>O<sub>4</sub>/C and WO<sub>2</sub>/C nanocomposites.



Fig. S10. Cyclic voltammogram of the SLMoS<sub>2</sub>/C anode.

In the first discharge, a slope starts at about 1.5 V is indicative of the formation of  $Li_xMoS_2$ . The inconspicuous conversion is caused by the expanded (002) spacing, low crystallinity and disordered structure of  $SLMoS_2^{.1-3}$  The  $Li_xMoS_2$  then decomposes into Mo nanoparticles embedded in  $Li_2S$  matrix, giving a cathodic peak at about 0.5 V. The slope below 0.25 V is

assigned to the formation of a solid electrolyte interphase (SEI) film. During the anodic scanning,  $Li^+$  stored within the carbon and defects of SLMoS<sub>2</sub>/C are firstly released, together with the partially oxidation of Mo, leading to broad oxidation peaks centered at about 0.5 V and 1.3 V, respectively. The pronounced peak at about 2.3 V is associated with the oxidation of  $Li_2S$  to sulfur.<sup>4</sup> Therefore, MoS<sub>2</sub> converts to a mixture of sulfur and Mo metal after the first cycle. Accordingly in the following cycles, the reduction peak at 2.0 V can be attributed to the formation of  $Li_2S$ , and the association of  $Li^+$  ions with Mo is found at 1.0 V.<sup>5</sup>



Fig. S11. Cycling performance of the SLMoS<sub>2</sub>/C anode at a current density of 50 mA/g.



Fig. S12. Nyquist plot of SLMoS<sub>2</sub>/C (a) before and (b) after cycles.

The Nyquist plots SLMoS<sub>2</sub>/C anodes consist of one semicircle at high and medium frequencies and a straight line at low frequencies, corresponding to the charge transfer resistance at the electrolyte/electrode interface and the solid-state diffusion resistance of Li<sup>+</sup> in the electrode, respectively. After cycling, the interfacial charge-transfer resistance increases because of the SEI layer formed and destruction of the MoS<sub>2</sub> structure during cycling.

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