

Electronic Supporting Information for:

On the vacancy-controlled dealloying of rapidly solidified Mg-Ag alloys

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Experimental procedure:

Mg-Ag alloys with nominal compositions of $\text{Mg}_{65}\text{Ag}_{35}$, $\text{Mg}_{62}\text{Ag}_{38}$, $\text{Mg}_{58}\text{Ag}_{42}$, $\text{Mg}_{54}\text{Ag}_{46}$, and $\text{Mg}_{50}\text{Ag}_{50}$ (at.%) were prepared from pure Mg (99.9 wt.%) and pure Ag (99.9 wt.%) in a quartz crucible using a high-frequency induction furnace in an argon atmosphere. Using a single roller melt spinning apparatus, the prealloyed ingots were remelted in a quartz tube by high-frequency induction heating and then rapidly solidified onto a copper roller at a circumferential speed of $\sim 18 \text{ m s}^{-1}$.

The dealloying of the $\text{Mg}_{65}\text{Ag}_{35}$ and $\text{Mg}_{62}\text{Ag}_{38}$ ribbons was firstly performed in the 1 wt. % HCl solution at room temperature. Then the dealloying was continuously carried out in the same solution at $90 \pm 5 \text{ }^\circ\text{C}$ in order to further leach out the residual Mg in the samples. In comparison, no bubbles emerged when the $\text{Mg}_{58}\text{Ag}_{42}$ and $\text{Mg}_{54}\text{Ag}_{46}$ ribbons were immersed in the 1 wt. % HCl solution at room temperature. Thus the dealloying of the $\text{Mg}_{58}\text{Ag}_{42}$ and $\text{Mg}_{54}\text{Ag}_{46}$ alloy was directly carried out at $90 \pm 5 \text{ }^\circ\text{C}$. It is astonishing that the $\text{Mg}_{50}\text{Ag}_{50}$ alloy cannot be dealloyed even in the 10 wt.% HCl solution at $90 \pm 5 \text{ }^\circ\text{C}$.

Microstructural characterization of the rapidly solidified Mg-Ag alloys and as-dealloyed samples was performed using X-ray diffraction (XRD, Hitachi Rigaku D/max-RB) with $\text{Cu K}\alpha$ radiation, scanning electron microscopy (SEM, LEO 1530VP), and X-ray photoelectron spectroscopy (XPS, ESCALAB 250) using monochromatic $\text{Al K}\alpha$ radiation.

Electrochemical measurements:

Electrochemical measurements were performed in a standard three-electrode cell using an LK 2500A Potentiostat. A 1 M NaCl aqueous solution was chosen as electrolyte to avoid the interference of chemical dealloying. The Mg-Ag ribbons were directly used as the working electrode. The counter electrode was a Pt plate, while the reference electrode was a saturated calomel electrode (SCE). Prior to electrochemical measurements, the electrolytes were deaerated by bubbling with N_2 for 10 min.

Figures and Tables:

Table E1 Lattice parameter (nm) of different Mg-Ag alloys calculated using the Bragg Equation with different crystal planes.

alloy composition	crystal plane						average
	(100)	(110)	(111)	(200)	(210)	(211)	
Mg ₆₅ Ag ₃₅	0.3341	0.3344	0.3345	0.3344	0.3341	0.3343	0.3343
Mg ₆₂ Ag ₃₈	0.3339	0.3338	0.3338	0.3340	0.3340	0.3339	0.3339
Mg ₅₈ Ag ₄₂	0.3321	0.3316	0.3323	0.3324	0.3324	0.3324	0.3322
Mg ₅₄ Ag ₄₆	0.3311	0.3314	0.3310	0.3313	0.3311	0.3312	0.3312
Mg ₅₀ Ag ₅₀	0.3302	0.3302	0.3303	0.3305	0.3305	0.3306	0.3304

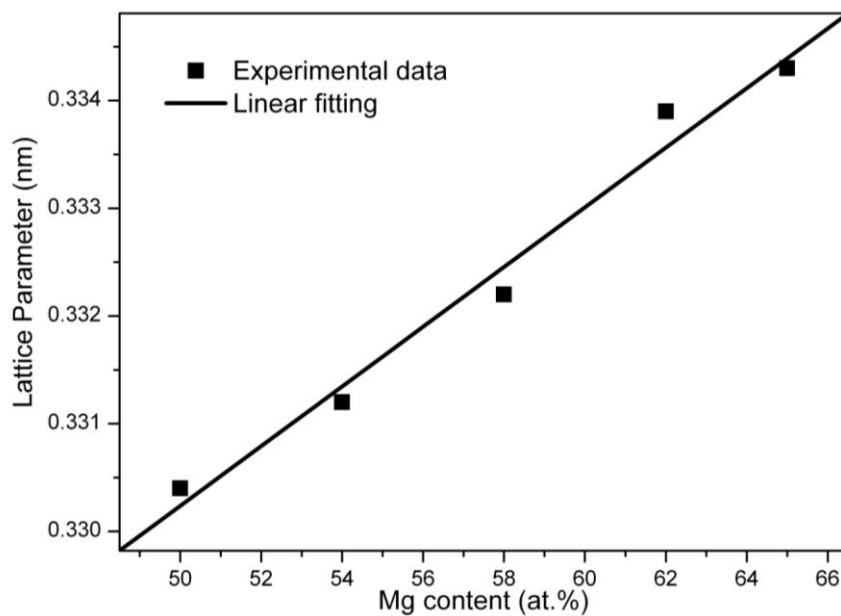


Fig. E1 Plot of lattice parameter vs. Mg content in the rapidly solidified Mg-Ag alloys.

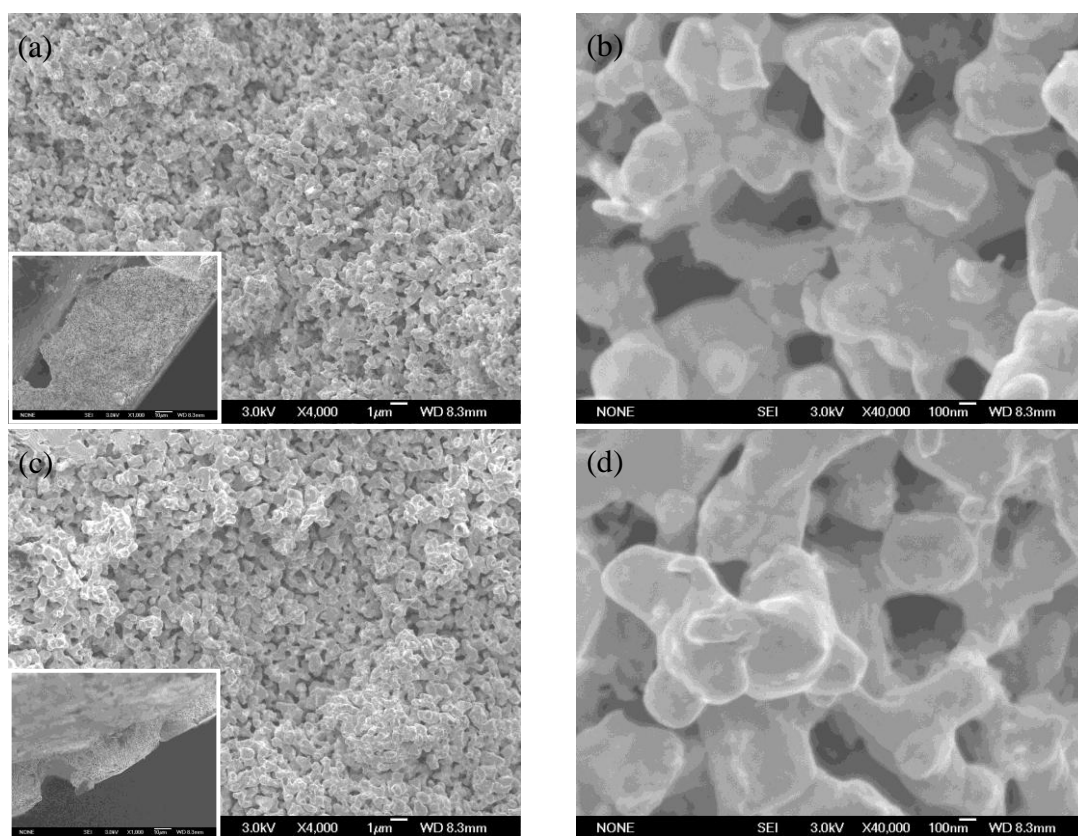


Fig. E2 The SEM micrographs showing the section-view microstructure of the np-Ag by dealloying (a, b) $Mg_{62}Ag_{38}$ and (c, d) $Mg_{54}Ag_{46}$ alloys in the 1 wt.% HCl solution.

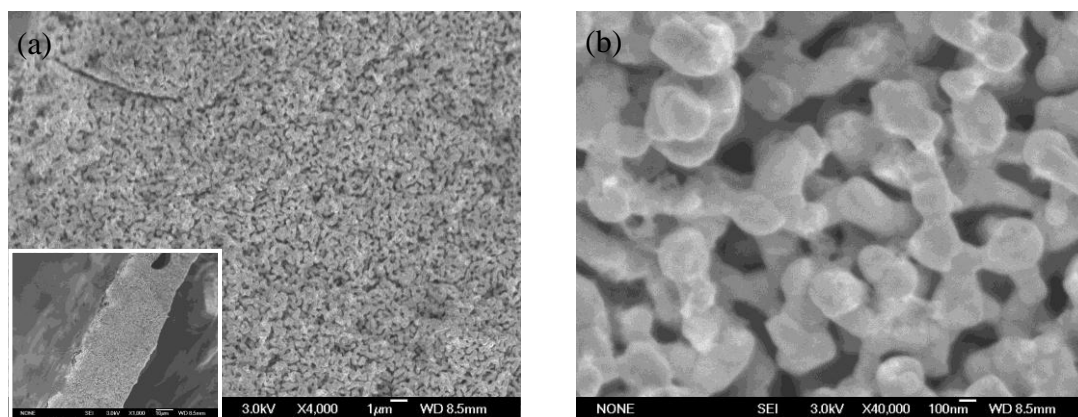


Fig. E3 The SEM micrographs showing the section-view microstructure of the np-Ag by dealloying the $Mg_{65}Ag_{35}$ alloy in the 1 wt.% HCl solution for 600 min.

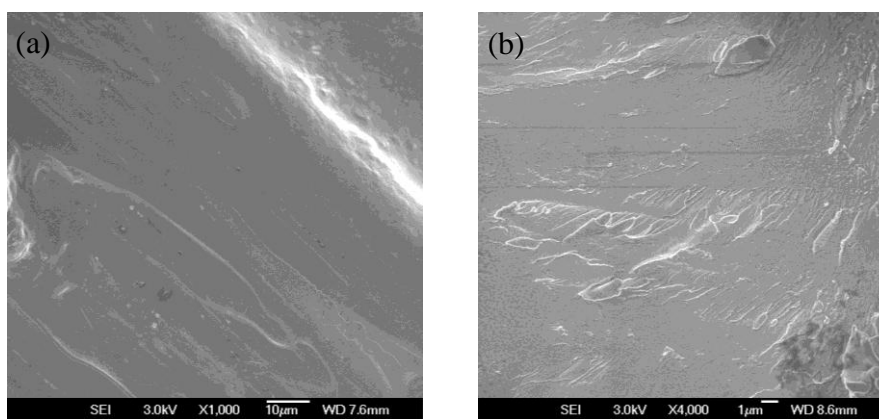


Fig. E4 The SEM micrographs showing the (a) surface-view and (b) section-view microstructure of the rapidly solidified Mg₆₅Ag₃₅ alloy.

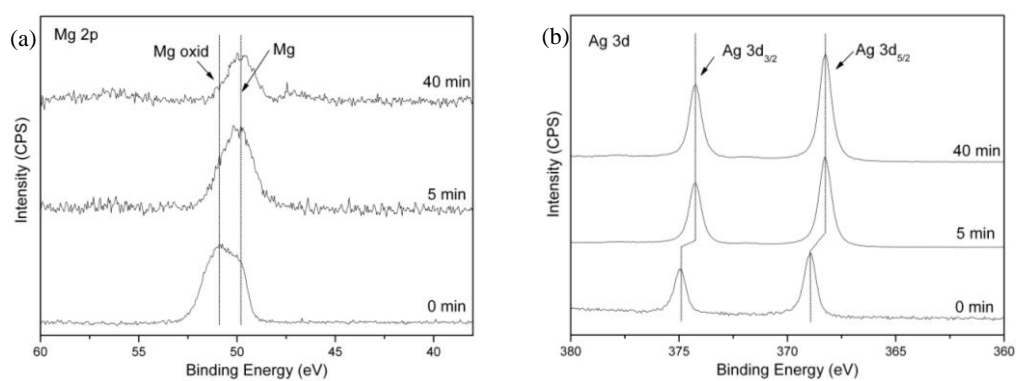


Fig. E5 (a) Mg 2p and (b) Ag 3d XPS spectra of the $Mg_{65}Ag_{35}$ alloy deallayed in the 1 wt.% HCl solution for 0, 5, and 40 min.

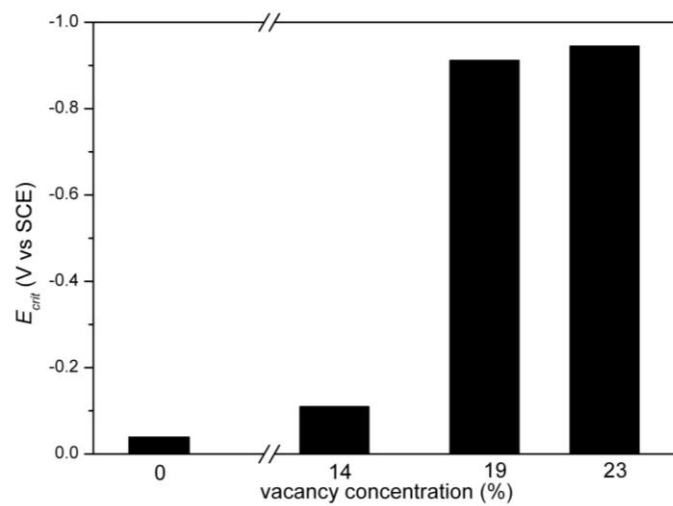


Fig. E6 The histogram of E_{crit} vs. vacancy concentration for the rapidly solidified Mg-Ag alloys.