

Supporting Information:

# CuO nanowires growthon Cu<sub>2</sub>O by in-situ thermal oxidation in air

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## SI-1. Derivation of the Eq.(6)

The relationship between the size of a embryo(or the number of molecules in a embryo) and the radius of the embryo is<sup>1</sup>

$$n = \frac{4\pi}{3} r^3 \bar{\rho} \quad (\text{S1})$$

Here  $\bar{\rho} = \rho N_A / M$ ,  $\rho$  is the density of CuO,  $N_A$  is Avagadro's number and  $M$  is molecular weight. Then the free energy of forming a nucleus can be expressed as

$$\Delta G = -n\Delta\mu + 4\pi r^2\gamma = -n\Delta\mu + 4\pi\gamma\left(\frac{3n}{4\pi\bar{\rho}}\right)^{2/3} \quad (\text{S2})$$

Second oder partial derivative of  $\Delta G$  at  $n=n^*$  is

$$\left. \frac{\partial^2 \Delta G}{\partial n^2} \right|_{n=n^*} = -\frac{8}{9}\pi\gamma\left(\frac{3}{4\pi\bar{\rho}n^2}\right)^{2/3} = -\frac{\bar{\rho}^2 \Delta\mu^4}{32\pi\gamma^3} \quad (\text{S3})$$

Substituting eq S2 and eq S3 into eq 3, one obtains

$$J = Z\beta^* N \exp\left(\frac{-\Delta G^* - E_a^{CuO}}{kT}\right)$$

$$= \left[ \frac{\overline{\rho}^2}{64\pi^2 k} \frac{\Delta\mu^4}{T\gamma^3} \right]^{1/2} \beta^* N \exp\left[\left(-\frac{16\pi\gamma^3}{3\overline{\rho}^2 \Delta\mu^2} - E_a^{CuO}\right)/kT\right] \quad (S4)$$

Under the condition of experimental temperature, the free energy change,  $\Delta\mu$ , can be expressed as<sup>2</sup>

$$\Delta\mu \approx \frac{L_v}{T_c}(T_c - T) \equiv a(T_c - T) \quad (S5)$$

Where  $L_v$  is the latent heat,  $T_c$  is the critical transformation temperature. Also in the temperature window of the growth of CuO nanowires, the interfacial tension,  $\gamma$ , is in proportion to  $(T_c - T)$  approximately<sup>3</sup>, in order to simplify the derivation process, a proportional constant  $b$  is setted

$$\gamma = b(T_c - T) \quad (S6)$$

As the density of nanowires is proportional to the nucleation rate, i.e.  $P_{NW} \propto J$ , we substitute eq S5 and eq S6 into eq S4

$$P_{NW} \propto \left(\frac{T_c - T}{T}\right)^{1/2} \exp\left[\left(-\frac{\eta b^3(T_c - T)}{3a^2} - E_a^{CuO}\right)/kT\right] \quad (S7)$$

After setting the fitting parameter  $\xi = \frac{\eta b^3}{3a^2}$ , the final result is obtained

$$P_{NW} \propto \left(\frac{T_c - T}{T}\right)^{1/2} \exp\left[\left(-\xi(T_c - T) - E_a^{CuO}\right)/kT\right] \quad (6)$$

## SI-2. The determination of $E_a^{CuO}$

The phase transition rate is determined by Gibbs free energy, nucleation rate, nucleation site number, diffusion field and so on. As a phenomenological treatment,

the relationship between phase transition rate V and annealing temperature is given by<sup>2</sup>

$$V = \frac{d\omega}{dt} = K \exp\left(-\frac{\Delta G^* + E_a^{CuO}}{RT}\right) \quad (S8)$$

Where  $\omega$  is the mass variation,  $t$  is the reaction time,  $K$  is a constant,  $R$  is the molar gas constant with a value of  $8.3145 \text{ Jmol}^{-1}\text{K}^{-1}$ . The eq S8 also can be rewritten as

$$\ln V = \ln\left(\frac{d\omega}{dt}\right) = \ln K - \frac{\Delta G^*}{RT} - \frac{E_a^{CuO}}{RT} \quad (S9)$$

At the experimental temperature, the critical free energy  $\Delta G^*$  approaches to zero. Then the slope of curve  $\ln(d\omega/dt)-(1/T)$  can be represented as

$$\tan \theta = \frac{\partial \ln(d\omega / dt)}{\partial (1/T)} = -\frac{\Delta G^*}{RT} - \frac{1}{RT} \times \frac{\partial \Delta G^*}{\partial (1/T)} - \frac{E_a^{CuO}}{RT} \approx -\frac{E_a^{CuO}}{RT} \quad (S10)$$

The data between the yellow dash in Figure S1a are chosen to fit the eq S10. So the the value of  $E_a^{CuO}$  is determined to be  $46.652 \text{ kJ/mol}$ .

Figure S1. (a) Thermo-gravimetric analysis(TGA) curves(red boxes) of Cu<sub>2</sub>O slices and the derivative curves of TGA(green line). (b) Arrhenius plot of ln(dω/dt) versus the inverse annealing temperature(1/T).

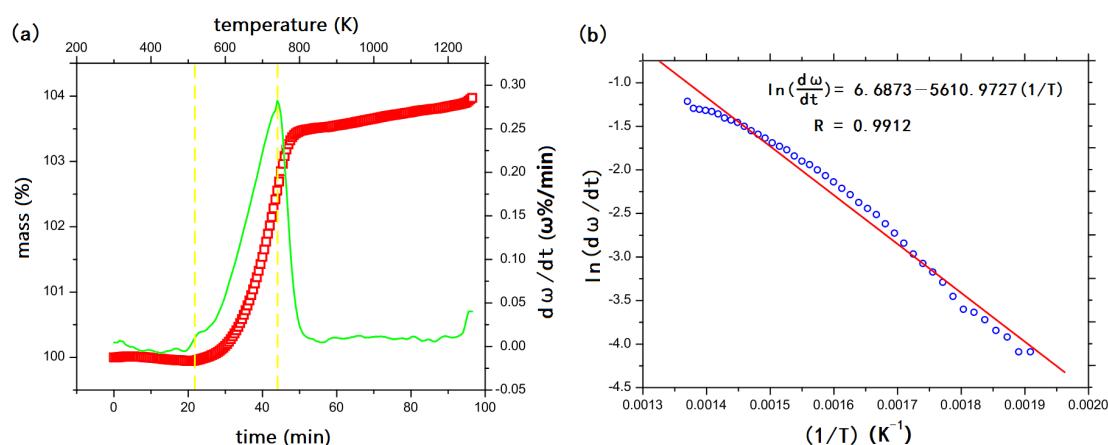


Figure S2. The inhomogeneous crystallines on the surface of CuO nanowire(in yellow dash boxes)

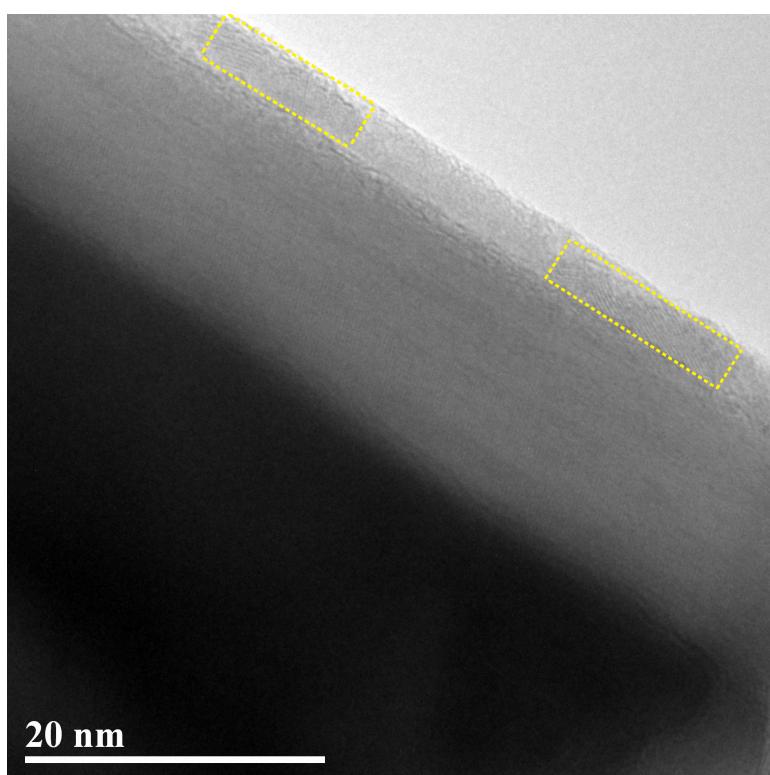
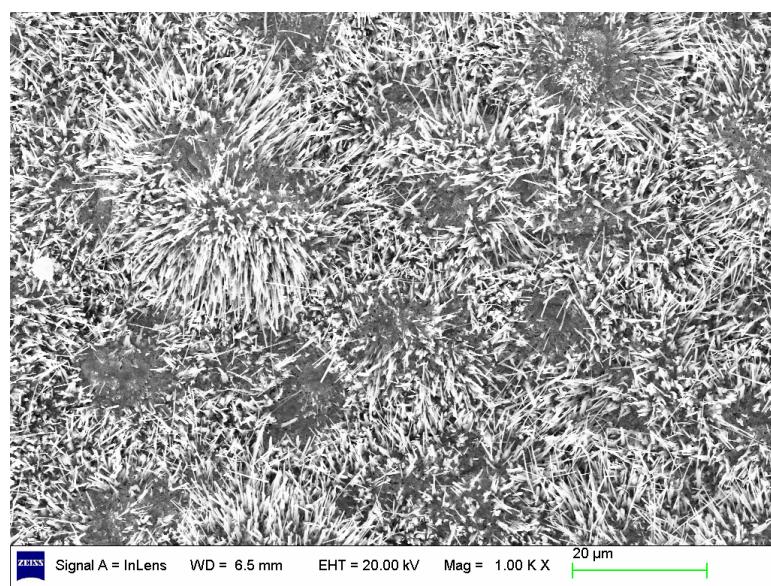


Figure S3. Morphology of the annealed compressed copper powder slices at 600 °C in air for 6h



## References

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