# **Supporting information**

Table	<b>S</b> 1	The	specific	surface	area	$(S_{BET})$	pore	volume	(Vp),	and	average	pore
diamet	er c	of the	samples	at differe	ent mi	lling sp	eed ar	nd with d	lifferen	$t C_3 N$	V4 loading	3.

Samples	ZnO		1%-	3%-C <sub>3</sub> N	<sub>4</sub> @ZnO		5%-	7%-
			C <sub>3</sub> N <sub>4</sub> @Zn				C <sub>3</sub> N <sub>4</sub> @ZnO	C <sub>3</sub> N <sub>4</sub> @Zn
			0					0
	Without	350	350 rpm	250	300	350	350 rpm	350 rpm
	milled	rpm		rpm	rpm	rpm		
S <sub>BET</sub>	10.3	13.0	16.0	12.3	12.1	12.5	8.5	5.4
$(m^{2}/g)$								
Pore	0.03	0.06	0.06	0.05	0.04	0.04	0.03	0.02
volume								
(cm <sup>3</sup> /g)								
Average	10.02	18.3	15.2	15.9	12.8	14.0	16.5	15.7
pore								
diameter								
(nm)								

# 1. Morphology and size of the sample



Fig.1 TEM image for (a) nano-ZnO (before milling), (b) g-  $C_3N_4@ZnO$  (milled) Figure 1 shows the TEM image for nano-ZnO and g- $C_3N_4@ZnO$  hybrid catalyst after ball milling process. The images show that the diameters of ZnO (Fig.1a) particles and g- $C_3N_4@ZnO$  (Fig.1b) materials are about 20 nm.

# 2. FT-IR spectra



Fig.2 FT-IR spectra of C<sub>3</sub>N<sub>4</sub>, ZnO and C<sub>3</sub>N<sub>4</sub>@ZnO (A resolution of 0.5 cm<sup>-1</sup>)

Fig.2 illustrates the FT-IR spectrum of  $C_3N_4$ , ZnO and g- $C_3N_4$ @ZnO, respectively. The FT-IR spectra of the synthesized g- $C_3N_4$  are similar to the results reported previously. The absorption band near 1635 cm<sup>-1</sup> is attributed to C–N stretching, while the three bands at 1240, 1320 and 1405 cm<sup>-1</sup> are originate from aromatic C–N stretching. The band near 810 cm<sup>-1</sup> is attributed to out-of plane bending modes of C–N heterocycles[1,2]. Abroad band near 3140 cm<sup>-1</sup> corresponds to the stretching modes of terminal NH<sub>2</sub> or NH groups at the defect sites of the aromatic ring [3]. The FT-IR spectra of g- $C_3N_4$ @ZnO photocatalysts are also similar to those of the main peaks of g- $C_3N_4$  wave-number. With the amount of doped g- $C_3N_4$  increasing, the absorbance band intensity of g- $C_3N_4$ @ZnO is gradually increase, which indicts the formation of a composite between ZnO and  $C_3N_4$ . The stronger valence bond interaction between  $C_3N_4$  and ZnO may be the key to enhance photocatalytic activity.

# 3. XRD patterns



Fig.3 The enlarged XRD patterns of C<sub>3</sub>N<sub>4</sub>, ZnO and C<sub>3</sub>N<sub>4</sub>@ZnO hybrid composite

Fig.3 shows the XRD patterns of different photocatalysts. Two pronounced peaks were found in  $g-C_3N_4$  at 27.4° and 13.1°, which can be indexed as (100) and (002) diffraction planes. The XRD peaks of the pure ZnO sample are in good agreement with the hexagonal wurtzite crystal phase of ZnO (JCPDS 65-3411). The diffraction peaks of  $g-C_3N_4$  cannot be found owing to the corresponding intensity of  $C_3N_4$  is

much lower than that of ZnO so that the changes of diffraction peaks of  $C_3N_4$  are not obvious in the patterns. The same results were obtained in previous reports [4,5]. This illustrates that g-C<sub>3</sub>N<sub>4</sub> is highly dispersed in the bulk phase of the catalyst. Furthermore, the XRD patterns reveal that the ZnO peak intensities decrease gradually with increasing g-C<sub>3</sub>N<sub>4</sub> content in the composite. No other new crystal phases are found in Fig.3, demonstrating that the C<sub>3</sub>N<sub>4</sub>@ZnO sample presents a twophase composition: C<sub>3</sub>N<sub>4</sub> and ZnO.

#### 4. Raman spectra



Fig.4 Raman spectra of g-C<sub>3</sub>N<sub>4</sub>@ZnO and pure ZnO. (Excitation wavelength: 514 nm)

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higher inset: Characterization of the pure ZnO ( raw material) and ZnO ( milled )
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Raman spectra were shown in Fig 4. Samples have four peaks (332.7 cm<sup>-1</sup>, 386.2 cm<sup>-1</sup>, 439.3 cm<sup>-1</sup> and 583.5 cm<sup>-1</sup>). Compared with pure ZnO, the peaks of g-C<sub>3</sub>N<sub>4</sub>@ZnO at 332.7 cm<sup>-1</sup> and 583.5 cm<sup>-1</sup> become strong, while the peak at of 386.2 cm<sup>-1</sup> is not observed and that of 439.3 cm<sup>-1</sup>does not change, which might be attributed to defects that formed in mechanical process. With the increasing amount of C<sub>3</sub>N<sub>4</sub> in the catalyst, the peaks become wider, which imply the hybridism of C<sub>3</sub>N<sub>4</sub> and ZnO.

## 5. Fluorescence emission spectra



Fig.5 Fluorescence emission spectra of different samples (Excitation wavelength: 325 nm)

Photocatalysts generate electrons and holes after being activated by light, and recombination of some electrons and holes can release energy in the form of fluorescence emission. Lower fluorescence emission intensity implies lower electronhole recombination rate [6]. Using an ultraviolet light with a 325 nm wavelength as the excitation source, the fluorescence emission spectra of ZnO hybridization with different contents of g-C<sub>3</sub>N<sub>4</sub> are shown in Fig.5. ZnO has three peaks: UV emission peaks (380 nm), for free carrier interband recombination, blue light emitting (450 nm), for the transition of oxygen vacancy of shallow donor level to valence band, defect emission peaks (470~600nm), caused by the transition of valence band. ZnO has the greatest relative intensity of emission spectra, which means that electrons and holes of ZnO are easy to recombine. The relative intensity of the  $g-C_3N_4$  (a)ZnO is lower than that of ZnO, it is clear that hybridization is helpful to inhibit the recombination of electrons and holes and improve the photocatalytic activity. With the increasing amount of  $C_3N_4$  content from 1wt % to 7 wt%, the peaks of 450 nm and 480 nm become strong, which is caused by the defects formed in mechanical process, indicating the high rate of combination of photo-induced electron and hole.

# References

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