Electronic Supplementary Information (ESI)

Enhanced Room Temperature Ionic Conductivity in the

LiBH₄·¹/₂NH₃-Al₂O₃ Composite

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Experimental Methods:

LiBH₄ (\geq 95%, Sigma-Aldrich), LiNH₂(95%, Sigma-Aldrich) and LiOH (\geq 98%, Sigma-Aldrich) were used directly without extra purification. γ -Al₂O₃ nanopowders (99.99%, Beijing Deke Daojin Science and Technology Co. Ltd) were dried at 350 °C for 24 h under vacuum to remove the adsorbed water and oxygen. To synthesize LiBH₄· $\frac{1}{2}$ NH₃-Al₂O₃ composite, 1 g mixture of LiBH₄, LiNH₂, LiOH and Al₂O₃ was mechanically milled in a 50 mL sealed stainless-steel milling jar at 300 rpm for 0.5 h with a ball-to-sample mass ratio of 20:1 using a planetary ball mill (PMQ0.4L). Subsequently, the mixture was annealed at 60°C for 20 h under Ar atmosphere. The molar ratio of LiBH₄, LiNH₂ and LiOH was fixed as 2:1:1. The addition amount of Al₂O₃ was set as 30, 45, 53, 60, 63, 67 and 75 wt% Al₂O₃, respectively. All sample

handlings were performed in an Ar-filled glovebox (MIKROUNA, China, $H_2O < 0.01$ ppm and $O_2 < 0.01$ ppm).

X-ray diffraction (XRD) measurements were carried out using Cu Ka (λ =1.5416 Å) radiation with the 20 range of 20–70° and a scan rate of 0.026°·s⁻¹ using EMPYREAN X-ray diffractometer. Investigated powder was sealed by 3M 7413D tape to prevent the air exposure. Differential scanning calorimetry (DSC) measurements were performed using a Mettler-Toledo TGA/DSC2 instrument (N₂ flow: 50 mL min⁻¹, ramp: 5 °C min⁻¹, temperature range: –40 to 80 °C) and the samples were filled in Al pan in Ar-filled glovebox. Fourier-transform infrared (FTIR) spectra were collected using Thermo Fisher Nicolet iS50 ATR spectrometer with the resolution of 4 cm⁻¹. The morphologies of the samples were observed using Titan G2 transmission electron microscopy (TEM) with a vacuum transfer chamber.

The X-ray photograph spectroscopy (XPS) spectra were recorded on Thermo Scientific K-Alpha+, XPS system. The samples transfer and handing process were protected under inert gas. The binding energies of B 1s, Li 1s, N 1s, O 1s and Al 2p calibrated with C 1s peak (BE = 284.6 eV) as a standard.

Ionic conductivity was measured by electrochemical impedance spectroscopy (EIS) using Mulit Autolab M204 electrochemical workstation. Au films were used as blocking electrodes and frequency varies from 1 Hz to 1 MHz. The powder samples were pressed into pellets in diameter of 0.6 mm with thickness of \sim 1 mm under a pressure of 0.3 GPa. The temperature range for the EIS measurement was set as 30 to 80 °C with a step of 10 °C and dwell time of 2 h. The ionic conductivity was obtained

using the following equation

$$\sigma = \frac{d}{RS} \tag{1}$$

where d is the thickness of the electrolyte, R is the resistance which is determined as the intersection of curve with the Z' axis the and S is the contact area between the electrode and electrolyte.

Cyclic voltammetry (CV) test at a scan rate of 0.5 mV s⁻¹ was performed at room temperature (RT) from –0.5 to 4.0 V with a Li| LiBH₄· $\frac{1}{2}$ NH₃-Al₂O₃ |SS (SS: stainless steel) cell using an electrochemical work station (DH7000, Donghua, China). The galvanostatic plating-stripping test was conducted using an Li|LiBH₄· $\frac{1}{2}$ NH₃-Al₂O₃|Li symmetric cell at the constant current density of ±0.1 mA cm⁻² at 30 °C.



Fig. S1 (a), (b) and (c): TEM images of the γ -Al₂O₃ nanopowders; (d) the SAED pattern.

 γ -Al₂O₃ nanopowders show the typical particle morphology with sharp edge, containing short rods and hexagons. SAED pattern in (d) displays two concentric rings assigned to the (004) and (044) crystal planes of γ -Al₂O₃, respectively.



Fig. S2 (a) ATR-FTIR spectra of the LiBH₄· $\frac{1}{2}$ NH₃-Al₂O₃ composites with different amounts of Al₂O₃, compared with the pristine LiBH₄· $\frac{1}{2}$ NH₃ and (b) magnified curves of 3500-3300 cm⁻¹ wavenumber on the part of N-H vibration.



Fig. S3 Portion of LiBH₄· $\frac{1}{2}$ NH₃ in amorphous state of pristine LiBH₄· $\frac{1}{2}$ NH₃ and LiBH₄· $\frac{1}{2}$ NH₃-Al₂O₃ composites calculated in the heating/cooling process.



Fig. S4 (a) B 1s XPS spectra of LiBH₄·½NH₃-Al₂O₃, (b) Li 1s, (c) N 1s, (d) O 1s and (e) Al 2p XPS

spectra of LiBH₄· $\frac{1}{2}$ NH₃-75 wt% Al₂O₃ compared with the pristine LiBH₄· $\frac{1}{2}$ NH₃ without Al₂O₃ addition.



Fig. S5 Nyquist plots of (a) $LiBH_4 \cdot \frac{1}{2}NH_3$ and (b) $LiBH_4 \cdot \frac{1}{2}NH_3-60$ wt%Al₂O₃ at room temperature.



Fig. S6 (a) Temperature-dependent Li-ion conductivities and (b) Li-ion conductivities at 30 °C of as a function of Al_2O_3 amount of the LiBH₄·½NH₃-Al₂O₃ composite in cooling process.



Fig. S7 Tests of a $LiBH_4\cdot {}^{1}\!{}^{2}\!NH_3$ -60 wt% Al_2O_3 pellet heated at 170 °C for 0.5 h in air.