

Supporting information for:

Optimum in the thermoelectric efficiency of

nanostructured Nb-doped TiO₂ ceramics: from

polarons to Nb-Nb dimers

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Determination of the amount of dimer.

When the niobium content in the material is lower than the limiting concentration before the appearance of the dimers ($x \leq c_{\text{Nb}}^{\text{L}}$), a linear relationship is observed between the lattice parameters of the crystalline structure and the concentrations of its constituent elements (Vegard's Law). The evolution of the lattice parameter c can be described according to equation 1.

$$c = xc_{\text{Nb}_2\text{O}_5} + (1 - x)c_{\text{TiO}_2} \quad (1)$$

In this case, x represents the concentration in Nb, $c_{\text{Nb}_2\text{O}_5}$ the lattice parameter c of Nb_2O_5 assuming that it crystallizes in the space group $P4_2/mnm$ (this parameter is determined by a simple linear refinement and is equal to 0.313 nm) and c_{TiO_2} the lattice parameter c of TiO_2 (0.2959 nm) in the space group $P4_2/mnm$ (rutile).

When the Nb concentration exceeds the c_{Nb}^{L} limit but does not exceed the Nb solubility limit in the rutile lattice (7.5 mol% of Nb), we observe a deviation from Vegard's Law. To take into account the influence of the Nb-Nb dimers, we describe the evolution of the parameter c by the following law (equation 2).

$$c = x[2\alpha c_{\text{Nb-Nb}} + (1 - 2\alpha)c_{\text{Nb}_2\text{O}_5}] + (1 - x)c_{\text{TiO}_2} \quad (2)$$

In this case, x represents the concentration in Nb, α the dimer fraction, $c_{\text{Nb}_2\text{O}_5}$ the lattice parameter c of Nb_2O_5 assuming that it crystallizes in the space group $P4_2/mnm$, c_{TiO_2} the lattice parameter c of TiO_2 (0.2959 nm) in the space group $P4_2/mnm$ (rutile) and $c_{\text{Nb-Nb}}$ the lattice parameter c of the material during the formation of dimers (0.280 nm^{S1}).

Copy of the pcr file that was used for the Rietveld refinement.

```
COMM AVC-B-076-03
! Current global Chi2 (Bragg contrib.) =      2.954
! Files => DAT-file: AVC-B-076-03.dat,  PCR-file: AVC-B-076-03
!Job Npr Nph Nba Nex Nsc Nor Dum Iwg Ilo Ias Res Ste Nre Cry Uni Cor Opt Aut
   0   7   1   0   1   0   0   0   0   0   0   1   0   0   0   0   0   0   1
!
! Resolution file for Pattern#   1
/Users/averchere/ownCloud/Othello/FullProf/AVC-B-076-03/LaB6_D8-t3_Soller2,5.irf
!Ipr Ppl Ioc Mat Pcr Ls1 Ls2 Ls3 NLI Prf Ins Rpa Sym Hkl Fou Sho Ana
   0   0   1   0   1   0   4   0   0  -3  10  -1   0   1   0   0   0   0
!
! Lambda1  Lambda2      Ratio      Bkpos      Wdt      Cthm      muR      AsyLim      Rpolarz      2nd-muR -
> Patt# 1
  1.540560  1.544390   0.50000    42.000    20.0000   1.0000   0.0000    60.00    0.0000   0.0000
!
!NCY  Eps  R_at  R_an  R_pr  R_gl      Thmin      Step      Thmax      PSD      Sent0
   60  0.10  1.00  1.00  1.00  1.00      4.1229    0.020475    80.0052    0.000    0.000
!
! Excluded regions (LowT  HighT) for Pattern#   1
      0.00      20.00
!
!
      12      !Number of refined parameters
!
! Zero      Code      SyCos      Code      SySin      Code      Lambda      Code MORE ->Patt# 1
  0.00000    0.0   0.01032    61.0   0.00000    0.0  0.000000    0.00   0
! Background coefficients/codes for Pattern#   1 (Polynomial of 6th degree)
      76.366      -65.933      89.968      -57.978      0.000      0.000
      21.00      51.00      81.00      71.00      0.00      0.00
!-----
! Data for PHASE number:   1 ==> Current R_Bragg for Pattern#   1:      5.50
!-----
Rutile
!
!Nat Dis Ang Pr1 Pr2 Pr3 Jbt Irf Isy Str Furth      ATZ      Nvk Npr More
   2   0   0  0.0  0.0  1.0   0   0   0   0   0      255.676   0   7   0
!
!
P 42/m n m      <--Space group symbol
!Atom  Typ      X      Y      Z      Biso      Occ      In Fin N_t Spc /Codes
Ti     TI      0.00000  0.00000  0.00000  0.31359  0.15549   0   0   0   0
      0.00      0.00      0.00      0.00      111.00
O      O      0.30524  0.30524  0.00000  0.64888  0.33772   0   0   0   0
      91.00      91.00      0.00      0.00      121.00
!-----> Profile Parameters for Pattern #   1
! Scale      Shapel      Bov      Str1      Str2      Str3      Strain-Model
  0.71755E-02  0.00000    0.00000    0.00000    0.00000    0.00000      0
      11.00000    0.000    0.000    0.000    0.000    0.000
!      U      V      W      X      Y      GauSiz      LorSiz Size-Model
      0.004503  0.000000    0.000000    0.000000    0.105902    0.001131    0.000000    0
      0.000      0.000      0.000      0.000      101.000      0.000      0.000
!      a      b      c      alpha      beta      gamma      #Cell Info
      4.594127  4.594127    2.961263    90.000000    90.000000    90.000000
      31.00000    31.00000    41.00000    0.00000    0.00000    0.00000
! Pref1 Pref2 Asy1 Asy2 Asy3 Asy4 S_L D_L
  0.00000  0.00000  0.00000  0.00000  0.00000  0.00000  0.04286  0.00000
      0.00      0.00      0.00      0.00      0.00      0.00      0.00
! 2Th1/TOF1 2Th2/TOF2 Pattern to plot
      20.000      80.005      1
```

Copy of the irf file that was used for the Rietveld refinement.

```
! LaB6_D8-t3_soller_2.5
!      U      V      W      X      Y      Z
0.001971 -0.001543 0.000864 0.007410 0.022966 0.0
0.001971 -0.001543 0.000864 0.007410 0.022966 0.0
```

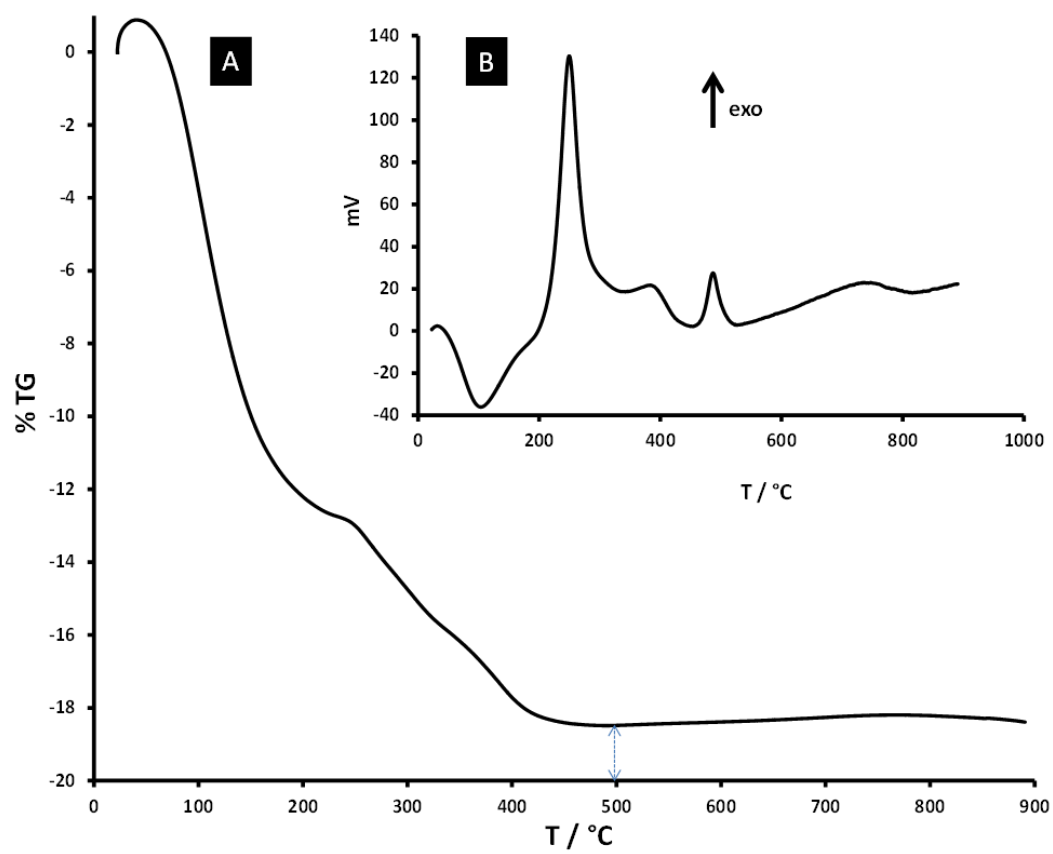


Fig. S1 (A) TGA and (B) TDA curves of the $\text{TiO}_2\text{:Nb}$ obtained from $\text{Ti}(\text{OEt})_4 + x [\text{Nb}(\text{OEt})_5]_2$.

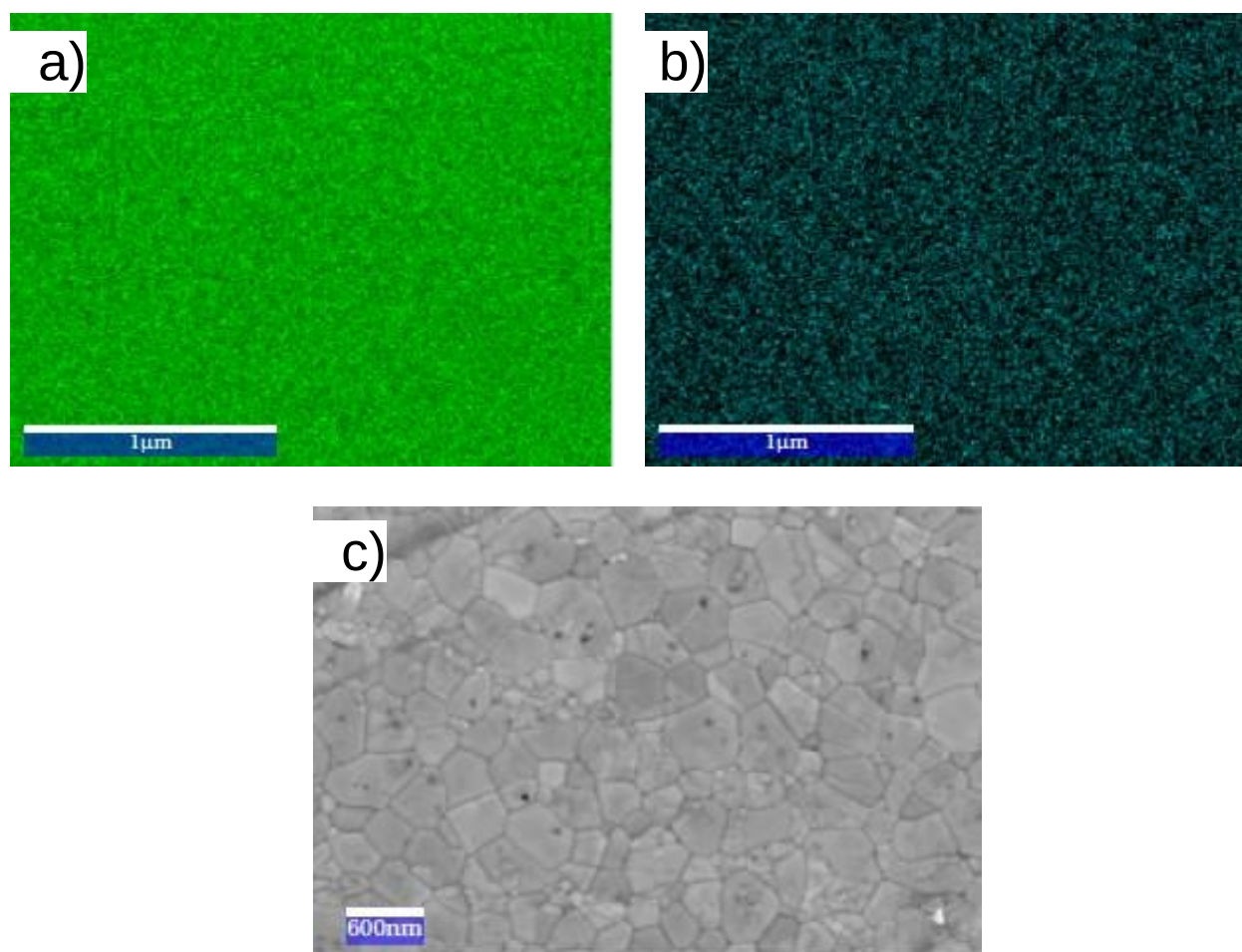


Fig. S2 SEM images and elemental mapping of a TiO_2 pellet doped with 2 mol% of niobium after sintering at 1233 K. The mapping of elements Ti-K (panel **a**) and Nb-L (panel **b**) shows that niobium is uniformly distributed in the TiO_2 lattice and creates a solid solution. SEM image of the mapping area of the sintered pellet at 1233 K (panel **c**).

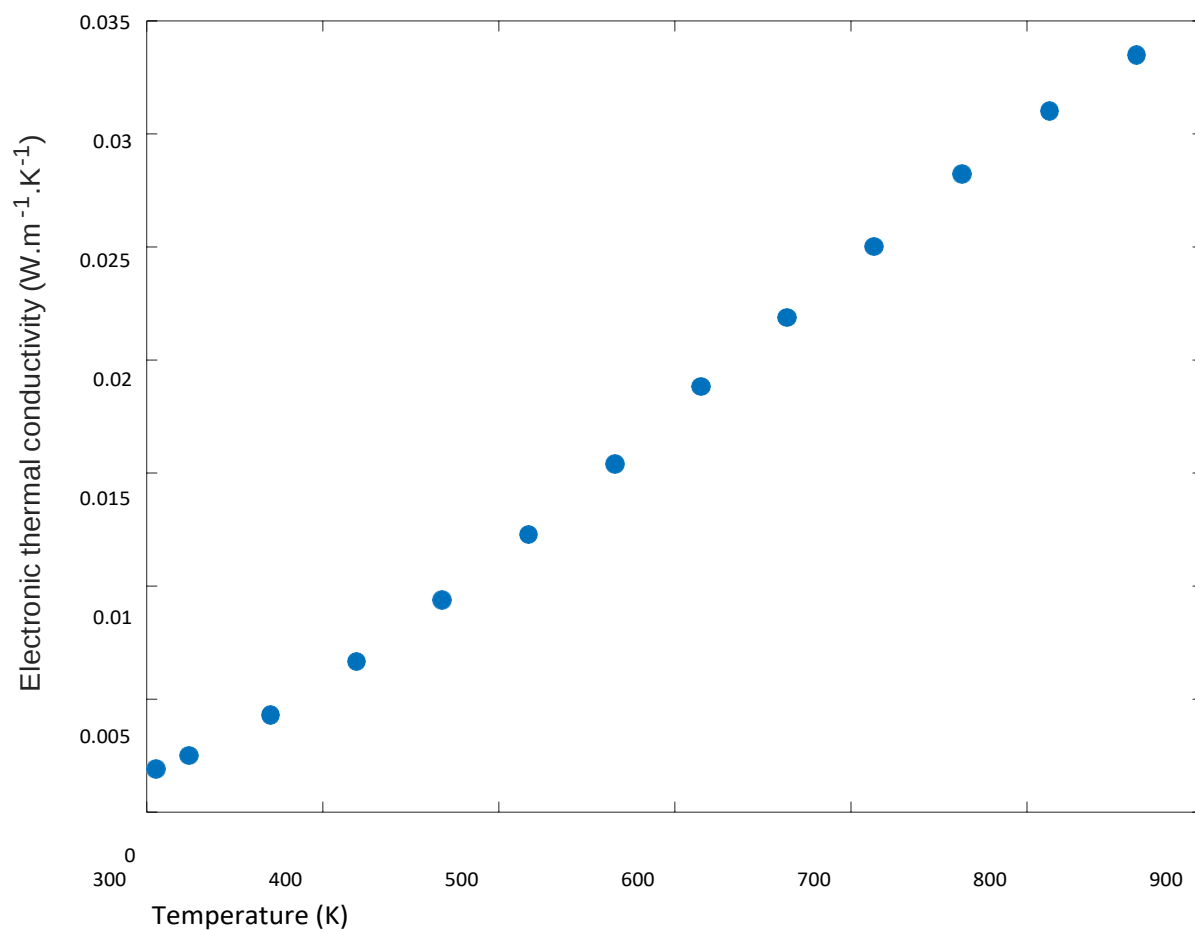


Fig. S3 Electronic thermal conductivities of the $\text{Ti}_{0.972}\text{Nb}_{0.028}\text{O}_2$ pellet sintered at 1123 K according to the Wiedemann-Franz law. The electronic thermal conductivity of four materials constitutes about 2% of total thermal conductivities (• : $\text{Ti}_{0.972}\text{Nb}_{0.028}\text{O}_2$).

References

(S1) K. Sakata, *J. Phys. Soc. Jpn.*, 1969, **26**, 1067–1067.