

Electronic Supplementary Information

Discovery of high-performance thermoelectric copper chalcogenide using modified diffusion-couple high-throughput synthesis and automated histogram analysis technique

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Lattice dynamics calculations

Lattice dynamics were obtained by Density-Functional Theory (DFT) calculations with Vienna ab initio simulation package (VASP)^{1,2}, using the frozen phonon method which is implemented in the Phonopy package. The projector-augmented wave (PAW) method as implemented in the Perdew–Burke–Ernzerhof (PBE)-type generalized gradient approximation (GGA) was used as the exchange–correlation functional³. Based on the crystallographic information obtained by the X-ray refinement (model 2), we firstly built an order $2 \times 2 \times 2$ supercell of Cu_3SnS_4 . Then, two Cu atoms in the

supercell are replaced by Sn atoms to form the chemical composition ($\text{Cu}_{22}\text{Sn}_{10}\text{S}_{32}$) that is close to $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ based on special quasirandom structures (SQS's) calculations.⁴ The plane-wave energy cutoff was set at 450 eV and a $3 \times 3 \times 3$ Monkhorst-Pack k mesh was used for crystal structure optimization to realize the convergence of Hellmann–Feynman forces. It is adopted that convergence criteria of 5×10^{-5} eV/Å for relaxation process of unit cell and 5×10^{-7} eV/Å for static calculation of displaced supercell at gamma point in accurate precision. The spin-orbit coupling (SOC) effect has not been considered.

Table S1 Crystallographic information of $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ prepared by melting-annealing-ball milling method (model 1). Here the composition is normalized to 16 atoms in the unit cell.

Composition	$\text{Cu}_{6.28}\text{Sn}_{1.72}\text{S}_{8.00}$
Formula weight	859.77
Space group	$I2m$ (No. 121)
a /Å	5.4163(3)
c /Å	10.833(1)
Unit cell volume /Å ³	317.79(7)
$F(000)$ /e	396.1
Z	1
μ/cm^{-1}	502.62
Calculated density (g cm^{-3})	4.492(1)
Radiation, wavelength(Å)	$\text{CuK}\alpha$, 1.54056
T (K)	295
Data range 2θ (°)	10 - 110
No. of reflections	152
No. of refined structure parameters	7
Profile function	Pseudo-Voigt

Refinement mode	Full profile
R_i	0.0393
R_p	0.0720
R_{wp}	0.0489
Goodness of fit	1.850

Table S2 Crystallographic information of $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ prepared by melting-annealing-ball milling method (model 3). Here the composition is normalized to 16 atoms in the unit cell.

Composition	$\text{Cu}_{5.56(2)}\text{Sn}_{2.33(4)}\text{S}_{8.10(2)}$
Formula weight	890.3
Space group	$I2m$ (No. 121)
$a / \text{\AA}$	5.4162(2)
$c / \text{\AA}$	10.8335(7)
Unit cell volume / \AA^3	317.80(5)
$F(000) / e$	407.5
Z	1
μ / cm^{-1}	589.3
Calculated density (g cm^{-3})	4.65
Radiation, wavelength(\AA)	Cu $K\alpha$, 1.54056
T (K)	295
Data range 2θ ($^\circ$)	10 - 110
No. of reflections	152
No. of refined structure parameters	7
Profile function	Pseudo-Voigt
Refinement mode	Full profile
R_i	0.032
R_p	0.050
R_{wp}	0.067
Goodness of fit	1.76

Table S3 Selected Bond Lengths (Å) and Angles (°) for Cu₇Sn₃S₁₀ in the model 2.

Cu1-S1	2.319(7)		
Cu2-S1	2.334(4)		
Cu3-S1	2.334(4)		
S1-Cu1-a S1 7)	111.86(2)	a S1-Cu2-b S1 7) [0010]	109.58(2)
S1-Cu1-b S1 7) [0000]	104.78(2)	a S1-Cu2-c S1	109.25(2)
S1-Cu1-c S1 7) [0000]	111.86(2)	b S1-Cu2-c S1	109.58(2)
a S1-Cu1-b S1 7) [0000]	111.86(2)	S1-Cu3-a S1 6)	111.99(2)
a S1-Cu1-c S1	104.78(2)	S1-Cu3-a S1 6) [0010]	108.23(2)
b S1-Cu1-c S1	111.86(2)	S1-Cu3-c S1 6) [0000]	108.23(2)
S1-Cu2-a S1 7)	109.58(2)	a S1-Cu3-b S1 6) [0-100]	108.23(2)
S1-Cu2-b S1 7) [0100]	109.25(2)	a S1-Cu3-c S1	108.23(2)
S1-Cu2-c S1 7) [0110]	109.58(2)	b S1-Cu3-c S1	111.99(2)

Symmetry codes: (1) x, y, z; (2) -y, x, -z; (3) -x, -y, z; (4) y, -x, -z; (5) -x, y, -z; (6) y, x, z; (7) x, -y, -z; (8) -y, -x, z; (9) 1/2+x, 1/2+y, 1/2+z; (10) 1/2-y, 1/2+x, 1/2-z; (11) 1/2-x, 1/2-y, 1/2+z; (12) 1/2+y, 1/2-x, 1/2-z; (13) 1/2-x, 1/2+y, 1/2-z; (14) 1/2+y, 1/2+x, 1/2+z; (15) 1/2+x, 1/2-y, 1/2-z; (16) 1/2-y, 1/2-x, 1/2+z.

Table S4 Parameters used to fit the heat capacity (C_p) data of Cu₇Sn₃S₁₀ by using a Debye and two Einstein modes. The Debye temperature (Θ_D) is derived by the equation

$$\Theta_D = (12\pi^4 N k_B / 5\beta)^{1/3}.$$

Fitting parameters	One Debye + one Einstein
ϕ (10^{-4} J mol ⁻¹ K ⁻²)	6.90
β (10^{-5} J mol ⁻¹ K ⁻⁴)	9.32
A (J mol ⁻¹ K ⁻¹)	2.94
Θ_E (K)	83.13
Θ_D (K)	275
R^2	0.99997
χ^2	0.000000126

Table S5 Parameters used to fit the lattice thermal conductivity of $\text{Cu}_7\text{Sn}_3\text{S}_{10}$.

Fitting parameters	$\text{Cu}_7\text{Sn}_3\text{S}_{10}$
L (μm)	0.53
A (10^{-41} s^3)	0.603
B ($10^{-18} \text{ s K}^{-1}$)	11.265
C (10^{36} s^{-3})	4.541
ω/ν (THz)	10.9/1.73
R^2	0.99828
χ^2	0.02738

* ω is angular frequency and ν is frequency, and their units are the same THz. (i.e. $\omega=2\pi\nu$)

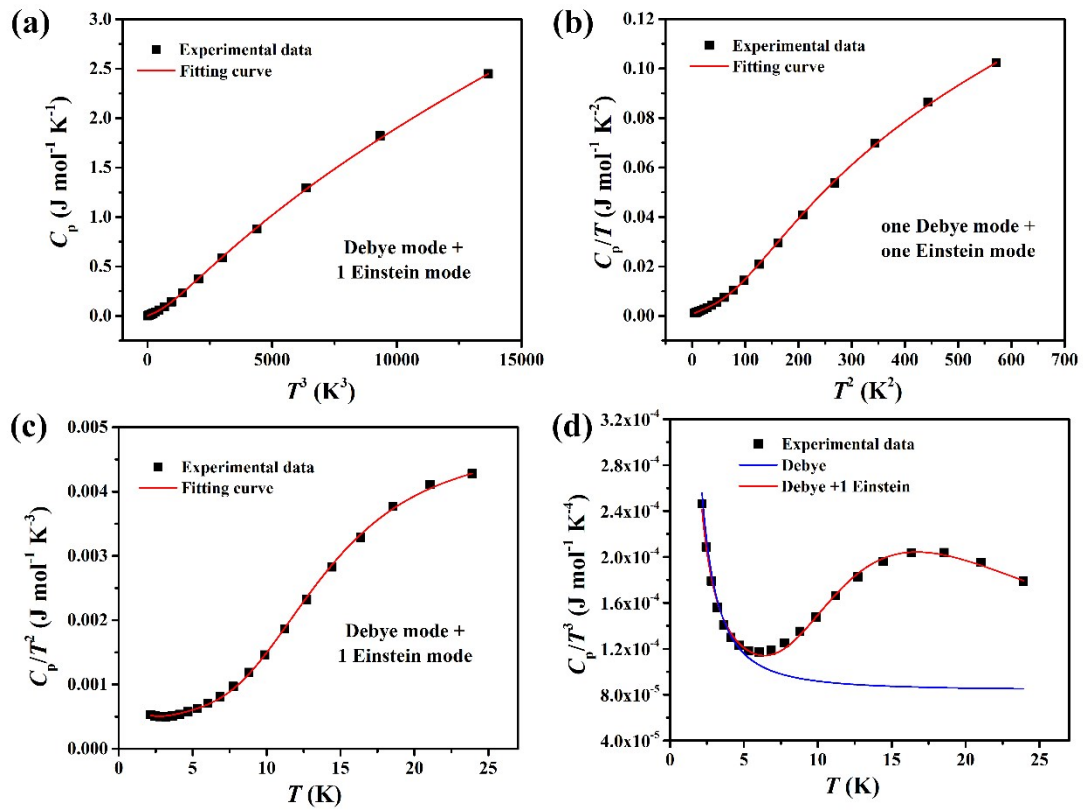


Fig. S1 Heat capacity fitting for $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ from 2 to 25 K. (a-c) C_p versus T^3 , C_p/T versus T^2 , and C_p/T^2 versus T for $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ by using one Debye mode + one Einstein mode. (d) C_p/T^3 versus T for $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ from 2 K to 25 K fitted by Debye mode and one Debye mode + one Einstein mode, respectively.

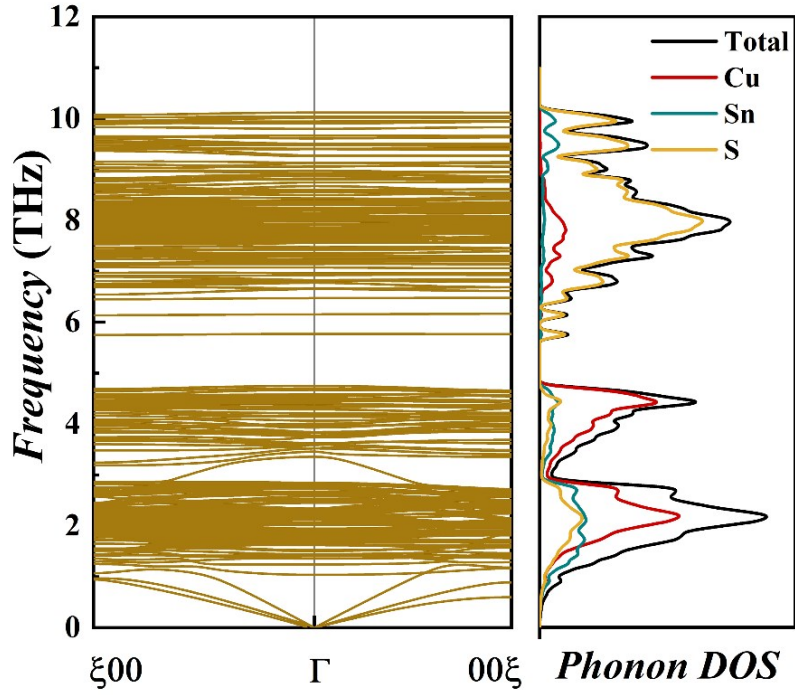


Fig. S2 Phonon dispersions and projected phonon density of states for $\text{Cu}_7\text{Sn}_3\text{S}_{10}$.

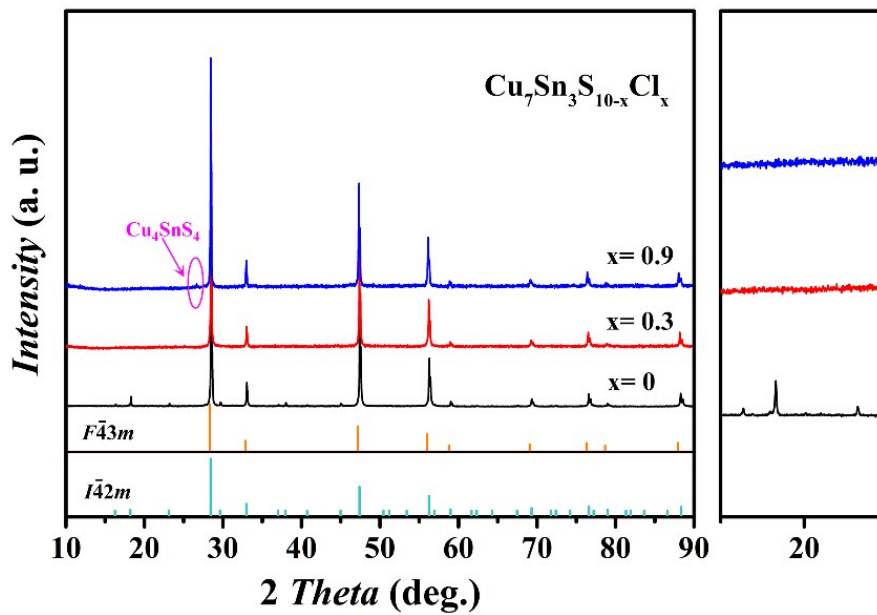


Fig. S3 Powder X-ray diffraction patterns for $\text{Cu}_7\text{Sn}_3\text{S}_{10-x}\text{Cl}_x$ ($x=0, 0.3, \text{ and } 0.9$) samples. The appearance of diffraction peaks belonging Cu_4SnS_4 indicates that the Cl-doping content in $\text{Cu}_7\text{Sn}_3\text{S}_{10}$ is less than $x = 0.9$.

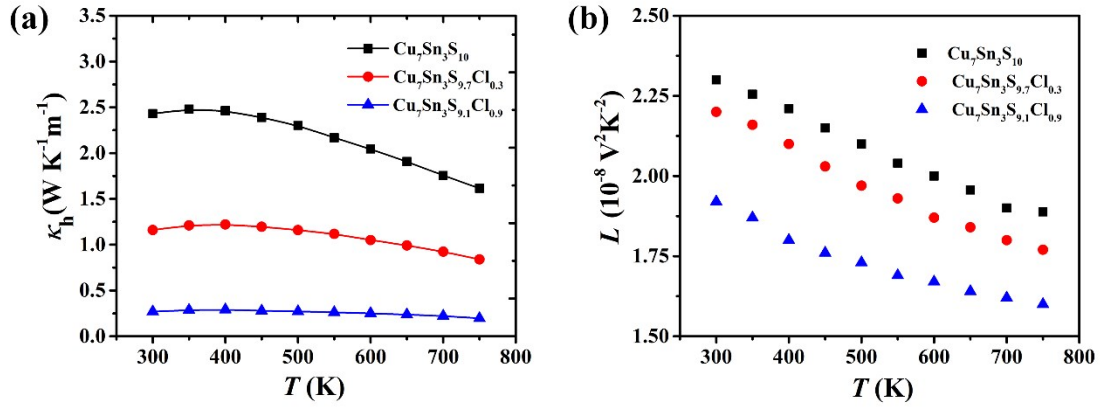


Fig. S4 Carrier thermal conductivity (κ_h) and Lorenz number (L) derived by the single parabolic band model for $\text{Cu}_7\text{Sn}_3\text{S}_{10}$, $\text{Cu}_7\text{Sn}_3\text{S}_{9.7}\text{Cl}_{0.3}$, and $\text{Cu}_7\text{Sn}_3\text{S}_{9.1}\text{Cl}_{0.9}$ samples from 300 K to 750 K.

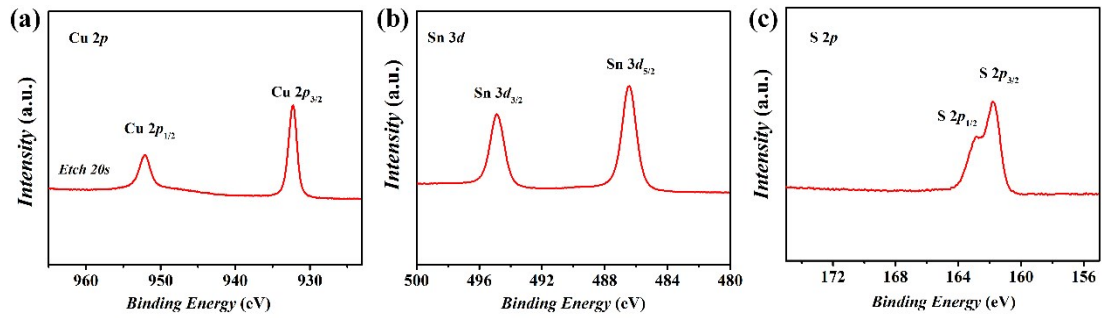


Fig. S5 XPS spectra of $\text{Cu}_7\text{Sn}_3\text{S}_{10}$: (a) Cu 2p, (b) Sn 3d, and (c) S 2p.

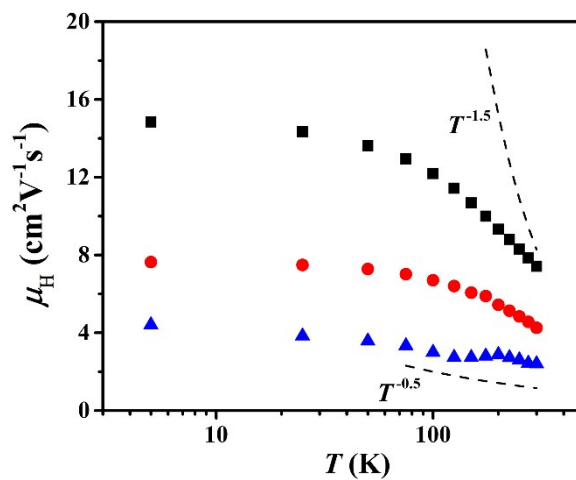


Fig. S6 Temperature dependence of Hall carrier mobility (μ_H) for $\text{Cu}_7\text{Sn}_3\text{S}_{10-x}\text{Cl}_x$ ($x=0, 0.3, \text{ and } 0.9$) samples from 5 K to 300 K. The dashed lines represent the $\mu_H \propto T^{-1.5}$ and $\mu_H \propto T^{-0.5}$ relationships, respectively.

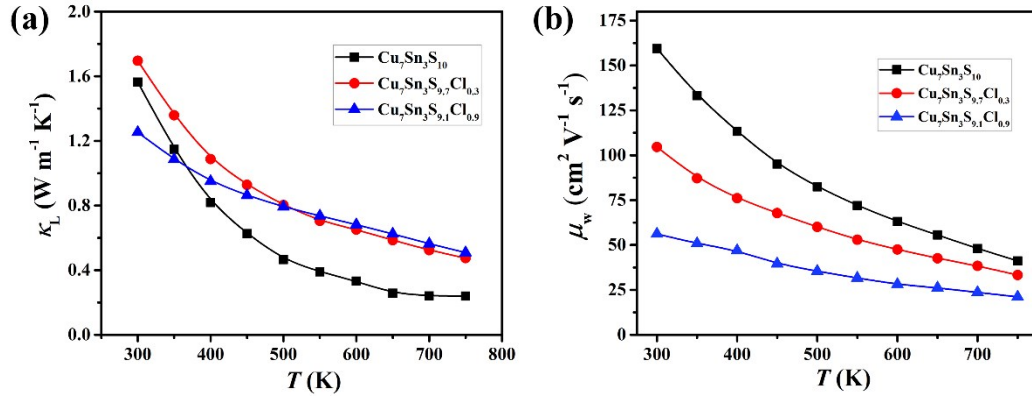


Fig. S7 (a) Lattice thermal conductivity κ_L as a function of temperature for Cu₇Sn₃S_{10-x}Cl_x. It decreases with temperature as expected from phonon-phonon scattering and is not strongly influenced by the Cl content. (b) The weighted mobility μ_w as a function of temperature for Cu₇Sn₃S_{10-x}Cl_x. It decreases with temperature as expected from phonon scattering and decreases with Cl content.

References

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