

Supporting information for “Hierarchical anti-reflective laser-induced periodic surface structures on amorphous Si films via ultrafast laser nanotexturing.”

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1. LIPSS periods predicted by Sipe and van Driel model.

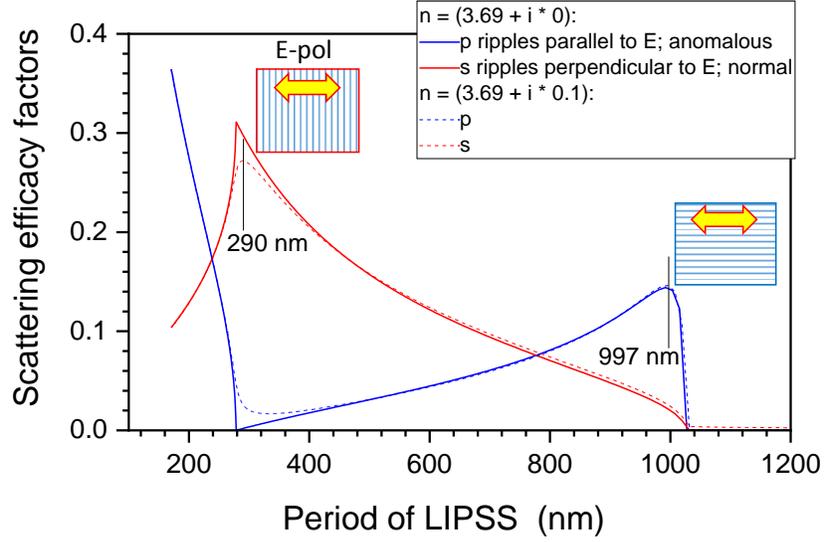


Figure 1: Normal and anomalous LIPSS periods for Si of refractive index $n = 3.69$ (at $\lambda = 1026$ nm) and with absorption $n = 3.69 + i \times 0.1$ calculated for s- and p-polarized irradiation according to Sipe and van Driel model. Explicit formulae is given in [H. Iwase, et.al. Opt. Express 17,4388 (2009)]. The scattering efficacy factor calculated by the model correlates with surface LIPSS patterns. For the normal incidence s- and p-pol. are same and the two cases represents two possibilities for ripples (LIPSS) to be aligned to the polarisation of E-field (anomalous ripples) and perpendicular (normal) as schematically shown by thumbnail insets. The period of normal ripples is expected as $\Lambda_n \approx \lambda/n$ and the anomalous $\Lambda_a \approx \lambda$ at the peak efficacy factors, respectively.

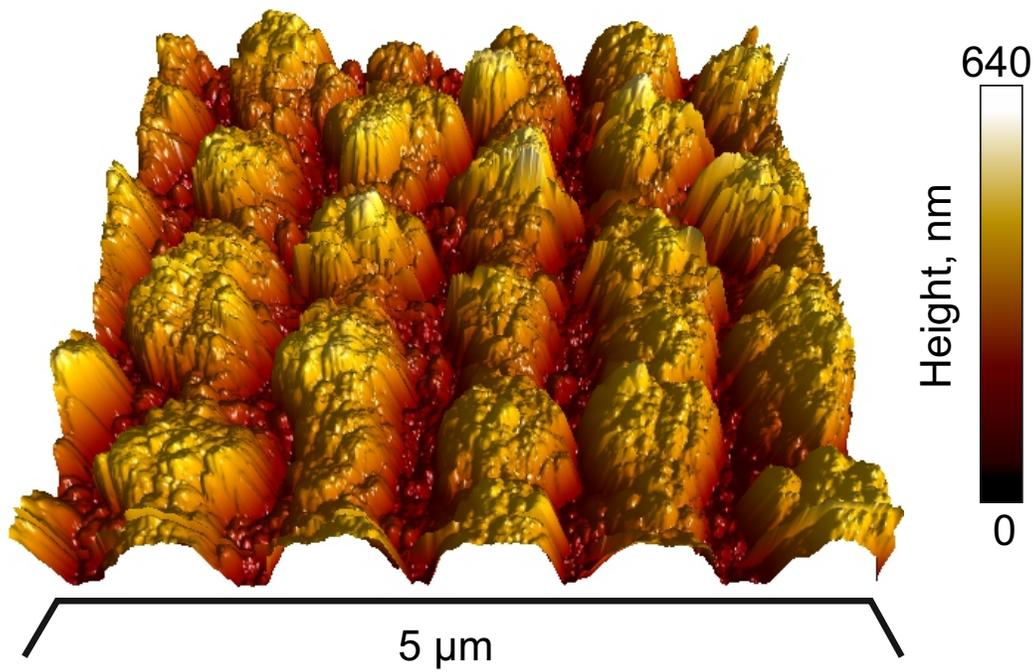


Figure 2: AFM image of LIPSS morphology ($E=1.5 \mu\text{J}$, $V=3 \mu\text{m/s}$) showing the maximal height of conical-shaped structures around 640 nm.

2. AFM analysis of LIPSS.

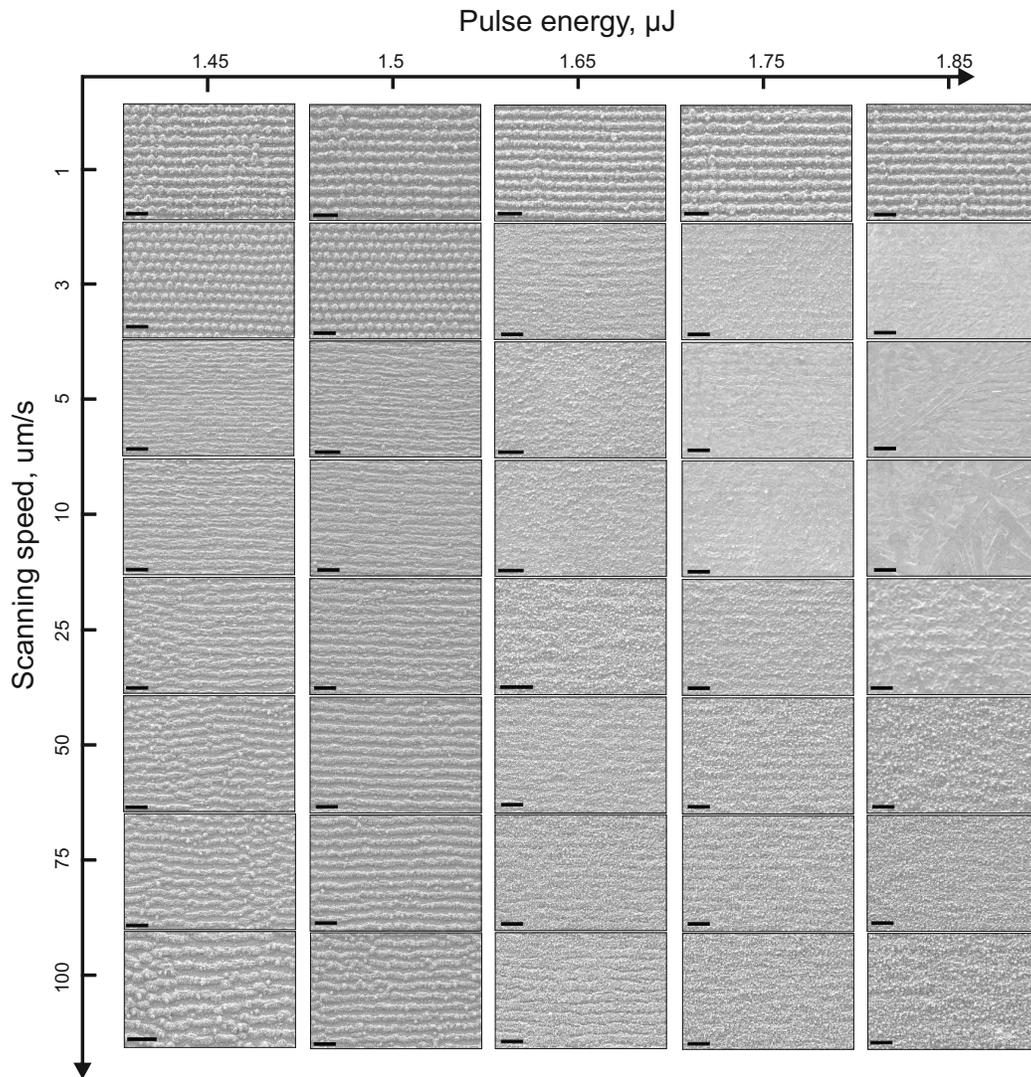


Figure 3: Diversity of the LIPSS morphologies is illustrated by series of top-view SEM images of LIPSS produced on 360-nm thick glass-supported a-Si film at variable pulse energies (1.45 - 1.85 μJ) and scanning speeds (1 - 100 $\mu\text{m/s}$).

3 **3. Variation of LIPSS morphology versus laser processing param-**
 4 **eters.**

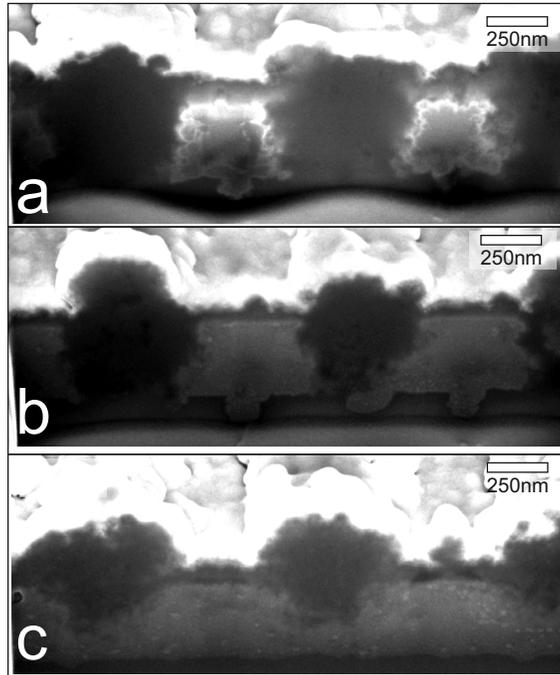


Figure 4: (a-c) True-color SEM images of FIB cross-section cuts of LIPSS fabricated on the a-Si film surface at various scanning speeds V of $1 \mu\text{m/s}$ (a), $3 \mu\text{m/s}$ (b), and $50 \mu\text{m/s}$ (c).

5 4. True-color SEM contrast of FIB cross-section cut images.

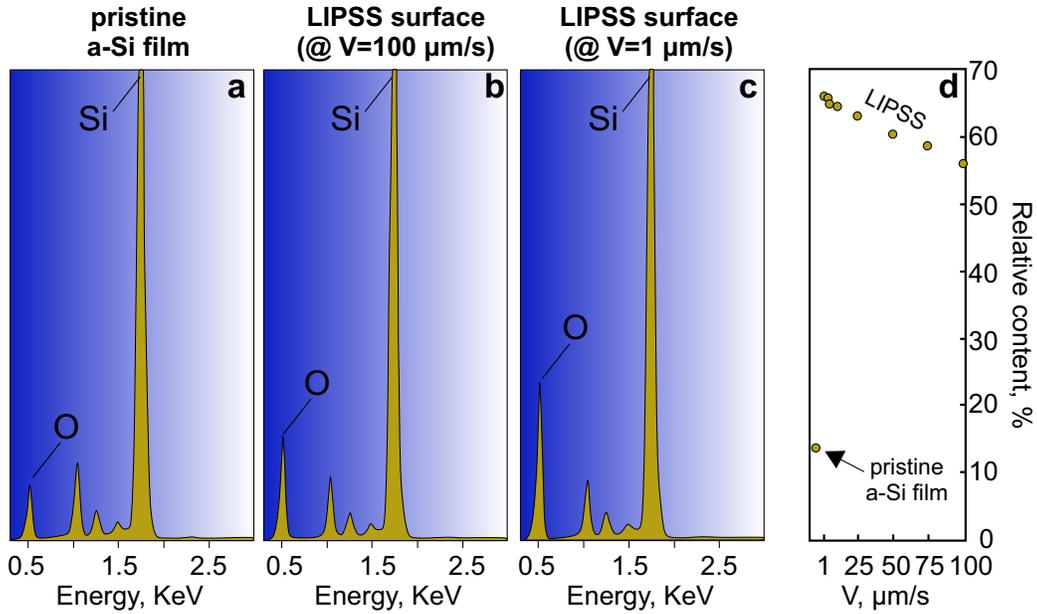


Figure 5: (a-c) Comparative EDX spectra of pristine a-Si film (a), LIPSSs produced at $V=100$ (b) and $1 \mu\text{m/s}$ (c) at $E=1.5 \mu\text{J}$. (d) Relative content of oxygen calculated from EDX spectra versus the scanning speed V for LIPSSs at $E=1.5 \mu\text{J}$. Relative content of oxygen obtained from EDX spectra of a-Si film ($\approx 14\%$) is provided as reference. Each dot is the average of at least 30 similar spectra.

6 5. EDX analysis

7 Here, to confirm the increasing amount of oxygen in laser-processed areas
 8 we analyzed their chemical composition with energy-dispersive X-ray spec-
 9 troscopy modality (Oxford Instruments, X-max) of the scanning electron
 10 microscope (Carl Zeiss, Ultra55+). For all measurements, the acceleration
 11 voltage and signal acquisition area was fixed at 20 kV and $50 \mu\text{m}^2$, respec-
 12 tively. Relative content of oxygen measured in the pristine a-Si film (≈ 14
 13 %) was used as reference. Representative EDX spectra of pristine a-Si film
 14 as well as LIPSS produced at printing speed of $V=1$ and $100 \mu\text{m/s}$ are pro-
 15 vided in Fig. S3(a-c). Fig. S3(d) summarizes the relative content of oxygen
 16 calculated from LIPSS produced at various scanning speed V .

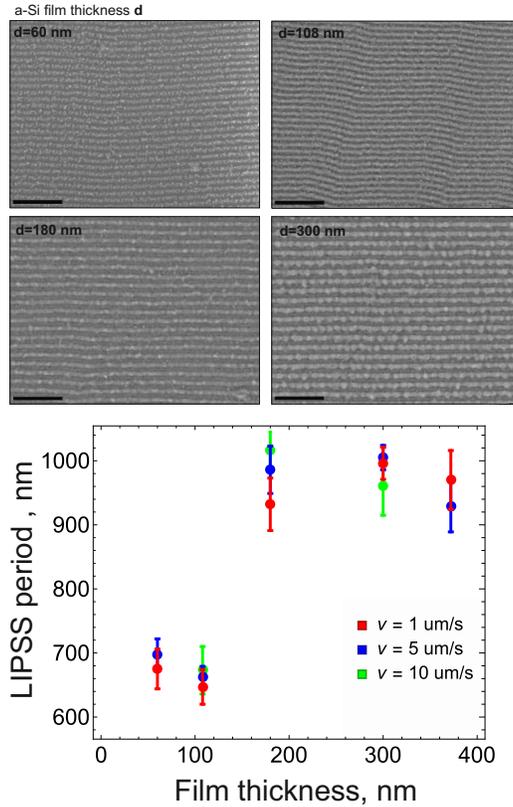


Figure 6: Top-view SEM images of near-wavelength LIPSS formed on a-Si film at fixed pulse energy of $1.45 \mu\text{J}$ and scanning speed of $3 \mu\text{m/s}$ and different a-Si film thickness d : 60 nm, 108 nm, 180 nm and 300 nm. (Bottom) Dependence of LIPSS period on Si film thickness measured for surface morphologies produced at scanning speeds $V = 1, 5$ and $10 \mu\text{m/s}$.)

17 6. LIPSS period versus a-Si film thickness

18 To illustrate the step-like decrease of the near-wavelength period with a-Si
 19 film thickness, we carried out series of experiments. The a-Si films of variable
 20 thickness d ranging from 60 to 360 nm were deposited using magnetron sput-
 21 tering under identical conditions at constant evaporation speed. LIPSS were
 22 fabricated under identical conditions at fixed pulse energy $E = 1.45 \mu\text{J}$ and
 23 variable scanning speed V . Morphology of the produced LIPSS was further
 24 analyzed with SEM imaging allowing to identify LIPSS period summarized
 25 in Fig. S4(bottom) as a function of film thickness d .

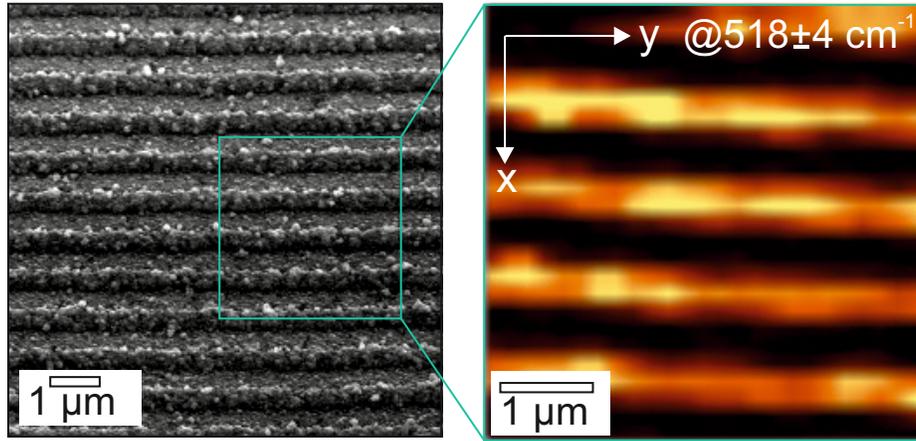


Figure 7: Correlated SEM and Raman ($@518\pm4\text{ cm}^{-1}$) images of the LIPSS surface produced on 180-nm thick a-Si film at $V=3\text{ }\mu\text{m/s}$ and $E=1.5\text{ }\mu\text{J}$.

26 **7. Composition of LIPSS produced on the surface of 180-nm thick**
 27 **a-Si film.**

28 In this section we show that LIPSS produced on the surface of twice
 29 thinner a-Si film ($d=180\text{ nm}$) preserve similar homogeneity and periodic-
 30 ity around 980 nm, while Raman mapping indicates similar structural and
 31 chemical composition - nc-Si pillars arranged parallel to the LIPSS orienta-
 32 tion. This provide the way to tailor the height of the nc-Si pillars, which is
 33 obviously limited by the initial a-Si film thickness.

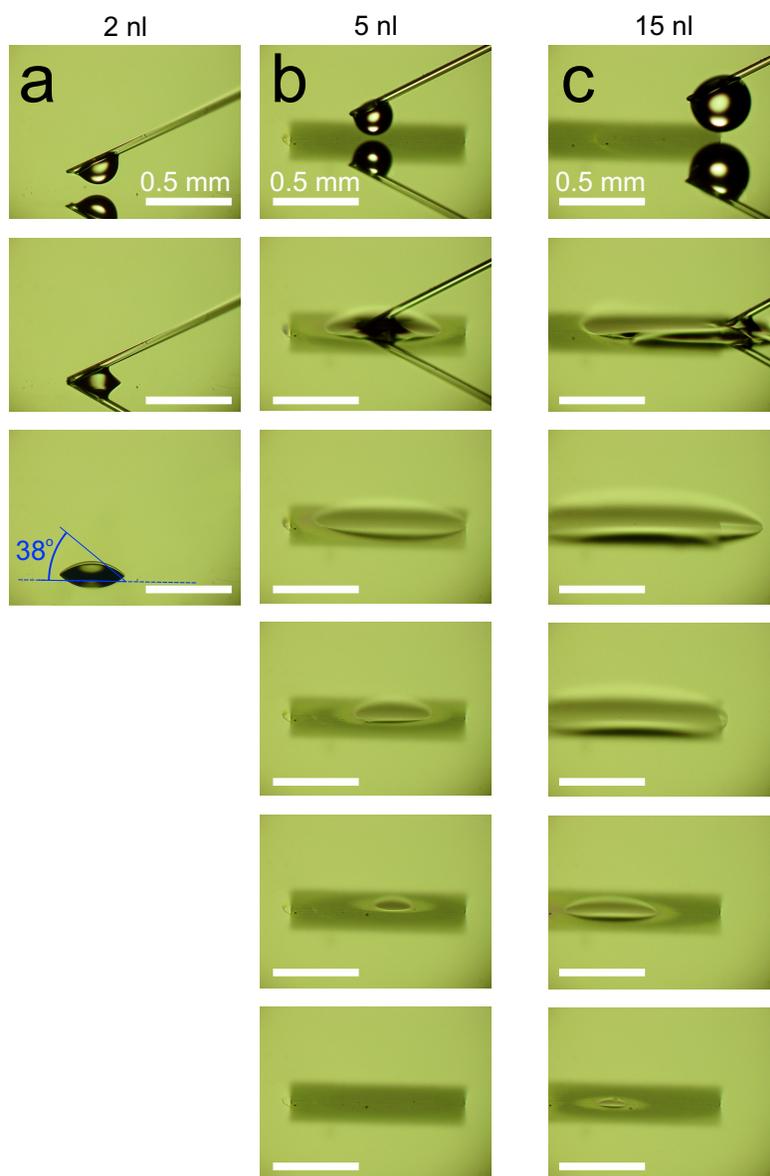


Figure 8: Optical photographs showing the deposition and spreading of the water droplets on a surface of pristine a-Si film (a), onto the 3D conical-shaped LIPSS at the center of patterned area (b), onto the boundary between similar LIPSS area and pristine a-Si film (c.)

34 **8. Wetting properties of LIPSS.**

35 In this section, we highlight good wetting properties of LIPSS that facili-
 36 tate analyte deposition in sensing experiments. Series of optical photographs

37 indicates that the contact angle for a nL-volume water droplet reduces from
38 38° for pristine a-Si film to almost zero for LIPSS showing their good super-
39 hydrophilic properties (Fig. S7(a,b)). Upon its contact with the LIPSS, the
40 water droplet promptly spreads over the texture surface, and does not move
41 beyond its borders. Being deposited near the boundary between LIPSSs and
42 pristine a-Si (Fig. S7(c)), the capillary forces push the liquid towards the
43 textured surface (even the part situated on the smooth a-Si), where the liq-
44 uid finally evaporates providing analyte pre-concentration. Noteworthy, the
45 left-side LIPSSs boundary is not shown owing to limited field of view of the
46 optical system that was shifted to the right to show the liquid behavior on
47 the right-side boundary.

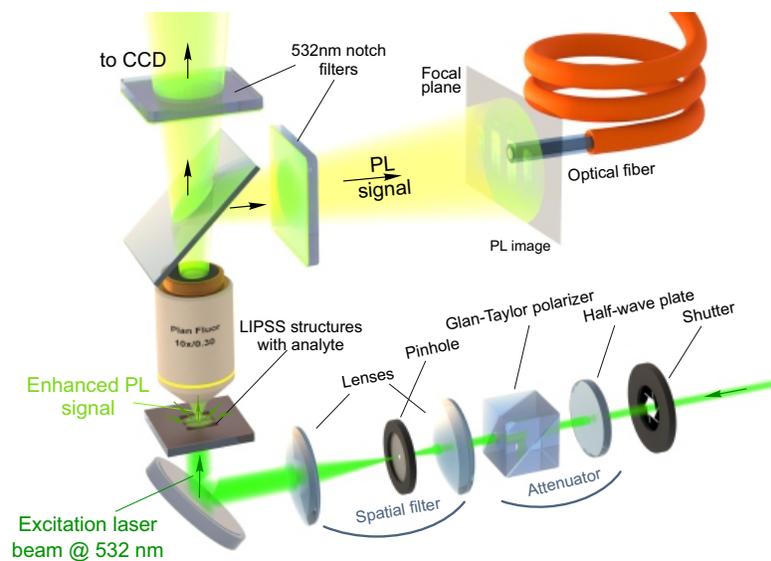


Figure 9: Schematic of the experimental setup used to assess applicability of LIPSS as SEF-based sensor of $[\text{Au}^{3+}]$ ions.

48 **9. Experimental setup for SEF sensing experiments.**

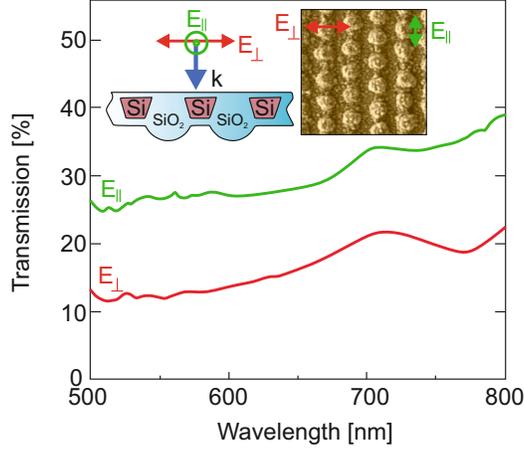


Figure 10: Difference of visible-range transmission through the 3D conical-shaped LIPSS ($V=3 \mu\text{m/s}$, $E=1.5 \mu\text{J}$) for white light polarized either parallel (green) or perpendicular (red) to the LIPSS orientation. Top insets schematically illustrate excitation conditions as well as provide reference top-view SEM image of the tested LIPSS.)

49 10. Polarization-resolved transmission of 3D conical-shaped LIPSS.

50 Laser sources generating linearly polarized radiation are typically used
 51 for sensing experiments. In our sensing experiments shown in this paper,
 52 LIPSS morphologies represent periodical arrangement of nc-Si pillars capped
 53 with 3D SiO_2 protrusions were used. As any type of the grating, our textures
 54 are expected to have anisotropy of the optical properties, which means that
 55 the polarization direction of the pump laser radiation in sensing experiments
 56 should be properly adjusted to ensure maximal performance. To illustrate
 57 this feature, here we measure visible-range FTIR transmission through the
 58 LIPSS (similar to those used for sensing experiments) probed with broad-
 59 band visible light radiation polarized either parallel (green) or perpendicular
 60 (red) to the LIPSS orientation (Fig. S5). These measurements showed de-
 61 creased transmission for light polarized perpendicular to nc-Si pillars orienta-
 62 tion. Noteworthy, the reflection coefficient for both polarization orientations
 63 remained almost the same. Unfortunately, experimental limitations of the
 64 optical setup do not allow us to probe in a similar way near-IR part of the
 65 spectrum, where the anisotropy of optical properties is expected to be pro-
 66 moted by the LIPSS periodicity, which will be the subject of our further
 67 studies.