

Electronic Supplementary Information

Synergistic Effect of Hierarchical Nanopores in Co-doped cobalt oxides 3D Flowers for Electrochemical Energy Storage

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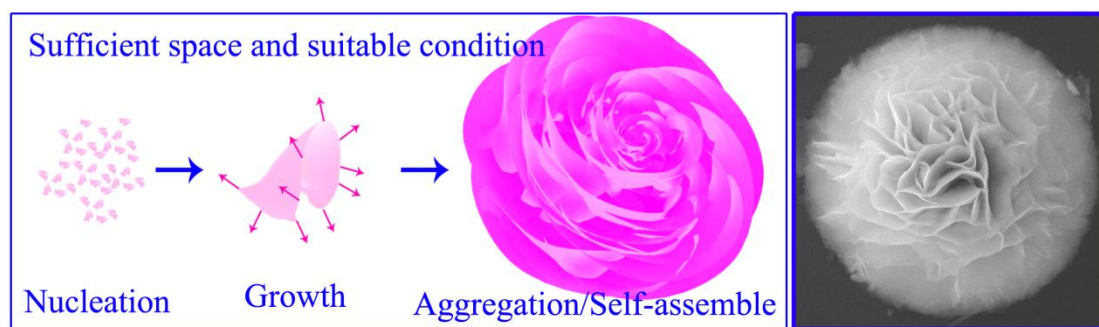
This file contains Supplementary Scheme 1, Tabel S1, Figures S1, S2, S3, S4, S5,
S6, S7.

Formation mechanism of precursor Co(OH)₂ 3DFs

The different morphologies of Co(OH)₂ are tailored by the dissociation rates of the ethanolamine based on different concentrations. The concentration plays a key role in the kinetic reaction of counter ions of Co²⁺ and dissociated OH⁻. Small nuclei will be created at the sites of initial Co(OH)₂ suspensions, and followed by aggregation/self-assembly into larger particles/structures with a particular morphology, which is driven by the force to minimize the surface energy. The pink precipitates which began to appear at the bottom of the glass vial after 6 h (with 15 ml theethanolamine) were

investigated by SEM, showing a 3DF morphology. It reveals that the process of nucleation is very slow and the process of growth and aggregation/self-assembly into 3DFs is very fast, showing a slow nucleation and fast growth mechanism. At low OH^- concentration, there is a relatively smaller number of nuclei and a sufficient time and room for particle growth. This leads to the formation of more petals and larger size flower-like 3D $\text{Co}(\text{OH})_2$ ($\text{Co}(\text{OH})_2$ 3DFs), while the energetically favored state is clarified by the Ostwald ripening process.¹⁻² On the other hand, at high OH^- concentration, a larger number of nuclei were created and there remained relatively limited Co^{2+} in the reaction solution, which further grew into a less petal and smaller size $\text{Co}(\text{OH})_2$ 3DFs.

The assumption of the plausible reaction pathways for the transformation mechanism is as following: the $\beta\text{-Co}(\text{OH})_2$ nanosheets synthesized used this method are not stable under heating, easily transformed to Co_3O_4 crystals. During the calcined in an Ar environment, the Ar crashed the Co_3O_4 crystal and took out the O^{2-} ion producing O vacancies.³⁻⁴ Then the Co_3O_4 could be decomposed into CoO ⁵ or even Co. Subsequently, $\text{Co}_3\text{O}_4/\text{CoO}/\text{Co}$ composites were steered by Kirkendall Effect, which can take place in the metal/metal oxide system.⁶ When the reaction front localized at the $\text{Co}/\text{Co}_3\text{O}_4$ or Co/CoO composites interface, the reduced Co metallic ions diffuse through outward the growing $\text{Co}_3\text{O}_4/\text{CoO}$ composites nanosheets aggregate into nanoparticles and generate a nanopore inside the $\text{Co}_3\text{O}_4/\text{CoO}$ composites,⁷ forming the hierarchically nanoporous $\text{Co}/\text{Co}_3\text{O}_4\text{-CoO}$ nanostructure.



Scheme 1 Schematic diagram showing the formation mechanism of $\text{Co}(\text{OH})_2$ 3DFs and the corresponding SEM image.

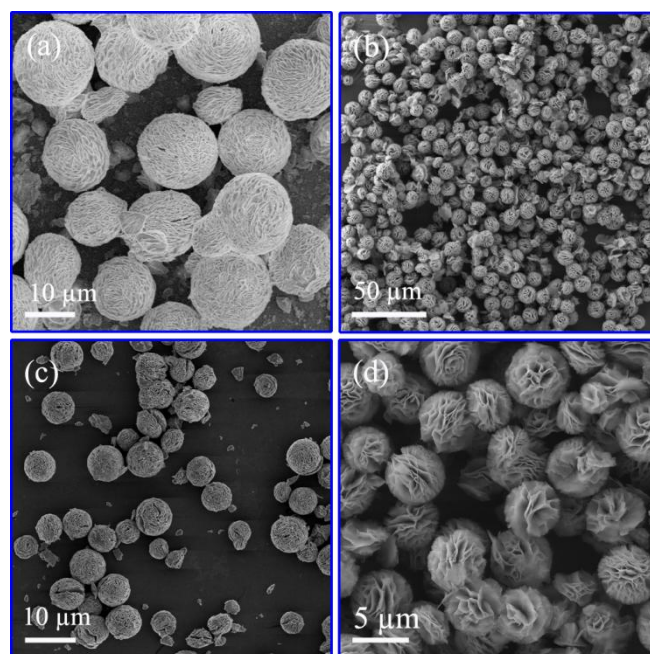


Figure S1 The calcination precursors $\beta\text{-Co}(\text{OH})_2$ 3DFs, which are prepared with different stoichiometry of the ethanolamine: (a) 10 ml (named as CH1), (b) 20 ml (named as CH2), and their corresponding calcined samples, $\text{Co}/\text{Co}_3\text{O}_4\text{-CoO}$ 3DFs, as shown in (c) C1 and (d) C2.

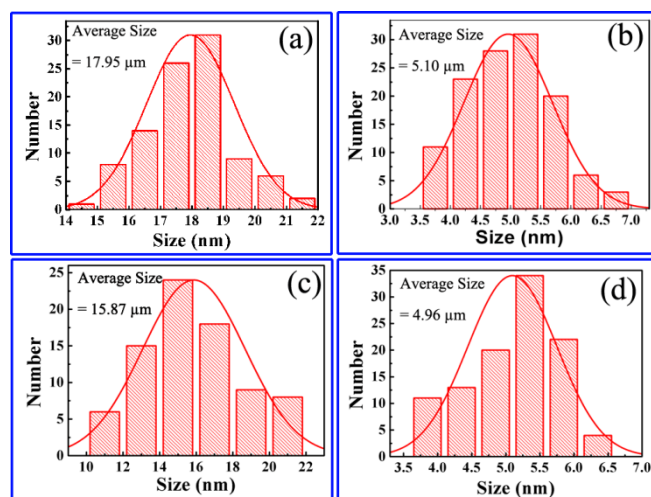


Figure S2 The size distributions of the precursors β -Co(OH)₂ 3DFs: (a) CH1 specimen; (b) CH2 specimen; (c) and (d) their corresponding calcined Co/Co₃O₄-CoO 3DFs, respectively.

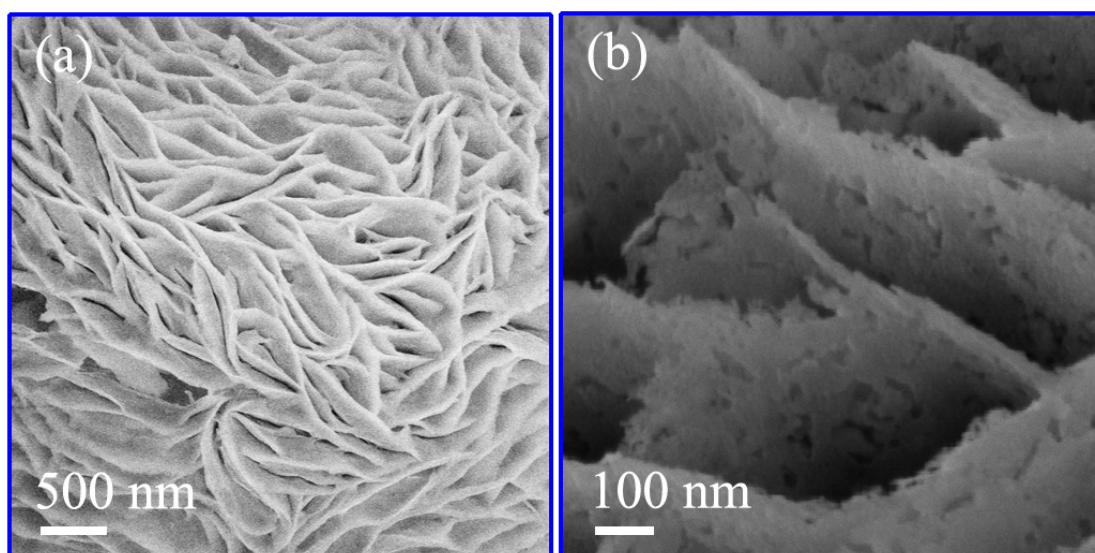


Figure S3 The SEM images of the morphologies of the nanosheets of (a) the β -Co(OH)₂ 3DFs (CH1) and (b) the Co/Co₃O₄-CoO 3DFs (C1).

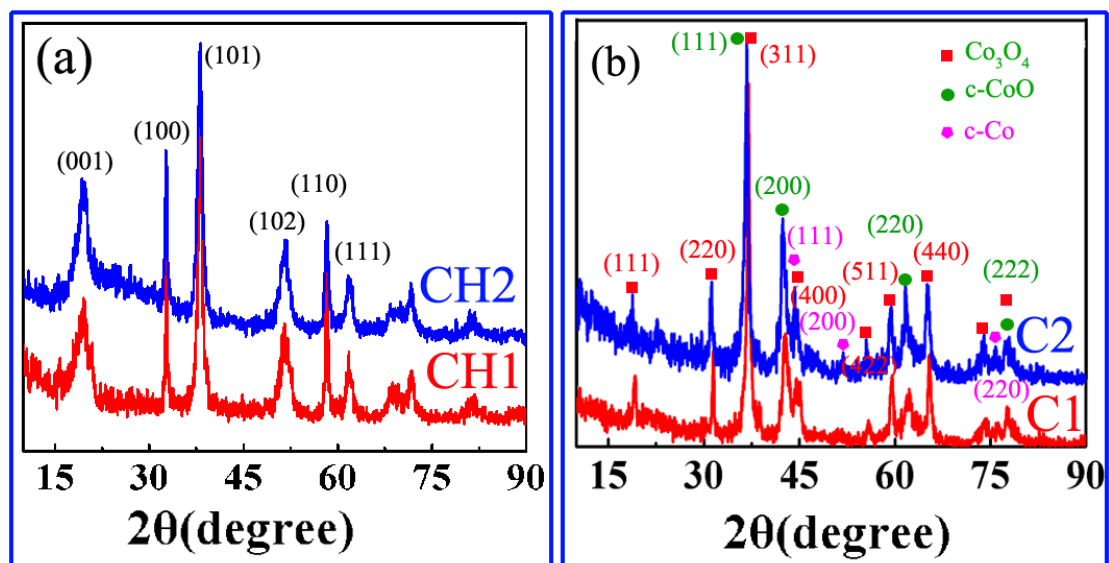


Figure S4 (a) XRD spectra of precursors β -Co(OH)₂ 3DFs (red for CH1 specimen and blue for CH2); (b) XRD spectra of the corresponding calcined Co/Co₃O₄-CoO 3DFs (red for C1 and blue for C2).

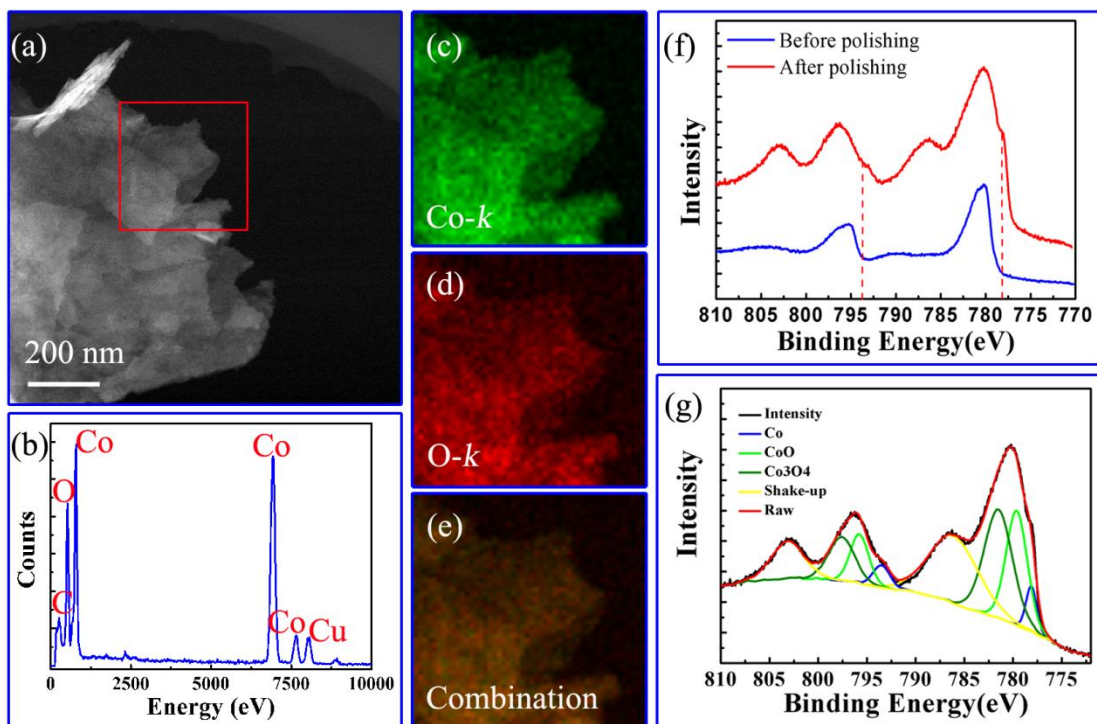


Figure S5 Chemical characterization of the uniformly nanoporous Co-Co₃O₄-CoO 3DFs (specimen C2): (a) HAADF-STEM image of the nanosheets used for elemental

mapping analysis; (b) EDX spectrum; (c) cobalt mapping; (d) oxygen mapping; (e) their combination; (f) XPS spectra before and after polishing; (g) the fitted XPS spectra after polishing.

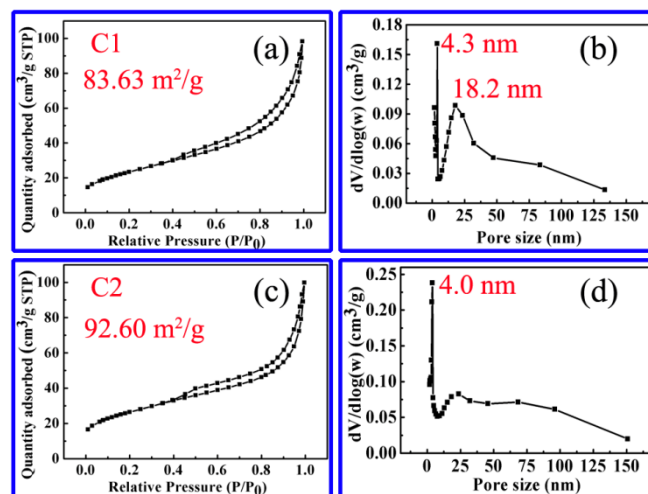


Figure S6 (a) Nitrogen (N_2) adsorption-desorption isotherm of the hierarchically nanoporous Co/Co_3O_4-CoO 3DFs (C1) and (b) the corresponding pore size distribution; (c) Nitrogen (N_2) adsorption-desorption isotherm of the uniformly nanoporous Co/Co_3O_4-CoO 3DFs (C2) and (d) the corresponding pore size distribution.

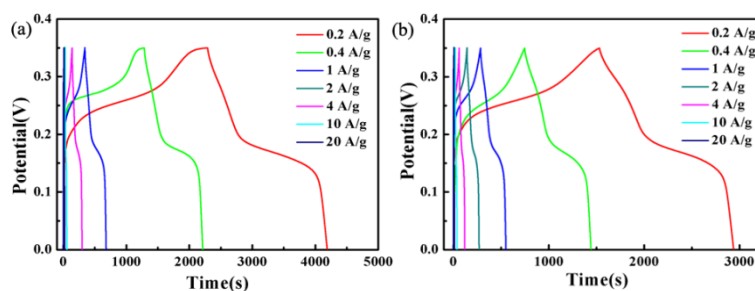


Figure S7 Galvanostatic charge/discharge curves of the hierarchically nanoporous Co/Co_3O_4-CoO 3DFs (a) and uniformly nanoporous Co/Co_3O_4-CoO 3DFs (b) at different current densities.

Tabel S1 The comparison of the performance of nanoporous Co/Co₃O₄-CoO 3DFs and other electrolyte materials

Electrode materials	Specific capacitance(F/g)	Current density(A/g)	Ref
Co ₃ O ₄	604	4	16
CoO/Co ₃ O ₄	362.8	0.2	47
Co ₃ O ₄	599	2	64
Co ₃ O ₄	358	2	65
Co ₃ O ₄	754	2	66
Co ₃ O ₄ on rGO/CNTs	378	2	67
Co ₃ O ₄ - CNFs	556	1	68
Co ₃ O ₄	150	1	69
Co ₃ O ₄	401	0.5	70
CoO	600	0.5	71
a-CoMoO ₄	90	2	72
CoO	311.8	1	73
CoMoO ₄	380	1	74
Co/Co ₃ O ₄ -CoO 3DFs	902.3	1	This work

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