

Electronic Supplementary Information (ESI)

Influence of Hidden Halogen Mobility on Local Structure of CsSn(Cl_{1-x}Br_x)₃ Mixed-Halide Perovskites by Solid-State NMR

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EXPERIMENTAL

Materials

Starting materials were purchased from the following commercial sources and were used without further modification: CsCl (Terochem Laboratories Ltd. 99.7%), CsBr (Sigma, 99.99%), SnCl₂ (Alfa Aesar, >99%), SnBr₂ (Alfa Aesar, 99.2%), SnBr₄ (Sigma, 99%), HBr (Anachemia, 48%), and H₃PO₂ (Sigma, 50 wt.% in H₂O).

High-Temperature Synthesis of CsSn(Cl_{1-x}Br_x)₃ (0 ≤ x ≤ 1): Various members of the solid solution CsSn(Cl_{1-x}Br_x)₃ were prepared by reactions at high temperature. CsX and SnX₂ (X = Cl, Br) were combined in stoichiometric ratios on a 0.5-g scale, finely ground using an agate mortar and pestle, pressed into pellets, and loaded into fused-silica tubes which were evacuated under a pressure of 10⁻³ mbar and sealed. The tubes were heated at 1.5 K/min to either 673 or 723 K (depending on composition), held at that temperature for 15 h, and then cooled to room temperature at 5 K/min. The samples were stored in glass vials and further characterized under ambient conditions.

Cubic CsSnCl₃ was obtained as a metastable phase at room temperature as follows: a sample of CsSnCl₃ prepared as described above was heated to 673 K, kept at this temperature for 15 h, cooled to 573 K at 5 K/min, and then quenched in an ice-water bath.

Mechanochemical Synthesis of CsSnBr₃: A mixture of 1.5 mmol each of CsBr and SnBr₂ was ground using an agate mortar and pestle for 10 min. Within an argon-filled glove box, the mixture was transferred to a 50-mL zirconia grinding vessel (containing ca. 50 g of zirconia balls with 3–8 mm diameter), which was sealed with parafilm to minimize exposure to air. The sample was ground in a Changsha Deco DECO-PBM-V-0.4L electric planetary ball mill at a rotation frequency of 550 rpm for 0.5 h. The vessel was opened to scratch its inner wall by using a clean spatula and sealed again under an inert atmosphere (Ar glove box). This process was repeated four times for a total of 2.5 h of grinding time.

Solvent Synthesis of CsSnBr₃: A mixture of 1 mmol each of CsBr and SnBr₂ was placed in a 40-mL glass vial to which 4.5 mL of concentrated HBr and 0.5 mL of H₃PO₂ were added. The mixture was heated under a nitrogen atmosphere to 120 °C on a hot plate, with continuous stirring by a magnetic stir bar. A black precipitate formed immediately. After the mixture was heated for 0.5 h, it was cooled over 1 h to room temperature. The precipitate was filtered under reduced pressure, washed with isopropyl alcohol, dried for 0.5 h, and quickly packed into a 4-mm o.d. ZrO₂ rotor for the solid-state NMR experiments.

Solvent Synthesis of Cs₂SnBr₆: A mixture of CsBr (2 mmol) and SnBr₄ (1 mmol) was placed in a 10-mL glass vial to which 5 mL of concentrated HBr was added. The mixture was heated to 120 °C on a hot plate, with continuous stirring by a magnetic stir bar. A white precipitate formed immediately. After the mixture was heated for 1 h, it was cooled over 1 h to room temperature. The precipitate was filtered using a Buchner funnel, washed with 95% ethanol, dried overnight, and stored in a vial under ambient conditions.

Powder X-ray Diffraction: Powder XRD patterns were collected on a Rigaku Ultima IV diffractometer equipped with a Co $K\alpha$ radiation source ($K_{\alpha 1}$, 1.78900 Å; $K_{\alpha 2}$, 1.79283 Å) operated at 38 kV and 38 mA, and a D/Tex Ultra detector with a Fe filter to eliminate K_{β} radiation (1.62083 Å). The samples were placed on zero background plates. Data were collected in continuous scan mode between 5 and 90° in 2θ with a step size of 0.0200°. Profile fitting was performed using the FullProf suite of software and unit cell parameters were refined.

Energy-dispersive X-ray Spectroscopy and Field Emission Scanning Electron Microscopy: Samples were examined on Zeiss Sigma 300 VP field emission scanning electron microscope equipped with dual silicon drift detectors for energy-dispersive X-ray spectroscopy to determine chemical compositions.

UV-Visible Diffuse Reflectance Spectroscopy: Diffuse reflectance spectra were collected on a Cary 5000 UV-vis-NIR spectrophotometer between 200 and 800 nm and calibrated with a Spectralon (>99%) reflectance standard. The diffuse reflectance spectra were converted to absorption spectra using the Kubelka-Munk function, $\alpha/S = (1-R)^2/2R$, where α is the Kubelka-Munk absorption coefficient, S is the scattering coefficient, and R is the reflectance. Direct bandgaps were extrapolated from the intercepts in Tauc plots of $(\alpha h\nu)^2$ vs E (eV).

Solid-State Nuclear Magnetic Resonance (NMR) Spectroscopy:

I. Cesium-133 NMR Spectroscopy: Solid-state ^{133}Cs NMR measurements were performed at 11.75 T (^1H , 500 MHz) on a Bruker Avance NEO 500 spectrometer, under magic angle spinning (MAS) conditions using a 4 mm H/X MAS Bruker probe with $\omega_0/2\pi(^{133}\text{Cs}) = 65.6$ MHz. All samples were packed into 4 mm o.d. ZrO_2 rotors and NMR data were acquired using a Bloch decay pulse sequence using a short tip angle pulse of 1.38 μs ($\pi/2$ pulse = 5.5 μs , solution $\gamma B_1/2\pi = 45.5$ kHz) with an acquisition time of 100 ms and an optimized recycle delay of 10 to 1800 s. The nuclear spin-lattice relaxation time (T_1) values of ^{133}Cs nuclei for all samples were measured using an inversion recovery pulse sequence. The T_1 values were calculated by fitting the peak intensity values using a three-parameter exponential decay function: $I_{(t)} = I_{(\infty)} + Ae^{-t/T_1}$, where $I_{(t)}$ and $I_{(\infty)}$ are the NMR signal intensities measured at time t and at t infinity, respectively, and T_1 are A are the spin-lattice relaxation time and pre-exponential constant, respectively. All ^{133}Cs NMR spectra discussed here were referenced by setting a 0.1 M CsNO_3 (aq.) solution at $(^{133}\text{Cs}) = 0.00$ ppm.

II. Tin-119 NMR Spectroscopy: Solid-state ^{119}Sn NMR measurements were performed at 11.75 T (^1H , 500 MHz) on a Bruker Avance NEO 500 spectrometer under both MAS and non-spinning conditions using a 4 mm H/X MAS Bruker probe with $\omega_0/2\pi(^{119}\text{Sn}) = 186.5$ MHz. All samples were packed into 4 mm o.d. ZrO_2 rotors and NMR data were acquired using a 4.0 μs $\pi/2$ ($\gamma B_1/2\pi = 62.5$ kHz) either with a Hahn-echo pulse sequence $((\pi/2)_x - \tau_1 - (\pi)_y - \tau_2 - \text{ACQ}$, where τ represents the interpulse and refocusing delays) or with a Bloch pulse sequence with an acquisition time of 4-5 ms and an optimized recycle delay of 0.01-200 s.

Variable temperature (VT) ^{119}Sn NMR spectra were acquired between 230 and 418 K with the same instrument under non-spinning conditions using a Bruker VT unit and calibrated using the ^{207}Pb chemical shifts of MAPbCl_3 .² Dry $\text{N}_2(\text{g})$ was used as the VT gas and gas flow rates were adjusted to reach the target temperatures. The ^{119}Sn T_1 values were measured using an inversion recovery pulse sequence as discussed in the ^{133}Cs NMR experimental section. All ^{119}Sn NMR spectra were referenced by setting the ^{119}Sn signal of a tetracyclohexyl-tin(IV) powder to $\delta(^{119}\text{Sn}) = -97.35$ ppm, a secondary reference with respect to $\text{Sn}(\text{CH}_3)_4$ at $\delta(^{119}\text{Sn}) = 0.00$ ppm.

Table S1. Elemental analysis of $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials measured by EDS.

Sample (Nominal composition)	Elemental atomic %				%Cl : %Br
	Cs	Sn	Cl	Br	
CsSnCl_3	19.3	19.4	61.3	-	100 : 0
$\text{CsSn}(\text{Cl}_{0.90}\text{Br}_{0.10})_3$	18.9	19.5	54.6	7.0	89 : 11
$\text{CsSn}(\text{Cl}_{0.67}\text{Br}_{0.33})_3$	18.8	19.4	39.6	22.2	64 : 36
$\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$	18.8	18.6	29.4	33.2	47 : 53
$\text{CsSn}(\text{Cl}_{0.33}\text{Br}_{0.67})_3$	18.3	19.1	19.0	43.6	30 : 70
$\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$	19.4	18.7	6.0	55.9	10 : 90
CsSnBr_3	17.3	18.0	-	64.7	0 : 100

Table S2. Unit cell constants (a) and direct bandgap values for $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials.

Sample	Unit cell constant, a (Å)	Direct bandgap (eV)
CsSnCl_3 (cubic)	5.5894(5)	2.83
$\text{CsSn}(\text{Cl}_{0.90}\text{Br}_{0.10})_3$	5.6108(3)	2.71
$\text{CsSn}(\text{Cl}_{0.67}\text{Br}_{0.33})_3$	5.6642(3)	2.42
$\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$	5.6991(6)	2.26
$\text{CsSn}(\text{Cl}_{0.33}\text{Br}_{0.67})_3$	5.7353(7)	2.10
$\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$	5.7830(2)	1.87
CsSnBr_3	5.8031(3)	1.79

Table S3. Room temperature solid-state ^{133}Cs NMR experimental parameters, chemical shifts, fwhm and spin-lattice relaxation (T_1) values for $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials. The ^{133}Cs NMR spectra were acquired at 11.75 T using a Bloch pulse sequence.

Sample	ν_{rot} (kHz)	Recycle delay (s)	^{133}Cs fwhm (Hz) ± 5	$\delta(^{133}\text{Cs})$ (ppm) ± 0.1	$T_1(^{133}\text{Cs})$ (s)
CsSnCl_3 (cubic)	13.0	300	138	64.7	14.8 ± 0.2
$\text{CsSn}(\text{Cl}_{0.90}\text{Br}_{0.10})_3$		60	92	68.6	9.8 ± 0.2
$\text{CsSn}(\text{Cl}_{0.67}\text{Br}_{0.33})_3$		100	82	73.7	15.3 ± 0.4
$\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$		60	114	74.7	19.7 ± 0.1
$\text{CsSn}(\text{Cl}_{0.33}\text{Br}_{0.67})_3$		60	149	73.4	25.0 ± 0.1
$\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$		60	125	67.7	35.3 ± 0.1
CsSnBr_3		60	70	64.0	34.9 ± 0.1

Table S4. Room temperature solid-state ^{119}Sn NMR experimental parameters, chemical shifts, fwhm and spin-lattice relaxation (T_1) values for $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials. The ^{119}Sn NMR spectra were acquired at 11.75 T using a Hahn-echo ($\nu_{\text{rf}} = 62.5$ kHz).

Sample	ν_{rot} (kHz)	Recycle delay (s)	$\delta(^{119}\text{Sn})$ (ppm)	^{119}Sn fwhm (kHz) ± 0.1	$T_1(^{119}\text{Sn})$ (s)
CsSnCl_3 (cubic)	10.0	10	-562 ± 1	1.8	3.5
$\text{CsSn}(\text{Cl}_{0.90}\text{Br}_{0.10})_3$		1	-544 ± 1	1.4	3.4
$\text{CsSn}(\text{Cl}_{0.67}\text{Br}_{0.33})_3$		1	-510 ± 1	1.7	2.4
$\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$		5	-487 ± 1	2.7	3.0×10^{-1}
$\text{CsSn}(\text{Cl}_{0.33}\text{Br}_{0.67})_3$		1	-458 ± 2	4.3	2.7×10^{-2}
$\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$		0.01	-365 ± 5	17.0	8.0×10^{-3}
CsSnBr_3		0.20	-289 ± 5	19.0	5.6×10^{-3}

Table S5. Room temperature solid-state ^{119}Sn NMR experimental parameters, chemical shifts, fwhm and spin-lattice relaxation (T_1) for CsSnBr_3 materials prepared by the solvent synthesis (SS), high temperature (HT) and mechanochemical synthesis (MCS) methods. The ^{119}Sn NMR spectra were acquired at 11.75 T using a Hahn-echo ($\nu_{\text{rf}} = 62.5$ kHz).

Sample	ν_{rot} (kHz)	Recycle delay (s)	$\delta(^{119}\text{Sn})$ (ppm)	^{119}Sn fwhm (kHz)	$T_1(^{119}\text{Sn})$ (s)
CsSnBr_3 (SS)	0.0	0.05	-386 ± 2	7.5 ± 0.2	0.025 ± 0.001
CsSnBr_3 (HT)		0.2	-284 ± 5	21.0 ± 0.5	0.007 ± 0.001
CsSnBr_3 (MCS)		0.2	-295 ± 10	34.0 ± 1.0	0.003 ± 0.001

Table S6. Variable-temperature ^{119}Sn NMR parameters for CsSnBr_3 (SS) acquired under non-spinning sample conditions at 11.75 T using a $4.0 \mu\text{s}$ $\pi/2$ Bloch pulse sequence.

T(set) (K)	T(calibrated) (± 2) (K)	$\delta_{\text{iso}}(^{119}\text{Sn})$ \dagger (ppm)	Ω (ppm) \dagger	κ \dagger	^{119}Sn fwhm (kHz) ± 0.1	^{119}Sn T_2^* (ms)	^{119}Sn T_1 (ms)
233	229.5	-393	85	-0.1	14.7	0.068	635(200)
253	250.9	-393	75	-0.1	13.6	0.074	390(85)
273	273.6	-390	75	-0.25	10.6	0.094	123(12)
292.4	295.4	-386	35	-1.0	7.2	0.139	25(1)
313	318.7	-371	0	NA	6.5	0.154	14(1)
333	341.5	-364	0	NA	6.1	0.164	8.8(5)
353	366.4	-350	0	NA	5.7	0.175	4.12(15)
373	385.5	-334	0	NA	6.0	0.167	2.16(9)
393	407.6	-319.6	0	NA	5.4	0.185	1.14(5)
403	417.9	-317.7	0	NA	4.5	0.222	0.92(2)

NA – not applicable; \dagger - δ_{iso} , Ω and κ are isotropic chemical shift, span and skew, respectively,

$$\text{where } \delta_{\text{iso}} = \frac{\delta_{11} + \delta_{22} + \delta_{33}}{3}, \Omega = (\delta_{11} - \delta_{33}) \text{ and } \kappa = 3\left(\frac{\delta_{22} - \delta_{\text{iso}}}{\Omega}\right)$$

Supplementary note 1:

The variable temperature ^{119}Sn spin-lattice relaxation time (T_1) is related to the absolute temperature values for CsSnBr_3 (SS) as shown in Table S6. $\log_{10}[T_1/\text{s}]$ linearly depends on $[1000/\text{T}(\text{K})]$ (i.e., an Arrhenius relationship) within the temperature range of 230 to 418 K. The slope of the Arrhenius fit is related to the activation energy as $E_a = (2303 \cdot \text{R}) \cdot (\text{slope})$, where $\text{R} = 8.314 \text{ J}/(\text{mol} \cdot \text{K})$. A slope value of $1.51 \pm 0.06 \text{ K}$ was obtained from the least-squares Arrhenius fit; hence $E_a = 28.9 \pm 1.2 \text{ kJ/mol}$ or $0.30 \pm 0.01 \text{ eV}$

Table S7. Solid-state ^{133}Cs NMR acquisition parameters and experimental conditions used for the CsSnBr_3 areal degradation study (see Figure 9b in the manuscript). The ^{133}Cs NMR spectra were acquired at 11.75 T using a Bloch pulse sequence.

Material	T_1 (s)	ν_{rot} (kHz)	Recycle delay (s)	# of scans	Acquisition time (min)
CsSnBr_3 (degraded)	-	5 kHz	1800	4	120
CsSnBr_3	34.9		60	4	4
Cs_2SnBr_6	170		500	4	33
CsBr	631		300	4	20

Table S8. Solid-state ^{119}Sn NMR acquisition parameters and experimental conditions used for the CsSnBr_3 areal degradation study (see Figure 9c in the manuscript). The ^{119}Sn NMR spectra were acquired at 11.75 T using a Hahn-echo ($\nu_{\text{rf}} = 62.5$ kHz).

Material	T_1 (s)	ν_{rot} (kHz)	Recycle delay (s)	# of scans	Acquisition time (min)	
CsSnBr_3 (degraded)	region-1	-	0	0.2	512	1.7
	region-1	-	10	0.2	512	1.7
	region-2	-	10	100	512	853
	region-3	-	10	100	512	853
	region-4	-	12	0.01	2,000,000	736
CsSnBr_3	0.006	10	0.2	1024	3.4	
Cs_2SnBr_6	12.3	10	100	32	53	
SnO_2	3.1 ^{ref.3}	10	30	32	15	
SnBr_4	1.95 ^{ref.4}	0	3	512	25.6	
SnBr_2 (trace metal grade)	SnBr_2 region	1.12 ^{ref.4}	13	1.5	512	12.8
	β -Sn region	0.00012 ^{ref.4}	12	0.005	2,000,000	537

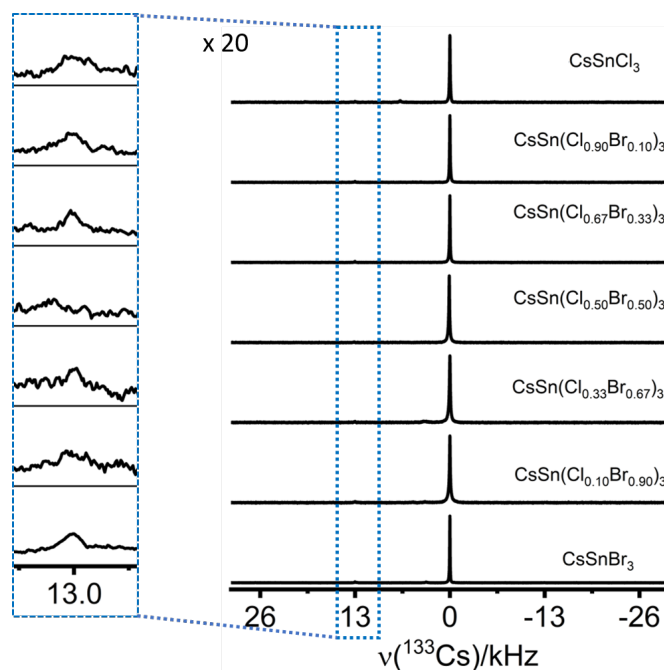


Figure S1. Room temperature solid-state ^{133}Cs NMR spectra of $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials at 11.75 T with a magic angle spinning frequency of 13 kHz. Expansion shows the corresponding spinning side band.

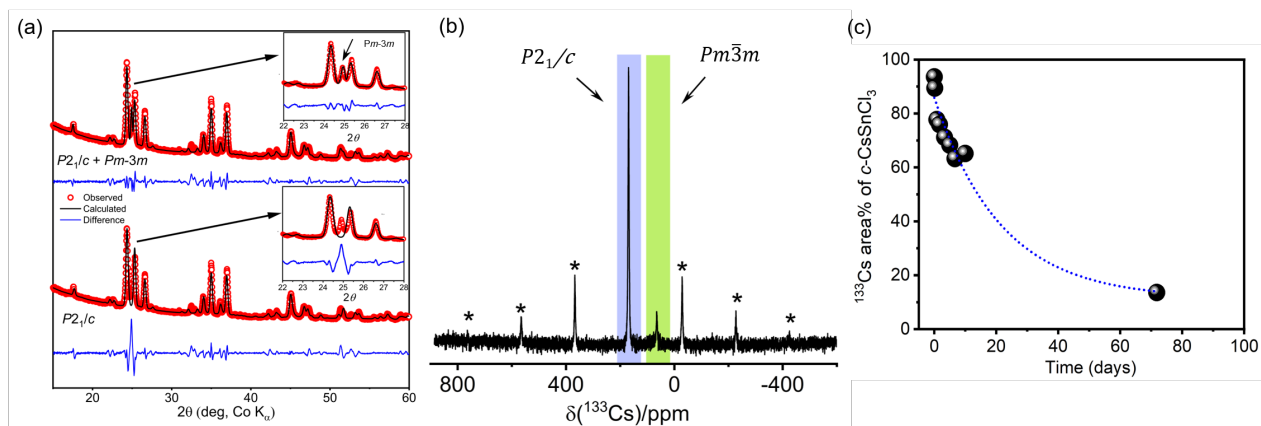


Figure S2. Room temperature PXRD (a) and ^{133}Cs NMR spectra (b) of CsSnCl_3 materials prepared by the high-temperature sealed-tube method followed by slow-cooling (5 K/min). PXRD patterns were collected within 24 h of synthesis and ^{133}Cs NMR spectra were acquired at 11.75 T after six days of synthesis. The asterisks (*) in (b) indicate spinning sidebands for monoclinic CsSnCl_3 . Plot of ^{133}Cs NMR peak area for the cubic CsSnCl_3 phase as a function of time (up to 72 days) showing mono-exponential like decay kinetics for the cubic to monoclinic phase transition in CsSnCl_3 (c). The data points between 10 to 72 days are missing because of the sudden institutional lockdown (~60 days) during the first wave of COVID-19 pandemic in early 2020.

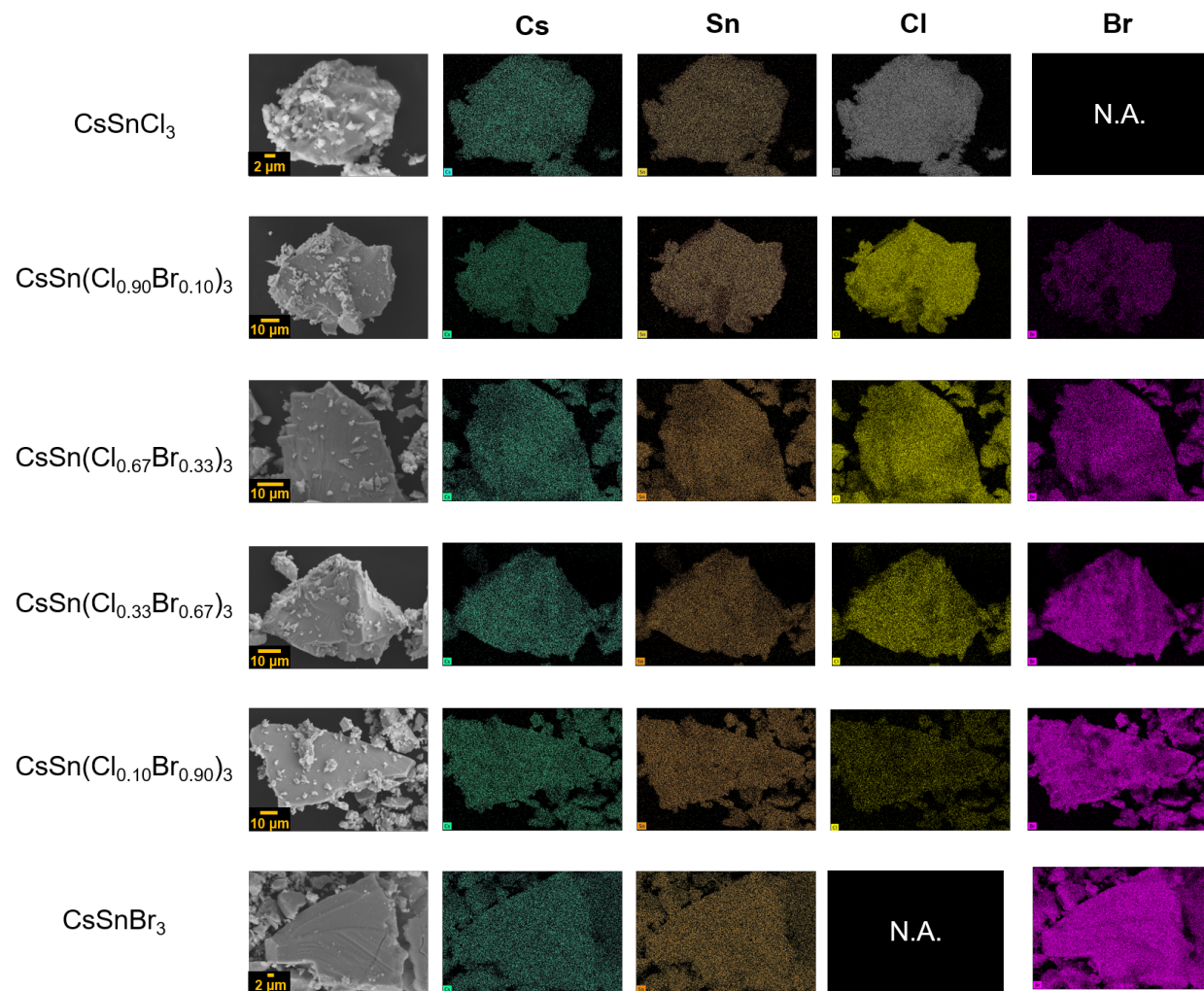


Figure S3. FESEM images and the corresponding EDS elemental mapping for Cs, Sn, Cl and Br for the $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials (top to bottom: $x = 0.00, 0.10, 0.33, 0.67$ and 1.00).

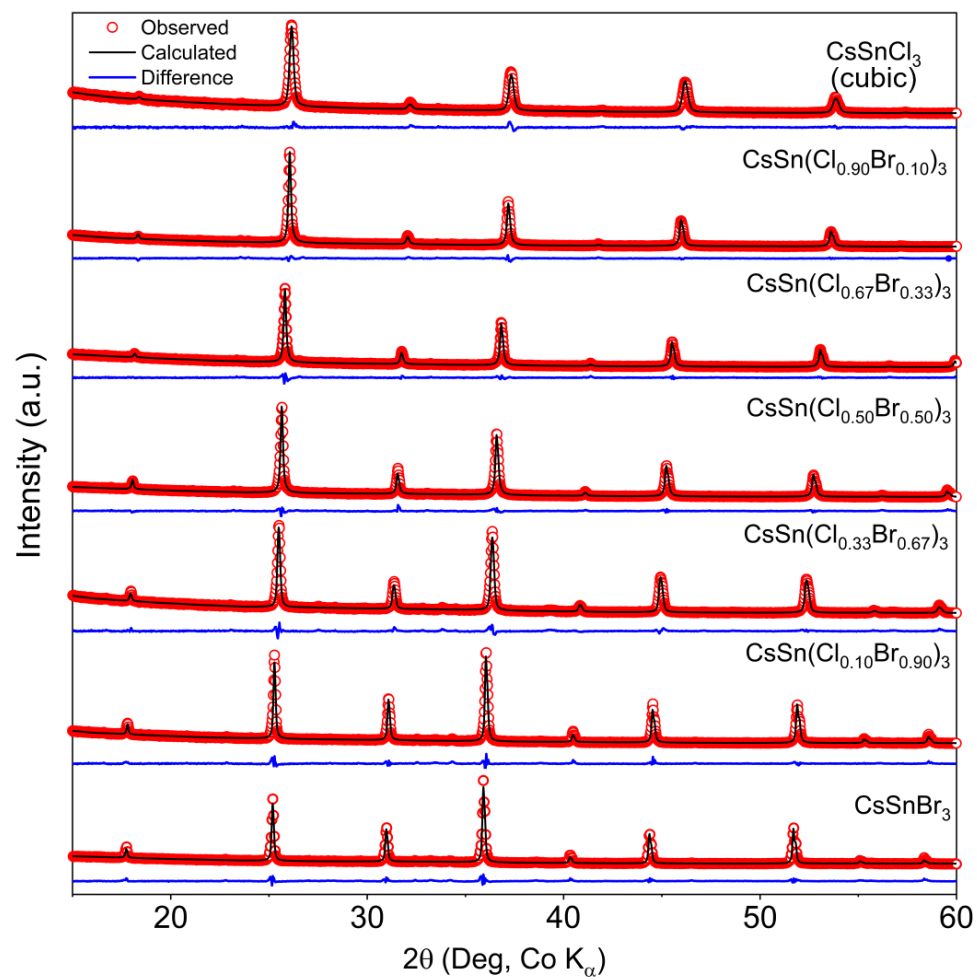


Figure S4. Room temperature experimental and fitted PXRD diagrams for the $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ series. All the diffraction data are fitted assuming a cubic ($Pm-3m$) space group symmetry.

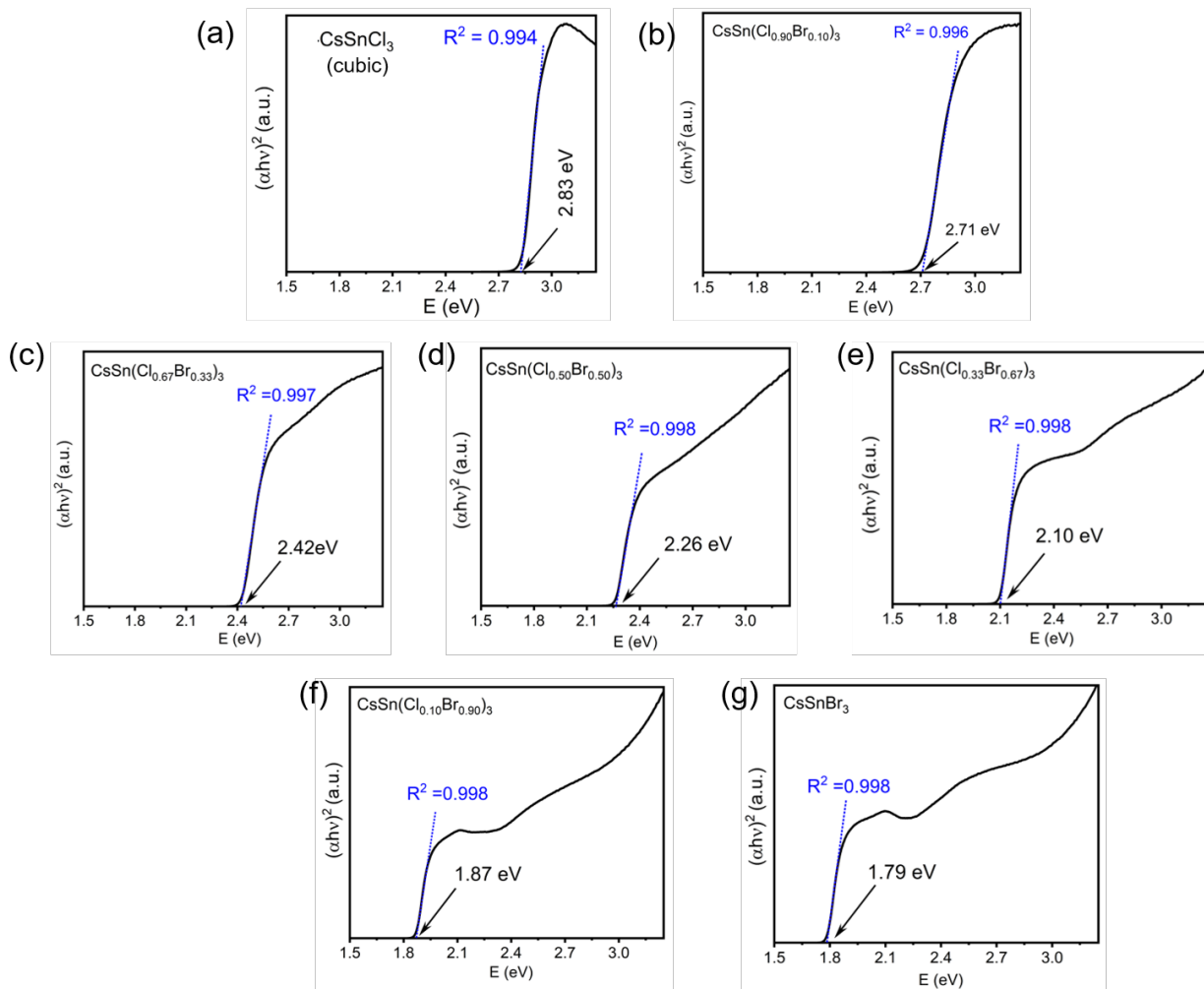


Figure S5. Tauc plots showing direct bandgaps of the cubic phases of $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ materials; *c*- CsSnCl_3 (a), $\text{CsSn}(\text{Cl}_{0.90}\text{Br}_{0.10})_3$ (b), $\text{CsSn}(\text{Cl}_{0.67}\text{Br}_{0.33})_3$ (c), $\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$ (d), $\text{CsSn}(\text{Cl}_{0.33}\text{Br}_{0.67})_3$ (e), $\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$ (f) and CsSnBr_3 (g).

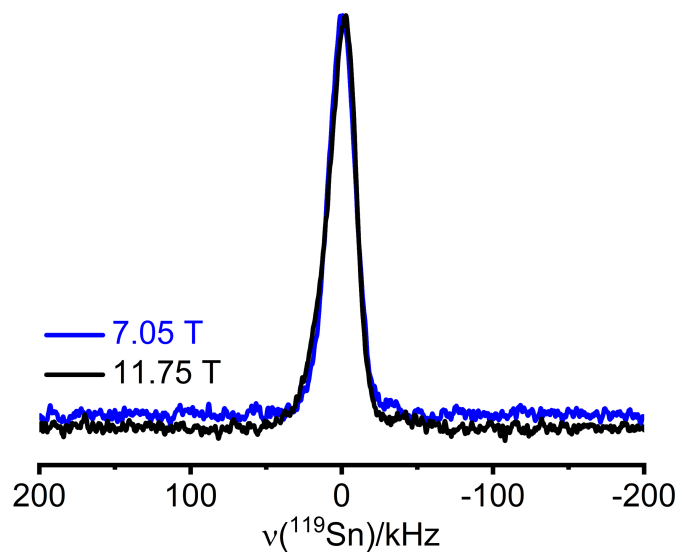


Figure S6. Solid-state ^{119}Sn NMR spectra of CsSnBr_3 at 7.05 and 11.75 T under non-spinning sample conditions. The scales are vertically normalized.

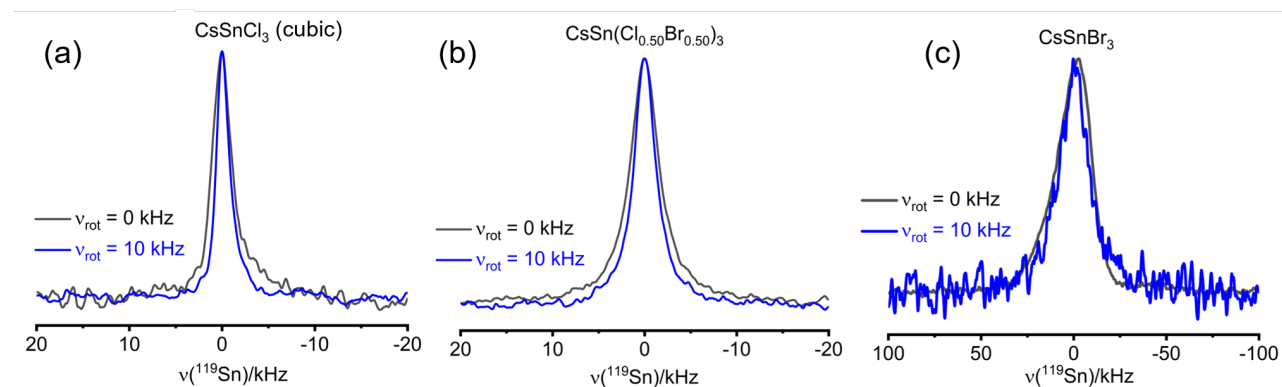


Figure S7. Solid-state ^{119}Sn NMR spectra of *c*- CsSnCl_3 (a), $\text{CsSn}(\text{Cl}_{0.50}\text{Br}_{0.50})_3$ (b) and CsSnBr_3 (c) at 11.75 T acquired with spinning frequencies of 0 and 10 kHz. The scales are vertically normalized.

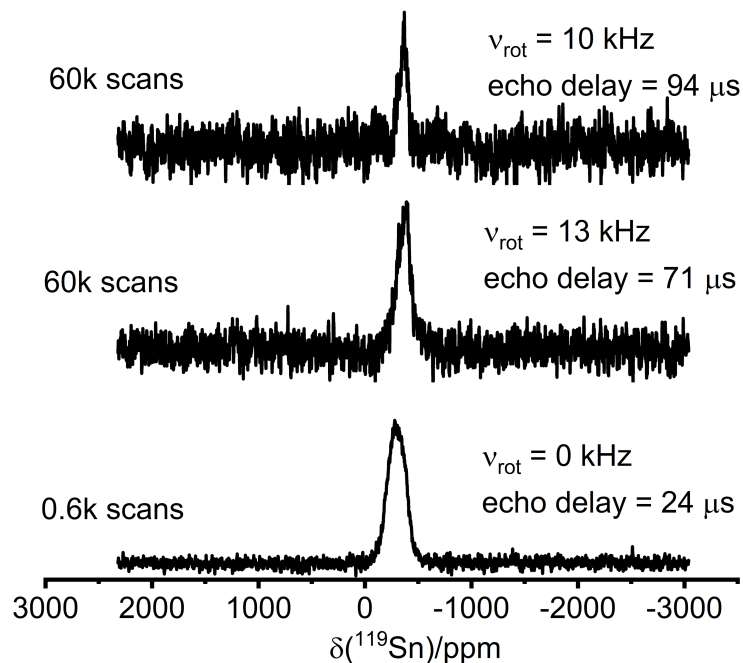


Figure S8. Solid-state ^{119}Sn NMR spectra of $\text{CsSn}(\text{Cl}_{0.10}\text{Br}_{0.90})_3$ at 11.75 T acquired with spinning frequencies between 0 to 13 kHz with the Hahn-echo pulse sequence and with various echo-delays as indicated. The scales are vertically normalized.

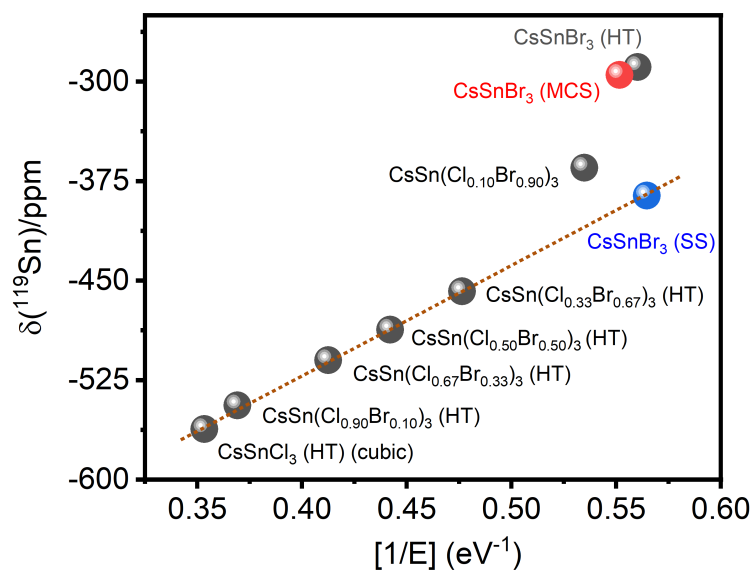


Figure S9. Solid-state ^{119}Sn NMR chemical shifts vs the inverse of direct bandgap values for the $\text{CsSn}(\text{Cl}_{1-x}\text{Br}_x)_3$ series.

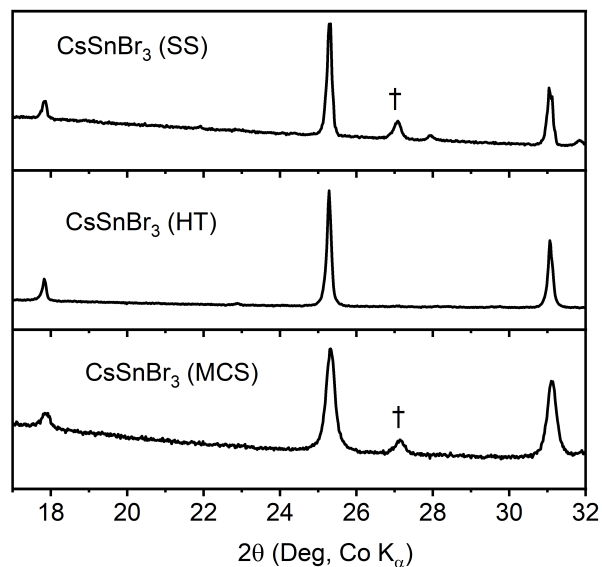


Figure S10. Room temperature PXRD patterns for the CsSnBr_3 parent material synthesized by the solvent synthesis (SS), high temperature (HT) and mechanochemical synthesis (MCS) methods. The dagger (†) indicates signal from Cs_2SnBr_6 .

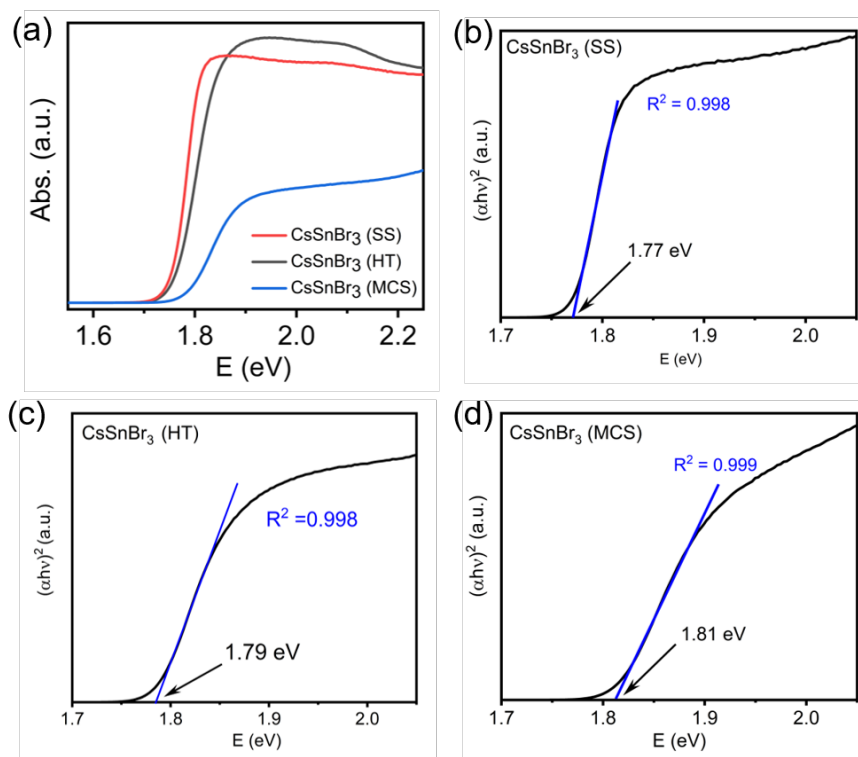


Figure S11. UV-Vis absorption spectra (a) and Tauc plots showing direct bandgaps (b-d) for the CsSnBr_3 parent material synthesized by the solvent synthesis (SS), high temperature (HT) and mechanochemical synthesis (MCS) methods.

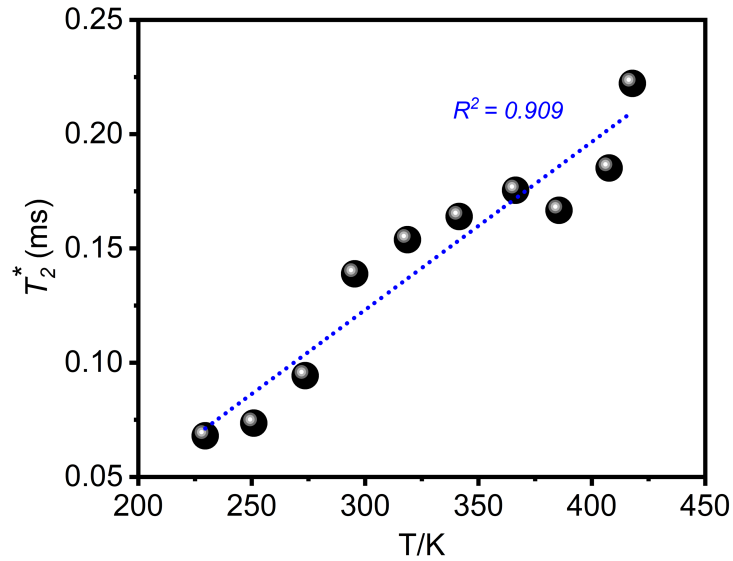


Figure S12. ^{119}Sn T_2^* (i.e., $\frac{1}{f_{\text{whm}}}$) relaxation time as a function of absolute temperature (230-418 K) for the CsSnBr_3 (SS) material. ^{119}Sn NMR spectra were acquired at 11.75 T under non-spinning sample conditions. The data were fit by a least-squares method with the following equation: $T_2^*/\text{ms} = -0.09756(0.02587) + 7.35516(0.77322) \times 10^{-4} \cdot T/\text{K}$

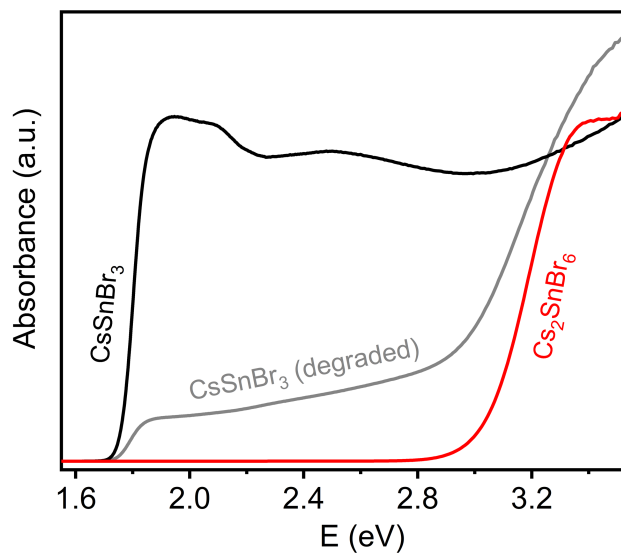


Figure S13. UV-Vis absorption spectra for a CsSnBr_3 sample that was stored under ambient laboratory conditions over 300 days (degraded CsSnBr_3), pristine CsSnBr_3 and Cs_2SnBr_6 .

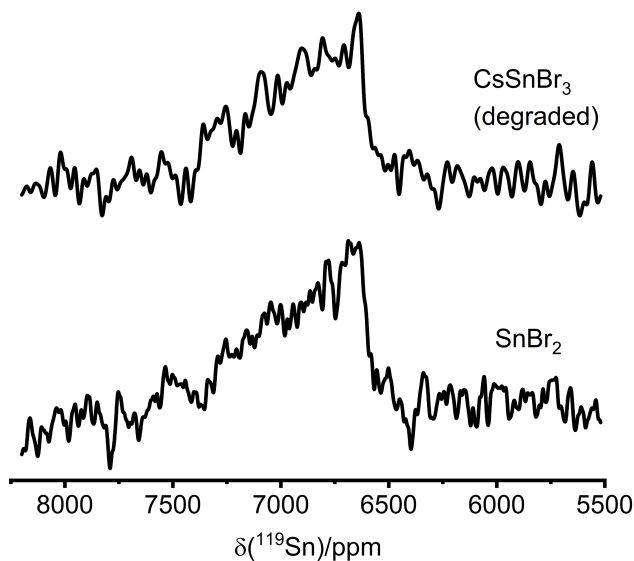


Figure S14. Solid-state ^{119}Sn NMR spectra of the degraded CsSnBr_3 parent and SnBr_2 starting precursor at 11.75 T acquired with spinning frequencies of 12 kHz with 2,000,000 scans each.

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