## **Supporting Information**

Revealing Failure Mechanism of Transition-Metal Chalcogenides towards Copper Current Collector in Secondary Batteries

Guannan Zu<sup>a</sup>, Gencai Guo<sup>a</sup>, Hongyi Li<sup>\*a</sup>, Yue Lu<sup>b</sup>, Ruzhi Wang<sup>a</sup>, Yuxiang Hu<sup>c</sup>,

Lianzhou Wang\*c and Jinshu Wang\*a

- <sup>a</sup> Key Laboratory of Advanced Functional Materials of Education Ministry of China, College of Materials Science and Engineering, Beijing University of Technology, Beijing, 100124, China. Email: lhy06@bjut.edu.cn, wangjsh@bjut.edu.cn. Tel.: +86-67391101
- <sup>b</sup> Institute of Microstructure and Properties of Advanced Materials, Beijing University of Technology, Beijing, 100124, China.
- <sup>c</sup> Nanomaterials Centre, Australian Institute for Bioengineering and Nanotechnology, and School of Chemical Engineering, The University of Queensland, Australia. Email: l.wang@uq.edu.au



Fig. S1 SEM cross section view of (a) Cu/1hTi/MoS<sub>2</sub>, (b) Cu/2hTi/MoS<sub>2</sub> (the white lines in (a) and (b) illustrated sputtered Ti layers).



Fig. S2 (a) SEM image and (b) cross section view of Cu/1hTi.

ruble bri representative experin	iterits on Three in Secondary butteries.	
Control groups	Conclusion	
(a)	The traditional Cu thin foils could also arise	
Cu <sub>thin foil</sub> /MoS <sub>2</sub> and Cu <sub>thin foil</sub> /2hTi/MoS <sub>2</sub>	the generation of Cu <sub>2</sub> S NPs.	
(b)	Ti was not the only choice for the	
Cu/MoS <sub>2</sub> and Cu/2hW/MoS <sub>2</sub>	construction of passivation layer between Cu	
	and MoS <sub>2</sub>	
(c)	Ti was not the only choice for the	
$Cu/MoS_2$ and $Cu/C/MoS_2$	construction of protection layer between Cu and MoS <sub>2</sub>	
(d)	The formation of Cu <sub>2</sub> S wasn't related to the	
Cu/MoS <sub>2</sub> (prepared at 25 °C)	preparation temperature.	
(e)	The influence of CuO (might generate during	
Cu/2hCu/MoS <sub>2</sub>	polishing) was validated, we found that the	
	generation of Cu <sub>2</sub> S wasn't related to CuO.	
(f)	The cycling performance was similar with	
Ti/MoS <sub>2</sub>	that of $Cu/Ti/MoS_2$ .	
(g)	WS <sub>2</sub> showed the same failure mechanism	
Cu/WS <sub>2</sub> and Cu/2hTi/WS <sub>2</sub>	with MoS <sub>2</sub> . The improvement strategy was	
	effective for $WS_2$ .	
(h)	FeS showed the same failure mechanism with	
Cu/FeS and Cu/2hTi/FeS	FeS. The improvement strategy was effective	
	for FeS.	
(i)	Based on the conventional slurry/coating	
Cu <sub>thin foil</sub> /MoS <sub>2 bulk powder</sub>	method, Cu <sub>2</sub> S could be observed in Cu <sub>thin</sub>	
	$_{\text{foil}}/\text{MoS}_{2 \text{ bulk powder}}$ resultant prepared by the	
	conventional slurry/coating method	
	2	

Table S1 Representative experiments on TMCs in secondary batteries.

- [a] All the metal layers (Ti, W, Cu) have been prepared for 2 h, with direct current power of 200 W at 400 °C. All the TMCs films have been prepared for 1 h, with radio frequency power of 100 W at 400 °C.
- [b] Cu foil-Φ16 mm, 0.5 mm in thickness, 99.95% in purity, Cu<sub>thin foil</sub>-Φ16 mm, 9 μm in thickness.
- [c] The batteries in Table S1 were tested at the current density of 1 C with the window of 0.1-3 V at room temperature.



Fig. S3 SEM images of (a) Cu<sub>thin foil</sub>/MoS<sub>2</sub>, (b) Cu<sub>thin foil</sub>/2hTi/MoS<sub>2</sub>, (c) Cu/2hW/MoS<sub>2</sub>,
(d) Cu/MoS<sub>2</sub> (prepared at room temperature), (e) Cu/2hCu/MoS<sub>2</sub>, (f) Ti/MoS<sub>2</sub>, (g) Cu/WS<sub>2</sub>, (h) Cu/2hTi/WS<sub>2</sub>, (i) Cu/FeS, (j) Cu/2hTi/FeS.



Fig. S4 CV curves of (a) Cu/WS<sub>2</sub> and (c) Cu/2hTi/WS<sub>2</sub>, (b) Cu/FeS and (d) Cu/2hTi/FeS. The CV curves were obtained at the scan rate of  $0.1 \text{ mVs}^{-1}$  and sweep window of 0-3 V.



Fig. S5 Cycling performance of different samples corresponding to Table S1.



Fig. S6 TEM images of the other  $Cu/MoS_2$  sample after 200-cycle test at 1 C. (a) low magnitude, (b) enlarged image of the red line area in (a), (c) enlarged image of the white line area in (a).



Fig. S7 TEM image of Cu/MoS<sub>2</sub> anode after cycling.



Fig. S8 TEM images of Cu/MoS<sub>2</sub> (prepared at 25 °C) anode in LIBs (a) before and (b) after cycling at 1 C. The inset image was the enlarged figure of the red area in (b).



Fig. S9 TEM images of (a) Cu/WS $_2$  and (b) Cu/2hTi/WS $_2$  anode in LIBs after cycling at 1 C.



Fig. S10 TEM images of (a) Cu/FeS and (b) Cu/2hTi/FeS anode in LIBs after cycling at 1 C.



Fig. S11 (a) The cycling performance of commercial bulk  $MoS_2$  powder at 0.5 and 1 C, SEM images of the mixture of  $MoS_2$  powder, carbon black and PVDF (b) before cycling, (c) after cycling, TEM images of  $MoS_2$  after cycling (d) low magnitude, (e) high resolution. The inset TEM image in Fig. S12e was the enlarged figure of the red line area.



Fig. S12 TEM images of Cu/2hCu/MoS<sub>2</sub> anode in LIBs (a) before and (b) after cycling at 1 C.



Fig. S13 TEM images of  $Ti/MoS_2$  anode in LIBs (a) before and (b) after cycling at 1 C.

Table S2 Formation energy of different substances according to the first-principles calculations

Structure	E (eV/formula)
MoS <sub>2</sub> (bulk)	-21.7940059
Cu (metal)	-3.72875892
Cu <sub>2</sub> S (bulk)	-11.74212656
Li (bulk)	-1.898217485
Li <sub>2</sub> S (bulk)	-11.96849764
Mo (metal)	-10.94908539
S(bulk)	-4.12142378562
S <sub>8</sub> (bulk, single atom)	-4.12329825219

$\Delta H= 3.89 eV$	Endothermic reaction
$\Delta H= 2.28 \text{ eV}$	Endothermic reaction
$\Delta H$ = -0.16 eV	Exothermic reaction
$\Delta H$ = -1.29 eV	Exothermic reaction
	$\Delta H= 3.89 \text{eV}$ $\Delta H= 2.28 \text{ eV}$ $\Delta H= -0.16 \text{ eV}$ $\Delta H= -1.29 \text{ eV}$





## Experimental section of MoS<sub>2</sub> in SIBs

The preparation procedures of Cu/MoS<sub>2</sub> and Cu/2hTi/MoS<sub>2</sub> have been described in the main manuscript. The SIBs adopted Na metal as the counter electrode and 1 M NaClO<sub>4</sub> dissolved in propylene carbonate/fluoroethylene carbonate (PC: 2% FEC) as the electrolyte. Glass fiber (GF/A: 1.6  $\mu$ m, Whatman, America) acted as the separator. The galvanostatic charge and discharge measurements were carried out on CT2001A model (LANHE, China) with the testing window of 0.1-3 V (vs. Na/Na<sup>+</sup>) at different current densities at room temperature. Cyclic voltammograms (CV) measurement was performed on a ParSTAT MC electrochemical working station (Ametek Advanced Measurement Technology, America) with the window of 0-3 V. The samples were used as anodes to assemble 2032-coin cells in an argon-filled glove box with both H<sub>2</sub>O and O<sub>2</sub> concentration below 0.5 ppm.



Fig. S16 CV curves of (a) Cu/MoS<sub>2</sub> and (b) Cu/2hTi/MoS<sub>2</sub> in SIBs, (c) cycling test and (d) rate

performance of Cu/MoS<sub>2</sub> and Cu/2hTi/MoS<sub>2</sub>.



Fig. S17 TEM images of (a) Cu/MoS<sub>2</sub> and (b) Cu/2hTi/MoS<sub>2</sub> anodes in SIBs after cycling at 1 C. After disassembling the cycled SIBs, TEM characterization was carried out, as shown in Figure S16. Cu<sub>2</sub>S NPs were observed on Cu/MoS<sub>2</sub> resultant (Figure S17a), while none was observed in the cycled Cu/2hTi/MoS<sub>2</sub> resultant (Figure S17b). This result indicated that the corrosive side reaction between Cu and S also occurred in TMCs-based anode in SIBs, which was the other convincing evidence of the corrosion mechanism in rechargeable batteries.