

Supplementary Information

Rational Understanding of the Catalytic Mechanism of Molybdenum Carbide on Polysulfide Conversion in Lithium-Sulfur Battery

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Exerimental Section

Preparation of Mo₂C@C: 4g C (TIMCAL 350G carbon, **Fig. S13** shows the SEM image of the TIMCAL 350G) was placed in a flask and put in an ultrasonic bath for 15 min to disperse the particles well. Subsequently, an aqueous solution of ammonium molybdate (1.56 g in 20 ml of distilled water) was added dropwise to the support, and the mixture was then homogenized for 30 min and dried at 110 °C overnight. Last, the sample was annealed in a tube furnace from ambient temperature to 850 °C at a rate of 5 °C min⁻¹ under Ar flow for 4 h. The Ar atmosphere was retained throughout the cooling of the reactor to room temperature, and the the Mo₂C@C was thus obtained.

Preparation of Mo₂C@C-S: The Mo₂C@C-S cathode was synthesized by a melting diffusion strategy. 400 mg sulfur was well mixed with 100 mg Mo₂C@C. The mixture

was sealed in a Teflon container and heated at 155 °C for 20 h to incorporate sulfur into the porous of Mo₂C@C. The S content of Mo₂C@C-S composite is 78.2 wt% (**Fig. S14**).

Preparation of Clean Mo₂C@C-S: 1 g Mo₂C@C-S were added to the 50 ml CS₂ solvent. Then the mixture was kept for 2h at room temperature. The suspension was centrifuged and washed by CS₂, and then dried at 60 °C for 24 h to obtain Clean Mo₂C@C-S.

Preparations of Mo₂C@C-S and C-S electrodes: Mo₂C@C-S or C-S: Super P: PVDF in the weight ratio 8:1:1 were dispersed in NMP to form a slurry, which was cast onto an Al foil. The electrode was dried overnight in an oven at 60 °C and then roll-pressed and punched a disc with a diameter of 12 mm. The sulfur loading of the electrode is about ~1.5 mg/cm².

Characterizations

XRD measurements were carried out on a Bruker D8-Advance powder X-ray diffractometer operating at 40 kV and 30 mA, using Cu-K α radiation ($\lambda = 1.5405 \text{ \AA}$). The morphology, particle size and elements information were obtained *via* SEM (Hitachi S-5500) coupled with an EDS (Ametek). HRTEM (JEOL 2100F) was used to study the detailed particle and crystal information, operating at an accelerating voltage of 200 kV. Surface areas were determined from nitrogen adsorption and desorption isotherms performed on a Quantachrome Autosorb-1 instrument. Before measurement, the samples were degassed at 80 °C on a vacuumline. TGA analysis was used to determine the sulfur content of the material, on a TA Instruments SDT

Q600, employing a heating rate of 10 °C/min from room temperature to 800 °C under a N₂ atmosphere. The XPS analysis was collected by centrifugation and dried under vacuum overnight. The XPS samples were sealed in a vial before being quickly transferred to the chamber of an ultrahigh vacuum Imaging XPS Microprobe system for analysis (Thermo VGScientific ESCALab 250). All spectra were fitted with Gaussian-Lorentzian functions and a Shirley-type background using CasaXPS software. The binding energies were calibrated to the C 1s peak assigned to C-H at 284.6 eV.

Electrochemical measurements

Assembly of electrochemical cells: All Li-S cells were assembled in a glovebox filled with Ar by a coin-cell design with the components as following: using a Li metal foil as anode, a Celgard 2400 membrane as separator, and 1 M LiTFSI and 0.1 M LiNO₃ in 1:1 (v/v) dissolved in 1,3-dioxacyclopentane (DOL) and 1,2-dimethoxyethane (DME) as electrolyte, Mo₂C@C-S or C-S electrode as cathode. CV measurements were carried out by a Versa STAT 3 potentiostat at a scan rate of 0.1 mV/s within a potential range of 1.6-2.8 V vs. Li⁺/Li. Galvanostatic cycling testing for cells with different cathodes were conducted with a LAND CT2001A at various constant current densities of 0.1 C-2 C within a potential range of 1.6-2.8 V vs. Li⁺/Li. EIS was performed with dc voltage kept at open-circuit voltage and an ac voltage of 5 mV in amplitude was applied in the frequency range 1000 kHz-0.1 mHz.

Li₂S nucleation tests

Li₂S nucleation was studied by loading host (Mo₂C@C or C) on carbon-fiber-paper (CP) current collectors, using Li₂S₈ as catholyte and lithium foil as the anodes. First, Li₂S₈ with a concentration of 0.20 mol/L was prepared by dissolving S and Li₂S (molar ratio 7:1) dissolved in tetraglyme, followed by vigorous stirring for 24 h. The CP was punched into circle disks with a diameter of 12 mm to load Mo₂C@C and C with the identical loading contents of 1.0 mg/cm², respectively. The cathodes and anodes were separated by Celgard 2400 separator. The 25 μ L Li₂S₈ electrolyte was first distributed into the cathode and then 20 μ L LiTFSI (1.0 mol/L) without Li₂S₈ was added onto the anode compartment. The batteries were galvanostatically discharged to 2.06 V at 0.112 mA and then maintained potentiostatically at 2.05 V until the current was below 0.01 mA. The entire process was recorded to evaluate the nucleation/growth rate of Li₂S based on Faraday's law.

Density functional theory calculations: First-principles calculations were performed within the framework of DFT, as implemented in the Vienna *ab initio* simulation package.^[1] We adopted the all-electron projector-augmented wave potential (PAW) with Perdew-Burke-Ernzerhof generalized gradient approximation as the exchange-correlation function.^[2,3] A energy cutoff of 500 eV for the plane-wave expansion and Monkhorst-Pack meshe of 4 $\times$ 4 $\times$ 1 were used to sample the Brillouin zone and ensure the total energy calculations converged with an error of less than 1 meV/atom.^[4] Structural relaxations were performed with forces converged to less than 0.01 eV/ $\text{\AA}$. A vacuum distance of 15 $\text{\AA}$ was employed to avoid interactions of neighboring images. The van der Waals interactions were considering for all the simulations using

the DFT-D3 method.^[5] A (2×1) Mo₂C (101) slab model with a thickness of 5.94 Å was used to investigate the adsorption behavior of LiSPs. The climb-image nudged elastic band (CI-NEB) method with 5 images was applied for computing diffusion and decomposition barriers.^[6] The force convergence criterion for optimization was set at 0.05 eV/Å.

Supplementary Result

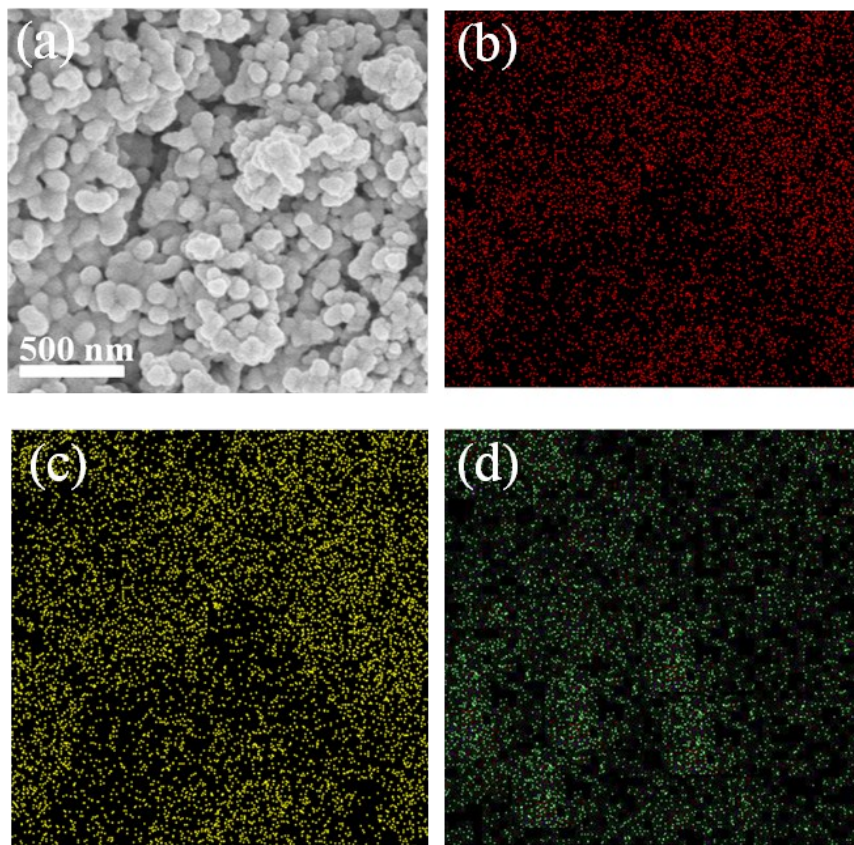


Fig. S1 (a) The Mo₂C@C-S; EDS mapping of (b) Mo, (c)S and (f) C in the SEM image.

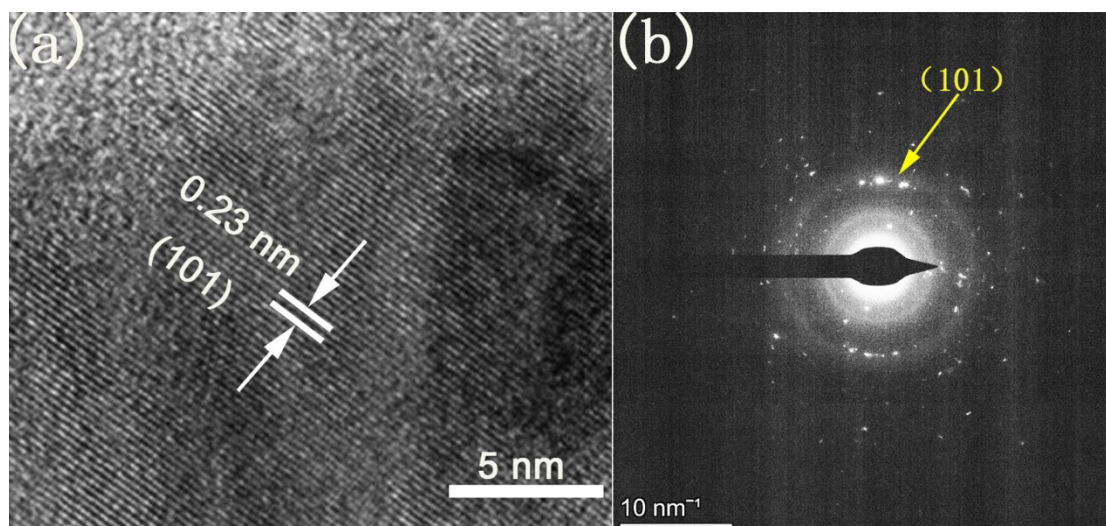


Fig. S2 The HRTEM and SAED patterns of the Mo₂C@C.

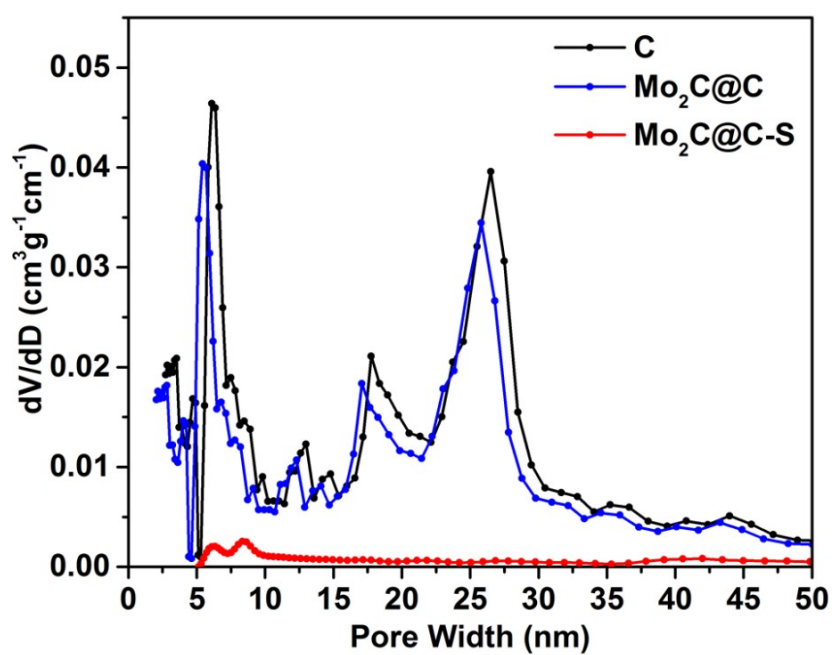


Fig. S3. The pore size distribution of the C, Mo₂C@C and Mo₂C@C-S.

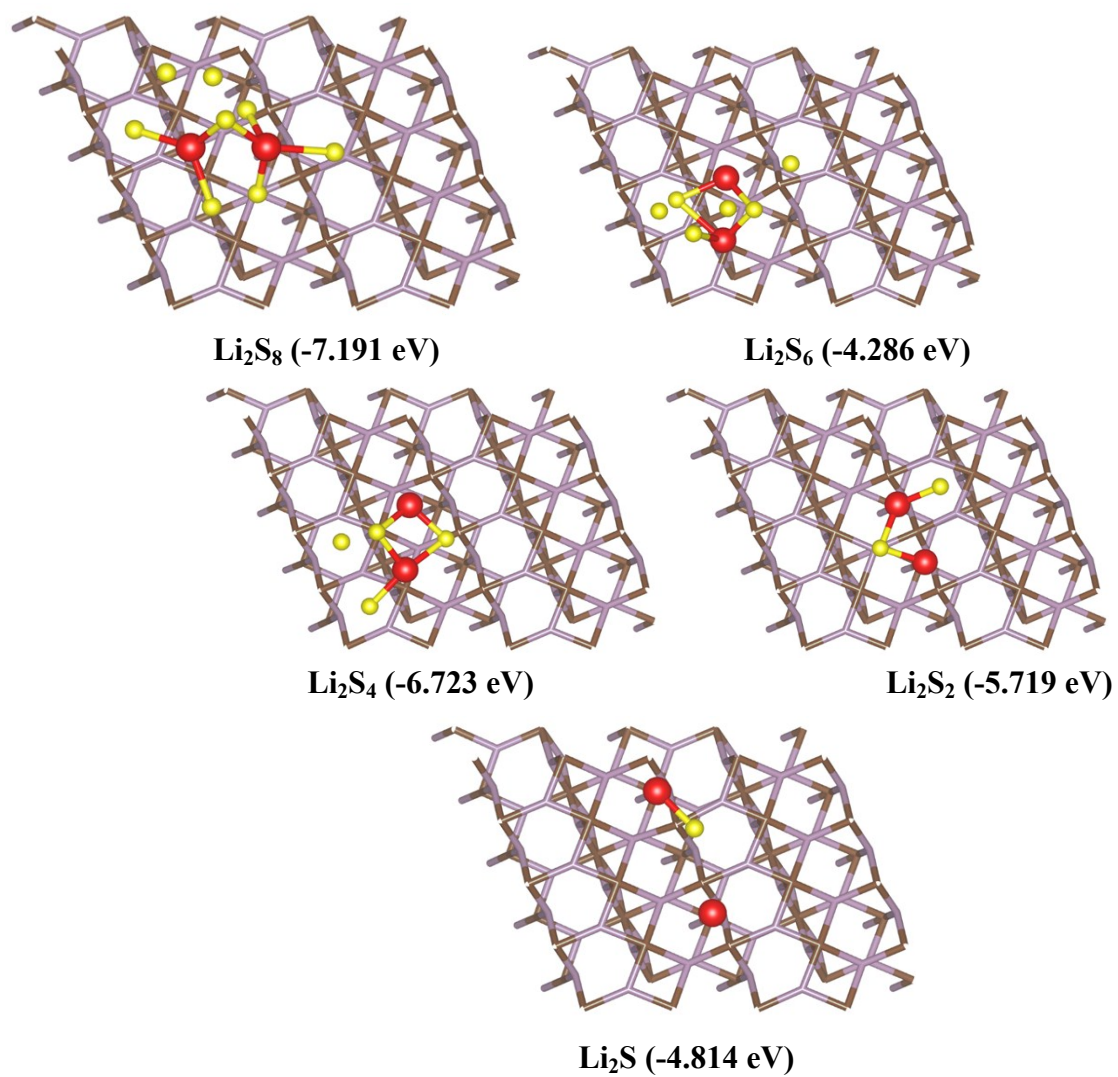


Fig. S4. Optimized geometries of LiPSs on bare Mo_2C (101) surface. The black, light purple, and yellow balls represent C, Mo, and S, respectively.

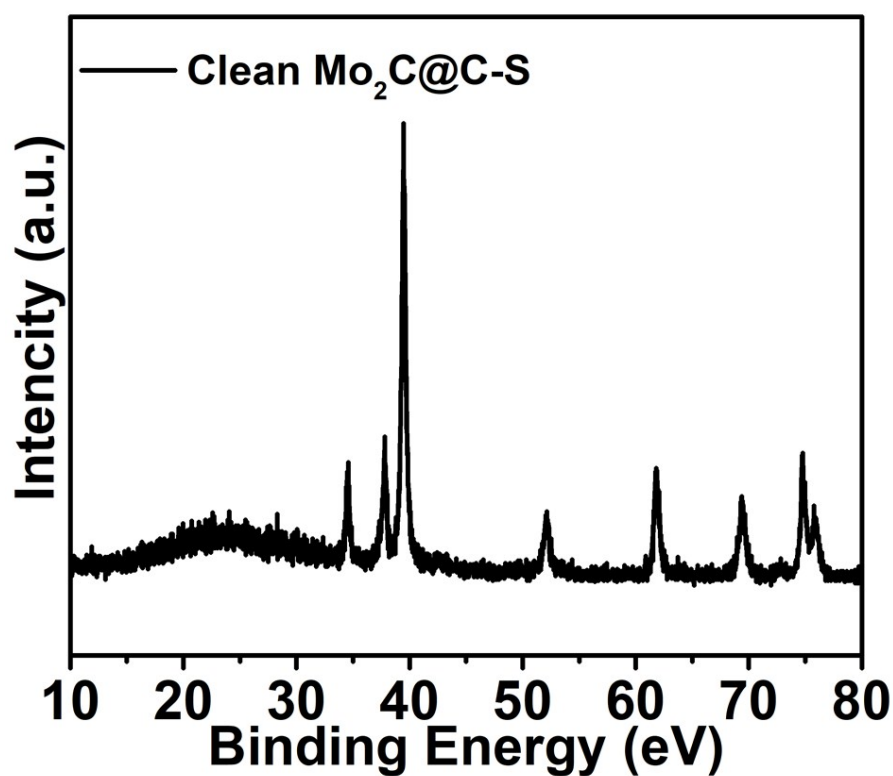


Fig. S5 The XRD of Clean $\text{Mo}_2\text{C}@\text{C-S}$.

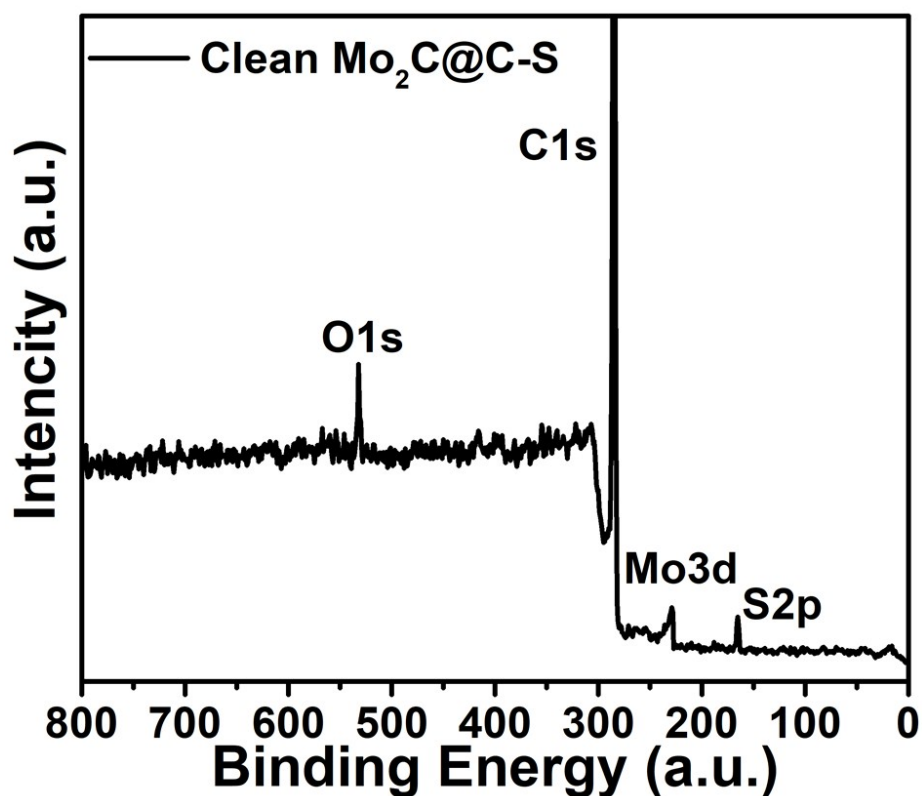


Fig. S6 XPS survey spectra of Clean $\text{Mo}_2\text{C}@\text{C-S}$.

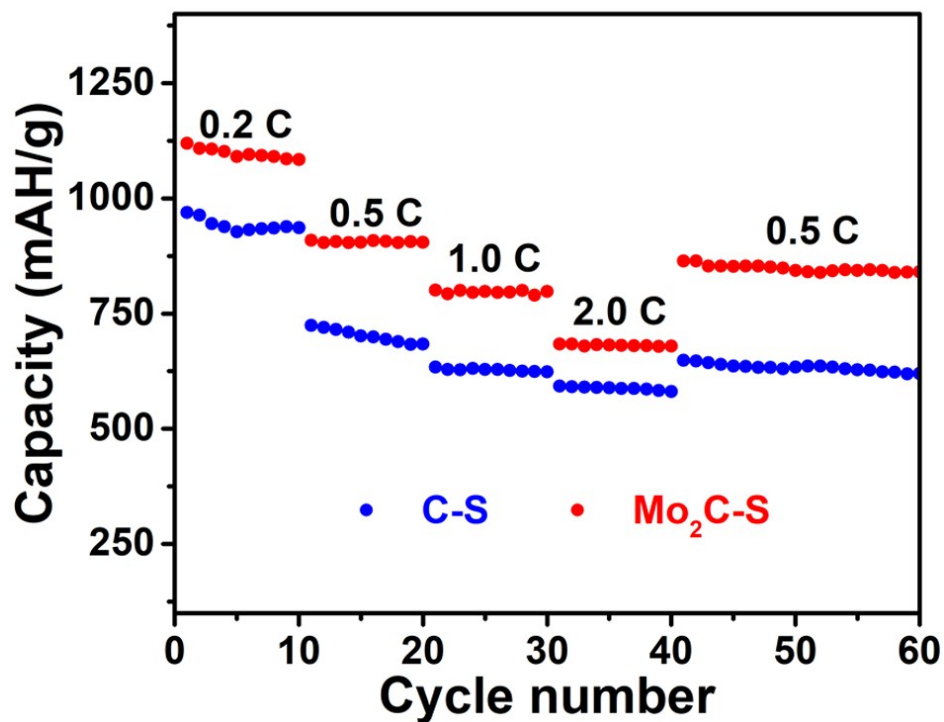


Fig. S7 The rate capabilities of the C-S and Mo₂C@C cathodes.

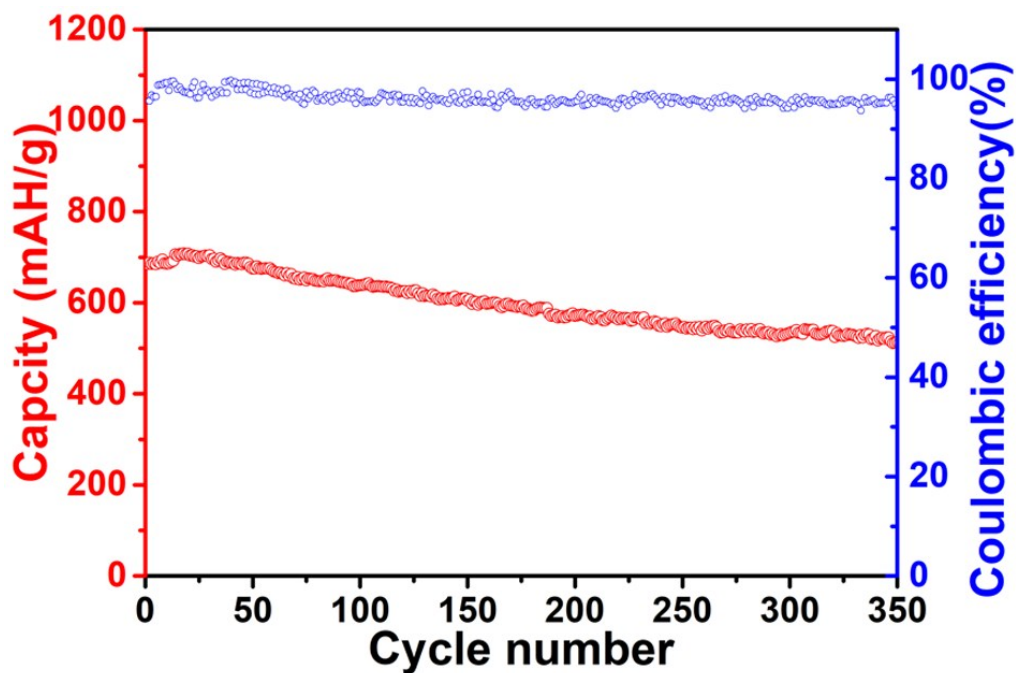


Fig. S8 The cycling performance of cells with Mo₂C@C-S over 350 cycles at a charge/discharge rate of 2 C.

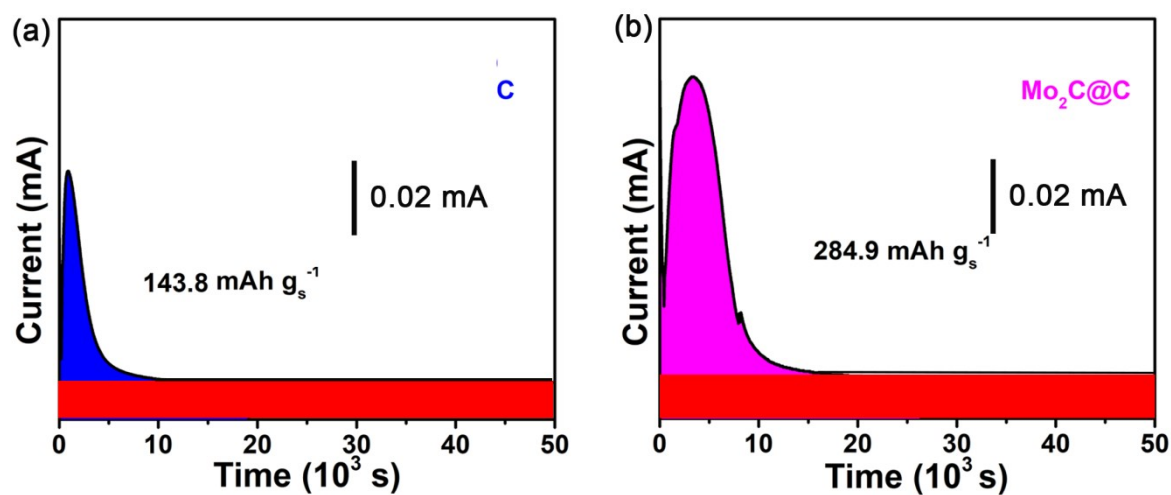


Fig. S9 (a, b) Potentiostatic discharge profiles of a Li_2S_8 /tetraglyme solution on C (a) and $\text{Mo}_2\text{C}@\text{C}$ (b) surfaces at 2.05 V, respectively.

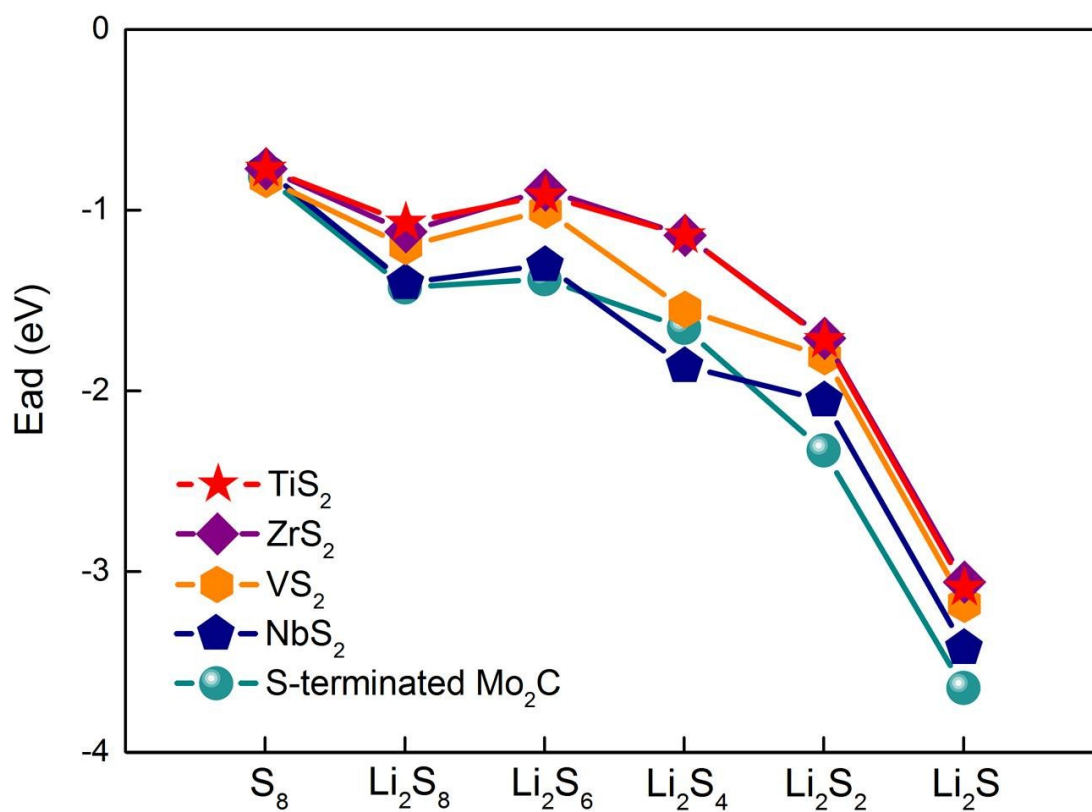


Fig. S10 Calculated adsorption energies of S_8 and LiPSs on S- Mo_2C (101).

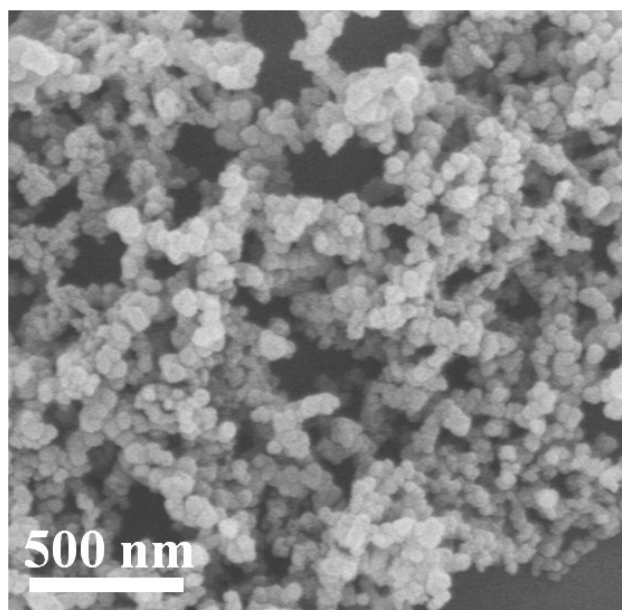


Fig. S13 SEM image of the C.

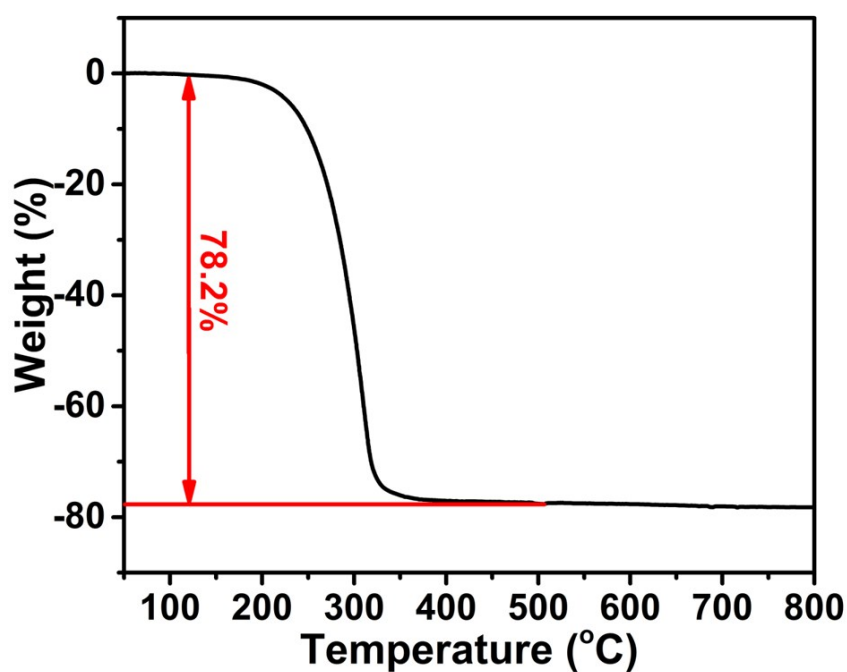


Fig. S14. The TGA curves of Mo₂C@C-S measured under a N₂ flow. The sulfur content is 78.6 wt. % for the Mo₂C@C-S composite.

Table S1. Summary of S 2*p*_{3/2} binding energies (eV) of Mo₂C@C-S and Clean Mo₂C@C-S, measured by high-resolution XPS

Sulfur species	Mo-S bonding (<i>p</i> _{3/2} %Area)	S-S bonding (<i>p</i> _{3/2} %Area)
Mo ₂ C@C-S	44.4% (162.8 eV)	22.2% (163.7 eV)
Clean Mo ₂ C@C-S	48.0% (162.7 eV)	18.0% (163.6 eV)

Table S2. Energy difference between intact and destructed Li_2S_x on S-Mo₂C (101).

	E
Li_2S_6	0.47
Li_2S_4	0.62

References

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- 5 S. Grimme, J. Antony, S. Ehrlich, H. Krieg, J. Chem. Phys. 2010, 132, 154104.
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