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Exploring the nature of the fergusonite–scheelite phase transition and ionic conductivity enhancement by Mo⁶⁺ doping in LaNbO₄

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Supporting Information

Figure S1. Powder x-ray diffraction profiles of as-made samples of LaNb_{1-x}Mo_xO_{4+0.5x} (LNMO). Indices marked above the simulated pattern correspond to a unit cell obtained by transforming the ICSD reference structure into a setting with *I*2/*b* symmetry (c > a > b; $\gamma < 90^\circ$) for better comparability with the tetragonal scheelite structure. Asterisks denote reflections that are considered to arise from structure modulations, by analogy to LaNb_{1-x}W_xO_{4+0.5x}.¹

Sample	Range division (°C)	α_{V} , low <i>T</i> range (×10 ⁻⁶ K ⁻¹)	α_V , high <i>T</i> range (×10 ⁻⁶ K ⁻¹)
LNMO00	500	16.4	9.3
LNMO08	600	11.8	11.7
LNMO12	650	11.9	15.5
LNMO16	600	11.0	14.1
LNMO20	450	10.4	13.4

Table S1. Average linear thermal expansion coefficients derived from linear fitting to the refined lattice volumes (manuscript Figure 1(d)) above and below a range division point chosen arbitrarily to maximise the goodness of fit in each region.



Figure S2. Comparison of lattice parameters of LNMO20 refined against VT-XRD data recorded upon heating and cooling. Error bars are smaller than the symbols.



Figure S3. Laboratory x-ray diffraction data (Cu K α radiation) recorded for LaNbO₄ at room temperature (blue line), and the fit obtained by symmetry distortion mode Rietveld refinement (red line) while refining the amplitudes of eight symmetry-adapted distortion modes. (Grey line: difference between observed and calculated plots; blue markers: positions of Bragg reflections. $R_{wp} = 3.66\%$, $R_{Bragg} = 2.80\%$)



Figure S4. Amplitudes of the eight distortion modes refined in the distortion mode analysis depicted in Figure S3.



Figure S5. Symmetry distortion mode Rietveld refinement fit to the same data shown in Figure S3, while refining only the amplitude of the most significant symmetry-adapted distortion mode (a2) out of eight possible modes. (Blue line: observed; red line: calculated; grey line: difference between observed and calculated; blue markers: positions of Bragg reflections. $R_{wp} = 4.20\%$, $R_{Bragg} = 3.28\%$)



Figure S6. Phonon dispersion curves calculated for fergusonite-type LaNbO₄ at 37 °C.



Figure S7. Phonon dispersion curves calculated for scheelite-type LaNbO₄ at (a) 477 °C, (b) 377 °C and (c) 177 °C. Line colours are assigned independently for each plot and do not necessarily represent corresponding phonon modes in different plots.



Figure S8. Section of a bond valence energy landscape (BVEL) map calculated for a DFToptimised $2 \times 2 \times 1$ supercell of undoped LaNbO₄ using 3DBVSMAPPER.² Black and white atoms mark the positions of O²⁻ interstitial sites denoted by Toyoura *et al.*³ as O_{int}-1 and O_{int}-2, respectively (these sites are drawn for indication only and were not included in the BVEL calculation). The BVEL map is drawn at an isolevel of +0.90 eV above the minimum energy value in the interconnected pathway. (Figure prepared using VESTA⁴)

LNMO16							
Observation	Nb	Мо	La	Mo/(Mo+Nb)	(Mo+Nb)/total		
Centre #1	10.86	2.19	13.19	0.168	0.497		
Centre #2	11.19	2.19	13.61	0.164	0.496		
Centre #3	11.19	2.21	13.71	0.165	0.494		
Centre #4	10.72	2.20	12.84	0.170	0.502		
Centre #5	10.77	2.03	12.90	0.159	0.498		
Centre #6	10.81	1.98	13.36	0.155	0.489		
Average				0.163	0.496		
Standard Dev.				0.005	0.004		
Edge #1	12.26	2.43	14.59	0.165	0.502		
Edge #2	11.12	2.23	13.70	0.167	0.494		
Edge #3	12.19	2.38	14.88	0.163	0.495		
Edge #4	11.53	2.44	14.17	0.175	0.496		
Average				0.168	0.497		
Standard Dev.				0.004	0.003		
Expected value				0.16	0.50		
Observation	Nb	Μο	la	Mo/(Mo+Nb)	(Mo+Nb)/total		
Centre #1	11 54	2.93	15 21	0.202	0 488		
Centre #2	11 94	2.98	15 52	0.200	0 490		
Centre #3	12 03	3 15	15.54	0.208	0.494		
Centre #4	11 57	2.86	15.23	0.198	0 487		
	11.57	2.00	10.20	0.100	0.407		
Standard Dov				0.004	0.430		
Stanuaru Dev.				0.004	0.003		

Average		0.196	0.496	
Standard Dev.		0.005	0.006	
Expected value	·	0.20	0.50	-
Table S2. Non-norm various points on the measurement points subsets based on the	nalised relative cation abund e surfaces of sintered pellets were chosen at random and eir proximity to the pellet edge	ances obtained of LNMO16 an divided arbitra ge.	from EDX spect nd LNMO20. Fo rily into "Centre	tra collected at or each sample, ?" and "Edge"

14.99

15.45

15.03

15.05

0.202

0.189

0.191

0.203

0.503

0.490

0.492

0.498

References

Edge #1

Edge #2

Edge #3

Edge #4

- C. Li, R. D. Bayliss and S. J. Skinner, Solid State Ionics, 2014, 262, 530-535. 1
- 2 M. Sale and M. Avdeev, J. Appl. Crystallogr., 2012, 45, 1054-1056.

3.06

2.81

2.78

3.03

12.12

12.02

11.75

11.88

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- 4 K. Momma and F. Izuma, J. Appl. Crystallogr., 2008, 41, 653-658.