# NIR driven fast macro-damage repair and shear-free reprocessing of thermoset elastomers via dynamic covalent urea bonds 

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## Experimental section

Materials: Amine-terminated polysiloxane (DMS-A12: $M_{\mathrm{n}}=900-1000 \mathrm{~g} / \mathrm{mol}$ ) were purchased from Gelest, Inc. Isophorone diisocyanate (IPDI), hexamethylene diisocyanate (HDI), 4,4'-methylenebis-(cyclohexyl isocyanate), 4,4'-diphenylmethane diisocyanate, and HDI trimer were obtained from best-reagent companies. All of the other regents were used as received.

Preparation of polydopamine nanoparticles (PDAs): For a typical synthesis of PDAs with an average size of 340 nm , the aqueous ammonia solution $\left(\mathrm{NH}_{4} \mathrm{OH}, 5 \mathrm{~mL}, 28-30 \%\right)$ was mixed with ethanol ( 80 mL ) and deionized water ( 180 mL ) under stirring at $30^{\circ} \mathrm{C}$ for 30 min . Dopamine hydrochloride ( 1 g ) was dissolved in deionized water $(20 \mathrm{~mL})$ and then added into the above mixture. The color of this solution immediately turned pale brown and gradually changed to dark brown. The reaction was allowed to proceed for 24 h . The PDAs were obtained by freeze drying after centrifugation and washed with water three times.

Preparation of PUA-PDMS/PDAs composites: 25 mg polydopamine nanoparticles was dispersed in 20 mL anhydrous tetrahydrofuran at room temperature and ultrasonicated for 20 min .4 g DMSA12 and 0.76 g IPDI were added to the solution and stirred for 1 h , then 0.2 g tri-HDI was added into the solution, the mixture was stirred at room temperature for 20 min , and then poured into a PTFE mold. The THF was evaporated at room temperature for 5 days. The PUA-PDMS composite elastomers with graphene or nanotube as fillers were prepared by using the same procedure just by replacing polydopamine nanoparticles with graphene or nanotube. The same procedure was also used for preparing the PUA-PDMS elastomer without PDAs fillers.

Self-healing experiments: The damage sample was exposed to an 808 nm near-infrared (NIR) light with diverse irradiation intensities and times (for scratches: $2 \mathrm{~W} / \mathrm{cm}^{2}, 5 \mathrm{~min}$; for hole repair: $\left.3 \mathrm{~W} / \mathrm{cm}^{2}, 10 \mathrm{~min}\right)$. The distance between the sample and the light was 2 cm .

Reprocessing experiments: The samples were first cut into small pieces and then put into a mold which was then irradiated by NIR light with an intensity of $150 \mathrm{~W} / \mathrm{cm}^{2}$. The distance between the mold and the light was 8 cm .

Characterization: The weight-average molecular weight of the synthesized PU-PDMS was determined by gel permeation chromatography (GPC; HLC-8320) with THF as an eluent phase and PMMA as a standard sample. PUA-PDMS/PDAs composites were characterized by IR spectroscopy on a Fourier transform infrared spectrometer (FTIR; Nicolet 560) in the frequency range of $400-4000 \mathrm{~cm}^{-1}$. NMR experiments were conducted by 1H NMR (Bruker ARX-400 at 400 MHz ) with DMSO-d ${ }^{6}$ as solvent. The morphology of polydopamine particles was observed on an FEI-inspect SEM with an acceleration voltage of 15 kV . Stress-relaxation analysis experiments were performed in tensile geometry on a DMA Q800 apparatus (TA Instrument), and rectangular samples were utilized (ca. $0.5 \mathrm{~mm}(\mathrm{~T}) \times 3 \mathrm{~mm}(\mathrm{~W}) \times 20 \mathrm{~mm}(\mathrm{~L})$ and a gauge length of $\sim 8 \mathrm{~mm})$. For the stress-relaxation analysis, the built-in stress relaxation mode was used. Samples were equilibrated at a set temperature for 5 min , and then subjected to a constant strain of $10 \%$. The stress decay over time was monitored. Mechanical tensile-stress experiments were conducted on an Instron 5567 machine (USA) at room temperature with a strain rate of $50 \mathrm{~mm} / \mathrm{min}$. The dimension of specimens was $35 \times 8 \times 0.7 \mathrm{~mm}^{3}$. At least four samples of each loading fraction were tested. Dynamic rheological measurements were carried out in a dynamic rheometer (Bohlin Gemini 2000, Malvern, British) in constant-strain mode by controlling the strain at $1 \%$. The diameter of the plate was 25 mm , and the gap was about 1 mm . All of the samples were tested from 60 to $180{ }^{\circ} \mathrm{C}$ at a heating rate of $3{ }^{\circ} \mathrm{C} / \mathrm{min}$ in the frequency of 1 Hz . The temperature variation under NIR light irradiation was recorded by a Testo 875i Infrared Thermal Imager.

## Computational Methods

The geometries of all the reactants, products and transition states (TS) were optimized at the b3lyp/6-311+G(d,p) level of theory (M062X method was also adopted, and the results were consistent). Frequency analysis was carried out on all stationary points to confirm their nature (minimum, TS) on the potential energy surface. The universal solvation model SMD was used to
mimic the solvent dimethyl sulfoxide (DMSO) for the calculation of the free energy of solvation.
All calculations were carried out with the Gaussian16 software package.


Figure S1. SEM image of the polydopamine nanoparticles.


Figure S2. Strain-stress curves of the original and repaired PUA-PDMS/PDAs composites.


Figure S3. Optical images of a PUA-PDMS/PDAs composite damaged by blade with many scratches, and after NIR exposure for 2 min .


Figure S4. Temperature variation of the PUA-PDMS/PDAs composites as a function of NIR exposure time.


Figure S5. Optical images showing the reprocessing in a "SCU" shape mold.


Figure S6. Optical images showing the fabrication process of a "田" shape product by NIR exposure. All of the overlapped areas are irradiated by NIR, which causes strong adhesion between each other due to the reversible urea bonds.


Figure S7. Stress-strain curve of the original and reprocessed PUA-PDMS elastomer without PDAs. Inset are the optical pictures of PUA-PDMS elastomer pieces (left) after being heated at $200^{\circ} \mathrm{C}$ for 5 min (right).


Figure S8. Optical pictures of small pieces of PUA-PDMS elastomers with graphene or carbon nanotube as fillers (left) after being irradiated by NIR light (right).


Figure S9. TGA curve of PUA-PDMS/PDA composites.


Figure S10. Relative FTIR peak intensity curves of the isocyanate group from a linear PUA-PDMS at about $2260 \mathrm{~cm}^{-1}$ which was normalized by C-H (3000-2800 $\mathrm{cm}^{-1}$ ) recorded as a function of temperature.




Figure S11. Schematic illustration of the mechanism of the reversible dissociation and reformation of a urea compound formed by IPDI and ethyl amine.


Figure S12. Intrinsic reaction coordinate of TS1.


Figure S13. Optical images of the PUA-PDMS/PDAs composite pieces made from hexamethylene diisocyanate (A) and 4,4'-methylenebis-(cyclohexyl isocyanate) (B) before and after NIR exposure.


Figure S14. Optical images of the PUA-PDMS/PDAs composite pieces made from 4,4'diphenylmethane diisocyanate before and after NIR exposure.


Figure S15. Energy profile of the decomposition reaction of the model urea compound formed by 4,4'-diphenylmethane diisocyanate and ethyl amine.


Figure S16. Storage modulus $\left(\mathrm{G}^{\prime}\right)$ and loss modulus $\left(\mathrm{G}^{\prime \prime}\right)$ of the PUA-PDMS/PDAs composites made from an aromatic isocyanate (4, $4^{\prime}$-diphenylmethane diisocyanate) as a function of temperature.



Figure S17. GPC curves of a linear PUA-PDMS. Black curve: a linear PUA-PDMS formed by IPDI and DMS-A12 with a 1:1 ratio; red curve: the linear PUA-PDMS was dissolved in toluene and then 1 equivalent DMS-A12 was added into the solution and incubated at $120^{\circ} \mathrm{C}$ for 12 h ; blue curve: another equivalent IPDI was added into the solution and stirred for 20 min , and then evaporate the toluene room temperature for 12 h and then $80^{\circ} \mathrm{C}$ for another 12 h .


Figure S18. Normalized stress-relaxation curves of PUA-PDMS/PDAs composite at various temperatures (Black: $90^{\circ} \mathrm{C}$, red: $100^{\circ} \mathrm{C}$, blue: $110^{\circ} \mathrm{C}$, dark cyan: $120^{\circ} \mathrm{C}$ ) as a function of time (A: $0-80 \mathrm{~min}$; B: 0-5 min). C: Fitting line of the relaxation times according to the Arrhenius equation, and the relaxation activation energy obtained from the slope.

The stress-relaxation properties of dynamic cross-linked elastomers conform to the Maxwell model, and the relaxation modulus can be described by the follow equation:

$$
E=E_{0} \mathrm{e}^{(-t / \tau)}
$$

where $E$ is the tensile stress relaxation modulus, $\tau$ is the relaxation time. The relaxation time $\tau$ could be determined as the time when $E / E_{0}=1 / \mathrm{e}$. Compared to traditional thermoset elastomers, the stress/modulus of dynamic-covalent cross-linked elastomers will relax to zero much faster. The relaxation times are controlled by the associative exchange reactions, and the temperature dependence of the relaxation time can be described by the following Arrhenius equation ${ }^{54}$ :

$$
\tau(\mathrm{T})=\tau_{0} \mathrm{e}^{(E \mathrm{a}, r / R T)}
$$

where $\tau(\mathrm{T})$ is the relaxation time at the temperature $T, \tau_{0}$ is a constant, $E_{\mathrm{a}, \mathrm{r}}$ is the relaxation activation energy and $R$ is the gas constant.

Table S1. The relative energy and Boltzmann distribution of the first 100 conformations with lower energy for chair/boat conformation

| chair conformation |  | boat conformation |  |  |  |
| :--- | ---: | ---: | :--- | ---: | ---: |
| RC label | rel. E (kJ/mol) | Boltzmann Dist | RC label | rel. E (kJ/mol) | Boltzmann Dist |
| M0001 | $\mathbf{0}$ | $\mathbf{0 . 1 7 9}$ | M0001 | $\mathbf{0}$ | $\mathbf{0 . 2 3 2}$ |
| M0002 | 1.31 | 0.105 | M0002 | 0.95 | 0.158 |
| M0003 | 1.78 | 0.087 | M0003 | 1.73 | 0.115 |
| M0004 | 2.23 | 0.073 | M0004 | 2.15 | 0.097 |
| M0005 | 2.58 | 0.063 | M0005 | 2.97 | 0.07 |
| M0006 | 2.66 | 0.061 | M0006 | 3.29 | 0.061 |
| M0007 | 2.76 | 0.059 | M0007 | 3.56 | 0.055 |
| M0008 | 3.07 | 0.052 | M0008 | 3.94 | 0.047 |
| M0009 | 3.9 | 0.037 | M0009 | 4.42 | 0.039 |
| M0010 | 3.99 | 0.036 | M0010 | 4.85 | 0.033 |
| M0011 | 4.31 | 0.031 | M0011 | 4.87 | 0.033 |
| M0012 | 4.78 | 0.026 | M0012 | 6.11 | 0.02 |
| M0013 | 4.81 | 0.026 | M0013 | 7.85 | 0.01 |
| M0014 | 5.37 | 0.02 | M0014 | 10.34 | 0.004 |
| M0015 | 5.43 | 0.02 | M0015 | 11.14 | 0.003 |
| M0016 | 14.27 | 0.55 | 0.13 | 0.001 | M0038 |


| M0040 | 15.16 | 0 | M0040 | 18.03 | 0 |
| :--- | ---: | ---: | :--- | ---: | ---: |
| M0041 | 15.18 | 0 | M0041 | 18.08 | 0 |
| M0042 | 15.37 | 0 | M0042 | 18.11 | 0 |
| M0043 | 15.55 | 0 | M0043 | 18.22 | 0 |
| M0044 | 15.67 | 0 | M0044 | 18.36 | 0 |
| M0045 | 15.67 | 0 | M0045 | 18.37 | 0 |
| M0046 | 15.77 | 0 | M0046 | 19.3 | 0 |
| M0047 | 16.14 | 0 | M0047 | 19.35 | 0 |
| M0048 | 16.17 | 0 | M0048 | 19.44 | 0 |
| M0049 | 16.18 | 0 | M0049 | 19.46 | 0 |
| M0050 | 16.5 | 0 | M0050 | 19.66 | 0 |
| M0051 | 16.58 | 0 | M0051 | 19.96 | 0 |
| M0052 | 17.03 | 0 | M0052 | 20.54 | 0 |
| M0053 | 17.2 | 0 | M0053 | 20.74 | 0 |
| M0054 | 17.44 | 0 | M0054 | 21.01 | 0 |
| M0055 | 18.02 | 0 | M0055 | 22.54 | 0 |
| M0056 | 18.03 | 27.61 | 0 | M0056 | 22.93 |
| M0057 | 18.5 | 0.17 | 0 | 0 | M0057 |


| M0083 | 28.46 | 0 | M0083 | 31.48 | 0 |
| :--- | :--- | :--- | :--- | ---: | ---: |
| M0084 | 28.52 | 0 | M0084 | 31.76 | 0 |
| M0085 | 29.32 | 0 | M0085 | 31.84 | 0 |
| M0086 | 29.88 | 0 | M0086 | 32.05 | 0 |
| M0087 | 30.01 | 0 | M0087 | 32.08 | 0 |
| M0088 | 31.19 | 0 | M0088 | 32.35 | 0 |
| M0089 | 31.82 | 0 | M0089 | 32.55 | 0 |
| M0090 | 31.87 | 0 | M0090 | 32.98 | 0 |
| M0091 | 33.84 | 0 | M0091 | 33.94 | 0 |
| M0092 | 34.24 | 0 | M0092 | 34.45 | 0 |
| M0093 | 35.92 | 0 | M0093 | 34.63 | 0 |
| M0094 | 37.49 | 0 | M0094 | 37.47 | 0 |
| M0095 | 37.88 | 0 | M0095 | 37.78 | 0 |
| M0096 | 38.44 | 0 | M0096 | 37.99 | 0 |
| M0097 | 39.89 | 0 | M0097 | 38.88 | 0 |
| M0098 | 40.51 | 0 | M0098 | 39.82 | 0 |
| M0099 | 40.63 | 0 | M0099 | 40.28 | 0 |
| M0100 | 41.64 | 0 | M0100 | 41.33 | 0 |

