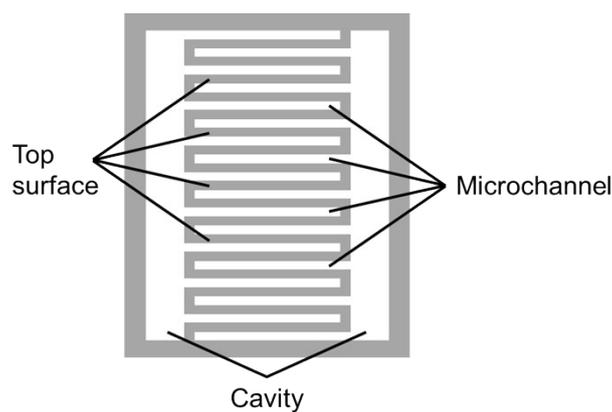


## Supplementary Material

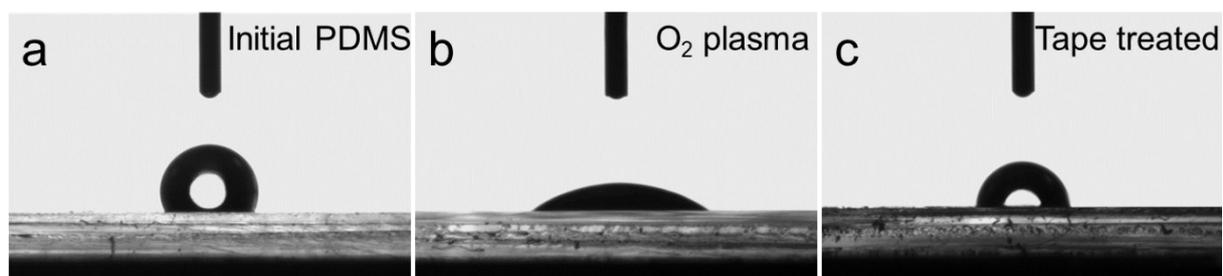
### Capillary force driven printing of asymmetric Na-ion micro supercapacitors

Jianmin Li, Yuanlong Shao\*, Chengyi Hou, Qinghong Zhang, Yaogang Li and Hongzhi Wang\*

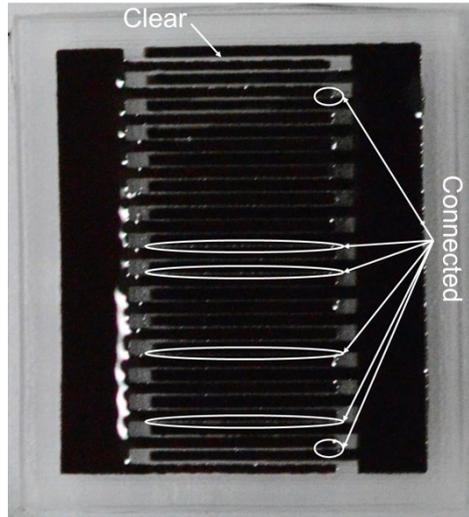
#### Figures



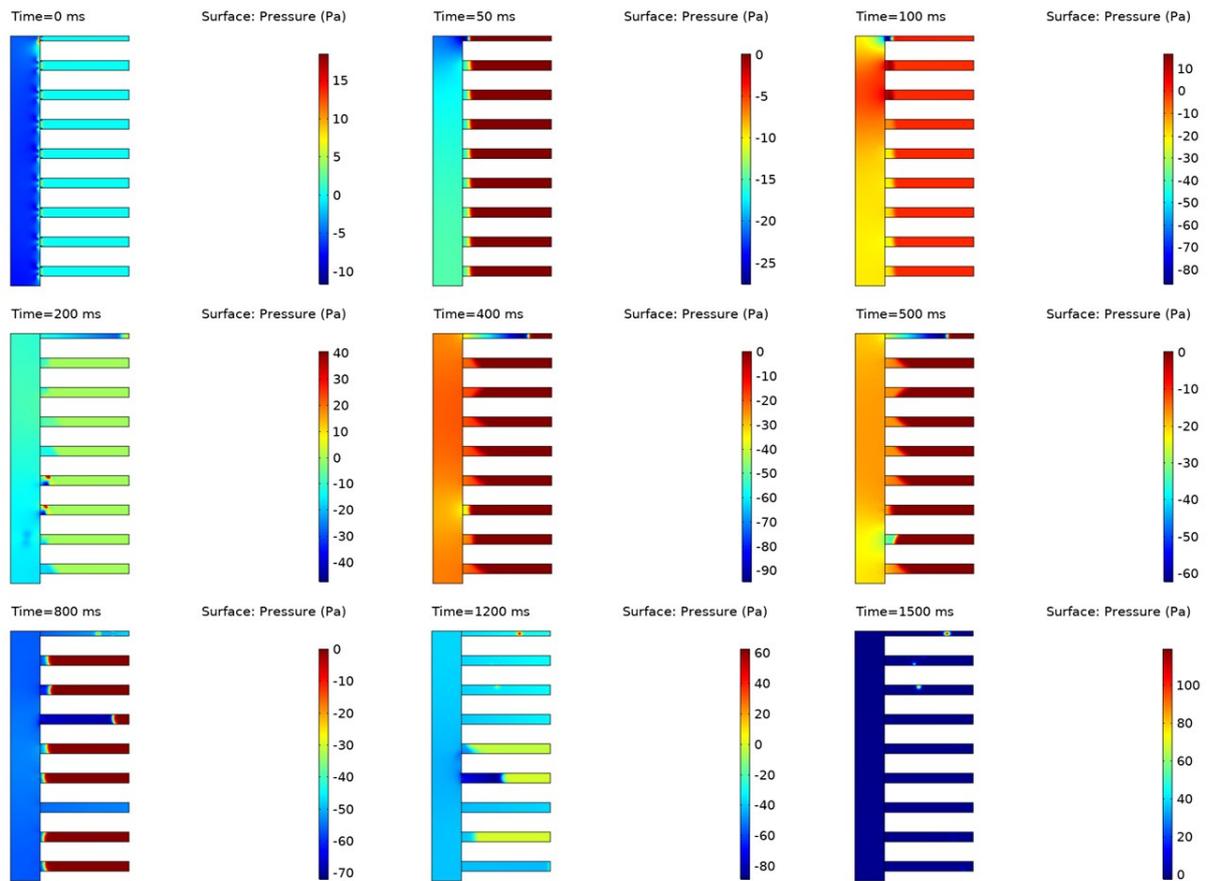
**Fig. S1** The schematic diagram of the PDMS template. The white part and gray part represent the microgroove and PDMS, respectively.



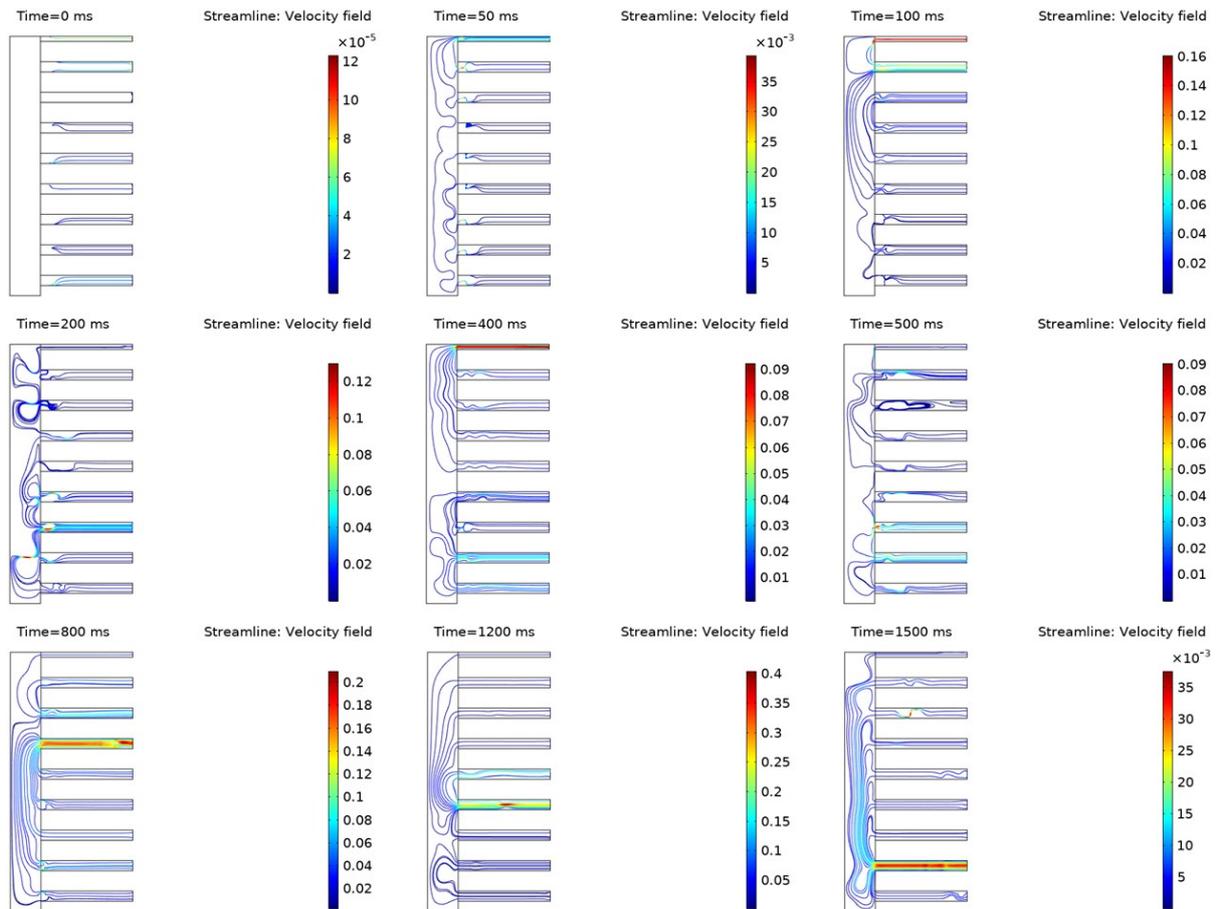
**Fig. S2** Contact angle tests of PDMS film at different surface conditions: the initial state (a), after O<sub>2</sub> plasma treating (b) followed by the tape treatment (c).



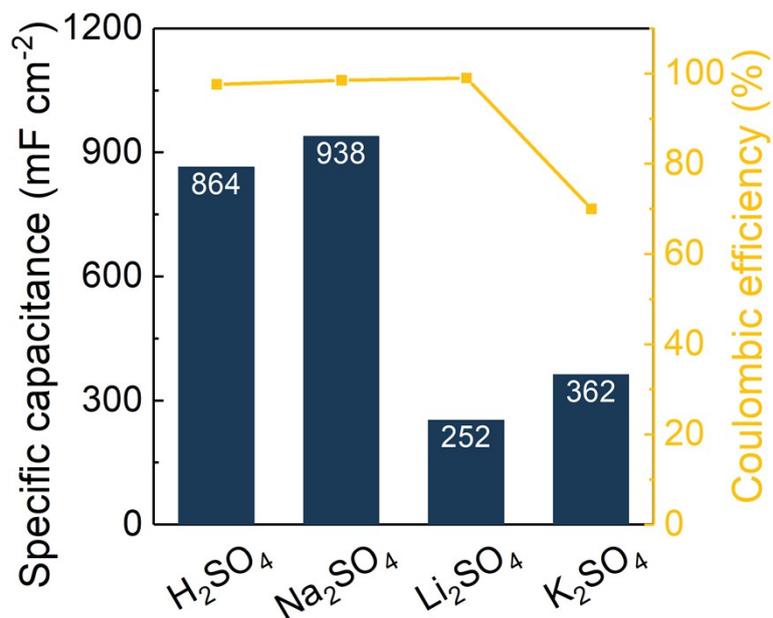
**Fig. S3** Digital image of the just assembled device without tape treatment. Most of the gaps are interconnected with each other.



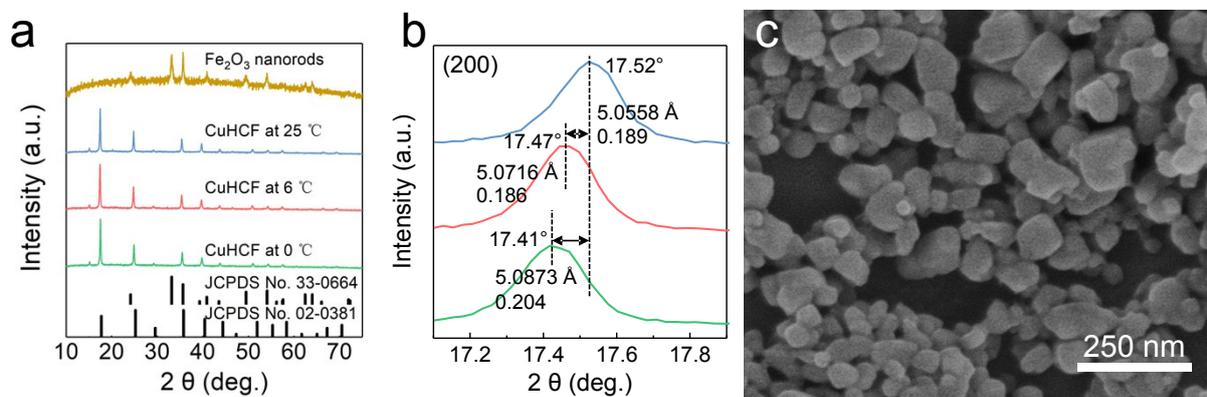
**Fig. S4** The pressure field distribution of the electrode ink in the capillary during the fabrication process.



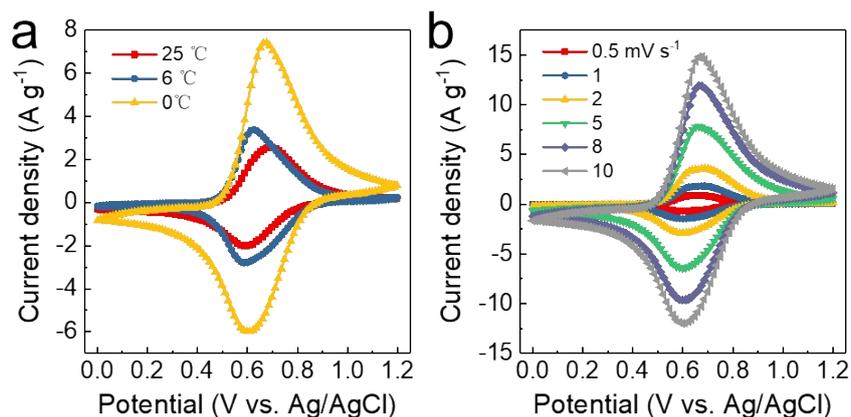
**Fig. S5** The velocity field distribution of the electrode ink in the capillary during the fabrication process.



**Fig. S6** Capacitance and coulombic efficiency of the SWCNTs/PEDOT:PSS/CuHCF film tested in different electrolytes.



**Fig. S7** XRD patterns of (a) Fe<sub>2</sub>O<sub>3</sub> NRs and CuHCF NPs synthesized at different temperatures, and (b) partially magnified (200) peak of the CuHCF NPs. (c) SEM image of the CuHCF NPs synthesized at 0 °C.



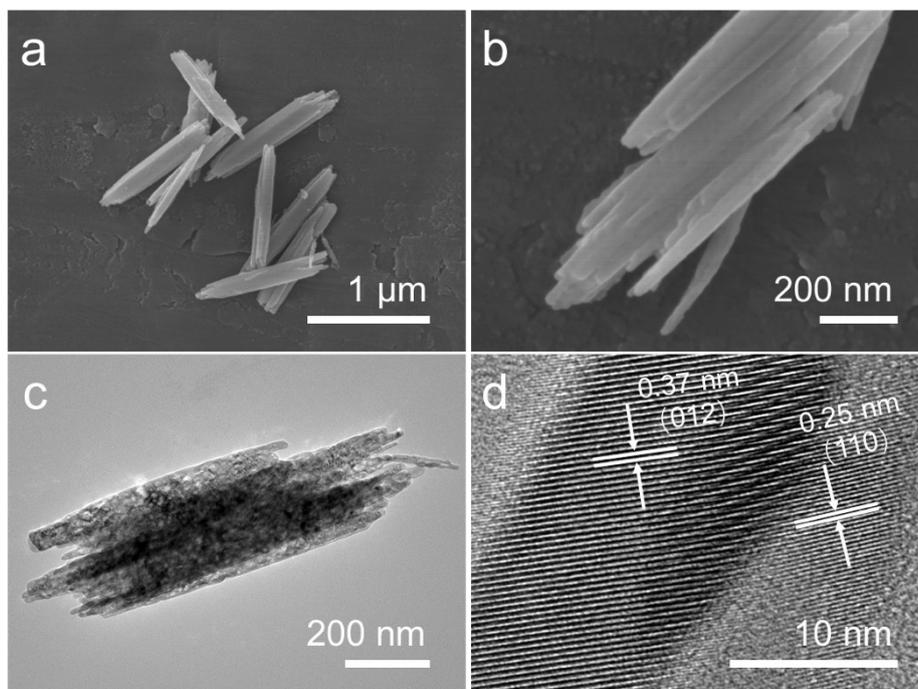
**Fig. S8** Cyclic voltammetry curves of the CuHCF NPs: (a) synthesized at different temperature, scan rate: 10 mV s<sup>-1</sup>; (b) synthesized at 0 °C under various scan rates.

To further enhance the electrochemical performance of the CuHCF NPs, we synthesized it at different temperatures (25°C, 6°C and 0°C), whose X-ray diffraction (XRD) patterns were shown in Fig. S7a. It is obvious that all of these three samples are well-crystallized cubic phase CuHCF, with the typical diffraction peaks located around 17.5, 25.2, 35.7 and 40.4°, indexing as (200), (220), (400) and (420) reflections (JCPDS No. 02-0381). We then magnified the (200) peak to ulteriorly evaluate the influence of temperature (see Fig. S7b), from which a left shift of the peak position is obvious as the temperature decreasing. Moreover, the interplanar spacing of these three samples were obtained by *Bragg equation*:

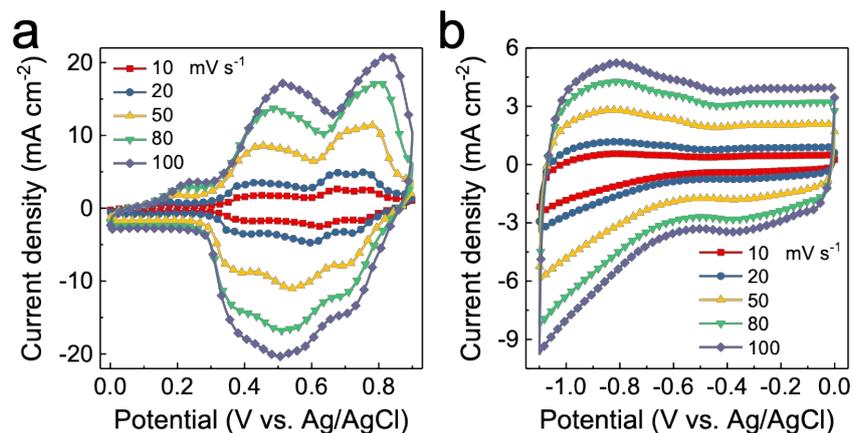
$$2d\sin\vartheta=n\lambda \quad (S1)$$

where  $d$  is the interplanar spacing,  $\vartheta$  is the diffraction angle,  $n$  is the diffraction series ( $n=1$ ),  $\lambda$  is the wavelength ( $\lambda=1.5406$  Å). Its interplanar spacing increased to 5.0873 Å at 0°C from 5.0558 Å at 25°C, which is beneficial for the hydrated Na<sup>+</sup> to insert. Fig. S7c exhibited the SEM image of the CuHCF NPs synthesized at 0°C, indicating the cubic morphology. As a result, the optimized electrochemical property was obtained for the CuHCF NPs synthesized at 0°C, as well as the outstanding rate capability (Fig. S8). Thus, the CuHCF NPs synthesized at 0°C was selected as the activated materials of positive electrode.

The XRD pattern of Fe<sub>2</sub>O<sub>3</sub> NRs is shown in Fig. S7a, from which the peaks located at 24.1, 33.2, 35.6 and 39.3° can be indexed as (012), (104), (110) and (006) reflections (JCPDS No. 33-0664) of hexagonal phase Fe<sub>2</sub>O<sub>3</sub>.



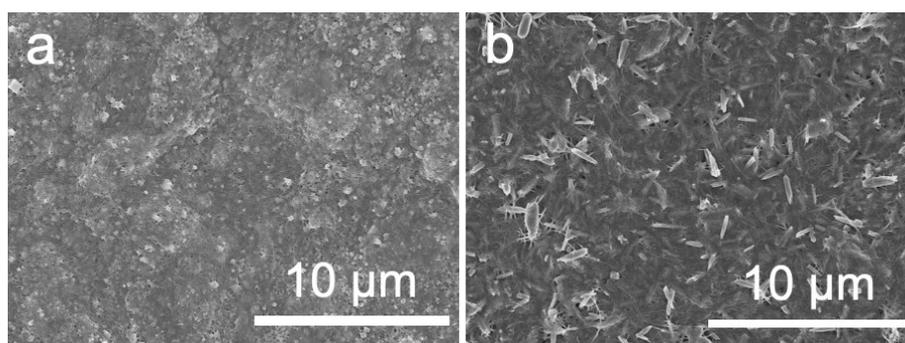
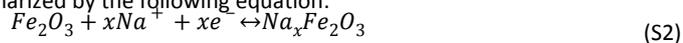
**Fig. S9** (a, b) SEM images, (c) TEM image and (d) HRTEM image of the as prepared Fe<sub>2</sub>O<sub>3</sub> NRs.



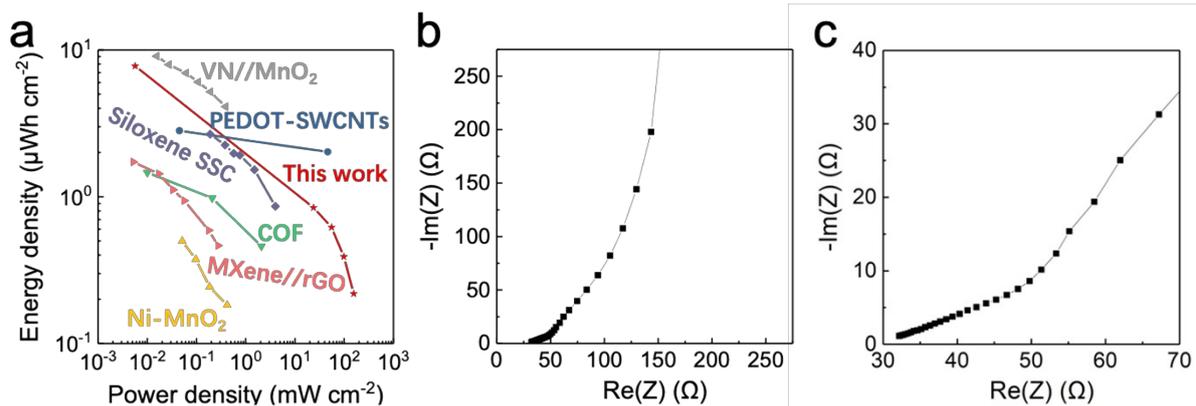
**Fig. S10** The cyclic voltammetry curves of the 2- $\mu\text{m}$  SWCNTs/PEDOT:PSS/CuHCF film (a) and SWCNTs/PEDOT:PSS/ $\text{Fe}_2\text{O}_3$  film (b).

Fig. S10a shows the CV curves (scan rate: 10-100  $\text{mV s}^{-1}$ ) of the as prepared SWCNTs/PEDOT:PSS/CuHCF film. It is obvious that the film showed a significant pseudocapacitive behavior, with two pairs of redox peaks, which are maintained even at a high scan rate of 100  $\text{mV s}^{-1}$ . Its specific capacitance was calculated to be 158  $\text{mF cm}^{-2}$  at 10  $\text{mV s}^{-1}$ , retaining 85.4% of its initial capacitance (135  $\text{mF cm}^{-2}$ ) under the scan rate of 100  $\text{mV s}^{-1}$ .

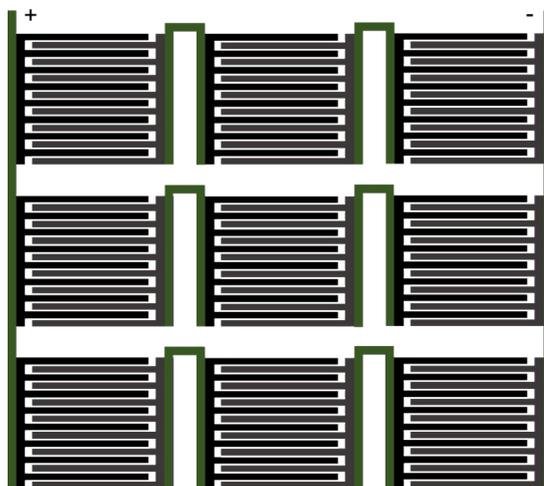
Also, we have prepared a SWCNTs/PEDOT:PSS/ $\text{Fe}_2\text{O}_3$  composite film, with a thickness of 2  $\mu\text{m}$ , to verify its potential to be used as the negative electrode of an AMSCs. From the CV curves shown in Fig. S10b, a wide voltage window of -1.1 V (-1.1~0 V) was clearly observed, as well as the intercalation-induced pseudocapacitive nature. It is striking that the ternary film can deliver a high areal specific capacitance of 77  $\text{F cm}^{-2}$  at scan rate of 10  $\text{mV s}^{-1}$ . Notably, as a result of its high conductivity, the ternary film could still maintain 67  $\text{F cm}^{-2}$  at 100  $\text{mV s}^{-1}$ , indicating a great rate performance. The energy storage mechanism could be summarized by the following equation:



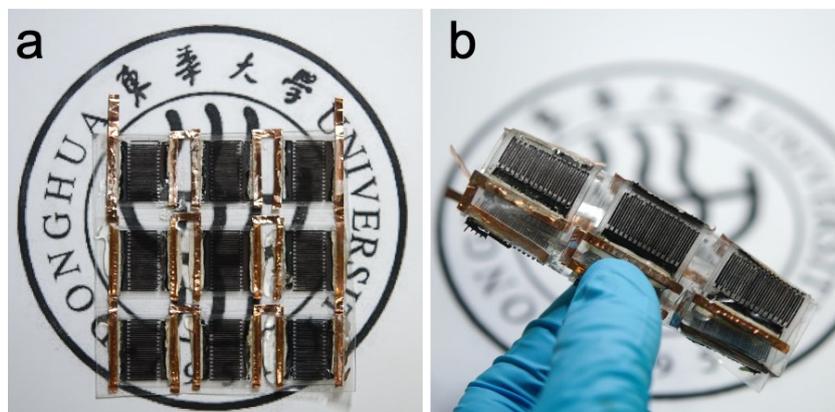
**Fig. S11** SEM images of positive electrode (a) and negative electrode (b).



**Fig. S12** Ragone plot (a), Nyquist plot (b), and enlarged Nyquist plot (c) of the as prepared AMSCs. SSC: symmetric supercapacitor, COF: Covalent Organic Frameworks, rGO: reduced graphene oxide.



**Fig. S13** The schematic diagram of the 9-device connected array.



**Fig. S14** Digital images of the 9-device connected array.