Supplementary Information

Enabling high-performance room-temperature sodium/sulfur batteries with few-layer 2H-MoSe₂ embellished nitrogen-doped hollow carbon spheres as polysulfide barrier

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Fig. S1 Low-magnification FESEM images of (a) SiO₂, (b) SiO₂@C, and (c) N-HCS.



Fig. S2 (a-c) FESEM and (d-f) TEM images of SiO₂, N-HCS, and S/2H-MoSe₂/N-HCS. Low-magnification FESEM images of (g) 2H-MoSe₂/N-HCS and (h) S/2H-MoSe₂/N-HCS.



Fig. S3 (a) STEM image of S/2H-MoSe₂/N-HCS. (b-f) Element distribution mappings of C, N, Mo, Se, and S.



Fig. S4 Schematic illustration of fabrication process of S/2H-MoSe₂/N-HCS.



Fig. S5 Fabrication procedure of 2H-MoSe₂/N-HCS/GO+GF and N-HCS/GO+GF separators.



Fig. S6 XRD pattern of S/2H-MoSe₂/N-HCSs.



Fig. S7 (a) TGA curves of S/2H-MoSe₂/N-HCS and S/N-HCS. (b) TGA curve of 2H-MoSe₂/N-HCS. Normally, when heating, $MoSe_2$ in air produces MoO_3 and SeO_2 due to the oxidation as follows:

 $2MoSe_2 + 7O_2 \rightarrow 4SeO_2 + 2MoO_3$

Afterward, the weight decreases due to the sublimation of SeO₂ and the removal of carbon by the oxidation.¹⁻⁵ The weight loss of 2H-MoSe₂/N-HCS is 59.8 wt%. 43.31 wt% weight loss for MoSe₂ is transformed into MoO₃, and 100 wt% weight loss for C is transformed into CO₂ in air.^{6,7} The accurate percentage of MoSe₂ in 2H-MoSe₂/N-HCS is calculated as follows: $M \times 59.8\% = M \times \alpha \times 43.31\% + M \times (1 - \alpha) \times 100\%$

M and α are corresponding to the weight of 2H-MoSe₂/N-HCS and the content of MoSe₂ in the composite, respectively. According to the formula, the content of MoSe₂ in 2H-MoSe₂/N-HCS is 70.6 wt%.

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Fig. S8 (a) N_2 adsorption-desorption isotherms and (b) pore size analysis of N-HCS, 2H-MoSe₂/N-HCS, and S/2H-MoSe₂/N-HCS.



Fig. S9 High-resolution XPS plot of C 1s in 2H-MoSe₂/N-HCS.



Fig. S10 High-resolution XPS plot of S 2p in S/2H-MoSe₂/N-HCS before cycling.



Fig. S11 CV curves of RT-Na/S batteries with (a) pristine GF and (b) N-HCS/GO+GF separators at the voltage range of 0.5 to 2.8 V (vs Na⁺/Na) and scan rate of 0.1 mV s⁻¹.



Fig. S12 Charge/discharge profiles of RT-Na/S batteries with (a) pristine GF and (b) N-HCS/GO+GF separators at 0.1 C.



Fig. S13 Capacity retention of RT-Na/S batteries with different separators at different current rates.



Fig. S14 Charge/discharge curves of RT-Na/S batteries with pristine (a) GF, (b) N-HCS/GO+GF, and (c) 2H-MoSe₂/N-HCS/GO+GF separators at various current rates.



Fig. S15 Cycling performance of $MoSe_2$ electrode at a current density of 0.5 C.



Fig. S16 (a) Low-magnification SEM image of 2H-MoSe₂/N-HCS/GO+GF separator. The corresponding elemental mappings in a selected region before cycling: (b) C, (c) N, (d) Se, and (e) Mo.



Fig. S17 (a) Low-magnification SEM image of 2H-MoSe₂/N-HCS/GO+GF separator after cycling. The corresponding elemental mappings: (b) C, (c) N, (d) Se, (e) Mo, (f) Na, and (g) S.



Fig. S18 Cycling performance of a RT-Na/S battery with a 2H-MoSe₂/N-HCS+GF separator at a current density of 0.5 C.



Fig. S19 Cycling performance of a RT-Na/S battery with a $MoSe_2/GO+GF$ separator at a current density of 0.5 C.



Fig. S20 Comparison of the electrochemical performance of RT-Na/S batteries with other reported literatures.

Fig. S21 CV curves of RT-Na/S batteries with (a) pristine GF and (b) N-HCS/GO+GF separators at different scan rates from 0.1 to 0.5 mV s⁻¹.

Fig. S22 Nyquist plots of RT-Na/S batteries with different separators (a) before cycling and (b) after ten cycles at 0.1 C (insets: the equivalent electrical circuits).

The electrochemical impedance spectroscopy was further generalized to analyze the solidstate ion diffusion behaviors by using Nyquist plots in the low-frequency region (Fig. S22b). Na-ion diffusion coefficient was calculated by the following equation:¹⁻³

$$D_{\rm Na}^{\ +} = R^2 \ T^2 / 2A^2 \ n^4 \ F^4 \ C_{\rm Na}^2 \ \sigma_{\omega}^2 \tag{1}$$

Where *R* is the gas constant, *T* represents the absolute temperature, *A* is the area of electrode, *n* is the number of the transferred electrons, *F* is the Faraday constant, and C_{Na} is the concentration of sodium-ion. Besides, σ_{ω} is Warburg coefficient, which is determined as the slope of Z' versus $\omega^{-1/2}$ (Fig. S23) in the low-frequency region based on the following equation:

$$Z' = R_e + R_{\rm ct} + \sigma_\omega \omega^{-1/2} \tag{2}$$

Where ω is the angular frequency, R_e is the resistance of electrolyte, and R_{ct} is the charge transfer resistance. Na-ion diffusion coefficients of RT-Na/S batteries with GF, N-HCS/GO+GF, and 2H-MoSe₂/N-HCS/GO+GF separators are 5.98 × 10⁻¹⁴, 2.0 × 10⁻¹³, and 2.84 × 10⁻¹³ cm² s⁻¹, respectively. D_{Na}^+ of a RT-Na/S battery with 2H-MoSe₂/N-HCS/GO+GF separator is higher than those of the other two RT-Na/S batteries with N-HCS/GO+GF and GF separators, indicating its improved rate capability (Fig. 4d).

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Fig. S24 SPSs permeation measurements in H-type cells with the pristine GF and N-HCS/GO+GF separators after different time.

Fig. S25 High-magnification SEM image of S/2H-MoSe₂/N-HCS after cycles.

Fig. S26 (a) XPS survey spectrum, (b) C 1s, (c) Mo 3d, (d) Se 3d, and (e) N 1s high-resolution XPS spectra of S/2H-MoSe₂/N-HCS after cycles.

Fig. S27 The optimized configurations and corresponding adsorption energies for binding of Na_2S , Na_2S_2 , Na_2S_4 , and Na_2S_6 to graphene and N-doped graphene.

Fig. S28 Partial density of states (PDOS) of Na₂S adsorbed on MoSe₂, including Na-*s*, S-*p*, Se-*p*, and Mo-*d* orbitals.

Fig. S29 Cycling performance of RT-Na/S batteries with 2H-MoSe₂/N-HCS/GO+GF separator at high sulfur loadings at 0.5 C.

Fig. S30 Photograph of a LED array panel with a "JLU" pattern lighted by a RT-Na/S battery with a 2H-MoSe₂/N-HCS/GO+GF separator.

Table S1 Comparison of the electrochemical performance between this work and other recently reported literatures for RT-Na/S batteries.

| Cathode material | Current density (mA g ⁻¹) | Initial capacity (mAh g ⁻¹) | Final capacity (mAh g ⁻¹) | Cycle number (N) | Sulfur content (wt%) | Ref. |
|---------------------------------------|---|---|---|------------------------|----------------------------|--------------|
| CS90-rGO | 200 | 650 | 498 | 50 | ~90 | S1 |
| MoS ₃ | 450 | 460 | ~180 | 1000 | 45 | S2 |
| CNF/AC-Na ₂ S ₆ | 335 | 750 | 550 | 100 | 50 | S3 |
| S@Co _n -HC | 100 | 1081 | 508 | 600 | 48 | S4 |
| S@iMCHS | 100 | 1215 | 292 | 200 | 46 | S5 |
| S@CNT/NPC | 837.5 | 601 | 410 | 500 | 43 | S6 |
| cZIF-8/S | 335 | 873 | 500 | 250 | 50 | S7 |
| S@Fe-HC | 100 | 1023 | 394 | 1000 | 40 | S8 |
| HSMC-Cu-S | 50 | 710 | 610 | 110 | 50 | S9 |
| FeS ₂ @NCMS/S | 100 | 1471 | 524 | 300 | 65.5 | S10 |
| Microporous carbon/sulfur | 837.5 | 1200 | 395 | 500 | 33.4 | S11 |
| CFC/S-2 | 167.5 | 360 | 120 | 300 | 24.4 | S12 |
| N, S-HPC/S | 230 | 381 | 378 | 350 | 22 | S13 |
| S/2H-MoSe ₂ /N- HCS | 837.5 | 1945 | 484 | 500 | 71.4 | This work |

*Results summarized from Fig. S20 in Supporting Information.

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Table S2 The values of R_{ct} of RT-Na/S batteries with different separators before cycling and after cycles.

| Separator | $R_{ct}(\Omega)$ Before cycling | $R_{ct}(\Omega)$ After ten cycles |
|-----------------------------------|---------------------------------|--------------------------------------|
| 2H-MoSe ₂ /N-HCS/GO+GF | 41.32 | 49.25 |
| N-HCS/GO+GF | 66.51 | 95.97 |
| GF | 112.41 | 221.32 |