

## Supporting Information

### **Ultrahigh capacity and cyclability of dual-phase TiO<sub>2</sub> nanowires with low working potential at room and subzero temperatures**

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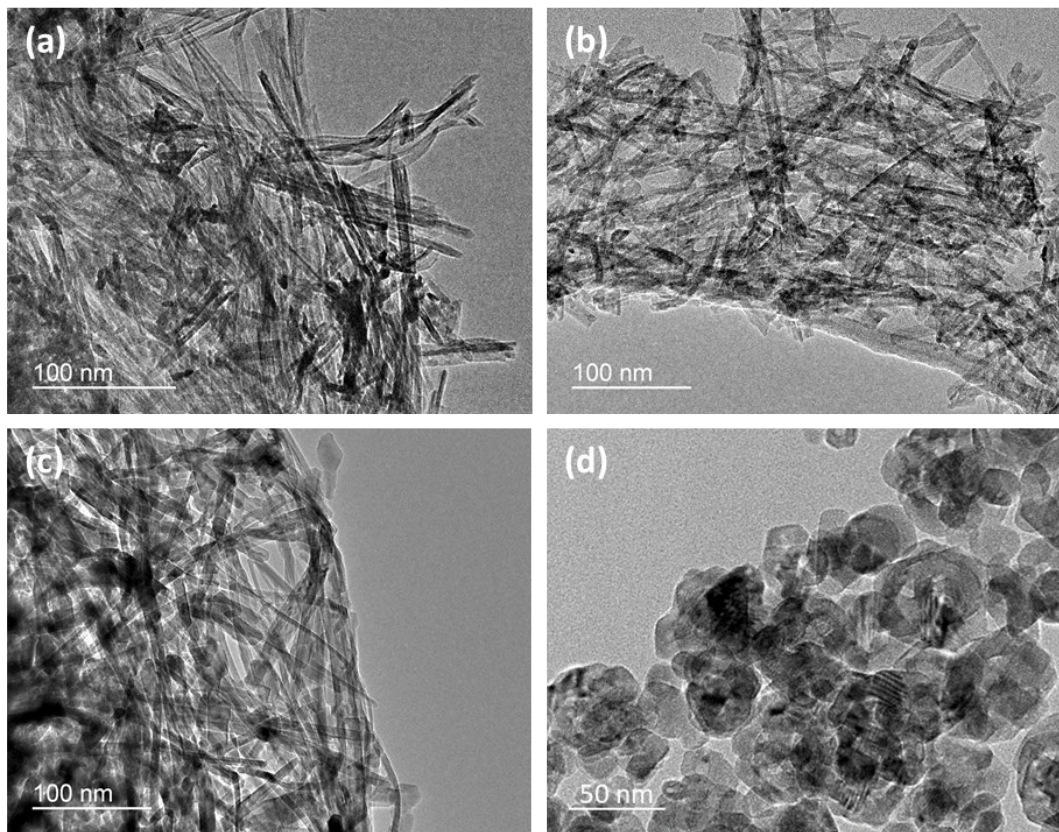
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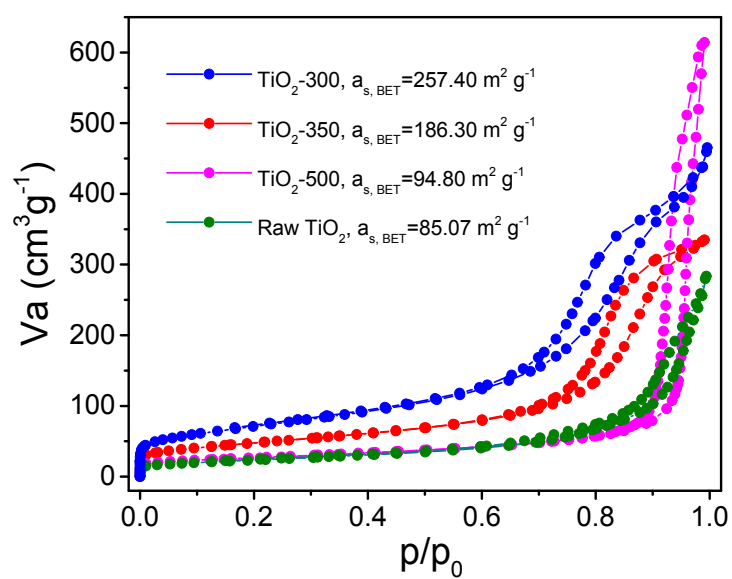
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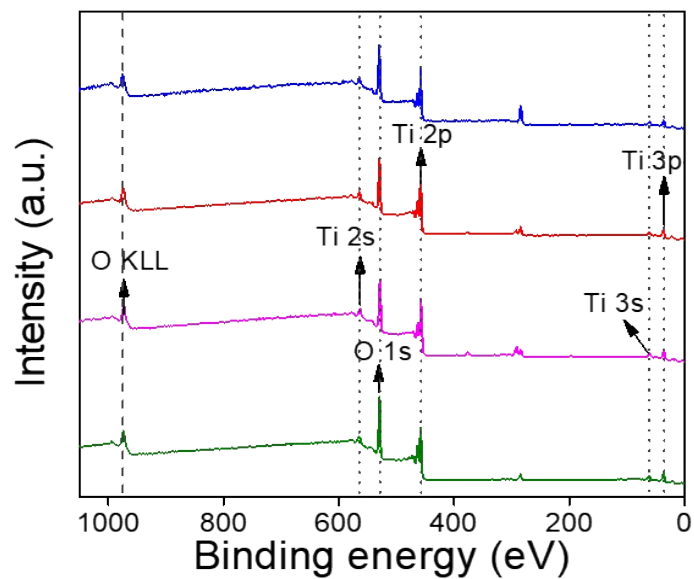
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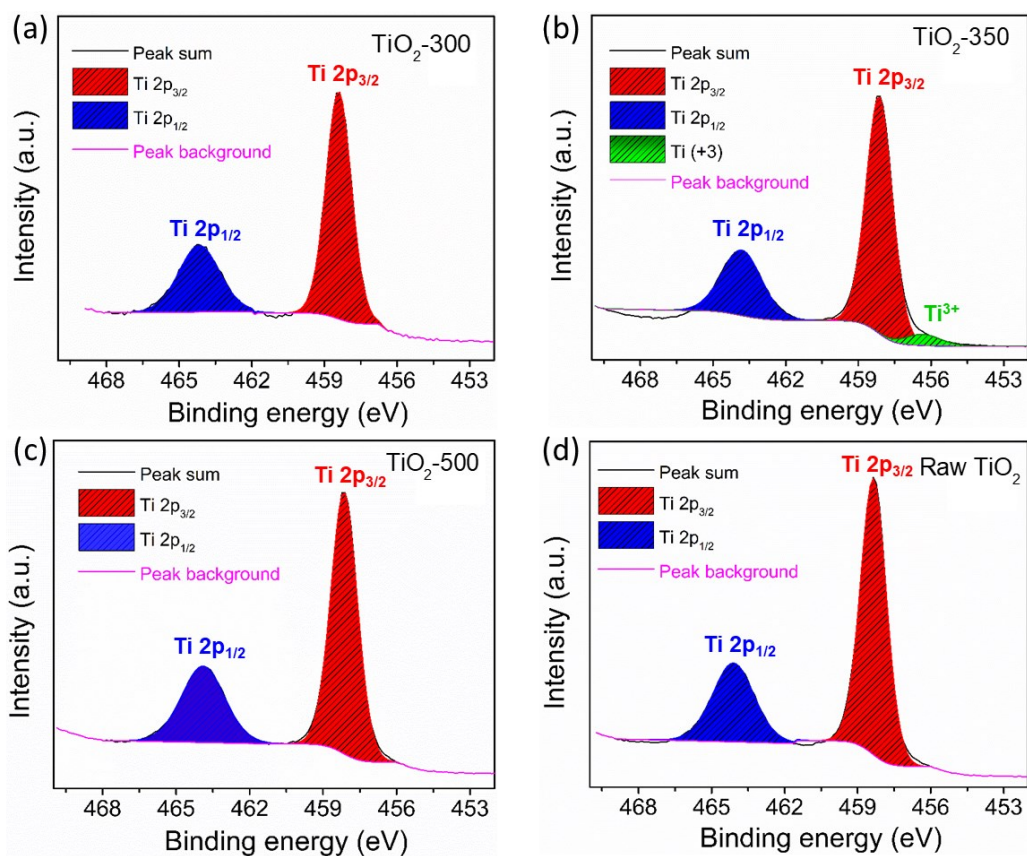
**Fig. S1.** TEM images of TiO<sub>2</sub>-300, TiO<sub>2</sub>-350, TiO<sub>2</sub>-500 nanowires, and raw TiO<sub>2</sub> nanoparticles, respectively.



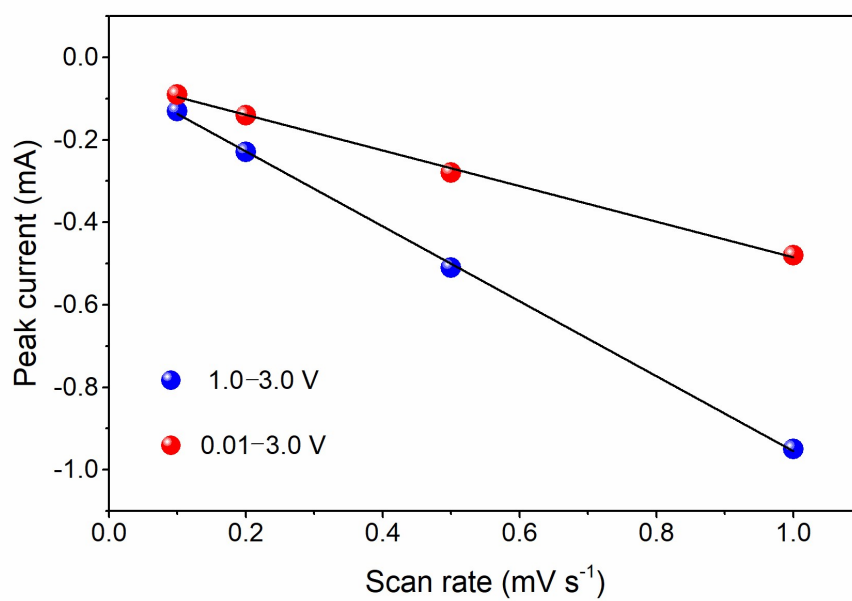
**Fig. S2.** Nitrogen adsorption/desorption isotherms of TiO<sub>2</sub>-300, TiO<sub>2</sub>-350, TiO<sub>2</sub>-500, and raw TiO<sub>2</sub> nanoparticles, respectively.



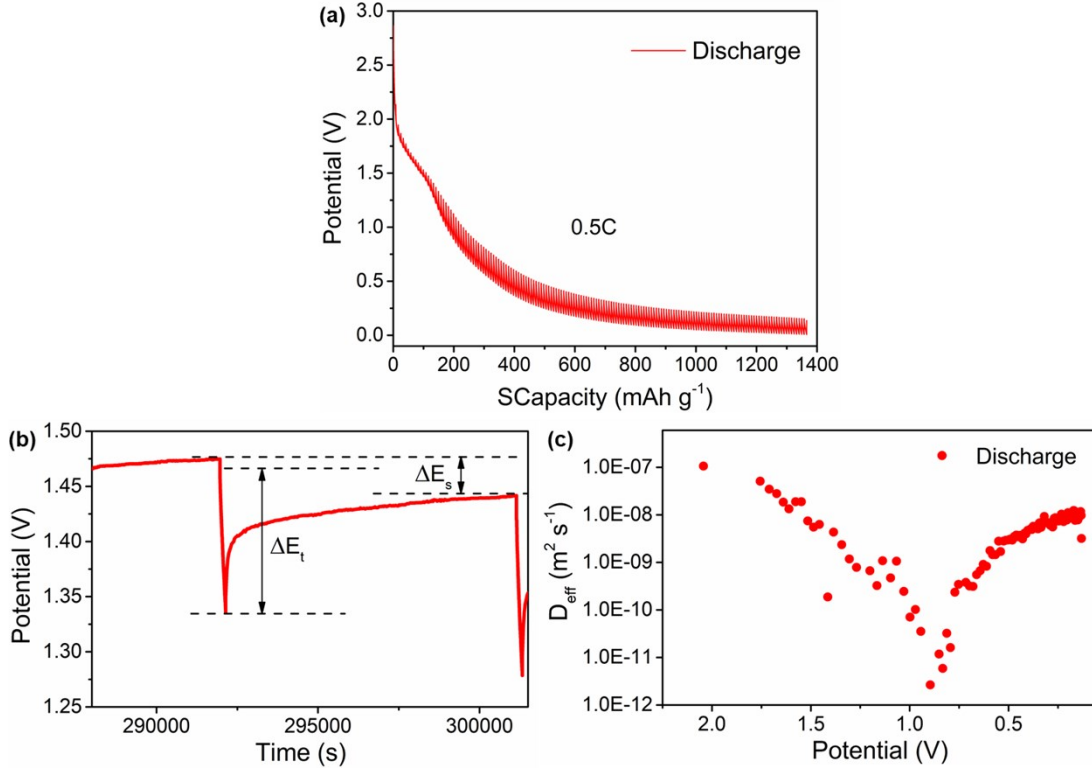
**Fig. S3.** XPS spectra of TiO<sub>2</sub>-300, TiO<sub>2</sub>-350, TiO<sub>2</sub>-500, and raw TiO<sub>2</sub> nanoparticles, respectively.



**Fig. S4.** XPS Ti 2p spectra of (a) TiO<sub>2</sub>-300, (b) TiO<sub>2</sub>-350, (c) TiO<sub>2</sub>-500 nanowires, and (d) raw TiO<sub>2</sub> nanoparticles.



**Fig. S5.** The relationship between the peak current and the scan rate of the TiO<sub>2</sub>-B phase in the TiO<sub>2</sub>-350 anode in the two potential windows.



**Fig. S6.** (a) GITT voltage profiles and equilibrium potentials during discharge in the enlarged potential window at 0.5C rate. (b) Voltage profile for a single titration of the GITT during discharge. (c) The apparent Li diffusion coefficient  $D_{app}$  at various states of discharge determined from the GITT voltage profiles.

We calculated the apparent Li diffusion coefficients  $D_{app}$  by assuming that  $\text{Li}^+$  diffusion obeys Fick's second law:

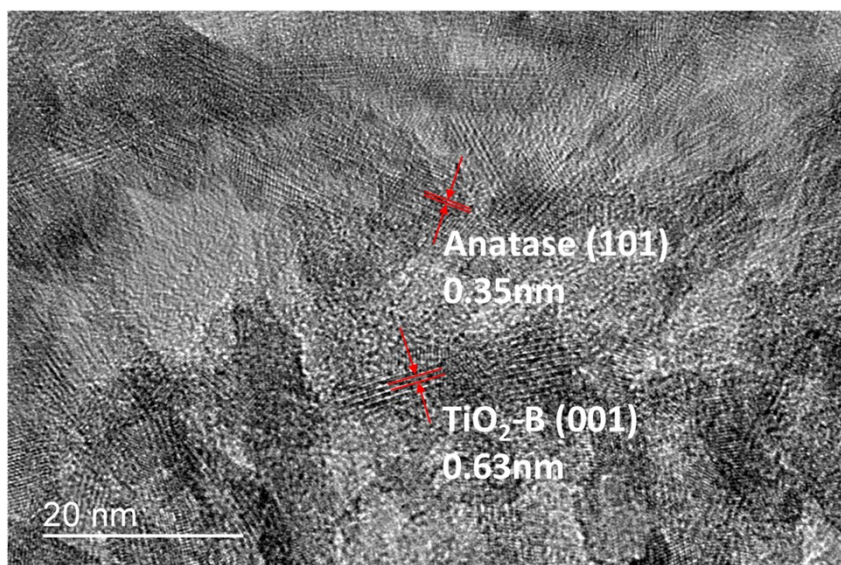
$$D_{app} = \frac{4}{\pi\tau} \left( \frac{m_B V_M}{M_B S} \right)^2 \left( \frac{\Delta E_s}{\Delta E_t} \right)^2 \quad (1)$$

where  $\tau$  donates current pulse time,  $m_B$ ,  $V_M$ , and  $M_B$  are the mass, molar volume and atomic weight of the electrode material  $\text{TiO}_2$ , respectively, and  $S$  presents the total contact area between the electrolyte and the electrode.

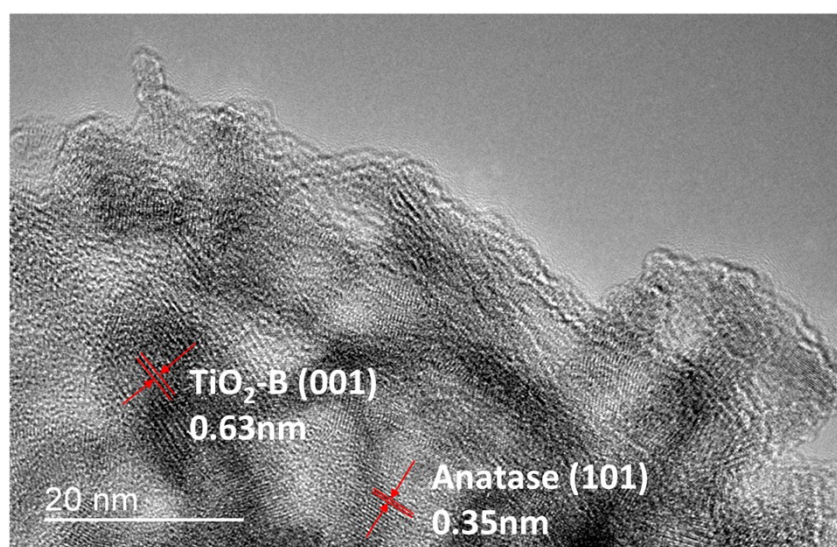
Before the GITT measurement, the cell was galvanostatic cycled at 0.5 C for 5 cycles between 0.01 V to 3 V.

During the GITT measurement, the cell was discharged at the rate of 0.5C (1 C = 335  $\text{mA g}^{-1}$ ) for 3 min, then followed by a relaxation for 150 min to reach the steady-state voltages. The potential window of GITT measurement ranges from 0.01 V to 3 V.

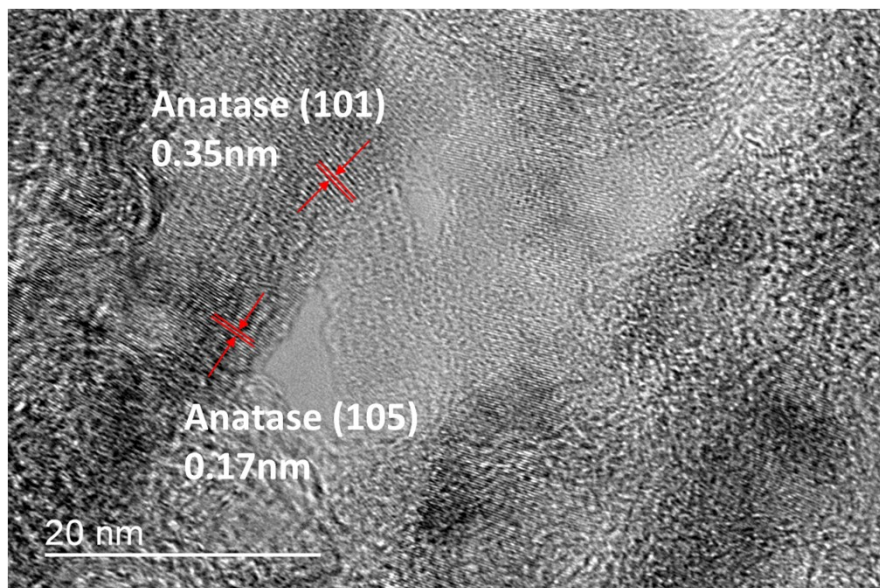




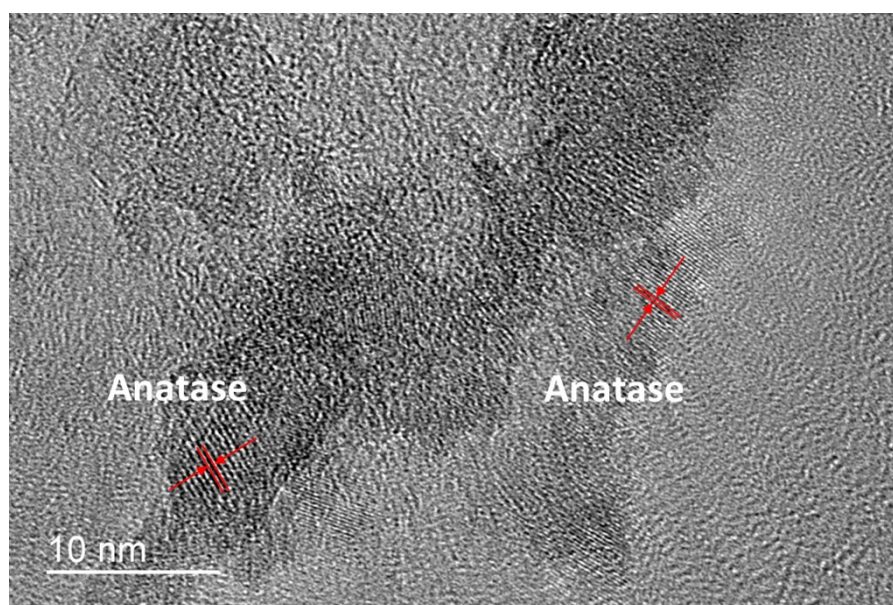
**Fig. S7.** HRTEM image of the TiO<sub>2</sub>-350 anode discharged from OCP to 1.0 V.



**Fig. S8.** HRTEM image of the TiO<sub>2</sub>-350 anode discharged from OCP to 0.5 V.

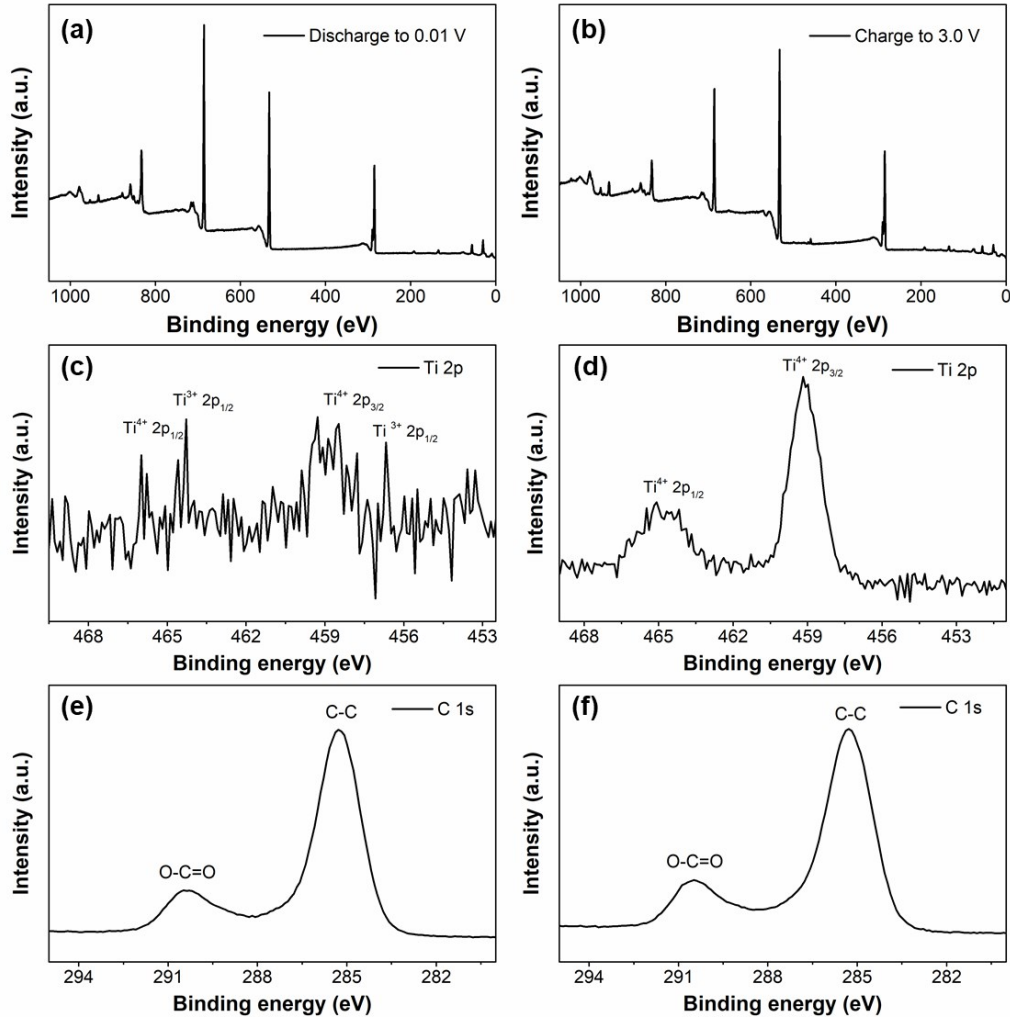


**Fig. S9.** HRTEM image of the TiO<sub>2</sub>-350 anode after cycling 100 cycles in the potential window of 0.01–3.0 V.



**Fig. S10.** HRTEM image of the TiO<sub>2</sub>-350 anode after cycling 500 cycles in the potential window of 0.01–3.0 V.





**Fig. S11.** Ex-situ XPS spectra of the TiO<sub>2</sub>-350 anode when (a) discharged to 0.01 V, and (b) charged to 3.0 V. High resolution spectra of Ti element when (c) discharged to 0.01 V, and (d) charged to 3.0 V, and high resolution spectra of C element when (e) discharged to 0.01 V, and (f) charged to 3.0 V, respectively.

Ex-situ XPS results show that the typical Ti<sup>4+</sup> 2p<sub>1/2</sub> and Ti<sup>4+</sup> 2p<sub>3/2</sub> peaks significantly diminish while Ti<sup>3+</sup> 2p<sub>3/2</sub> peak appears after discharging to 0.01 V, indicating the reduction of Ti<sup>4+</sup> to Ti<sup>3+</sup> which accounts for the ultrahigh capacity. When charging to 3.0 V, the characteristic Ti<sup>3+</sup> 2p<sub>3/2</sub> peak disappears while both Ti<sup>4+</sup> 2p<sub>1/2</sub> and Ti<sup>4+</sup> 2p<sub>3/2</sub> peaks recover. In addition, there is no other side reaction like formation of Ti-C bonding occurs during charging/discharging process. However, even assuming that all Ti<sup>4+</sup> ions reduce into Ti<sup>3+</sup> in the discharging process, this cannot contribute to such high capacity. Furthermore, it does not occur the reduction of Ti<sup>4+</sup> to Ti<sup>0</sup> at the end of discharging, indicating that the additional capacity is also not from the conversion reaction of Ti<sup>4+</sup> to Ti<sup>0</sup>. Herein, we propose that the additional capacity mainly derives from two parts: the intercalation lithiation and surface/interface Li ion storage behavior in the dual-phase TiO<sub>2</sub> nanowires.



**Table S1.** Initial CEs of anodes in the two potential windows of 1.0–3.0 V and 0.01–3.0 V, respectively.

<b>Anodes</b>	<b>Initial CEs in the potential window of 1.0–3.0 V (%)</b>	<b>Initial CEs in the potential window of 0.01–3.0 V (%)</b>
<b>TiO<sub>2</sub>-300</b>	33.03	31.51
<b>TiO<sub>2</sub>-350</b>	32.25	30.52
<b>TiO<sub>2</sub>-500</b>	36.31	33.56
<b>Raw TiO<sub>2</sub></b>	33.86	34.83

## References

- 1 W. Weppner and R. A. Huggins, J. Electrochem. Soc., 1977, 124, 1569-1578.