# Eshelby Untwisting 

Xiaodi Zhong, Hengyu Zhou, Chao Li, Alexander G. Shtukenberg, Michael D. Ward,* Bart Kahr* Department of Chemistry and the Molecular Design Institute, New York University, New York City, New York 10003, United States

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## References

Crystallography. Crystallographic information files (CIF) for benzil were retrieved from the Cambridge Structural Database.

- $P 3_{1} 21$ space group: CSD REFCODE BENZIL05; Deposition Number 1108768; M. More, G. Odou and J. Lefebvre, Acta Crystallogr. Sect. B: Struct. Sci., 1987, 43, 398.
- $\quad P 3_{2} 21$ space group: CSD REFCODE BENZIL06; Deposition Number 1859205; V. P. Charpe, A. Sagadevan and K. C. Hwang, Green Chem., 2020, 22, 4426, DOI: 10.1039/D0GC00975J.

Simulation parameters. Partial charges, atomic masses, lattice parameters and force field parameters are the same with a previous report from are laboratory describing twisting crystals in silico (see: C. Li, A. G. Shtukenberg, E. Efrati, L. Vogt-Maranto, P. Raiteri, J. D. Gale, A. L. Rohl and B. Kahr, J. Phys. Chem. C. 2020, 124, 15616-15624). Partial charges and atomic masses used here are contained in the tables below. Crystal lattice parameters for benzil were obtained from the CSD structures above.

Table S1. Partial atomic charges and masses for benzil

|  | Atom type | Charge | Mass (amu) |
| :--- | :--- | :--- | :--- |
| C1 | C | 0.4095 | 12.010 |
| C2 | CA | -0.0427 | 12.010 |
| C3 | CA | -0.1065 | 12.010 |
| C4 | CA | -0.1378 | 12.010 |
| C5 | CA | -0.1040 | 12.010 |
| C6 | CA | -0.1378 | 12.010 |
| C7 | CA | -0.1065 | 12.010 |
| O1 | O | -0.4459 | 16.000 |
| H1 | H | 0.1337 | 1.008 |
| H2 | H | 0.1342 | 1.008 |
| H3 | H | 0.1359 | 1.008 |
| H4 | H | 0.1342 | 1.008 |
| H5 | H | 0.1337 | 1.008 |
| C1C | C | 0.4095 | 12.010 |
| C2C | CA | -0.0427 | 12.010 |
| O1C | O | -0.4459 | 16.000 |
| C3C | CA | -0.1065 | 12.010 |
| C7C | CA | -0.1065 | 12.010 |
| C4C | CA | -0.1378 | 12.010 |
| H1C | H | 0.1337 | 1.008 |
| C6C | CA | -0.1378 | 12.010 |
| H5C | H | 0.1337 | 1.008 |
| C5C | CA | -0.1040 | 12.010 |
| H2C | H | 0.1342 | 1.008 |
| H4C | H | 0.1342 | 1.008 |


| H3C | H | 0.1359 | 1.008 |
| :--- | :--- | :--- | :--- |

Table S2. Force field parameters
S.2.1. Covalent Bond Parameters

| I-J | $\underline{\underline{k}}_{r}$ | $\underline{r}_{\text {eq }}$ |
| :--- | :--- | :--- |
| CA-CA | 20.7455 | 1.3870 |
| C-CA | 15.1645 | 1.4870 |
| C-O | 28.1001 | 1.2140 |
| C-C | 12.5800 | 1.5500 |
| CA-H | 14.9304 | 1.0870 |

S.2.2. Covalent Angle Parameters

| I-J-K | $\mathrm{k}_{\theta}$ | $\theta_{\text {eq }}$ |
| :--- | :--- | :--- |
| CA-CA-CA | 2.9132 | 119.9701 |
| C-CA-CA | 2.8031 | 120.1401 |
| CA-C-O | 2.9778 | 123.4401 |
| C-C-CA | 2.6769 | 118.3401 |
| C-C-O | 2.9124 | 120.9901 |
| CA-CA-H | 2.1014 | 120.0101 |

S.2.3. Covalent Dihedral Parameters

| I-J-K-L | $k_{1}$ | $\mathrm{k}_{2}$ | $\mathrm{k}_{3}$ |
| :--- | :--- | :--- | :--- |
| CA-CA-CA-CA | 0 | 0.1572 | 0 |
| C-CA-CA-CA | 0 | 0.1572 | 0 |
| CA-CA-C-O | 0 | 0.0434 | 0 |
| C-C-CA-CA | 0 | 0.0434 | 0 |
| CA-C-C-O | 0 | 0.0130 | 0 |
| CA-C-C-CA | 0 | 0.0130 | 0 |
| O-C-C-O | 0 | 0.0130 | 0 |
| CA-CA-CA-H | 0 | 0.1572 | 0 |
| H-CA-CA-H | 0 | 0.1572 | 0 |
| C-CA-CA-H | 0 | 0.1572 | 0 |

S.2.4. Covalent Improper Dihedral Parameters

| I-J-K-L | k | d | n |
| :--- | :--- | :--- | :--- |
| CA-C-CA-CA | 0.0477 | -1 | 2 |
| C-C-CA-O | 0.4553 | -1 | 2 |
| CA-CA-CA-H | 0.0477 | -1 | 2 |

S.2.5. Pair Potentials

| $\mathrm{I} / \mathrm{J}$ | $\varepsilon$ | $\sigma$ |
| :--- | :--- | :--- |
| CA | 0.0037 | 3.3997 |
| $H$ | 0.0007 | 2.5996 |
| C | 0.0037 | 3.3997 |
| 0 | 0.0091 | 2.9599 |

## Simulation details

Hexagonal rods of benzil (space groups $P 3_{1} 21$ and $P 3_{2} 21$ ) for simulations were constructed from the CIF files by Python codes, with lattice parameters determined at 100 K fixed at $a=b=8.356$ (3) Å and $c=13.375(6)$ Å. The sizes of the rods were 15 unit cells along the $c$ axis, and $4,6,8$, or 10 unit cells along $a$. Each unit cell was assigned a unique set of coordinates, and screw dislocations of Burgers vector $\mathbf{b}=<000 \pm 1>$ were generated by moving the coordinates of the unit cells on a chosen shear plane from 0 to $\mathbf{b}$. This dislocation core associated with this displacement extended evenly over 3 unit cells on that shear direction for all rods, regardless of rod size. For a righthanded screw dislocation, molecules outside the dislocation core were raised uniformly along [0001] from zero to $b$ when rotating by $2 \pi$ around the core, producing shears in the ( $2 \overline{1}^{1} 0$ ) or (1 $0 \overline{1}_{0}$ ) planes. The magnitude of the displacement, $z$, of the molecules along $c$ was calculated as:

$$
z=\frac{b}{2 \pi} \tan ^{-1} \frac{y}{x}
$$

Geometry optimizations and MD simulations using the Generalized Amber Force Field (GAFF) ${ }^{1,2}$ were conducted with the Large-scale Atomic/Molecular Massively Parallel Simulator (LAMMPS) code. ${ }^{3}$ The benzil nanorods were located in the centre of a simulation box, and the distances between the crystal surfaces and the box boundaries were > $15 \AA$ in orthogonal directions. Partial atomic charges were derived from a restrained electrostatic potential (RESP) calculation with the HF/6-31G* basis set using the R.E.D. web server. ${ }^{4,5,6,7}$ The General Utility Lattice Program (GULP) was used to calculate the improper four-body interaction torsional potentials with the introduction of equilibrium values in the angular dependencies torsional angle. ${ }^{8,9}$ Mechanical distortions were in the non-activated, energy barrier-free processes. Energy minimisation was achieved by conjugate gradient minimization with a relative energy tolerance of $10^{-8}$. Relaxation was sped by iterative cycles of geometry optimization, followed by Molecular Dynamics (MD) performed at 200 K with the NVE microcanonical ensemble controlled by a Langevin thermostat
with a timestep of 1 fs . The cut-off distance for pairwise Lennard Jones interactions was set to $12.5 \AA$, while the electrostatic interactions beyond $12.5 \AA$ were calculated in reciprocal space using the particle-particle particle-mesh $k$-space method. Trajectories were recorded every 500 fs for periods of 5 ns until the twisting angle was unchanging within <2\% over a period of 0.2 ns . Twisting angle calculations, potential energy visualizations and torsion angle statistics were completed using Python codes.

The LAMMPS input file and Python codes for creating crystals from CIF, calculating twisting angles, potential energy visualization, and torsion angle analysis can be found on https://github.com/hz2134-chem/KahrWardGroup.

## Calculation of the twist angles

Benzil molecules on the corners and edges of the crystals are more poorly coordinated. Calculation of twisting angles were based on molecules inside the crystal. Molecules on the surface were excluded. Unit cells within layers positioned at $1 / 4$ and $3 / 4$ along $c$ of the crystal were selected. For example, as in in Figure S1Figure-S1, the 4th layer and the 12th layer were selected, in a crystal with a height of 15 unit cells. In these two layers, the vectors (such as $v_{1}$ and $v_{2}$ ) connecting the third unit cells from the outermost surface (unit cells highlighted in Figure S1Figure S1), along six directions of the hexagonal cross section, were used for calculation of twisting angles in each frame. These angles were averaged over 0.1 ns , at the last stages of simulation when twisting angles were stabilized.


Figure S1. Scheme of calculation of twisting angles. Two layers on the right are extracted from the crystal positioned at $1 / 4$ and $3 / 4$ along $c$. Unit cells colored in yellow are used for calculating twisting angles, between matched vectors (otherwise translationally related) on two layers divided by the distance between the center of mass of two vectors along axial direction.


Figure S2. Representations of a $P 3_{2} 21$ benzil nanocrystal rod (dimensions $6 \times 6 \times 15$ unit cells) containing a righthanded dislocation obtained with molecular dynamics simulations at different temperatures, as viewed down the crystallographic c axis (upper row) and roughly perpendicular to the caxis (lower row). The corresponding Eshelby twisting angles (bottom) increase with increasing temperature. The twisting angle during the simulations at 10 K and 100 K did not change appreciably from the values observed just after the simulation was initiated, indicating that the twisting angle in each frame was near zero. Conversely, the twisting angles increase throughout the simulations at 200 K and 300 K , validating an Eshelby twist. At higher temperatures, the surface morphology became highly disordered, signifying melting and a reduction of crystalline order.


Figure S3. Twisting trajectories for $P 3_{2} 21$ benzil nanorods ( $6 \times 6 \times 15$ unit cells) with a right-handed dislocation, at different temperatures. At low temperatures ( 10 K and 100 K ), twisting is negligible and unchanging throughout the simulation, consistent local minima that frustrate mechanical deformation.


Figure S4. Twisting of $P 3_{2} 21$ benzil nanocrystal rods ( $6 \times 6 \times 15$ unit cells) with heterochiral shears in the ( $1 \overline{2} 10$ ) planes containing the [ ${ }^{10 \overline{1} 0}$ ] vector. The twisting angles for the rods with this dislocation are 0.22 deg $\AA^{-1}$ for the left-handed dislocation (left) and -0.060 deg $\AA^{-1}$ for the right-handed dislocation (right), both similar to the twisting angles of rods of same size and same space group sheared along the [ $2 \overline{1} \overline{1} 0]$ direction on a $(01 \overline{1} 0)$ plane.


Figure S5. Potential energy of benzil nanorods of size $6 \times 6 \times 15$, space group $P 3_{2} 21$, viewed along $c$. Left-handed dislocation (A) and right-handed dislocation (B). Vectors in shear planes are indicated in red lines. Models shown correspond to shear containing horizontal [ $2 \overline{1} \overline{1} \overline{1}_{0}$ ] vectors.




Figure S6. (top) Schematic representation of the three outermost layers of a benzil nanocrystal rod ( $6 \times 6 \times 15$ unit cells), as viewed down the $c$ axis. The layers are denoted by different colors that correspond to the legends in the histograms. (middle) Molecular structure of benzil, with the conformationally flexible angles denoted. (bottom) Histograms of the distribution of $\Psi$ angle for benzil molecules in the first, second and third ( $10 \overline{1}^{1} 0$ ) layers (from outermost layer to inner layers) of $6 \times 6 \times 15$ and $10 \times 10 \times 15$ nanocrystal rods obtained from Molecular Dynamics simulations at 200 K . The histograms reveal that $\Psi$ does not change with nanocrystal size in this regime and is independent of the presence and handedness of dislocations. Data for the $\Psi$ angle only is presented here. The other two angles exhibit nearly identical statistical distributions.

## References

1. J. Wang, R. M. Wolf, J. W. Caldwell, P. A. Kollman and D. A. Case, J. Comput. Chem., 2004, 25, 1157-1174.
2. A. Nemkevich, H.-B. Bürgi, M. A. Spackman and B. Corry, Phys. Chem. Chem. Phys., 2010, 12, 14916-14929.
3. S. Plimpton, J. Comput. Phys., 1995, 117, 1-19.
4. E. Vanquelef, S. Simon, G. Marquant, E, Garcia, G. Klimerak, J. C. Delepine, P. Cieplak and F.-Y. Dupradeau, Nucleic Acids Res., 2011, 39, W511-W517.
5. F. Wang, J. P. Becker, P. Cieplak and F. Y. Dupradeau, F. Y. R.E.D. Python: Object oriented programming for Amber force fields; Université de Picardie-Jules Verne, Sanford Burnham Prebys Medical Discovery Institute: 2013.
6. F. Y. Dupradeau, A. Pigache, T. Zaffran, C. Savineau, R, Lelong, N. Grivel, D. Lelong, W. Rosanski and P. Cieplak, Phys. Chem. Chem. Phys., 2010, 12, 7821-7839.
7. C. I. Bayly, P. Cieplak, W. Cornell and P. A. A. Kollman, J. Phys. Chem., 1993, 97, 10269-10280.
8. J. D. Gale and A. L. Rohl, Mol. Simul., 2003, 29, 291-341.
9. J. D. Gale, Z. Kristallogr. - Cryst. Mater., 2005, 220, 552-554.
