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## SI

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Table 1 shows the lifetime estimated using the first principle point defect computation and the SRH model for selected PV absorbers that have been examined extensively experimentally and from first principle computations. We analyzed the HSE point defect formation energies data for all possible growth conditions that are provided in literature, assuming all materials at all growth conditions are a p-type absorbers with conduction charge concentration  $10^{16}$  cm<sup>-3</sup>.

The point defects in PV absorbers that have been intensively studied both in the experiment and in first principle computations and the selected Cu-based PV materials were examined with the PBE exchange-correlation functional.<sup>23,24</sup> The Brillouin zone was sampled using a  $2 \times 2 \times 2$  k-point mesh. All intrinsic defects (the cation vacancies, the anion vacancies and the antisites) were considered. Their formation energies were computed at several chemical limits determined by the facets of the DFT phase diagram.<sup>25</sup> In order to overcome the effect on the defect formation energies of the underestimation of the band gap within PBE, the latter was extended to the HSE value. This was achieved by moving both

Table 1: The dominant deep defects and their transition states with respect to VBM, the theoretical life time computed with SRH model, and the experimental conversion efficiency for a set of tested structures with full HSE approach.

material	defect	transition state	theoretical life time(s)	theoretical efficiency $(\%)$	experimental efficiency $(\%)$
Si <sup>1</sup>	Vac Si	(2/0):0.02 (0/-2): 0.91	>> 1	33.3	$25.6^{2}$
$CdTe^{3}$	$\mathbf{V}_{Cd}$ , $\mathbf{Te}_{Cd}$	(-2/0):0.36 $(0,2):0.42$	$2.9  imes 10^{-10} - 0.19$	21.3 - 29	$22.1^{4}$
$GaAs^5$	$As_{Ga} Ga_{As}$	(2,1):0.63 (0,-1):0.4	$3.9 \times 10^{-11} - 3.2 \times 10^{-4}$	9.7-33.3	$28.8^{2}$
$Cu_2ZnSnS_4^{6}$	$\operatorname{Sn}_{Zn}$	(2,1):0.6	$1.84 \times 10^{-10}$ - $3.3 \times 10^{-7}$	10.2-20.9	$12.6^{2}$
$CuInSe_2$ <sup>7</sup>	$Cu_{In}$	(0,-1):0.37	$2.07 \times 10^{-6}$	22.7	$23.35^{8}$
$CuGeSe_2$ <sup>7</sup>	$Cu_{Ga}$	(0,-1):0.33	$6.67 \times 10^{-6}$	25.3	$23.35^{8}$
$MPI_3^{9}$	$I_i$	(-1,1)0.75	$2.31 \times 10^{-7}$ -0.002	24.6-27.7	$25.5^{10}$
InP	$\operatorname{Vac}_{In}$	(0,-1): 0.696769	>> 1	33.3	$22.1^{2}$
$Sb_2S_3^{11}$	$Sb_S S_{Sb}$	(3,2):0.5 (0,-1):0.67	$9.2 \times 10^{-12}$ - $1.2 \times 10^{-8}$	8.2 -14.1	$6.9^{12}$
$Sb_2Se_3^{13}$	$\operatorname{Vac}_{Se} \operatorname{Se}_{Sb}$	(1,0):0.85) Se <sub>Sb</sub> $(-1,1):0.62$	$10^{-16} - 1.9 \times 10^{-10}$	0-6.9	$9.2^{14}$
$Cu_2O^{15}$	$V_{Cu}$	(0,-1):0.42	$3.2 \times 10^{-7} - 3.2 \times 10^{-6}$	11.7-13.2	$9.54^{16}$
$CuSbS_2^{17}$	$Cu_{Sb}$ $Sb_{Cu}$	(0,-1):0.25 $(0,2):1.15$	$6.64 \times 10^{-10} \ 7.4 \times 10^{-7}$	11.5 - 19.5	$3.22^{18}$
$ZnSnP_2^{19}$	$V_P$	(0,1): 0.7 (0,-1):1	$1.2 \times 10^{-7} - 2.8 \times 10^{-6}$	19.1 - 21.7	$3.44^{20}$
$SnS^{21}$	$V_S$	(0,2) 0.64	$1.98{\times}~10^{-12}{-}6.63{\times}~10^{-10}$	0-8.6	$4.36^{22}$

band edges with respect to a common reference in GGA and HSE computations (more details are given in Refs.<sup>26,27</sup> The plots of the point defect formation energies of PV absorbers that have been intensively studied previously are shown below:

Table 2: The dominant deep defects and their transition states with respect to VBM and the theoretical life time computed with SRH model for the set of tested structures with GGA+HSE approach.

material	defect	transition state	theoretical life time(s)	experimental efficiency $(\%)$
Si	—	-	>> 1	$25.6^{2}$
CdTe	$V_{Cd}$	(-2, -1): 0.58	$2.34 \times 10^{-3} - \gg 1$	22.1 <sup>?</sup>
InP	—	_	$\gg 1$	$22.1^{2}$
GaP	$Ga_P, P_{Ga}$	(0,1):0.76, (1,2):0.717	$4.8 \times 10^{-11} - 2 \times 10^{-4}$	$2.42^{28}$
ZnTe	$V_{Zn}$	(0,-1):0.48	$5.53 \times 10^{-8} - 4.4 \times 10^{-6}$	$5.9^{29}$
$\mathrm{CuSbS}_2$	$\mathrm{Sb}_{Cu}$	(1,2):0.90594	$7.72 \times 10^{-9} - 3.07 \times 10^{-7}$	$3.22^{18}$
$\mathrm{Sb}_2\mathrm{S}_3$	$v_S$	(1,2):0.87	$1.56 \times 10^{-18} - 1.87 \times 10^{-12}$	$6.9^{12}$
$Sb_2Se_3$	$V_{Se}$	(0,2):0.65	$2.48 \times 10^{-11}$ - 4.11 $\times 10^{-9}$	$9.2^{14}$
SbSI	$V_S, V_I Sb_S$	(0,2):0.94,(-1,1):1.29,(1,3):0.87	$2.48 \times 10^{-26} - 1.21 \times 10^{-20}$	$3.5^{30}$
$\operatorname{SnS}$	$V_S$	$(1,2):0.543 \ 8.2 \ \times 10^{-13} \ -3.4 \ \times \ 10^{-10}$		$4.36^{22}$
SnSe	$\mathrm{Se}_{\mathrm{Sn}}$	(2,3): 0.731	$2.18 \times 10^{-18} - 1.46 \times 10^{-10}$	$6.44^{31}$
$\mathrm{Zn}\mathrm{Sn}\mathrm{P}_2$	$V_P$	(0,1):0.3	$1.08 \times 10^{-8} - 6.72 \times 10^{-7}$	$3.44^{20}$
Bi2S3	$V_S$	(0,1):1.1	$3.54 \times 10^{-18} - 3.4 \times 10^{-14}$	$5.9^{32}$
$Cu_2O$	_	_	$\gg 1$	$9.54^{16}$



Figure 1: GGA+HSE intrinsic defect formation energies as a function of the Fermi energy for  $Bi_2S_3$ , CdTe, Cu<sub>2</sub>O, CuSbS<sub>2</sub>, GaAs, and GaP. The zero in Fermi energy corresponds to the VBM, and the CBM is indicated with a vertical dashed line.



Figure 2: GGA+HSE intrinsic defect formation energies as a function of the Fermi energy for InP, Sb<sub>2</sub>S<sub>3</sub>, Sb<sub>2</sub>Se<sub>3</sub>, SbSi, Si, SnI<sub>2</sub>. The zero in Fermi energy corresponds to the VBM, and the CBM is indicated with a vertical dashed line.



Figure 3: GGA+HSE intrinsic defect formation energies as a function of the Fermi energy for SnS, SnSe, ZnSnP<sub>2</sub>, ZnTe. The zero in Fermi energy corresponds to the VBM, and the CBM is indicated with a vertical dashed line.



Figure 4: Theoretical efficiency from GGA+HSE approach for selected materials compared to their experimental efficiency. The range in theoretical efficiency corresponds to different growing conditions, hence different defects present in each material. The middle of the range is indicated by white circles. The green regions, indicating efficiencies lower (resp. larger) than 15%, correspond to correctly predicted low- (resp. high-) efficiency materials. The red regions correspond to false positives (upper left) and negatives (lower right).

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mpID	formula	SG	$E_{hull}$	$\mathrm{m}_{p1}$	$\mathrm{m}_{p2}$	$\mathrm{m}_{p3}$	$\mathrm{m}_{n1}$	$\mathrm{m}_{n2}$	$\mathrm{m}_{n3}$	$E_g$	$E_g^d$	$\Delta gap$	HSE-gap
17916	Y3CuGeSe7	$P6_{-3}$	0	0.659	0.659	3.915	0.28	0.645	0.645	1.118	1.118	0	1.9805
8446	K2CuP	Cmcm	0	2.175	2.744	3.624	0.227	0.754	0.812	1.142	1.223	0.081	1.9847
15684	K2CuAs	Cmcm	0	2.06	2.564	2.801	0.292	0.295	0.309	1.068	1.176	0.109	2.037
14205	K3Cu3As2	R-3m	0	1.272	1.272	1.922	0.36	0.36	0.761	1.296	1.317	0.037	2.0842
7439	K3Cu3P2	R-3m	0	1.236	1.236	2.144	0.392	0.392	0.705	1.309	1.345	0.055	2.1388
8017	AlCuTe2	I-42d	0	0.492	0.763	0.764	0.107	0.119	0.12	1.024	1.019	0	2.2581
3934	Cu3PS4	$Pmn2_{-}1$	0	0.734	1.304	1.478	0.268	0.448	0.466	1.02	1.033	0.012	2.3069
568954	Nd3CuGeSe7	$P6_{-3}$	0	0.765	0.765	4.176	0.357	0.804	0.804	1.194	1.205	0.012	2.0877
570226	Sm3CuGeSe7	$P6_{-3}$	0	0.702	0.702	3.861	0.336	0.741	0.741	1.165	1.17	0.008	2.0465
505558	Dy3CuGeSe7	$P6_{-3}$	0	0.641	0.641	3.651	0.29	0.636	0.636	1.113	1.115	0.002	2.0434
510011	La3CuGeSe7	$P6_{-3}$	0	1.135	1.135	4.399	0.381	0.869	0.869	1.185	1.222	0.038	2.2571
571347	Pr3CuGeSe7	$P6_{-3}$	0	0.796	0.796	4.275	0.376	0.848	0.848	1.206	1.219	0.014	2.1428
567428	Tb3CuGeSe7	$P6_{-3}$	0	0.659	0.659	3.68	0.293	0.648	0.648	1.127	1.127	0	2.059
7374	Ba(CuO)2	$I4_{-}1/amd$	0	1.008	3.786	3.796	0.301	0.302	0.815	1.379	1.379	0	2.5552
18685	SrCu2GeS4	$P3_221$	0	0.836	0.836	4.411	0.191	0.191	0.257	1.05	1.055	0.009	2.5616

1.5257	0	0.604	0.604	0.432	0.093	0.093	6.601	0.338	0.338	0	$P6_{-3}/mmc$	KCuTe	7436
3.306	0	0.444	0.444	0.248	0.248	0.248	3.188	3.188	3.188	0	Pa3	CuCl	23287
1.6776	0	0.362	0.362	0.133	0.127	0.122	2.16	1.98	0.541	0	$Pmn2_{-}1$	CdCu2GeS4	13982
1.6018	0.026	0.45	0.424	0.235	0.183	0.116	3.416	2.946	0.705	0	Ama2	BaCu2SnSe4	12364
1.5248	0	0.452	0.452	0.382	0.227	0.138	2.234	1.139	0.187	0	Pnma	BaCu4S3	654109
1.2931	0	0.82	0.82	0.879	0.704	0.352	1.352	1.103	0.477	0	Pnma	BaLaCuTe3	17063
1.236	0.032	0.634	0.617	0.538	0.538	0.207	2.057	2.057	0.688	0	$\rm P6_{-}3/mcm$	Ba6NaCu3Te14	569168
2.4162	0	0.561	0.561	0.252	0.252	0.252	1.784	1.784	1.784	0	F-43m	CuCl	22914
2.4315	0.008	0.974	0.971	0.245	0.193	0.193	3.3843	0.951	0.951	0	$P3_{-}121$	CuBiPbS3	17947
2.4026	0	0.98	0.98	0.249	0.19	0.186	2.771	2.18	0.534	0	$Pmn2_{-}1$	CdSi(CuS2)2	6449
2.25	0	0.706	0.704	0.152	0.151	0.137	0.952	0.951	0.533	0	I-42d	GaCuS2	5238
2.2932	0	0.892	0.896	0.136	0.133	0.121	1.037	1.037	0.494	0	I-42d	AlCuSe2	8016
1.6232	0.135	0.882	0.751	0.219	0.219	0.19	3.378	3.378	2.319	0	R-3m	K5CuSb2	27999
1.93	0	0.815	0.815	0.313	0.265	0.265	1.787	1.787	1.444	0	$P6_{-}3mc$	Cu6GeWS8	557225
1.707	0.055	0.714	0.678	0.793	0.716	0.159	1.438	0.585	0.57	0	Cmcm	Na2CuAs	15685
1.826	0.13	0.788	0.677	0.898	0.852	0.1	1.545	0.731	0.619	0	$\operatorname{Cmem}$	Na2CuP	7639
2.8971	0.091	1.864	1.794	0.892	0.892	0.733	4.779	0.935	0.935	0	$P6_{-3}$	Sm3CuGeS7	555978
3.0779	0	1.694	1.694	0.234	0.233	0.21	1.76	1.759	1.018	0	I-42d	AlCuS2	4979
2.7088	0	1.743	1.743	0.742	0.402	0.399	1.071	1.071	0.932	0	I-42d	CuBS2	12954

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29136	Sr6Cu3N5	$P4_2mc$	0	1.525	5.859	5.859	0.422	0.607	0.607	0.483	0.608	0.132	1.6424
9194	SmCuSeO	P4/nmm	1.18	0.653	0.653	2.936	0.224	0.224	0.339	1.286	1.286	0	2.6161
22863	CuI	P4/nmm	1.63	0.73	0.73	1.634	0.231	0.231	0.244	1.572	1.572	0	2.9193
570081	CuI	P-3m1	1.63	2	2	4.858	0.206	0.277	0.277	1.634	1.634	0	2.9916
32750	CuI	m R3m	1.63	1.09	1.09	1.393	0.163	0.173	0.173	1.164	1.164	0	2.758
673245	CuI	$\mathbf{Pc}$	1.63	0.895	0.916	1.054	0.164	0.173	0.176	1.145	1.145	0	2.7244
569346	CuI	$P6_{-}3mc$	1.63	1.053	1.053	1.513	0.159	0.179	0.179	1.217	1.217	0	2.7432
22895	CuI	F-43m	1.63	0.863	0.863	0.863	0.174	0.174	0.174	1.136	1.136	0	2.6699
5970	Ba(CuS)2	$\operatorname{Pnma}$	2.40	0.347	1.73	2.168	0.207	0.256	0.695	0.903	0.903	0	1.9518
542302	CuBi3PbS6	$ m Pmc2_{-}1$	4.05	1.118	1.876	2.196	0.212	0.55	0.627	0.622	0.622	0	1.4473
16179	SrCu2GeSe4	Ama2	4.22	0.425	3.078	3.774	0.208	0.284	0.435	0.592	0.723	0.135	1.8776
23431	CsCu2I3	Cmcm	4.27	1.236	2.206	2.361	0.268	0.283	0.313	1.901	1.901	0	3.364
21390	La2InCuSe5	Pnma	7.301	1.15	2.24	4.664	0.247	0.251	0.882	0.509	0.509	0	1.4084
23353	Cu2HgI4	I-42m	7.56	1.059	1.059	1.842	0.225	0.226	0.252	0.565	0.559	0	1.8524
624191	CuBiPbS3	$\operatorname{Pnma}$	9.30	0.766	0.942	1.097	0.276	0.323	0.431	0.633	0.651	0.038	1.735
7434	NaCuTe	P4/nmm	9.62	0.425	0.425	1.526	0.113	0.113	0.113	0.633	0.633	0	1.6734
15895	SiCu2S3	$C_{c}$	10.07	0.609	1.678	2.926	0.187	0.3	0.302	1.112	1.112	0	1.7747
18126	Tb3CuSnSe7	$P6_{-3}$	12.61	0.647	0.647	4.058	0.253	0.862	0.862	1.018	1.015	0	1.866

2.0263	0	0.753	0.753	0.438	0.283	0.187	2.617	0.975	0.263	70.87	$\operatorname{Pnma}$	Li2Cu4S3	766447
3.5525	0	1.981	1.981	0.352	0.27	0.267	2.219	0.537	0.532	69.33	$\operatorname{Pbcn}$	Li3CuS2	753737
1.4943	0.012	0.421	0.409	0.162	0.13	0.126	2.572	1.858	1.386	52.24	$Pmn2_{-1}$	Si(Cu4Se3)2	10428
2.6477	0.017	1.582	1.58	0.474	0.28	0.244	3.753	2.939	1.682	49.41	Cc	LiCuS	774736
2.5327	0	1.291	1.291	0.325	0.321	0.315	2.975	2.931	2.215	49.41	$\operatorname{Pbcn}$	LiCuS	766467
2.4835	0	1.076	1.076	0.298	0.246	0.244	1.312	1.208	0.419	49.41	$\operatorname{Pnma}$	LiCuS	753826
1.8792	0	0.598	0.598	0.204	0.203	0.179	3.245	2.004	1.595	43	$\mathrm{Pnm2_{-}1}$	Cu8GeS6	5546
2.4315	0	0.598	0.598	0.204	0.203	0.179	3.245	2.004	1.595	42.85	$Pmn2_{-}1$	BaCu2GeS4	5546
2.987	0	0.486	0.486	0.17	0.17	0.17	1.33	1.33	1.33	41.42	F-43m	CuBr	22913
2.987	0	0.448	0.448	0.167	0.166	0.163	1.449	1.438	1.437	41.42	I-4m2	CuBr	32880
2.5011	0	0.995	0.995	0.306	0.269	0.269	3.283	1.642	1.642	41.42	P4/nmm	CuBr	22917
1.6063	0	0.486	0.486	0.279	0.137	0.122	1.733	1.456	0.224	34.40	$\operatorname{Pnma}$	${\rm Ba}({\rm CuSe})2$	4473
2.753	0	1.525	1.525	0.392	0.209	0.208	4.401	1.261	1.255	32.65	$I4_1/amd$	Ca(CuO)2	754501
Hg is toxic	0	0.445	0.445	0.213	0.192	0.174	2.792	1.815	0.93	20.43	$Pna2_1$	CuHgSI	542426
2.3208	0	1.05	1.05	0.728	0.193	0.193	0.927	0.586	0.586	20.28	P3m1	$\mathrm{ErCuSe2}$	675180
2.103	0	0.851	0.851	0.606	0.145	0.145	4.864	0.365	0.365	19.71	$P6_{-3}/mmc$	LiCuS	774712
1.6203	0.098	0.612	0.518	0.129	0.129	0.129	0.804	0.804	0.804	18.78	F-43m	Al5CuSe8	37405
1.8972	0	0.99	0.993	0.809	0.809	0.232	4.236	0.671	0.671	17.91	$P6_{-3}$	Dy3CuSnSe7	510539
2.5154	0	1.18	1.18	0.324	0.197	0.197	2.615	0.599	0.599	15.42	P4/nmm	DyCuSeO	9304

3.0414	3.0661
0	0
1.642	1.614
1.642	1.614
0.479	0.47
0.324	0.318
0.324	0.318
4.89	4.35
0.977	0.945
0.977	0.945
94.6	98.12
P4/nmm	P4/nmm
6166 PrCuSO	42314 NdCuSO



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composition	Cu <sub>8</sub> GeS <sub>6</sub>
MP-id	5546
space group	Pnm2 <sub>1</sub>
hole effective mass	[1.595, 2.004, 3.245]
electron effective mass	[0.179, 0.203, 0.204]
HSE band gap (indirect/direct) (eV)	1.9/1.9
energy above hull (eV/atom)	0.044
life time (ns)	0.0



Energy (eV)













composition	K5CuSb2	
MP-id	27999	4
space group	R3m	
hole effective mass	[2.319, 3.378, 3.378]	
electron effective mass	[0.19, 0.219, 0.219]	
HSE band gap (indirect/direct) (eV)	1.44 / 1.59	
energy above hull (eV/atom)	0.0	
life time (ns)	2.6	<u> </u>
Chr. Sh Gen Chr.	<b>3</b> Þ	$-4 \xrightarrow{\  \  L \  B_1 \mid B \  \  Z \  \  \Gamma \  X \mid Q \  FP_1 \  \  Z \mid LP} Wave vector$
- 1.0 coefficient 10 coefficient 10	1 <b>5 6 1</b>	() () () () () () () () () ()
d.2.	/ /	Vac <sub>K</sub> $$ Vac <sub>Cu</sub> $$ Cu <sub>K</sub> $$ Cu <sub>K</sub>
		- Vac <sub>K</sub> $-$ Vac <sub>Sb</sub> $-$ Cu <sub>K</sub> $-$ K <sub>Cu</sub>
	ergy (eV)	—— Vас <sub>к</sub>

Г



















Figure 5: Theoretical efficiency from GGA+HSE approach for the 20 Cu based materials. The range in theoretical efficiency corresponds to different growing conditions, hence different defects present in each material. The middle of the range is indicated by diamonds.



Figure 6: Theoretical efficiency of typical PV absorbers and our outlined candidates versus their reserved HHI. The different colored regions correspond to typical limits. The color of the circles indicates the metal companionality (M.C.) in %.



Figure 7: Electron mobility at 300 K in Na<sub>2</sub>CuP as a function of the **k** and **q** meshes used for the integrations, see the main text. The **q** is the same as the **k** mesh for the matrix elements but a **q** mesh twice as dense in each direction is used for the energies in the delta distributions for the lifetimes (double-grid method of Ref.<sup>33</sup> The MRTA is used.



Figure 8: Electron mobility at 300 K in  $K_3Cu_3P_2$  as a function of the **k** and **q** meshes used for the integrations, see the main text. The **q** is the same as the **k** mesh for the matrix elements but a **q** mesh twice as dense in each direction is used for the energies in the delta distributions for the lifetimes (double-grid method of Ref.<sup>33</sup> The MRTA is used.



Figure 9: The defect formation energy as a function of Fermi level of intrinsic defects for KCuTe at all possible chemical potentials regions that determined from the facets of the phase diagram.



Figure 10: The defect formation energy as a function of Fermi level of intrinsic defects for NaCuTe at all possible chemical potentials regions that determined from the facets of the phase diagram.



Figure 11: The defect formation energy as a function of Fermi level of intrinsic defects for LiCuS at all possible chemical potentials regions that determined from the facets of the phase diagram.



Figure 12: COHP analysis (a,b,c,d,e,f) for all orbitals in pure bulk, averaged for each bond type for Al<sub>5</sub>CuSe<sub>8</sub>, AlCuTe, BaCu<sub>2</sub>S<sub>2</sub>, BaCu<sub>2</sub>Se<sub>2</sub>, BaCu<sub>2</sub>SnSe<sub>4</sub>, BaCu<sub>4</sub>S<sub>3</sub>. The VBM and CBM are represented by horizontal red dashed lines.



Figure 13: COHP analysis (a,b,c,d,e,f) for all orbitals in pure bulk, averaged for each bond type for CdCuGeS<sub>4</sub>, Cu<sub>2</sub>SiS<sub>3</sub>, Cu<sub>3</sub>PS<sub>4</sub>, Cu<sub>8</sub>SiSe<sub>6</sub>, Cu<sub>6</sub>GeWS<sub>8</sub>, Cu<sub>8</sub>GeS<sub>6</sub>. The VBM and CBM are represented by horizontal red dashed lines.



Figure 14: COHP analysis (a,b,c,d,e,f) for all orbitals in pure bulk, averaged for each bond type for ErCuSe<sub>2</sub>, K<sub>2</sub>CuP, K<sub>3</sub>Cu<sub>3</sub>As<sub>2</sub>, K<sub>3</sub>Cu<sub>3</sub>P<sub>2</sub>, K<sub>5</sub>CuSb<sub>2</sub>, KCuTe. The VBM and CBM are represented by horizontal red dashed lines



Figure 15: COHP analysis (a,b,c,d,e,f) for all orbitals in pure bulk, averaged for each bond type for  $Li_2Cu_4S_3$ , LiCuS,  $Na_2CuAs$ ,  $Na_2CuP$ , NaCuTe,  $Sr_6Cu_3N_5$ . The VBM and CBM are represented by horizontal red dashed lines



Figure 16: COHP analysis (a,b) for all orbitals in pure bulk, averaged for each bond type for  $SrCu_2GeSe_4$ ,  $Y_3CuGeSe_7$ . The VBM and CBM are represented by horizontal red dashed lines