# **Electronic Supplementary Information for:**

# Interlayer exciton emission in MoS<sub>2</sub>/VOPc inorganic/organic van der

## Waals heterostructure

Yuhan Kong<sup>a,b</sup>, Sk Md Obaidulla<sup>b,f</sup>, Mohammad Rezwan Habib<sup>b</sup>, Zukun Wang<sup>c</sup>, Rong

Wang<sup>d</sup>, Yahya Khan<sup>b</sup>, Haiming Zhu<sup>c</sup>, Mingsheng Xu<sup>b</sup>\*, Deren Yang<sup>e</sup>\*

<sup>a</sup> Department of Polymer Science and Engineering, Zhejiang University, Hangzhou 310027, P. R. China

<sup>b</sup> State Key Laboratory of Silicon Materials, School of Micro-Nano Electronics, Zhejiang University, Hangzhou 310027, P. R. China

E-mail: msxu@zju.edu.cn (M. S. Xu)

<sup>c</sup> Department of Chemistry, Zhejiang University, Hangzhou 310027, P. R. China

<sup>d</sup> ZJU Hangzhou Global Sci & Technol Innovat Ctr, Adv Semicond Res Inst, Hangzhou 311215, P. R. China

<sup>e</sup> State Key Laboratory of Silicon Materials, School of Materials Science and Engineering, Zhejiang University, Hangzhou 310027, P. R. China Email: <u>mseyang@zju.edu.cn (D. R. Yang)</u>

<sup>f</sup>Center of Excellence for Advanced Materials and Sensing Devices, Institute of Physics, Bijenička cesta 46, HR-10000 Zagreb, Croatia



**Figure S1.** Gaussian fitting of Raman spectra of VOPc/MoS<sub>2</sub> heterostructure: the calculated full width at half maximum (FWHM) corresponding to  $E_{2g}^1$  (386 cm-1) ~ 4.91 and  $A_{1g}$  (406 cm<sup>-1</sup>) ~7.09, respectively.



**Figure S2.** Gaussian fitting of Raman spectra of  $SnCl_2Pc/MoS_2$  heterostructure: the calculated FWHM corresponding to  $E_{2g}^1$  (386 cm-1) ~ 4.51 and  $A_{1g}$  (406 cm<sup>-1</sup>) ~7.24, respectively.



**Figure S3.** Gaussian fitting of Raman spectra of ML MoS<sub>2</sub>, the calculated FWHM corresponding to  $E_{2g}{}^{1}$  (386 cm-1) ~ 4.11 and  $A_{1g}$  (406 cm<sup>-1</sup>) ~6.02, respectively.



**Figure S4.** PL spectra (including individual constituent and sample background) of (a) VOPc/MoS<sub>2</sub> and (b) SnCl<sub>2</sub>Pc/MoS<sub>2</sub> heterostructures and right side corresponding to 5 different samples.



**Figure S5.** Peak fit of VOPc/MoS<sub>2</sub> heterostructure in which MoS<sub>2</sub> (668 nm), VOPc (880 nm), and the striking feature NIR emission peak at 805 nm (green line) are clearly resolved.



**Figure S6.** Peak fit of  $SnCl_2Pc/MoS_2$  heterostructure in which  $MoS_2$  (686 nm) and  $SnCl_2Pc$  (766 nm) are clearly resolved.



**Figure S7.** Ultraviolet photoelectron spectroscopy (UPS) spectrum (hv = 21.22 eV) (a,b) VOPc and VOPc/MoS<sub>2</sub> on pure highly doped Si substrate, (c,d) SnCl<sub>2</sub>Pc and SnCl<sub>2</sub>Pc/MoS<sub>2</sub> on pure highly doped Si substrate. The thickness of organic thin films is about 5 nm. Please note that the MoS<sub>2</sub> was firstly synthesized on SiO<sub>2</sub> surface and then transfer to Si substrate for fabricating heterostructures for UPS measurement.



**Figure S8.** (a, b) UV-vis absorbance spectrum of VOPc and SnCl<sub>2</sub>Pc films, respectively and corresponding their tauc plots (c, d) of  $(\alpha hv)^{1/2}$  as a function of energy based on  $(\alpha hv)^{1/2}=C$  ( $hv-E_g$ ), where  $\alpha$ , h, v, C, and  $E_g$  are the absorption coefficient, Planck's constant, the incident light frequency, proportionality constant, and the band-gap energy, respectively.

To find out the optical band gap of VOPc film, we plot the UV-vis absorption spectrum of the VOPc film by using the Tauc equation.<sup>[1]</sup> Thus, bandgap energy (~1.41 eV) of the VOPc film is estimated from **Figure S8b**. Based on the UPS results (**Figure S7a, b**), the HOMO level of the VOPc film is determined to be about 5.12 eV. From the UPS results, we can get the Fermi energy ( $E_F$ ) is around at 0.70 eV and the  $E_{cut-off}$  is around at 16.8 eV. So we can calculate the  $E_{HOMO}$ :  $E_{HOMO} = 21.22 - (16.8-0.70) = 5.12$  eV. From UV-vis and UPS measurements, we calculate the LUMO of VOPc film, about 3.71 eV, thus we calculate the HOMO/LUMO (5.12 eV/3.71 eV) energy levels of VOPc molecules. The HOMO/LUMO (5.41 eV/3.79 eV) energy level of SnCl<sub>2</sub>Pc molecules can also be calculated by the same method.<sup>1, 2</sup>

#### DFT predicted structure of MPc (M=VO, SnCl<sub>2</sub>)/MoS<sub>2</sub> heterostructure

The geometry of VOPc and SnCl<sub>2</sub>Pc molecules on monolayer MoS<sub>2</sub> are predicted by geometry optimization calculation using PBE functional as implemented in VASP. The initial structure of MoS<sub>2</sub>/MPc (M=VO, SnCl<sub>2</sub>) was designed by minimizing the lattice mismatch between MoS<sub>2</sub> monolayer and organic molecules. The strain due to the lattice mismatch were applied to the organic layer. After that the atomic positions of constituents in the heterostructures were relaxed by the PBE functional as implemented in VASP. The structure of VOPc/MoS<sub>2</sub> and SnCl<sub>2</sub>Pc/MoS<sub>2</sub> are relaxed to their minimum energy during the structural optimization. The orientations of the molecules with respect to MoS<sub>2</sub> in the molecule/MoS<sub>2</sub> heterostructure are depicted in Figure S9.



Figure S9. Optimized structure of (a) VOPc/MoS<sub>2</sub> (b)  $SnCl_2Pc/MoS_2$  heterostructure. d<sub>0</sub> denotes the interlayer spacing between MoS<sub>2</sub> and organic layer.

## State projected density of VOPc/MoS<sub>2</sub> heterostructure



**Figure S10.** Density and projected density of states of VOPc/MoS2 heterostructure projected. A, B, and D, E represent the states below and above the Fermi level. The states at the Fermi level is denoted by C. The corresponding charge density for A, B, C, D and E states are depicted.

### Effect of interlayer spacing on the density of states:



Figure S11. Variation of density of states with the interlayer spacing between  $MoS_2$  and VOPc layer in VOPc/MoS<sub>2</sub> heterostructure. The equilibrium interlayer separation between  $MoS_2$  and VOPc layer is  $d_0$  Å.



Figure S12. Variation of density of states with the interlayer spacing between  $MoS_2$  and  $SnCl_2Pc$  layer in  $SnCl_2Pc/MoS_2$  heterostructure. The equilibrium interlayer separation between  $MoS_2$  and  $SnCl_2Pc$  layer is  $d_0$  Å.



Wavelength (nm)

Figure S13. Transient absorption (a)  $SnCl_2Pc/MoS_2$ , (b) ML-MoS<sub>2</sub>, and (c)  $SnCl_2Pc$ .



Figure S14. TA kinetics of the VOPc signal in  $MoS_2/VOPc$  heterostructure (green) and pure VOPc only (red), under 800 nm excitation. The results show that in the heterostructure, the decay of the signal became faster, indicating the existence of charge transfer from VOPc to  $MoS_2$ .



**Figure S15.** TA kinetics of the MoS<sub>2</sub> signal in MoS<sub>2</sub>/VOPc heterostructure at 660 nm, under 800 nm excitation.



**Figure S16.** (a) PL spectra at the excitation wavelength of 532 nm and with varying excitation intensities. (b) Integrated PL intensity of MoS<sub>2</sub>, NIR peak at 805 nm and VOPc as a function of the laser intensity plotted on a log–log scale for excitation at 532 nm. (c) PL spectra at the excitation wavelength of 785 nm and with varying excitation intensities. (b) Integrated PL intensity of NIR peak at 805 nm and VOPc as a function of the laser intensity plotted on a log–log scale for excitation at 785 nm.

PL measurements were made at room temperature using a Renishaw inVia Raman microscope using excitation wavelengths of 532 nm and 785 nm. The laser beam was focused on to the samples via a  $50 \times$  objective lens producing a spot diameter of  $\sim 1 \,\mu$ m. The maximum laser power was kept at  $\sim 25 \,\text{mW}$  (532 nm) and 150 mW (785 nm) and it was varied using neutral density filters from 0.000001% to 100%.

We find that the PL intensity of  $MoS_2$ , new peak at 805 nm, and VOPc increases linearly with excitation power both under 532 nm and 785 nm laser beam. It is surprised that the PL intensity of  $MoS_2$  excesses that of the new 805 nm when the excitation fluence is 2.5 mw. More studies are needed to understand the underlying mechanisms.

### References

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