## **Supporting Information for**

## Bias-tunable two-dimensional magnetic and topological materials

Jie Li and Ruqian Wu\*

Department of Physics and Astronomy, University of California, Irvine, California 92697-4575,

USA.

\* E-mail: wur@uci.edu

## The detail of torque method

In the Torque method, the torque  $T(\theta)$  as a function of the polar angle  $\theta$  between the magnetization and the z is defined as,

$$T(\theta) = \frac{dE(\theta)}{d\theta} = \sum_{occ} \left\langle \psi_{ik} \left| \frac{\partial H_{so}}{\partial \theta} \right| \psi_{ik} \right\rangle$$
(1)

 $H_{so} = \sum_{i} \xi(r_i) l_i \cdot s$ with the spin-orbit coupling (SOC) Hamiltonian , and the summation goes over all occupied states. By integrating  $T(\theta)$ , we obtain the total energy  $E(\theta)$  as a function of the polar angle  $\theta$ , and the MAE is equal to the difference between the lowest and highest total energies.

	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn
$M@C_u$	1.579	1.462	1.635	2.245	2.308	2.185	2.485	2.46	2.179
$M@C_d$	1.717	1.528	2.025	2.117	2.706	1.704	2.401		
$M@H_{1-u}$	0.87	1.272	1.516	2.494	1.883	2.369	2.567	2.97	2.71
$M@H_{1-d}$	0.27	1.017	1.729	2.495	2.12	2.454	3.079		
M@H <sub>2-u</sub>	0.616	1.229	1.452	2.33	1.7	2.311	2.439	2.835	2.55
$M@H_{2-d}$	0.345	0.715	1.501	2.331	1.993	2.263	2.991		

Table S1. The binding energies (eV) of M@C,  $M@H_1$  and  $M@H_2$  (M= Ti - Zn).



**Fig. S1** The phonon spectrum and the relative total energy as function of AIMD time for  $Cr@C_u$ ,  $Cr@H_{1-u}$  and  $Cr@H_{2-u}$ , respectively. (a), (d) for  $Cr@C_u$ , (b), (e) for  $Cr@H_{1-u}$ , (c), (f) for  $Cr@H_{2-u}$ .

	Ti	V	Cr	Mn	Fe	Со	Ni
M@C	0.02	0.03	0.001	0.01	0.001	0.49	0.82
$M@H_1$	0.775	1.09	1.22	0.001	0.74	0.97	0.82
$M@H_2$	0.40	1.08	0.40	0.001	0.33	0.94	0.77

Table S2. The atomic displacements ( $\Delta d$ , in Å) of M@C, M@H<sub>1</sub> and M@H<sub>2</sub> (M= Ti-Ni).

	Ti	V	Cr	Mn	Fe	Со	Ni
M@C <sub>u</sub>	4.31	4.32	4.30	4.31	4.27	4.37	4.35
$M@C_d$	4.33	4.33	4.31	4.30	4.27	4.39	4.37
$M@H_{1-u}$	4.52	4.57	4.56	4.55	4.52	4.56	4.61
$M@H_{1-d}$	4.54	4.56	4.55	4.55	4.54	4.53	4.53
M@H <sub>2-u</sub>	4.50	4.54	4.58	4.54	4.53	4.56	4.60
M@H <sub>2-d</sub>	4.53	4.56	4.53	4.54	4.54	4.50	4.47

Table S3. The height (h, in Å) of the carbon cage in different lattices.

Ni). Ti V Cr Mn Fe Co Ni M@C 0.077 0.041 0.010 0.163 0.001 0.559 0.306  $M@H_1$ 1.000 0.449 0.211 0.001 0.205 0.161 0.450 0.597 0.779 0.423 0.001 0.011 0.501  $M@H_2$ 0.186

Table S4. The energy barriers (eV) between phase I nad II of M@C, M@H<sub>1</sub> and M@H<sub>2</sub> (M= Ti-



Fig. S2. Total energies of  $Cr@C_u$  and  $Cr@H_{1-u}$  versus the deposition density.



Fig. S3 (a) and (b) The side and top views of the charge redistribution of  $Cr@H_{1-u}$ . (c) and (d) The side and top views of the charge redistribution of  $Cr@H_{1-d}$ . (Red and blue colors indicate charge depletion and accumulation, the isosurface is  $2.0 \times 10^{-3} e/Å^{3}$ ).

	V@C <sub>u</sub>	V@C <sub>d</sub>	$V@H_{1\text{-}d}$	V@H <sub>2-d</sub>	Cr@C <sub>u</sub>	Cr@C <sub>d</sub>	Cr@H <sub>1-u</sub>	$Cr@H_{1-d}$
J <sub>1</sub> (meV)	3.38	2.44	1.13	1.11	0.07	0.06	0.32	1.08
	Cr@H <sub>2-u</sub>	Mn@C <sub>u</sub>	Mn@C <sub>d</sub>	Fe@H <sub>1-d</sub>	Fe@H <sub>2-d</sub>	Co@C <sub>d</sub>	Ni@C <sub>u</sub>	Ni@C <sub>d</sub>
J <sub>1</sub> (meV)	3.38	1.28	1.53	5.76	0.108	0.21	0.35	3.19
	Cu@C <sub>u</sub>	Cu@H <sub>2-u</sub>	Zn@C <sub>u</sub>					
J <sub>1</sub> (meV)	1.12	1.18	1.33					

Table S5. The exchange energies of M@C,  $M@H_1$  and  $M@H_2$  (M= V-Zn).



**Fig. S4** (a) and (b) The calculated MAEs and the renormalized magnetization as a function of temperature T for some 2D covalent crystals with ferromagnetic ordering.



**Fig. S5** (a) and (b) The projected density of states (PDOS) and the Fermi level dependent total and spin channel decomposed MAEs of Fe@H<sub>2-d</sub>, respectively. One may see that the large perpendicular MAEs mostly stem from the SOC interaction between  $d_{xy/(x^2-y^2)\downarrow}$  and  $d_{xz/yz\downarrow}$ orbitals of Fe atom, and the cross-spin SOC interaction between  $p_{x/y\uparrow}$  and  $p_{x/y\downarrow}$  orbitals of the carbon cage also contributes. The total MAE is rather sensitive to the shift of the Fermi level, indicating a noteworthy magnetoelectric effect for this system, i.e., the MAEs can be effectively tuned by adjusting the position of the Fermi level.



**Fig. S6** (a) and (b) The band structures of  $Ti@H_{1-u}$  and  $Cr@H_{1-u}$  from DFT calculations and the fitting band structures by using the Wannier90 package, respectively.



**Fig. S7** (a) Schematic phase transition of the of  $Cr@H_{1-u}$  lattice with two  $Cr@C_{28}$  molecules being converted to phase II (25%), one is at site 1 and the other randomly takes site A, B or C. (b)-(d) Their corresponding band structures. (e) The band structure of  $Cr@H_{1-u}$  with only one  $Cr@C_{28}$  molecule in the supercell being converted to phase II (12.5%).