

Supporting Information

Regulation Effects of Co^{2+} in Constructing of $\text{Cu}-\text{Ni}(\text{OH})_2@ \text{CoO}$ Nanoflower Clusters Heterojunction: A Critical Factor of Obtaining High-Performance Battery-type Hybrid Supercapacitor

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Experimental section

Pre-treatment of Ni-foam: First, several nickel foams (3 cm × 4 cm) were put in 6 M HCl solutions for 10 min to eliminate the surface oxides. Then, they were transferred into 20 mL acetone for 10 min to experience a process of degreasing. Finally, after 30 min ultrasonic cleaning and washing by deionized water, the nickel foams was taken to an oven with 60 °C for 1 h.

Preparation of Cu-Ni(OH)₂@CoO NFCs electrodes: The Cu-Ni(OH)₂@CoO NFCs composite is synthesized by a two-step hydrothermal method. In the first hydrothermal process, 0.3 g cupric sulfate (CuSO₄·5H₂O) and 2 g urea were added into 30 mL deionized water under stirring to form a homogeneous and transparent solution. The precursors were made by immersing nickel foams in aqueous solution and transferred to a Teflon-lined stainless steel autoclave, then kept at 120 °C for 6 h. In the second steps, after washed by deionized water several times, the obtained electrode material is transferred to a Teflon-lined stainless steel autoclave which contains 30 ml of cobaltous nitrate hexahydrate (Co(NO₃)₂·6H₂O, 0.3g) aqueous solution, then kept at 100 °C for 6 h. Ultimately, the synthetized sample was washed by deionized water several times and then transferred into an oven maintained at 60 °C overnight.

Assembly of flexible all-solid Hybrid Asymmetric Supercapacitors: 6 g polyvinyl alcohol (PVA) and 3 g KOH were added into 60 mL deionized water to prepare gel electrolytes for HASC. First, the above solution is heated to 100 °C for 30 min to obtain a homogeneous solution, then the positive electrode materials (Cu-

$\text{Ni(OH)}_2@\text{CoO}$ NFCs, 1 cm^2) and negative electrode materials (porous carbon coated nickel foam, 1 cm^2) were immersed into the PVA/KOH electrolytes for 1 min. After repeated soaking for three times, the electrodes were transferred to a petri dish and stand for 12 h before being assembled into a flexible all-solid HASC. And the specific assembly procedures of flexible all-solid HASC could be found in our previous work.

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Characterization methods

A Rigaku Smartlab X-ray diffraction (XRD) equipment was applied to collect the diffraction data of the obtained samples using $\text{Cu K}\alpha$ ($\lambda = 0.1544$ nm) radiation. Moreover, a Renishaw Invia Raman spectrometer with a 20 mW Ar^+ laser source of 532 nm was used to collect the Raman spectra. The morphologies and structures of samples were investigated by a field emission scanning electron microscope (FESEM) (Regulus 8220). In addition, a transmission electron microscopy (TEM, Tecnai F30, STWIN) coupled with an energy-dispersive X-ray spectroscopy (EDS) were conducted to detect the microstructure of active materials. The X-ray photoelectron spectra (XPS) were also measured on a Thermo Scientific ECSALab 250Xi X-ray photoelectron spectrometer with an $\text{Al K}\alpha$ X-ray radiation (1486 eV) to investigate the chemical states and surface composition.

Electrochemical measurements

Evaluation of Supercapacitive Performances: All electrochemical tests were based on a CHI 660E electrochemical workstation (CH Instrumental Co, Ltd, Shanghai), and the electrochemical property of individual electrode is studied by a three-electrode electrochemical set-up (Pt foil as the counter electrode, saturated calomel electrode (SCE) as the reference electrode and the self-supported $\text{Cu-Ni(OH)}_2@\text{CoO}$ NFCs as the working electrode, respectively). In this system, an aqueous solution of 3 M KOH was used as electrolyte, and the stability of electrode

was tested under the scan rate of 50 mV s⁻¹. Besides, the range of frequency in the electrochemical impedance spectroscopy (EIS) measurement is from 0.01 HZ to 100 KHz. Meanwhile, the galvanostatic charge-discharge (GCD) tests were carried out from -0.1 V to 0.4 V and the areal specific capacitance (C_s) were calculated by the equation as follows:

$$C_s = It / A\Delta V \quad (1)$$

Where C_s is the areal specific capacitance (mF·cm⁻²), I is the discharge current (A), t is the discharge time (s), A represents the area of self-supported electrodes (cm²) and ΔV is the potential window (V).

Besides, the energy density and power density of Ni(OH)₂@CoO NFCs//porous carbon HASC are calculated by the two equations as follows:

$$E = 1/2 C_s V^2 \quad (2)$$

$$P = E/t \quad (3)$$

Where E represents the energy density (mWh·cm⁻²), C_s is the areal specific capacitance (F·cm⁻²), V means the operating voltage (V), P is the power energy (mW·cm⁻²), t refers to the discharge time (s).

Also, to achieve $q_+ = q_-$, the mass ratio of positive electrode active material to negative electrode active material is determined from the following equation:

$$m_+ : m_- = \left(\int \frac{idV}{v} \right)_- : \left(\int \frac{idV}{v} \right)_+ \quad (4)$$

The mass ratio of the positive electrode active material to the negative electrode active materials was determined from the cyclic voltammetry acquired at 10mV/s for both electrodes.

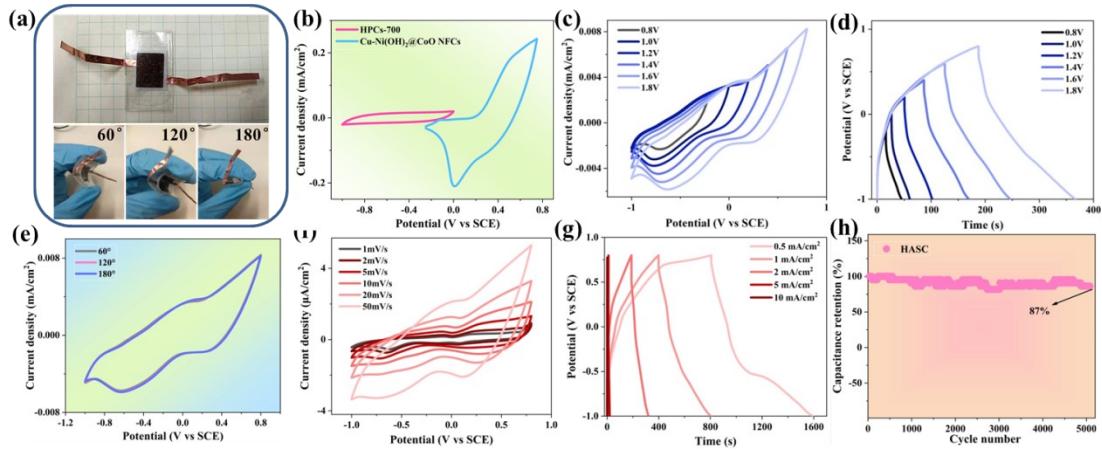


Figure S1. (a) The photo of obtained flexible all-solid supercapacitor device with different bend angles; (b) CV curves of Cu-Ni(OH)₂@CoO NFCs and porous carbon at a scan rate of 100 mV·s⁻¹ in a three-electrode system; (c) CV plots of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC at different potential windows; (d) GCD curves of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC at different potential windows from 0.8 to 1.8 V; (e) CV plots of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC at different bend angles; (f) CV plots of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC at various scan rates; (g) GCD plots of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC at various current density; (h) a function of specific capacitance of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC to current density.

A flexible solid-state HASC was fabricated to explore the practical application of electrode materials. In the assemble procedure, the as-prepared Cu-Ni(OH)₂@CoO NFCs, porous carbon from our previous work¹ and PVA/KOH gel were chosen as positive electrode, negative electrode and electrolyte, respectively. The outstanding flexibility of assembled HASC device is displayed in the **Figure. S1a**, and the shape of CV plots at different bend angles (**Figure. S1e**) is nearly unchanged at all. As shown in **Figure. S1b**, the potential windows of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC

were determined through the test of voltage range at a scan rate of 100 mV s⁻¹. As depicted in **Figure. S1c** and **S1d**, the CV and GCD tests were conducted at different potential windows from 0.8 V to 1.8 V. No obvious polarization could be observed when the potential window was less than 1.8 V (**Figure. S1c**), suggesting that the operating voltage of HASC could be broadened up to 1.8 V. As displayed in **Figure. S1f**, a series of redox peaks appeared at the CV plots in different scan rates, indicating that both pseudo-capacitance and double-layer capacitance take effect in the process of electrode reactions.² Furthermore, the CV curves of HASC device nearly remain its initial shape, suggesting a good reversibility.³ Additionally, based on **Figure. S1g** and **Equation 1**, the mass loading of the positive and negative electrode in the assembled HASC are 0.0022 g and 0.0153 g, and the areal specific capacitances of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC can be calculated to be 996.6 mF cm⁻² at current density of 1.0 mA cm⁻².

Energy density and power density are two important parameters in evaluating supercapacitive performance. The obtained Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC device exhibits a superior energy density of 0.45 mWh cm⁻² when its power energy up to 9 mW cm⁻², exceeding most of transition-metal based asymmetric supercapacitors. The comparatively detailed parameters with some recent publications can be found in the *Ragone* plots (**Figure. S9**) and **Table S4**. Furthermore, the values of R_s and R_{ct} derived from the EIS measurement in **Figure. S8** are treated as the criterion of evaluating the charge transfer process, which demonstrates a superior electrochemistry property for Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC.⁴ In

addition, the durability of HASC was explored through charge and discharge cycle test at a current density of 1 mA cm^{-2} . No abnormal fluctuations are witnessed during testing, and the capacitance still remains *ca.*87% after 5000 cycles (**Figure. S1h**). Besides, the coulombic efficiency of HASC can reach to 92.4%, indicating an excellent reversibility (**Figure. S10**).⁵ To sum up, the $\text{Cu-Ni(OH)}_2@\text{CoO}$ NFCs//porous carbon flexible solid-state HASC device displays an admirable performance.

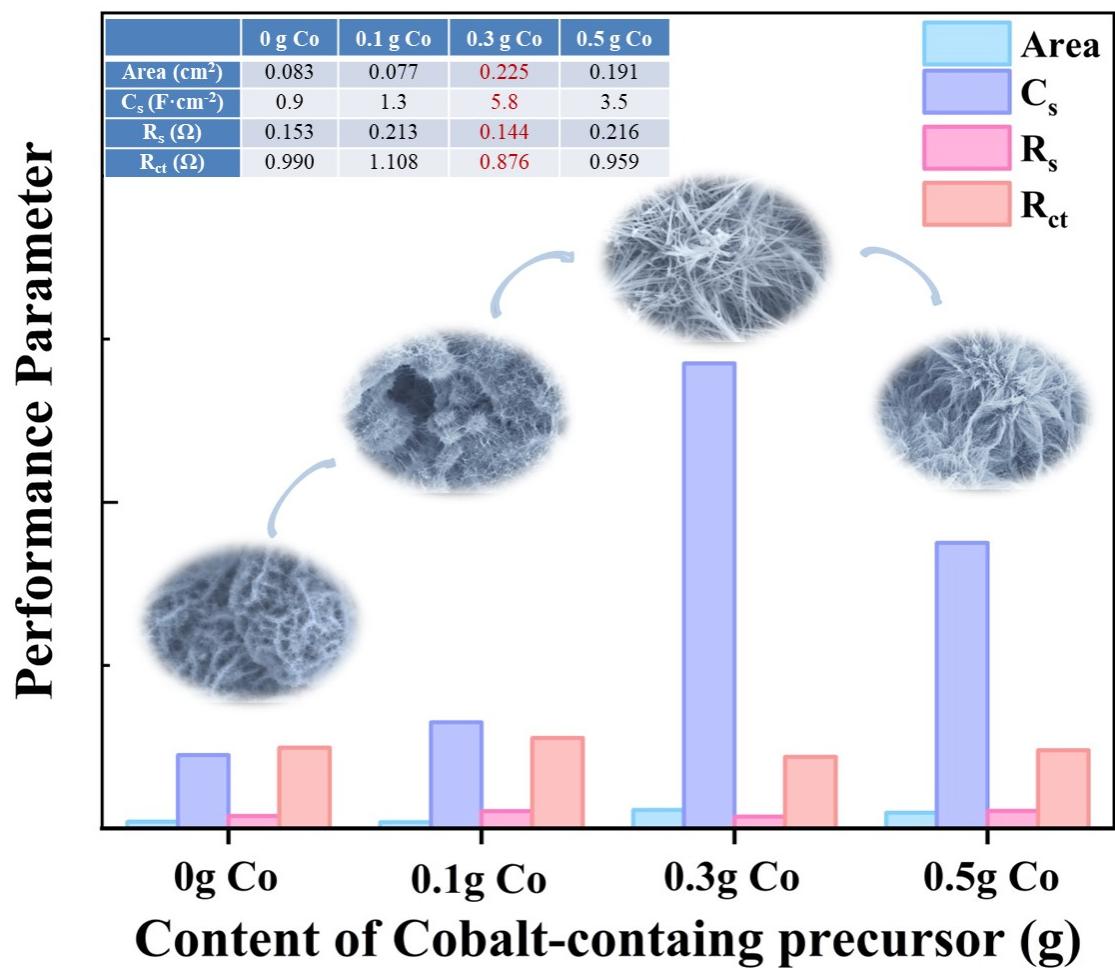


Figure S2. Relationship between properties parameters (Area: the area of one CV cycle; C_s : areal specific capacitance of electrode materials; R_s : the equivalent series resistance; R_{ct} : charge transfer resistance) and content of precursor Co-containing.

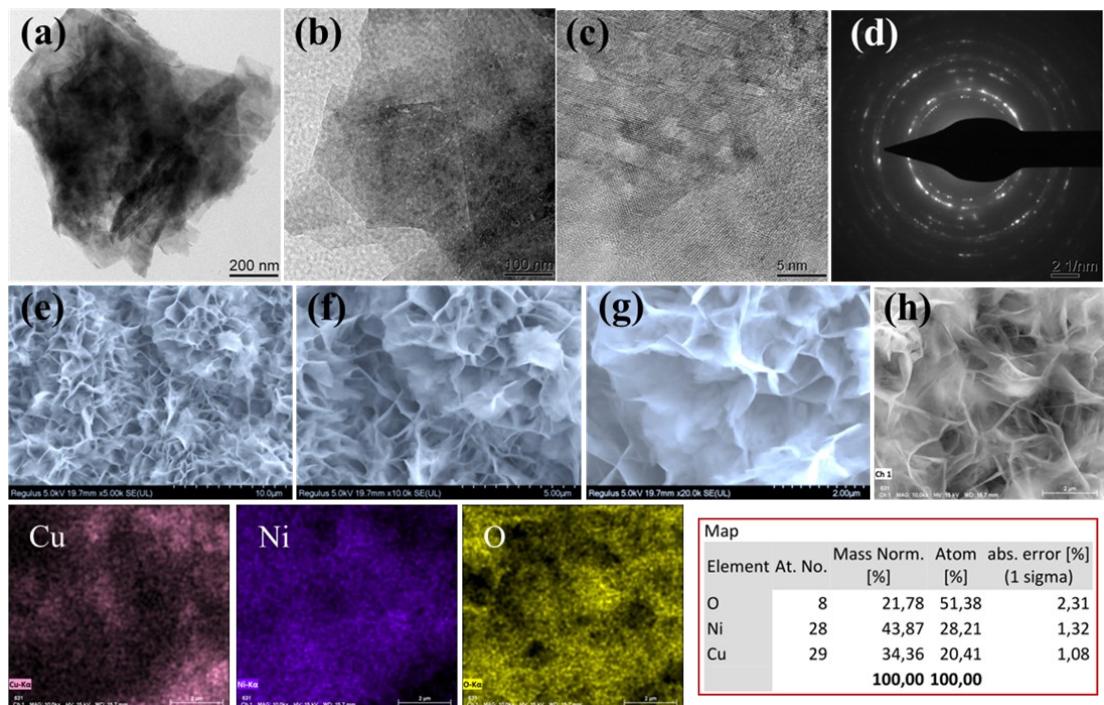


Figure S3. TEM images (a-d); SEM images (e-g) and EDS analyses (h) of typical Cu-Ni(OH)₂ NAs active materials with different magnifications and locations of sample.

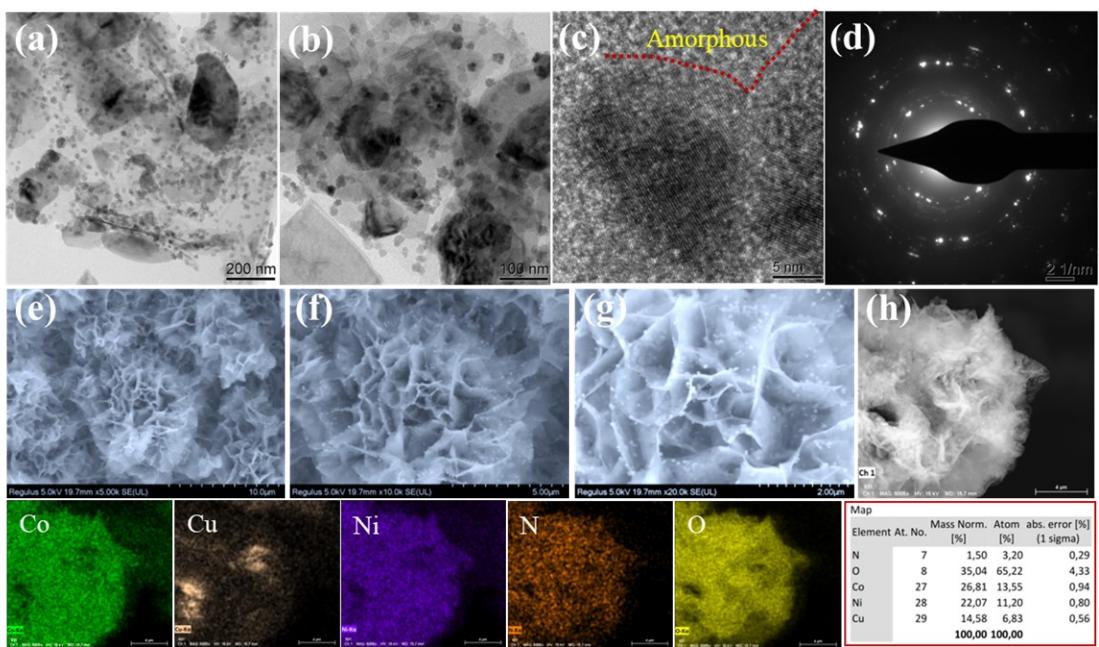


Figure S4. TEM images (a-d); SEM images (e-g) and EDS analyses (h) of typical Cu-Ni(OH)₂@CoO NAs; active materials with different magnifications and locations of sample.

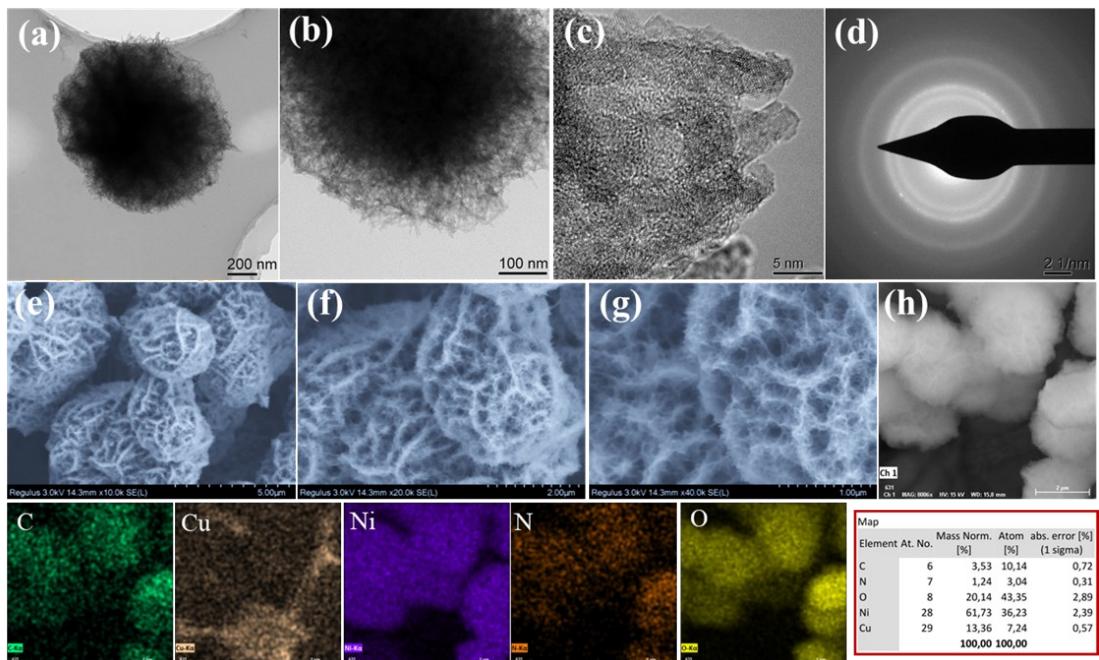


Figure S5. TEM images (a-d); SEM images (e-g) and EDS analyses (h) of typical Cu-Ni(OH)₂ NBs active materials with different magnifications and locations of sample.

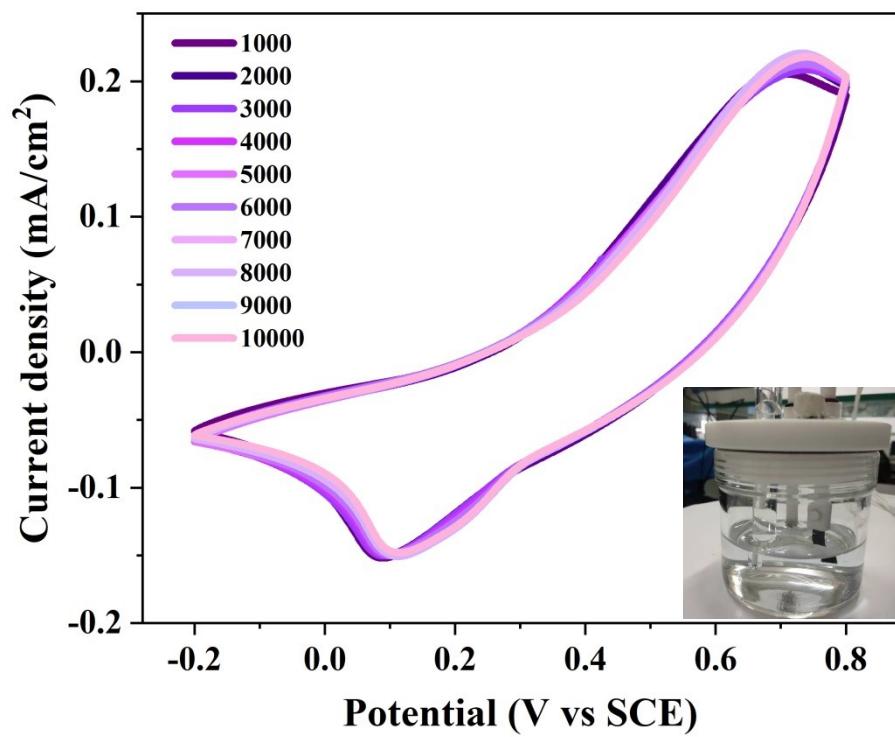


Figure S6. Comparison of CV curves at different cycle numbers for $\text{Cu-Ni(OH)}_2@\text{CoO}$ NFCs electrode.

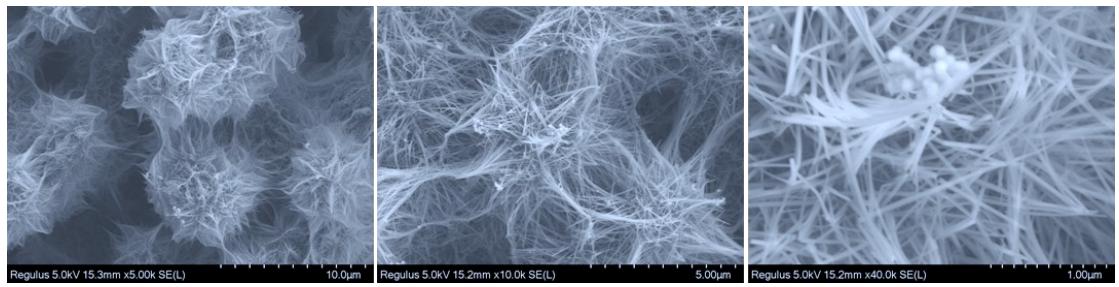


Figure S7. SEM images of Cu-Ni(OH)₂@CoO NFCs after 10000 CV cycles.

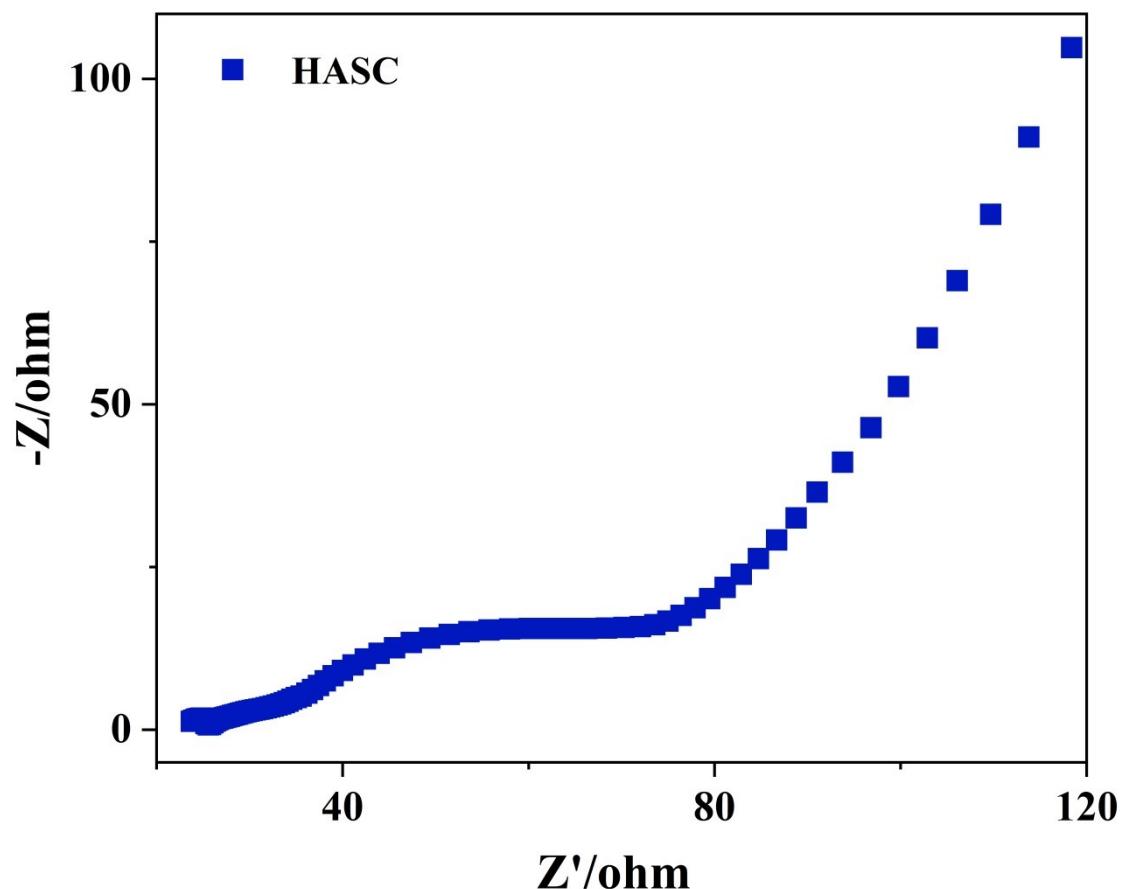


Figure S8. Nyquist plots of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC.

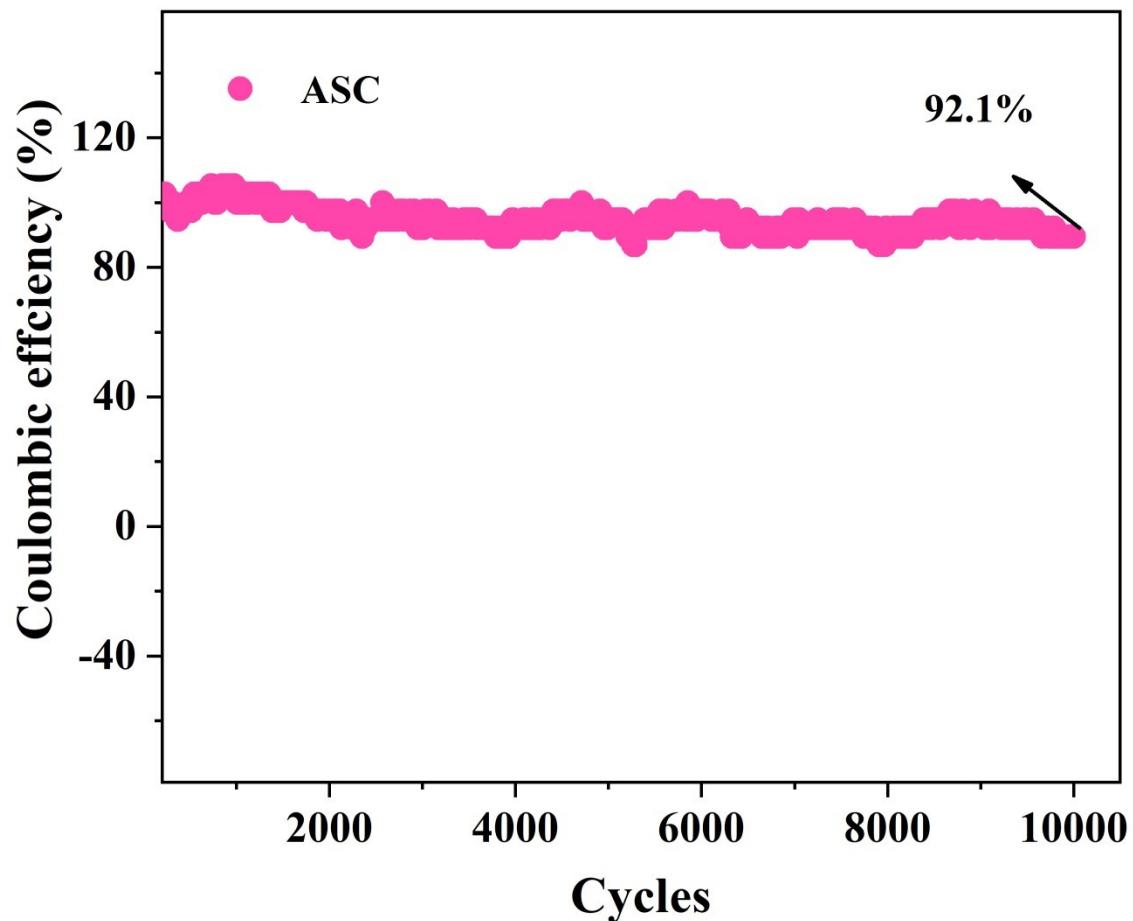


Figure S9. Coulomb efficiency of Cu-Ni(OH)₂@CoO NFCs//porous carbon HASC as a function of cycle number.

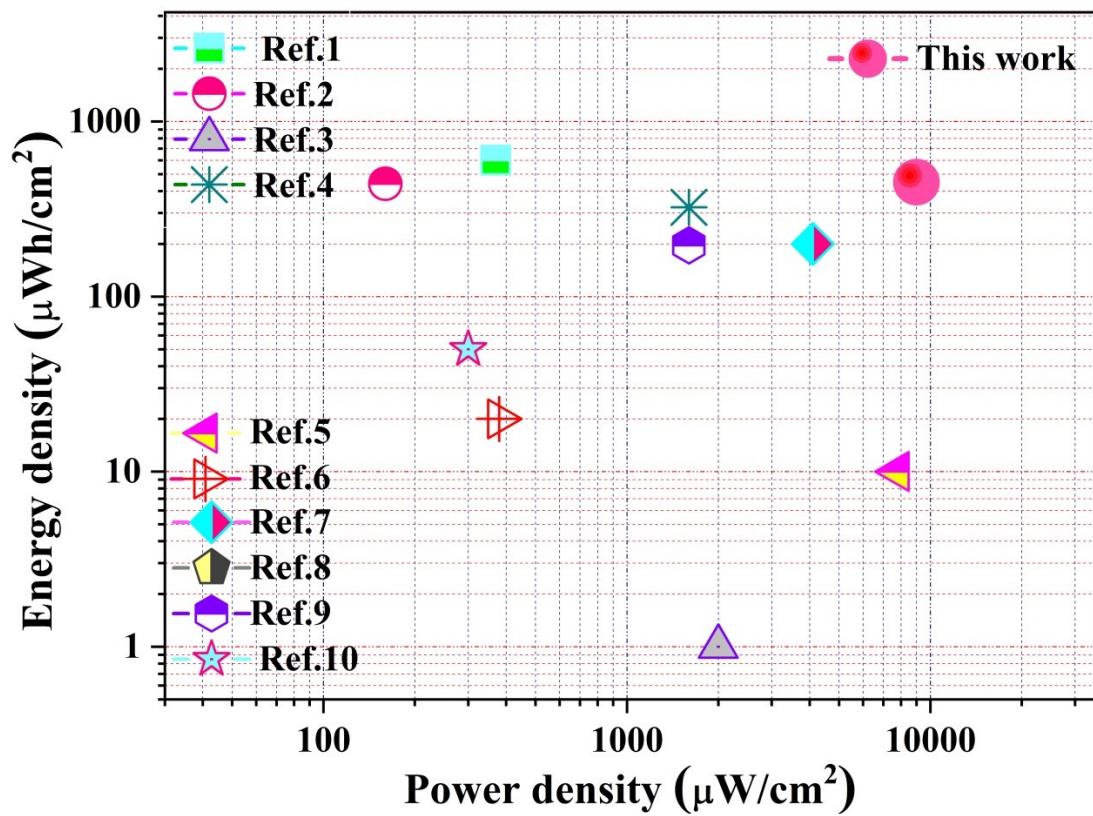


Figure S10. Ragone plots showing the comparison of $\text{Cu}-\text{Ni}(\text{OH})_2@\text{CoO}$ NFCs//porous carbon HASC with other Ni/Co-based devices.

Table S1. Comparison of several electrode materials in the terms of conductivity

	<i>Resistivity (μΩ·m)</i>	<i>Conductivity (S/m)</i>
Cu-Ni(OH)₂ NAs	3.98313	251059.08401
Cu-Ni(OH)₂@CoO NAs	5.10301	195962.92514
Cu-Ni(OH)₂ NBs	4.42779	225846.52941
Cu-Ni(OH)₂@CoO NFCs	3.17578	314883.03737

Table S2. The values of R_s and R_{ct} for Cu-Ni(OH)₂ NAs, Cu-Ni(OH)₂@CoO NAs, Cu-Ni(OH)₂ NBs and Cu-Ni(OH)₂@CoO NFCs electrodes

	Cu-Ni(OH)₂ NAs	Cu-Ni(OH)₂@CoO NAs	Cu-Ni(OH)₂ NBs	Cu-Ni(OH)₂@CoO NFCs
R_s (Ω)	0.142	0.162	0.158	0.131
R_{ct} (Ω)	1.132	1.035	1.125	1.023

Table S3. Comparison of Ni/Co-based materials applied to supercapacitors

Electrode	Electrolyte	Areal specific capacitance	current density	Reference
Ni/Co-oxyhydroxides	6 M KOH	8.87 F·cm ⁻²	5 mA·cm ⁻²	[6]
CoO@Ni(OH)₂ core–shell	2 M KOH	1418.2 F·g ⁻¹	1 A·g ⁻¹	[7]
Ni(OH)₂/CoO/rGO	3 M KOH	1056 F·g ⁻¹	5 mV·s ⁻¹	[8]
VCO/CNFs	1 M KOH	1.83 F·cm ⁻²	8 mA·cm ⁻²	[9]
CoO@CoS/Ni₃S₂	2 M KOH	2506 F·g ⁻¹	1 A·g ⁻¹	[10]
CoO-GHBs/CC	5 M KOH	2238 F·g ⁻¹	1 A·g ⁻¹	[11]
CoO/NiO–Cu@CuO	6 M KOH	2.03 F·cm ⁻²	2 mA·cm ⁻²	[12]
CoO/Co-Cu-S	2 M KOH	2300 F·g ⁻¹	2 A·g ⁻¹	[13]
CuCoO-H	3 M KOH	405.36 F·g ⁻¹	20 A·g ⁻¹	[14]
Cu-Ni(OH)₂@CoO NFCs	3 M KOH	5.8 F·cm⁻²	1 mA·cm⁻²	This work

Table S4. Comparison of Ni/Co-based materials assembled to HASC in the field of supercapacitors

HASC	C _s (mF cm ⁻²)	E (mWh cm ⁻²)	P (mW cm ⁻²)	Reference
Ni@Ni(OH)₂/ACC	1910	0.6	0.37	[15]
Ni₂Co₄@NiCo₂O₄/CFP	1250	0.44	0.16	[16]
NiCo₂O₄ NG @ CF	25	0.001	2	[17]
PNH@NeC LDH//AC	21	0.323	1.6	[18]
NiO/Ni(OH)₂/PEDOT//AC	404	0.01	7.8	[19]
NiO@MnCo-LDH//Ni/AC	368	0.02	0.38	[20]
3D-NiMoO₄/Ni@CW//rGO/CF	504	0.2	4.1	[21]
Co(OH)₂@Ni/Ag-mesh	248	0.00042	0.00833	[22]
blanket-like Co(OH)₂/CoOOH/Co₃O₄/Cu(OH)₂	888	0.196	1.6	[23]
NiO//α-Fe₂O₃	229	0.05	0.3	[24]
Cu-Ni(OH)₂@CoO NFCs//Porous carbon	996.6	0.45	9.0	This work

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