Supplementary Information

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3 Title

Formation mechanism of Ruddlesden-Popper faults in compressive-strained ABO₃ perovskite
 superlattices

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Fig. S1 (a) Formation of RP fault as a result of atomic steps on the surface of the LaSrAlO₄ substrate. Due to the rock-salt type structure of LaSrAlO₄, the atomic step is displaced with regard to the underlying layer by a/2[111] (in pseudocubic axes). Subsequent deposition leads to an RP fault on the (010) plane. (b) Formation of RP fault due to nonstoichiometry. During the growth, the excessive A-site cations must be accommodated by the formation of an AO island on the fully coalesced AO layer. To minimize the electrostatic repulsion between the neighboring A-site cations, the AO island is displaced by a/2[111] with respect to the underlying crystal, i.e., rock-salt type. The subsequent epitaxial growth results in an antiphase domain encompassed in RP faults on {100} planes.



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Fig. S2 (a) Atomic model of PrNiO₃ viewed in [100] direction. The crystal can be described by a pseudocubic symmetry with the lattice parameter of $a_{PNO,pseudo} = 3.82$ Å (b) Atomic model of PrAlO₃ viewed in [100] direction with $a_{PAO,pseudo} = 3.76$ Å. (c)

39 Atomic structure of the LaSrAlO₄ substrate. LaSrAlO₄ possesses tetragonal symmetry with a = b = 3.75 Å and c = 12.65Å.



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Fig. S3 (a) AC-HRTEM image acquired at the substrate-superlattice interface in [100] projection. Negative C_s imaging conditions have been applied with the spherical aberration coefficient C_s tuned to $-15 \mu m$ and an overfocus of c.a. 5 nm, resulting in brightatom contrast. The number on the left denotes the numbering of unit cells along the growth direction [001]. (b) Measured out-ofplane lattice parameter d_{001} and in-plane lattice parameter d_{010} as a function of the number of unit cells along the growth direction. Each data point is averaged over 40 unit cells along the in-plane direction, and the error bar corresponds to the standard error. (c) AC-HRTEM image acquired at the superlattice surface. (b) Measured out-of-plane lattice parameter d_{001} and in-plane lattice parameter d_{010} as a function of the number of unit cells along the growth direction. agreeing well with the [4 u.c.//4 u.c.] PrNiO₃/PrAIO₃ bilayer structure.



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Fig. S4 (a) AC-HRTEM image acquired at the RP fault in [100] projection. Negative Cs imaging conditions has been applied with 53 the spherical aberration coefficient C_s tuned to - 15 μ m and an overfocus of c.a. 5 nm, resulting in bright-atom contrast. The red arrows denote the RP fault plane. (b) In-plane strain map ε_{xx} showing a local expansion of the in-plane lattice parameter d_{020} . (c) 54 AC-HRTEM image acquired at the same position as in (a). An underfocus of c.a. - 1 nm has been applied, resulting in dark-atom 55 56 contrast. (d) Comparison between the experimental and simulated image, showing clear contrast distinction between Pr and Ni/Al 57 cations.

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Fig. S5 (a) AC-HRTEM image acquired at the RP fault in [001] projection (i.e., plan view). The red arrows indicate the RP fault plane. (b) In-plane strain map ε_{xx} showing a local expansion of the in-plane lattice parameter d_{020} . 62

(a) (b) (c) RP fault RP fault 100111 1-10101

Fig. S6 (a) AC-HRTEM image acquired at the origin of the RP faults observed in Fig. 3 and S4. Negative C_s imaging conditions has been applied with the spherical aberration coefficient C_s tuned to $-15 \,\mu$ m and an overfocus of c.a. 5 nm, resulting in bright-atom

contrast. (b) In-plane strain map ε_{xx} showing a local expansion of the in-plane lattice parameter d_{020} and two dislocation cores. (c)

AC-HRTEM image acquired at the same position as in (a). An underfocus of c.a. - 1 nm has been applied, resulting in dark-atom

contrast.

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74 Fig. S7 AC-HRTEM images acquired with dark-atom contrast showing RP faults (indicated by the red arrows) separated by only 1
75 unit cell. No further broadening was observed till the superlattice surface.

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