B₂N Monolayer: a Direct Band-Gap Semiconductor with High and Highly

Anisotropic Carrier Mobility

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Supplementary methods

Carrier mobility calculations

The carrier mobility of two-dimensional materials based on the deformation potential theory was estimated by using the following expression^{1–3}:

$$\mu = \frac{e\hbar^3 C_{2D}}{k_B T m^* m_d E_1^2}$$

Here, C_{2D} , E_1 and m^* represent the elastic modulus of the longitudinal strain, the deformation potential constant of the VBM for holes or CBM for electrons, and the carrier effective mass in the

$$C_{2D} = \frac{2(E - E_0)}{S_0 \left(\frac{\Delta l}{l_0}\right)^2}$$

transport direction, respectively. The C_{2D} is defined as $\binom{l_0}{l_0}$, where E, E_0, S_0, l_0 represent the total energy for the compressed/dilated structure, the total energy at equilibrium, the equilibrium lattice area, the equilibrium lattice constant in the transport direction and the deformation of l_0 ,

$$\frac{\Delta E}{\Delta l}$$

respectively. The E_1 equals to U_0^{\prime} , in which *E*, *l*, and Δl are donated as the energy change of CBM or VBM under compression or tensile strain, the lattice constants in the transport direction, and the deformation of l_0 . m^* is the effective mass in transport direction, and m_d is determined by $m_d = \sqrt{m_x^* m_y^*}$ (m_x^* is perpendicular to m_y^*). The temperature *T* is set to be 300 K, and $k_{\rm B}$ is Boltzmann constant.

Absorption spectra calculations

Absorption spectra is expressed by $A(\omega) = 1 - e^{-\alpha(\omega) \cdot \Delta z}$, where $\alpha(\omega) = \frac{\omega \varepsilon_2}{cn}$ is the absorption

$$n = \left| \frac{\varepsilon_1^2 + \varepsilon_2^2 + \varepsilon_1}{2} \right|.$$

coefficient, $\sqrt{2}$ is the index of refraction, ω is the light frequency, ε_1 and ε_2 are the real and imaginary parts of the dielectric function, *c* is the speed of light in vacuum, Δz represents the unitcell size in the *c* direction. The electronic structures were obtained from HSE06 level. The *k*-mesh is doubled and the number of bands is twice that of the optimized calculation. To further validate the effectiveness of our calculation, we have calculated the absorption spectra of cd-silicon using the same parameters, and the results is in agreement with the reported ones⁴.

Detail on the structure prediction

The particle swarm optimization (PSO) method within the evolutionary algorithm as implemented in the Crystal structure AnaLYsis by Particle Swarm Optimization (CALYPSO) code was employed to find the lowest energy structures of B_xN_y ($1 \le x, y \le 4$) monolayers. Unit cells containing 1- 4 formula units (f.u.) were considered. In the first step, random structures with certain symmetry are constructed in which atomic coordinates are generated by the crystallographic symmetry operations. Local optimizations using the VASP code were done with the conjugate gradients method and stopped when Gibbs free energy changes became smaller than 1×10^{-5} eV per cell. After processing the first generation structures by PSO. 40% of the structures in the new generation are randomly generated. A structure fingerprinting technique of bond characterization matrix is applied to the generated structures, so that identical structures are strictly forbidden. These procedures significantly enhance the diversity of the structures, which is crucial for structural global search efficiency. In most cases, structural searching simulations for each calculation were stopped after generating 1000~ 1200 structures (e.g., about 20 ~ 30 generations).

Supplementary Tables

Table SI. The calculated elastic constants (C_{ij} , N/m), Young's modulus (E_Y , N/m) and Poisson's ratio (P_r) of B₂N, B₃N and B₄N monolayers.

	<i>C</i> ₁₁	<i>C</i> ₂₂	<i>C</i> ₁₂	<i>C</i> ₆₆	$E_{\rm Yx}$	$E_{\rm Yy}$	P _{rx}	P _{ry}
B ₂ N	208.72	161.40	31.62	48.23	202.53	156.61	0.19	0.15

B ₃ N	157.16	171.14	13.66	90.34	156.07	169.95	0.08	0.09
B ₄ N	186.85	88.14	29.77	85.47	176.79	83.40	0.34	0.16

Madariala	x direction		y direction		Band gap	Direct or	Defe
Materials -	μ_{e}	μ_h	μ_e	μ_h	(eV)	indirect	Keis
graphene	339000	322000	320000	351000	0	D	5
graphdiyne	208100	19700	172200	19100	0.46	D	6
phosphorene	1100	640	80	10000	1.51	D	7
Silicene	258000	223000	257000	222000	0	D	8
As	21	66	21	66	2.47	ID	9
Kagome-P	1090	20	1090	1540	1.64	ID	10
Kagome-As	580	40	570	1430	1.9	ID	10
Kagome-Sb	510	120	510	4570	1.71	ID	10
Sb	45	34	15	16	2.38	ID	11
Be ₂ N ₆	10000	1090	9270	1210	1.71	ID	12
<i>h</i> -BN	34	36	34	36	5.6	D	13
BP	49960	13700	68810	26050	0.85	D	9
C_3N	448	155	529	2205	1.22	ID	14
C_2N	41	30	730	332	2.46	D	14
SiS	1400	366	9490	1300	1.22	D	15
Si ₃ S	15970	8890	106	360	0.94	D	16
Si ₃ O	12920	7340	136	887	1.19	D	16
SiAs	1250	390	310	1120	2.37	D	17
PC_6	294000	248000	3160	164000	0.84	D	18
P_3Cl_2	740	700	56890	26450	2.41	D	19
CaP ₃	10000	80	1000	780	1.15	D	20
ScN	117000	14	12200	137	1.39	D	21
GaSe	73	1181	58	70	2.94	ID	22
GaTe	992	128044	600	547	1.88	ID	22
GeS	2430	116	320	60	2.29	ID	23
GeSe	4032	323	1555	76	1.59	D	23
GeAs	2380	280	1170	550	2.04	D	17
GeTe	1049	294	603	1066	2.35	D	24
As_2S_3	253	10	168	13	3.11	ID	25
MoS_2	72	184	73	212	1.81	D	26
BCN	176	88	23	164	3.19	ID	27
BC_2N	52550	14820	3700	270	1.6	D	28
BCP	1590	1610	2290	840	1.18	D	27
PC_6N	36	13700	30	10	2.56	ID	29
Ti ₂ CO ₂	611	74100	254	22500	1.28	D	30
Zr_2CO_2	83	2300	1096	1695	1.76	D	30
Hf_2CO_2	126	2192	2270	1598	1.79	D	30
TiNF	500	2100	550	1000	1.52	D	31

Table SII. Carrier mobility and band gap of some 2D materials.

TiNCl	550	1900	1100	2500	1.42	D	31
TiNBr	800	1800	1700	2500	1.40	D	31
CrPbTe ₃	62	129	31	79	0.25	ID	32
CuLiSe	3019	266	3019	266	1.60	D	33
CuLiTe	2766	494	2766	494	1.85	D	33
$Zn_2C_2P_2$	145	98689	10024	2361	1.16	D	34
MoSSe	530	58	490	55	2.02	D	35
PdPS	312	13	55	249	2.12	Ι	36
PdPSe	197	0.34	2.04	344.33	1.95	Ι	36

Table SIII. The optimized structural parameters of B_2N , B_3N , and B_4N monolayers.

Space group	Compound	<i>a</i> , <i>b</i> , <i>c</i> (Å) <i>α</i> , <i>β</i> , <i>γ</i> (deg)	Atomic position		
			B1(1a) (0, 0, 0.50057)		
			B2(1a) (0.33334,0.33335,0.50060)		
			B3(1a) (0.62705, 0.47124, 0.50058)		
			B4(1a) (0.90168, 0.62706, 0.50053)		
		<i>a</i> =6.8226	B5(1a) (0.47122, 0.90175, 0.50056)		
		<i>b</i> =6.8228	B6(1a) (0.05291, 0.20754, 0.50058)		
D1	DN	<i>c</i> =25.1132	B7(1a) (0.73957, 0.05286, 0.50057)		
1 1	D 41 N 3	<i>α</i> =92.771	B8(1a) (0.20755, 0.73955, 0.50057)		
		<i>β</i> =94.2301	N1(1a) (0.14486,0.56511, 0.50055)		
		<i>γ=60.0012</i>	N2(1a) (0.29002, 0.14489, 0.50061)		
			N3(1a) (0.56513, 0.29001, 0.50062)		
			N4(1a) (0.70238, 0.86015, 0.50052)		
			N5(1a) (0.43747, 0.70240, 0.50059)		
			N6(1a) (0.86016, 0.43743, 0.50054)		
		<i>a</i> = <i>b</i> =7.8011	B1(4h) (0.92162, 0.57838, 0.5)		
P4/mbm	B_3N_2	<i>c</i> =34.7066	B2(8j) (0.79886, 0.85661, 0.5)		
		$\alpha = \beta = \gamma = 90$	N(8j) (0.03779, 0.23684, 0.5)		
		<i>a</i> =10.3074	$D_1(A_2) = (0.18220, 0.17110, 0.5)$		
4	DN	<i>b</i> =3.9085	B1(4e)(0.18529, 0.17110, 0.5) B2(4e)(0.01(52, 0.81220, 0.5))		
Amm2	B ₂ IN	c=29.6758	B2(4e)(0.91052, 0.81529, 0.5)		
		$\alpha = \beta = \gamma = 90$	N(4e)(0.17930, 0.34204, 0.3)		
			B1(2 <i>a</i>) (0.04209, 0.5, 0.49559)		
		-17 1227	B2(2 <i>a</i>) (0.29435, 0.5, 0.50877)		
	B ₃ N	u = 17.1527	B3(2a) (0.38528, 0.5, 0.48601)		
Cm		D=2.0239	B4(2 <i>a</i>) (0.16873, 0, 0.50239)		
Cm		c = 51.8302	B5(2 <i>a</i>) (0.44426, 0, 0.48942)		
		$\alpha - \gamma - 90$	B6(2 <i>a</i>) (0.35416, 0, 0.50881)		
		p=92.0100	N1(2 <i>a</i>) (0.21026, 0.5, 0.50455)		
			N2(2a) (0.08415, 0, 0.49779)		
C2/m	B_4N	<i>a</i> =10.37839	B1(4 <i>i</i>) (0.45919, -0.5, 0.50157)		

<i>b</i> =2.6918	B2(4 <i>i</i>) (0.68824, -0.5, 0.49117)
<i>c</i> =60.8509	B3(4 <i>i</i>) (0.64321, -1, 0.49385)
<i>α=γ=</i> 90	B4(4 <i>i</i>) (0.27345, 0, 0.49812)
$\beta = 100.7124$	N(4 <i>i</i>) (0.42552, 0, 0.50286)

Table SIV. The energy differences between AA and AB stacking calculated by nonlocal optB86b-vdW density functional, optB88-vdw density functional and DFT-D2 method.

Energy	Methods							
(eV/atom)	optB86b	optB88	DFT-D2	PBE				
E_{AA}	0.02117	0.01528	0.02539	-0.001				
E_{AB}	0	0	0	0				

Supplementary Figures



Figure S1. Low-energy structures and the energy differences (eV/atom) relative to ground state structures.





Figure S2. Phonon dispersion curves of (a) B₂N, (b) B₃N and (c) B₄N monolayers

Figure S3. Snapshots of the final frame of each molecular dynamic simulation at 300, 1000 and 1500 K for (a) B_2N , (b) B_3N and (c) B_4N after 6 ps of simulated annealing.



Figure S4. Three-dimensional electron localization functions (isosurface value = 0.75) of (a) B_2N , (b) B_3N and (c) B_4N .



Figure S5. Orbit-resolved band structures and PDOS for *h*-BN.



Figure S6. The three stacking configurations of trilayer B_2N and the energy differences (eV/atom) relative to ABC stacking using the optB68b-vdW functional.



Figure S7. The band structure of ABC stacking of B₂N at the HSE06 level.



Figure S8. Adsorption spectra of B_2N , B_3N and B_4N monolayers compared to that of phosphorene, diamond silicon (cd-Si) and MoS₂ monolayer.



Figure S9. The crystal structures of predicted B₃N and B₄N monolayers and their energy differences.



Figure S10. Partial charge density of the VBM and the CBM for B_3N (a-b) and B_4N (c-d), respectively. Band decomposed charge densities calculations suggest that both VBM and CBM of B_3N are mainly attributed by the π bonds from hybridized B-B *p* orbital electrons, while VBM and CBM of B_4N are contributed by the σ bonds and π bonds from hybridized B-B *p* orbital electrons, respectively.



Figure S11. The band structure of B_4N monolayer along the $Y \rightarrow \Gamma$ direction using (a) 50 K-points and (b) 3000 K-points at the PEB level.



Figure S12. (a) The total energy difference of B_2N monolayer at the strain between -1% to 1% along the *x* and *y* directions. Energy shift of CBM and VBM for B_2N with respect to the vacuum energy as a function of lattice dilation

and compression along the (b) x and (c) y directions.



Figure S13. (a) The total energy difference of AB stacking of B_2N at the strain between -1% to 1% along the *x* and *y* directions. Energy shift of CBM and VBM for bilayer B_2N with respect to the vacuum energy as a function of lattice dilation and compression along the (b) *x* and (c) *y* directions.

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