## Supporting Information

## Toughened PLA-b-PCL-b-PLA Triblock Copolymer based Biomaterials: Effect of Self-Assembled Nanostructure and Stereocomplexation on the Mechanical Properties

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Figure S1: ${ }^{1} \mathrm{H}-\mathrm{NMR}$ spectra of homopolymers and the 20D-11CL-20D triblock copolymer
*The number average molecular weight $(\mathrm{Mn})$ of the homopolymers/block copolymers is determined from ${ }^{1} \mathrm{H}-\mathrm{NMR}$ (Figure S1) by using the following equations (S1-i to S1-v):

Degree of polymerization of dihydroxyl terminated PCL, $D P_{P C L}=\frac{e}{j}+2$

Degree of polymerization of dihydroxyl terminated PLA, $D P_{P L A}=\frac{l}{m}+2$

Number average molecular weight of dihydroxyl terminated PCL,

$$
\begin{align*}
M n_{P C L} & =\left(M o_{P C L} \times D P_{P C L} \times 2\right)+M o_{D M G} \\
& =\left[114.14 \times\left(\frac{e}{j}+2\right) \times 2\right]+202.34 \tag{S1-iii}
\end{align*}
$$

Number average molecular weight of dihydroxyl terminated PLA,

$$
\begin{align*}
M n_{P L A} & =\left(M o_{P L A} \times D P_{P L A} \times 2\right)+M o_{D M G} \\
& =\left[72 \times\left(\frac{l}{m}+2\right) \times 2\right]+202.34 \tag{S1-iv}
\end{align*}
$$

Number average molecular weight of triblock copolymer (PLA-PCL-PLA)

$$
\begin{align*}
M n_{\text {triblock }} & =\left[\left(M o_{P C L} \times D P_{P C L}\right)+M o_{D M G}\right]+\left[\left(M o_{P L A} \times D P_{P L A} \times 2\right)+M o_{D M G}\right] \\
& =\left[\left\{114.14 \times\left(\frac{e}{j}+2\right) \times 2\right\}+202.34\right]+\left[\left\{72 \times\left(\frac{l}{m}+2\right) \times 2\right\}+202.34\right] \tag{S1-v}
\end{align*}
$$

where, $M o=$ molar mass of repeating unit

$$
M_{D M G}=\text { molar mass of initiator }
$$







Figure S2: ${ }^{13} \mathrm{C}$-NMR spectra of homopolymers and the 20D-11CL-20D triblock copolymer


Figure S3: GPC curves of triblock copolymers


Figure S4: Normalized crystallinity of PCL in the triblock copolymers and blend specimens as determined from WAXS results. Note that this crystallinity can be considered as the normalized one with respect to the PCL content.


Figure S5: Normalized crystallinity of sc (stereocomplex crystallites) in the blend specimens as determined from WAXS results. Note that this crystallinity can be considered as the normalized one with respect to the sc content.



Figure S6: WAXS profiles for (a) the specimen 20D-11CL-20D and (b) PLA homopolymer. The panel (c) indicates subtracted profile. Note here that the amorphous halo of PLA (b) showing peaks at $11.0,14.6$ and $22.8 \mathrm{~nm}^{-1}$ was subtracted from the WAXS profile for the triblock copolymer 20D-11CL-20D (a) to completely remove the contribution of amorphous halo due to PLA to yield (c), where the peaks $11.3,13.7$ and $22.2 \mathrm{~nm}^{-1}$ are assigned to the amorphous halo of PCL, whereas, the diffraction peaks are observed at 14.9 and $16.6 \mathrm{~nm}^{-1}$. Namely $(c)=(a)-(b) \bullet$ factor and the factor was estimated as follows:

The WAXS profile of PLA (b) was subtracted from that of 20D-11CL-20D (a) such that there was no overestimation of the subtraction which would lead to the negative value of the scattering intensity.

As the content of PLA was $79 \%$ in 20D-11CL-20D specimen, the subtraction factor was chosen as 0.8 so as to give $(c)=(a)-(b) \cdot 0.8$, which would remove completely the amorphous halo of PLA. The factor was adjusted until the complete removal of amorphous PLA halo was achieved.

The similar method was used to completely remove the amorphous halo of PLA for the other specimens.

Table S1: Summaries of the DSC results

| Specimen code name | $\begin{gathered} \mathrm{T}_{\mathrm{m}, \mathrm{PCL}} \\ \left({ }^{\circ} \mathrm{C}\right) \end{gathered}$ | $\begin{gathered} \Delta \mathrm{H}_{\mathrm{m}}, \mathrm{PCL} \\ (\mathrm{~J} / \mathrm{g}) \end{gathered}$ | $\mathrm{T}_{\mathrm{cc}}$, PDLA <br> $\left({ }^{\circ} \mathrm{C}\right)$ | $\begin{gathered} \Delta \mathrm{H}_{\mathrm{cc}}, \text { PDLA } \\ (\mathrm{J} / \mathrm{g}) \end{gathered}$ | $\mathrm{T}_{\mathrm{m}}$, pDLA <br> $\left({ }^{\circ} \mathrm{C}\right)$ | $\begin{gathered} \Delta \mathrm{H}_{\mathrm{m}, \text { PDLA }} \\ (\mathrm{J} / \mathrm{g}) \end{gathered}$ | $\begin{gathered} \mathrm{T}_{\mathrm{m}}, \text { scPLA } \\ \left({ }^{\circ} \mathrm{C}\right) \end{gathered}$ | $\begin{gathered} \Delta \mathrm{H}_{\mathrm{m}, \mathrm{scPLA}} \\ (\mathrm{~J} / \mathrm{g}) \end{gathered}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 50D | - | - | 103.6 | 41.3 | 175.4 | 49.8 | - | - |
| 10D-11CL-10D | 54.5 | 34.3 | 72.8 | 13.8 | 169.8 | 36.6 | - | - |
| 20D-11CL-20D | 53.8 | 11.9 | 83.3 | 24.7 | 173.2 | 38.9 | - | - |
| 9D-23CL-9D | 59.5 | 43.0 | 68.1 | 4.5 | 169.4 | 27.5 | - | - |
| 19D-18CL-19D | 56.6 | 33.5 | 77.6 | 21.8 | 174.5 | 38.0 | - | - |
| 50CL | 64.2 | 78.8 | - | - | - | - | - | - |
| B1 ${ }^{1)}$ | 53.5 | 23.5 | - | - | 167.2 | 3.9 | 221.5 | 45.3 |
| B2 ${ }^{2)}$ | 54.0 | 17.3 | 94.7 | 18.8 | 172.8 | 27.7 | 222.0 | 25.3 |
| B3 ${ }^{3)}$ | 59.5 | 52.9 | - | - | 168.3 | 8.6 | 218.2 | 22.0 |
| B4 ${ }^{4}$ | 59.2 | 39.3 | 99.8 | 17.3 | 170.4 | 15.6 | 219.8 | 32.5 |

[^0]Crystallinity of PCL from DSC:
$\mathrm{X}_{\mathrm{c}, \mathrm{PCL}}(\%)=\frac{\Delta H_{m, P C L}}{\Delta H^{\circ}{ }_{m, P C L} \times W_{P C L}} \times 100$
wherse, $\Delta H^{\circ}{ }_{m, P C L}=139.5 \mathrm{~J} / \mathrm{g}$ and
$W_{P C L=}$ weight fraction of PCL

Table S2: Summaries of the stress-strain results

| Specimen code name | Yield strength (MPa) | Ultimate tensile strength (MPa) | Elongation at break (\%) | $\begin{gathered} \text { Young's } \\ \text { modulus (MPa) } \end{gathered}$ | Tensile toughness ( $\mathrm{MJ} / \mathrm{m}^{3}$ ) |
| :---: | :---: | :---: | :---: | :---: | :---: |
| 50D | $39.7 \pm 1.9$ | $39.7 \pm 1.9$ | $6.8 \pm 2.0$ | $1387.2 \pm 83.4$ | $2.0 \pm 0.7$ |
| 10D-11CL-10D | $12.9 \pm 0.8$ | $12.9 \pm 0.8$ | $544.7 \pm 91.2$ | $508.2 \pm 91.9$ | $51.5 \pm 12.3$ |
| 20D-11CL-20D | $36.2 \pm 1.4$ | $36.2 \pm 1.4$ | $708.3 \pm 199$ | $792.7 \pm 74.4$ | $160.3 \pm 19.9$ |
| 9D-23CL-9D | $15.9 \pm 3.2$ | $15.9 \pm 3.2$ | $598.5 \pm 151$ | $677.5 \pm 62$ | $69.3 \pm 29.4$ |
| 19D-18CL-19D | $21.1 \pm 0.6$ | $21.3 \pm 0.8$ | $949.6 \pm 70.9$ | $580.2 \pm 57.9$ | $147.3 \pm 16.3$ |
| B1 ${ }^{1)}$ | $9.9 \pm 0.8$ | $15.6 \pm 1.4$ | $540.6 \pm 58$ | $345.6 \pm 54$ | $65.4 \pm 11.6$ |
| B2 ${ }^{2)}$ | $32.5 \pm 3.1$ | $32.5 \pm 3.1$ | $661.2 \pm 121.5$ | $830.1 \pm 56.1$ | $152.9 \pm 22.3$ |
| B3 ${ }^{3)}$ | $13.8 \pm 1.3$ | $13.8 \pm 1.3$ | $430.5 \pm 42$ | $602.5 \pm 72$ | $38.5 \pm 8.7$ |
| B4 ${ }^{4}$ | $20.1 \pm 2.4$ | $20.1 \pm 2.4$ | $455.3 \pm 37.3$ | $813.3 \pm 130.8$ | $70.4 \pm 7.8$ |
| 50CL | $13.1 \pm 0.8$ | $26.9 \pm 1.6$ | $1683 \pm 79.4$ | $143.6 \pm 19.7$ | $273.4 \pm 19.7$ |

${ }^{1)} 10 \mathrm{D}-11 \mathrm{CL}-10 \mathrm{D} / 10 \mathrm{~L}-11 \mathrm{CL}-10 \mathrm{~L}$ (50/50) blend specimen 2) $20 \mathrm{D}-11 \mathrm{CL}-20 \mathrm{D} / 20 \mathrm{~L}-10 \mathrm{CL}-20 \mathrm{~L}$ (50/50) blend specimen ${ }^{3}$ ) 9D-23CL-9D/8L-20CL-8L (50/50) blend specimen ${ }^{4}$ ) 19D-18CL-19D/16L-22CL-16L (50/50) blend specimen


Figure S7: Plots of $\tan \delta$ vs Temperature $\left({ }^{\circ} \mathrm{C}\right)$ for (a) the neat triblock copolymers and (b) enantiomeric blends, as determined from dynamic mechanical analysis


Figure S8: Representative plots of $\mathrm{E}^{\prime \prime}$ vs PLA content (\%) at (a) $30^{\circ} \mathrm{C}$ and (b) $110^{\circ} \mathrm{C}$, for the neat triblock copolymers and enantiomeric blends, as determined from dynamic mechanical analysis


Figure S9: 1d SAXS profiles of (a) neat triblock copolymers and (b) triblock copolymer blends (as-prepared specimens).

Table S3: $d$-spacing of the microdomains and long period of the PCL crystalline lamellar stacking for the neat block copolymers and their blends, as evaluated from SAXS results

| Specimen code name | $d$-spacing of the <br> microdomains, $(\mathrm{nm})$ | Long period of the PCL <br> crystalline lamellar stacking, <br> $(\mathrm{nm})$ |
| :---: | :---: | :---: |
| 50CL | 15.8 | - |
| 10D-11CL-10D | 45.9 | 15.5 |
| 20D-11CL-20D | 52.5 | 19.6 |
| 9D-23CL-9D | 63.9 | 19.6 |
| 19D-18CL-19D $^{\text {B1 }}{ }^{\text {1) }}$ | 61.3 | 22.6 |
| B2 ${ }^{\text {2) }}$ | 36.4 | 13.6 |
| B3 $^{\text {3) }}$ | 43.5 | 14.9 |
| B4 $^{\text {4) }}$ | 51.3 | 16.1 |

${ }^{1}$ ) $10 \mathrm{D}-11 \mathrm{CL}-10 \mathrm{D} / 10 \mathrm{~L}-11 \mathrm{CL}-10 \mathrm{~L}$ (50/50) blend specimen
2) $20 \mathrm{D}-11 \mathrm{CL}-20 \mathrm{D} / 20 \mathrm{~L}-10 \mathrm{CL}-20 \mathrm{~L}$ ( $50 / 50$ ) blend specimen
${ }^{3)} 9 \mathrm{D}-23 \mathrm{CL}-9 \mathrm{D} / 8 \mathrm{~L}-20 \mathrm{CL}-8 \mathrm{~L}$ (50/50) blend specimen
${ }^{4}$ ) 19D-18CL-19D/16L-22CL-16L (50/50) blend specimen


Figure S10: 1d SAXS profiles of the triblock copolymers measured at $210^{\circ} \mathrm{C}$

Crystallinity of PCL, $X_{c, P C L}=\frac{2 l_{c}}{l_{a}+2 l_{c}}$

Table S4: Weight fraction and volume fraction of PCL in the triblock copolymer system

| Specimen code name | Weight fraction of PCL | Volume fraction of PCL |
| :---: | :---: | :---: |
| 10D-11CL-10D | 0.35 | 0.38 |
| 20D-11CL-20D | 0.21 | 0.24 |
| 9D-23CL-9D | 0.56 | 0.59 |
| 19D-18CL-19D | 0.32 | 0.35 |

Weight fraction of PCL $=\frac{\text { Weight of PCL block }}{\text { Total weight of block copolymer }}$
Volume fraction $=\frac{M_{P C L} / \rho_{P C L}}{M_{P C L} / \rho_{P C L}+{ }^{M_{P L A}} / \rho_{P L A}}$
$\rho_{P C L}=X_{c} \rho_{c}+\left(1-X_{c}\right) \rho_{a}$, where $\rho_{c}=1.2 \mathrm{~g} / c c,{ }_{1} \rho_{a}=1.08 \mathrm{~g} / c c \quad$ (S13-iii)
$\rho_{P L A}=X_{c} \rho_{c}+\left(1-X_{c}\right) \rho_{a}$, where $\quad \rho_{c}=1.29 \mathrm{~g} / c c\{$ Auras, $2004 \# 34\}, \quad 2 \quad \rho_{a}=1.25 \mathrm{~g} / c c$ (S13-iv)
where $M_{P C L}$ and $M_{P L A}$ denote the molecular weights of PCL and PLA respectively.

## References

(1) Crescenzi, V.; Manzini, G.; Calzolari, G.; Borri, C. Thermodynamics of fusion of poly- $\beta$-propiolactone and poly- $\epsilon$-caprolactone. comparative analysis of the melting of aliphatic polylactone and polyester chains. European Polymer Journal 1972, 8, 449-463.
(2) Auras, R.; Harte, B.; Selke, S. An overview of polylactides as packaging materials. Macromolecular bioscience 2004, 4, 835-864.


[^0]:    ${ }^{1)}$ 10D-11CL-10D/10L-11CL-10L (50/50) blend specimen
    2) $20 \mathrm{D}-11 \mathrm{CL}-20 \mathrm{D} / 20 \mathrm{~L}-10 \mathrm{CL}-20 \mathrm{~L}(50 / 50$ ) blend specimen
    ${ }^{3)} 9 \mathrm{D}-23 \mathrm{CL}-9 \mathrm{D} / 8 \mathrm{~L}-20 \mathrm{CL}-8 \mathrm{~L}(50 / 50)$ blend specimen
    ${ }^{4}$ ) 19D-18CL-19D/16L-22CL-16L (50/50) blend specimen

