

Supporting Information

**Spatial charge separation and high-index facet dependence
in polyhedral Cu₂O type-II surface heterojunctions for
photocatalytic activity enhancement**

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Experimental Section

Computational details. Utilizing the projected augmented wave (PAW) method to study density functional calculations, executed in the Vienna *ab initio* simulation package (VASP) code.^[1] The exchange-correlation functional adopted the generalized gradient approximation of Perdew, Burke, and Ernzerhof (PBE).^[2] In order to minimize the interaction between adjacent slabs, the periodic boundary condition with a vacuum space of 15 Å was used. The kinetic energy cutoff was programmed to be 400 eV. Monkhorst–Pack uncommon k-point meshes^[3] of 3×3×1 and 1×2×1 was recommended to {001} and {332} facets in the optimization process, while 7×7×1 and 3×7×1 were embraced by DOS calculations.

The surface free energy of Cu₂O {111} and {332} facets was calculated by *ab initio* thermodynamic approach, according to the surface linking with an oxygen atmosphere, which was described by an oxygen pressure, p , and temperature, T . Since surface system is modeled by symmetrical slabs with two equivalent surfaces, the surface free energy at given (T, p) can be expressed as:

$$\gamma_{\text{surf}}(T, p) = \frac{1}{2A} [G_{\text{slab}}(T, p, N_{\text{Cu}}, N_{\text{O}}) - N_{\text{Cu}}\mu_{\text{Cu}}(T, p) - N_{\text{O}}\mu_{\text{O}}(T, p)] \quad (1)$$

Here, A is the area exposed to a unit cell, G_{slab} is the Gibbs free energy of the slab, μ_{Cu} and μ_{O} are the chemical potentials of copper and oxygen, while N_{Cu} and N_{O} are the number of Cu and O atoms. Thermodynamically the surface with the lowest $\gamma_{\text{surf}}(T, p)$ under certain conditions (T, p) corresponds to the most stable surface.

It has been proved in eq (1)^[4,5] that Gibbs free energies of solids can be expressed as a first approximation total energies. Thus $\gamma_{\text{surf}}(T, p)$ can be approximated by:

$$\gamma_{\text{surf}}(T, p) \approx \frac{1}{2A} \left[E_{\text{slab}} - \frac{N_{\text{Cu}}}{2} E_{\text{Cu}_2\text{O}}^{\text{bulk}} - \Delta N_{\text{O}}^{\text{stoich}} \mu_{\text{O}}(T, p) \right] \quad (2a)$$

where

$$\Delta N_{\text{O}}^{\text{stoich}} = (N_{\text{O}} - \frac{N_{\text{Cu}}}{2}) \quad \text{and} \quad (2b)$$

$$E_{\text{Cu}_2\text{O}}^{\text{bulk}} \approx 2\mu_{\text{Cu}}(T, p) + \mu_{\text{O}}(T, p). \quad (2c)$$

The E_{slab} and $E_{\text{Cu}_2\text{O}}^{\text{bulk}}$ are total energies of the slab and of the formula unit of Cu_2O bulk, respectively. The $\Delta N_{\text{O}}^{\text{stoich}}$ is the number of excess O atoms.

Before establishing the function between oxygen chemical potential (μ_{O}) and $\gamma_{\text{surf}}(T, p)$, the oxygen poor limit (O_{lean}) and oxygen rich limit (O_{rich}) should be considered. The O_{lean} can be regarded as μ_{O} corresponding to the decomposition of bulk Cu_2O into bulk Cu and O_2 gas, nevertheless at O_{rich} oxygen gas condenses on the surface. The O_{lean} and O_{rich} limits are determined by eqs (3a) and (3b), respectively.

$$\mu_{\text{O}}^{\text{min}} = E_{\text{Cu}_2\text{O}}^{\text{bulk}} - E_{\text{Cu}}^{\text{bulk}} \quad \text{and} \quad \mu_{\text{Cu}}^{\text{max}} = E_{\text{Cu}}^{\text{bulk}} \quad (3a)$$

$$\mu_{\text{O}}^{\text{max}} = \frac{1}{2} E_{\text{O}_2} \quad \text{and} \quad \mu_{\text{Cu}}^{\text{min}} = \frac{1}{2} \left[E_{\text{Cu}_2\text{O}}^{\text{bulk}} - \frac{1}{2} E_{\text{O}_2} \right] \quad (3b)$$

where $E_{\text{Cu}}^{\text{bulk}}$ and E_{O_2} are total energies of Cu atom in the Cu-bulk and isolated O_2 molecule, respectively. Half the total energy of O_2 molecule is chosen as the zero reference for $\mu_{\text{O}}(p, T)$.

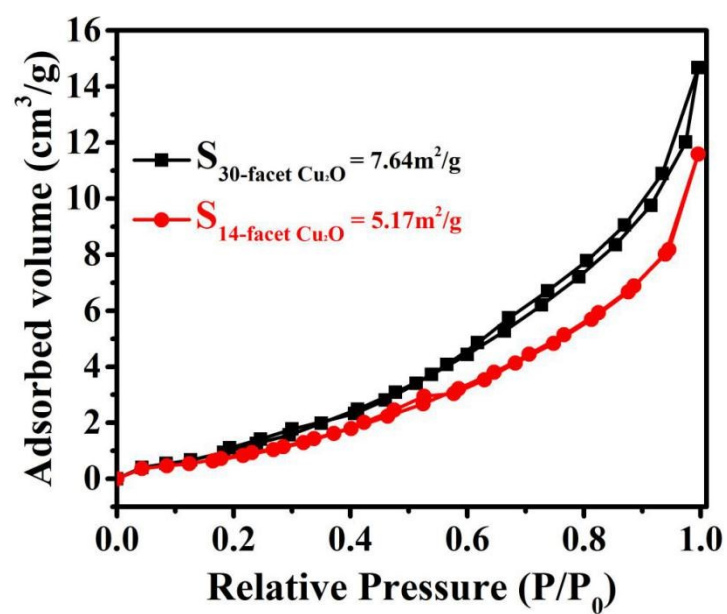


Fig. S1 N₂ adsorption/desorption isotherms for the 30-facet and 14-facet Cu₂O, respectively.

References

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