## **Supporting Information**

## Carbon-Doping Induced Energy Band Modification and Vacancy in SnS<sub>2</sub> Nanosheets for Room-temperature ppb-Level NO<sub>2</sub> Detection

Ruozhen Wu<sup>a</sup>, Juanyuan Hao<sup>\*a,b</sup>, Tingting Wang<sup>a</sup>, Shengliang Zheng<sup>a</sup> and You Wang \*<sub>a,b</sub>

<sup>a</sup>School of Materials Science and Engineering, Harbin Institute of Technology, Harbin 150001, P. R. China.

<sup>b</sup>Key Laboratory of Micro-Systems and Micro-Structures Manufacturing, Ministry of Education, Harbin 150001, P. R. China

\*E-mail: jyhao@hit.edu.cn and y-wang@hit.edu.cn



**Fig. S1** Homemade chemical gas sensing analysis system. The gas test box is placed in a cyclic damp heat test instrument (left) with the controllable temperature and relative humidity.



Fig. S2 TEM image of pristine  $SnS_2$ . The morphology of pristine  $SnS_2$  showed hexagonal plate-like structure with uniform lateral size of ~400 nm.



Fig. S3 TEM image of the  $C-SnS_2-32$  sample. It indicates the obvious mesoporous nanostructures in the  $C-SnS_2-32$  nanosheets.



**Fig. S4** N<sub>2</sub> sorption isotherms at 77 K of (a) C-SnS<sub>2</sub>-32 and (b) pristine SnS<sub>2</sub>. The BET surface area of C-SnS<sub>2</sub>-32 nanosheets was measured to be 26.6 m<sup>2</sup>g<sup>-1</sup>, which was 2.5 times larger than that of pristine SnS<sub>2</sub> (10.7 m<sup>2</sup>g<sup>-1</sup>), which indicates that the adsorption capacity of NO<sub>2</sub> is improved on the C-doped SnS<sub>2</sub> surface.



**Fig. S5** High-resolution O 1s XPS spectra for the C-SnS<sub>2</sub>-32. The peak centred at 531.9 eV is assigned to the surface adsorbed oxygen.<sup>[1,2]</sup>



**Fig. S6** High-resolution C 1s XPS spectra for C-SnS<sub>2</sub>-8 and C-SnS<sub>2</sub>-16. The clear signal of C-Sn bond is also found in both of C-SnS<sub>2</sub>-8 and C-SnS<sub>2</sub>-16, confirming the presence

of C doping in SnS<sub>2</sub>.



**Fig. S7** The peak intensity ratios of Sn-C with C-O and C-C for C-doped  $SnS_2$ . The peak intensity ratios of the Sn-C/C-C and the Sn-C/C-O increase from C-SnS<sub>2</sub>-8 to C-SnS<sub>2</sub>-32, indicating the content of Sn-C bond increases with C dopants concentration while the C-C and C-O peaks does not change.



**Fig. S8** FT-IR spectra of pristine  $SnS_2$ , C- $SnS_2$ -8, C- $SnS_2$ -16, and C- $SnS_2$ -32. The peak at ~520 cm<sup>-1</sup> for C-doped  $SnS_2$  is assigned to the Sn-C stretching, which is absent in pristine  $SnS_2$ , supporting that C is doping into  $SnS_2$  crystals.



**Fig. S9** Baseline resistance of pristine  $SnS_2$ , C- $SnS_2$ -8, C- $SnS_2$ -16, and C- $SnS_2$ -32. The baseline resistance sharply declined after C doping, from ~3000 M $\Omega$  of pristine  $SnS_2$  to ~130 M $\Omega$  of C- $SnS_2$ -32, indicating that the C doping dramatically enhanced the electron concentration and the conductivity of  $SnS_2$ .



Fig. S10 Dynamic response-recovery curve of  $C-SnS_2-32$  to diverse  $NO_2$  concentrations from 4 ppm to 10 ppm. The result indicates that the response value of  $C-SnS_2-32$  sensor increases with increasing  $NO_2$  concentration and the signal completely returns to initial state after  $NO_2$  release.



Fig. S11 A temporal trace of experimentally recorded noise of response for .the sample C-SnS<sub>2</sub>-32. The result indicates that the root-mean-square value of response noise is approximate ~0.12%.



**Fig. S12** UV-Vis diffuse reflectance spectroscopy of the  $C-SnS_2-32$  and pristine  $SnS_2$ . The inset of is Tauc plot with indirect bandgap fitting. The characterization of UV-Vis diffuse reflectance spectroscopy indicates the conspicuously narrowed bandgap structure from 2.11 to 1.75 eV after C doping.



**Fig. S13** KPFM measured work function area scans of the  $C-SnS_2-32$  and pristine  $SnS_2$ . The measurement of KPFM indicates the elevation of Fermi energy level from 4.81 to 4.61 eV after C doping.

## References

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