

## Supporting Information

### Decoding the charge carrier dynamics in triple cation based perovskites solar cells

Mohd Taukeer Khan,<sup>1,\*</sup> Naveen Harindu Hemasiri,<sup>2</sup> Samrana Kazim,<sup>2,3</sup> Shahzada Ahmad<sup>2,3\*</sup>

<sup>1</sup>Department of Physics, Faculty of Science, Islamic University of Madinah, Prince Naifbin Abdulaziz, Al Jamiah, Madinah 42351, Kingdom of Saudi Arabia

<sup>2</sup>BCMaterials, Basque Center for Materials, Applications and Nanostructures, UPV/EHU Science Park, 48940, Leioa, Spain

<sup>3</sup>IKERBASQUE, Basque Foundation for Science, 48009, Bilbao, Spain

\*Correspondence: [shahzada.ahmad@bcmaterials.net](mailto:shahzada.ahmad@bcmaterials.net); [khanmtk@iu.edu.sa](mailto:khanmtk@iu.edu.sa)

#### 1. Variation of fitting parameters with applied bias

EIS was analyzed by the equivalent circuit (EC) shown in Figure 2d. In EC  $R_S$  ascribed to the series resistance of contacts for electrical measurements and was found to be in the range of 9 – 10  $\Omega$ . The capacitance  $C_1$  corresponds to geometrical capacitance and bulk dielectric relaxation of the perovskites layer was determined from CPE1 through the relation [1]:

$$C_1 = \left[ \left( \frac{1}{R_S} + \frac{1}{R_1} \right)^{p_1-1} T_1 \right]^{1/p_1}$$

The capacitance  $C_2$  corresponds to low-frequency mechanisms including ionic transport, interfacial charge accumulation, and transport and was evaluated from CPE2 through the relation:

$$C_2 = \left[ \left( \frac{1}{R_S + R_1} + \frac{1}{R_2} \right)^{p_2-1} T_2 \right]^{1/p_2}$$

The time constant  $\tau_1$  and  $\tau_2$  were calculated from the relation:  $\tau = RC$  by using corresponding values of resistance and capacitance.

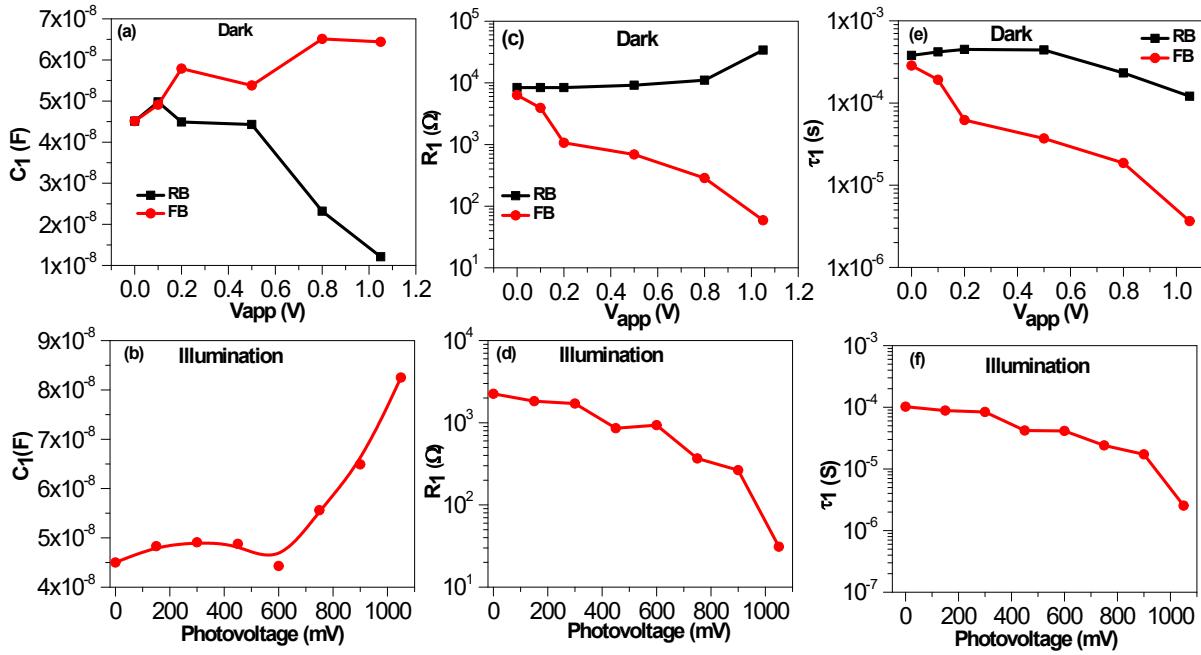


Figure S1. Variation of geometrical capacitance  $C_1$  (a, b), charge transport resistance  $R_1$  (c, d), and time constant  $\tau_1$  (e, f) as a function of applied bias and photovoltage.

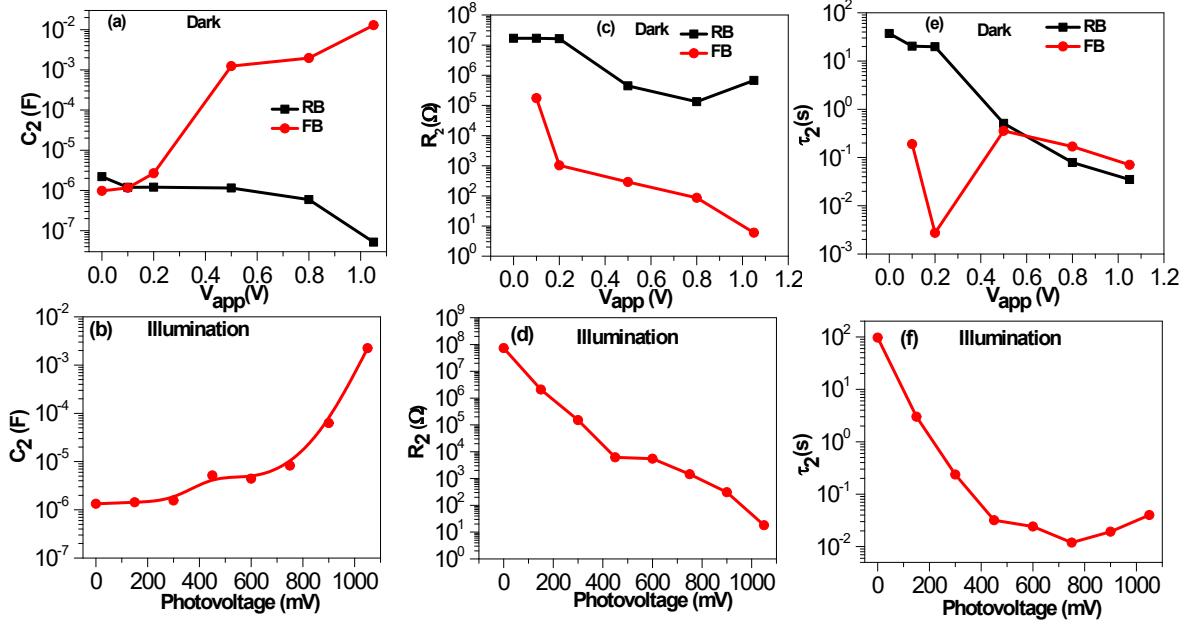


Figure S2. Variation of interfacial capacitance  $C_2$  (a, b), recombination resistance  $R_2$  (c, d), and time constant  $\tau_2$  (e, f) as a function of applied bias and photovoltage.

## 2. Dielectric constant measurement

Room temperature  $C-f$  spectra of perovskites layer sandwiched between FTO and Au electrodes are shown in Fig. S3. The relative dielectric constant ( $\epsilon_r$ ) of the perovskites layer was evaluated to be 44 through the relation:  $C = \epsilon_0 \epsilon_r A/d$ .

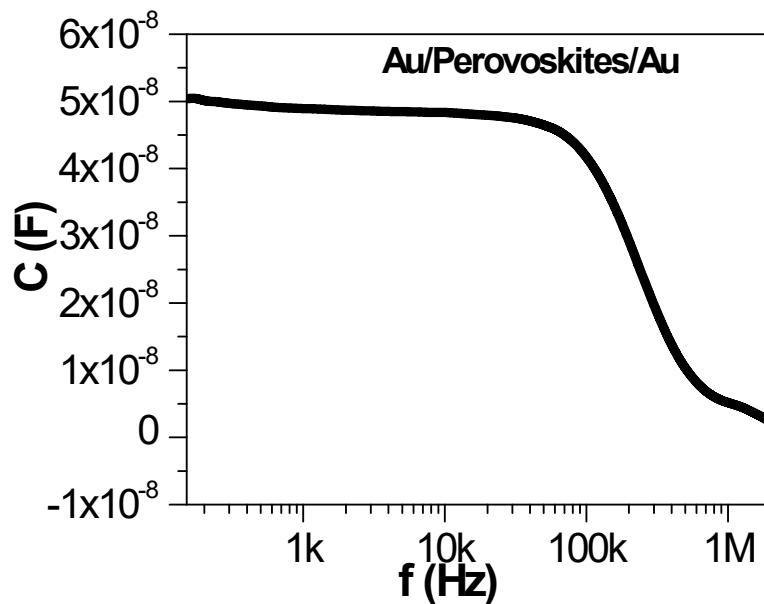


Figure S3.  $C-f$  spectra of FTO/perovskites/Au at room temperature.

### 3. Evaluation of accumulated ionic concentration at the interface

The variation of accumulation capacitance at the perovskites/TiO<sub>2</sub> interface as a function of photovoltage (illumination) is illustrated in Fig. S4.

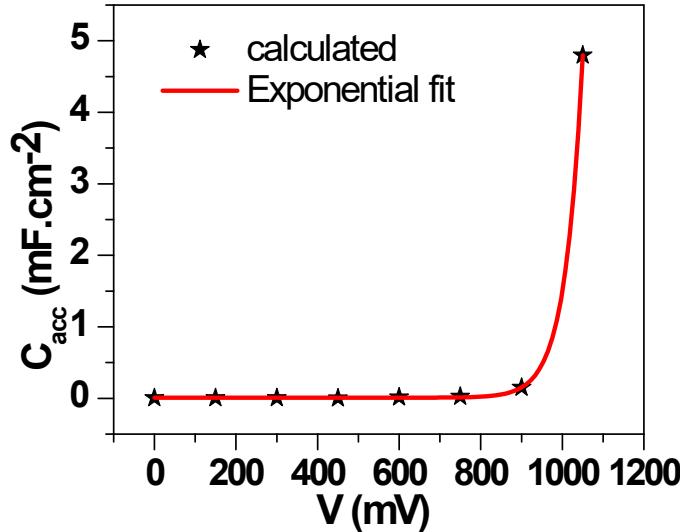
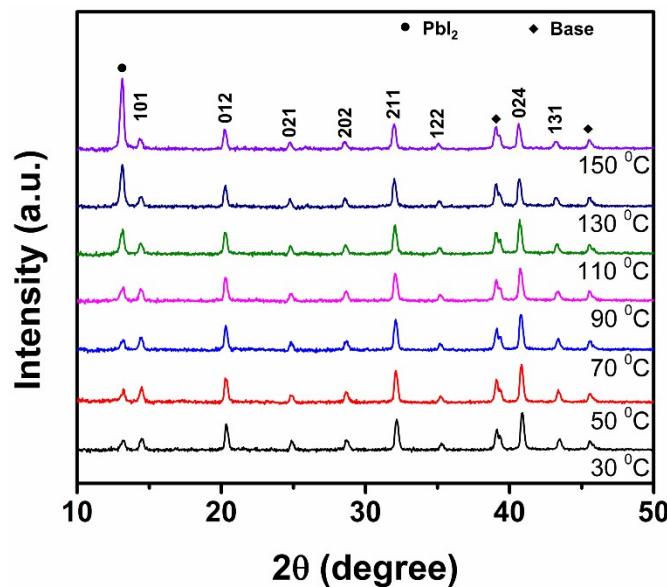


Fig. S4 Exponential fit (solid red line) of the evolution of accumulation capacitance as a function of photovoltage.

#### 4. Temperature-dependent XRD perovskite film

Temperature-dependent X-ray diffraction (XRD) pattern of triple cation  $\text{Cs}_{0.1}(\text{FAPbI}_3)_{0.81}(\text{MAPbBr}_3)_{0.09}$  perovskite films was measured in the temperature range 30 °C – 150 °C to check phase transition. With an increase of temperature above 90 °C, the intensity of peak corresponds to  $\text{PbI}_2$  i.e. (101) increases indicating degradation of perovskites layer with temperature. No change in structure can be observed in the investigated  $J-V$  and immittance spectroscopy temperature range, confirming the change in electrical properties with temperature is not attributed to phase transitions in the perovskites layer.

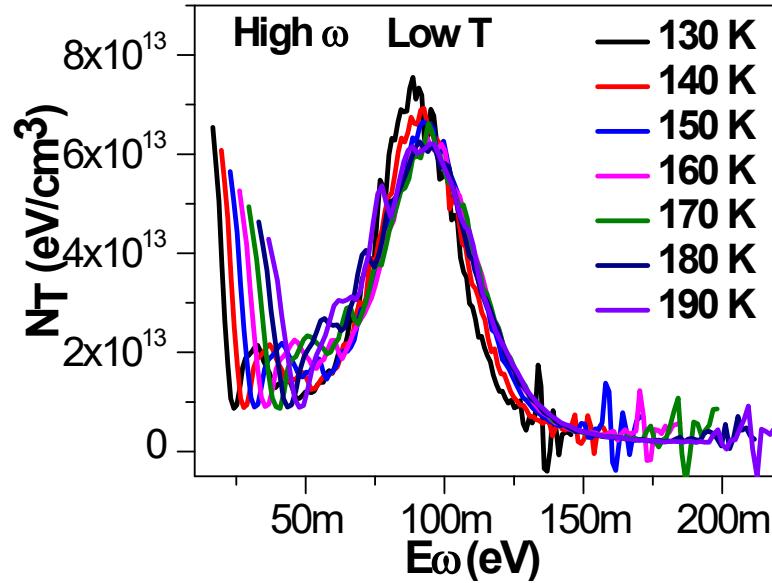


**Figure S5.** Temperature-dependent XRD of  $\text{Cs}_{0.1}(\text{FAPbI}_3)_{0.81}(\text{MAPbBr}_3)_{0.09}$  perovskite.

## 5. Trap density at low temperature

Figure S6 shows the trap density profile as a function of demarcation energy ( $E\omega$ ) evaluated through the relation:

$$N_T = \left( -f \frac{dC}{df} \right) \frac{V_{bi}}{qWk_B T}, \quad E_\omega = k_B T \ln \left( \frac{\beta T^2}{\omega} \right)$$



**Figure S6.** The temperature-dependent density of traps at low frequencies.

**Table S1.** Comparison of transport parameters extracted from thermal admittance spectroscopy.

Device	V <sub>bi</sub> (V)	N <sub>D</sub> (cm <sup>-3</sup> )	Trap DOS (eV <sup>-1</sup> cm <sup>-3</sup> )	E <sub>A</sub> (meV)	ε <sub>r</sub>	Ref.
FTO/c-TiO <sub>2</sub> /mp-TiO <sub>2</sub> /Cs <sub>0.1</sub> (FAPbI <sub>3</sub> ) <sub>0.81</sub> (MAPbBr <sub>3</sub> ) <sub>0.09</sub> / Spiro-OMeTAD/Au	0.88	2.35×10 <sup>21</sup>	7.60×10 <sup>13</sup> 2.45×10 <sup>15</sup>	91 HF LT 254 LF HT	44	Present work
FTO/c-TiO <sub>2</sub> /mp-TiO <sub>2</sub> / (FAPbI <sub>3</sub> ) <sub>0.85</sub> (MAPbBr <sub>3</sub> ) <sub>0.15</sub> / Spiro-OMeTAD/Au	1.05		8.84×10 <sup>16</sup>	124	21	3
FTO/c-TiO <sub>2</sub> /mp-TiO <sub>2</sub> / MAPbI <sub>3</sub> /Spiro-MeTAD/Au	0.69		1.37×10 <sup>17</sup>	83	32	
FTO/c-TiO <sub>2</sub> /mp-TiO <sub>2</sub> / MAPbI <sub>3</sub> /Spiro-MeTAD/Au			2.76 × 10 <sup>18</sup> 5.57 × 10 <sup>17</sup>	17 21	30	4
FTO/c-TiO <sub>2</sub> /mp-TiO <sub>2</sub> / MAPbI <sub>3</sub> /Spiro-MeTAD /Au	1.0	1.4×10 <sup>18</sup>		250	32.5	5
FTO/c-TiO <sub>2</sub> /MAPbI <sub>3</sub> / Spiro- MeTAD/Au		2.4×10 <sup>17</sup>		450		
ITO/PEDOT/MAPbI <sub>3</sub> / PC <sub>61</sub> BM/C <sub>60</sub> /BCP/Al			1.8×10 <sup>17</sup>			6
FTO/PEDOT/MAPbI <sub>3</sub> /PCBM/BCP		6.93×10 <sup>14</sup>				7
ITO/PEDOT/MAPbI <sub>x</sub> Cl <sub>3-x</sub> / PC <sub>61</sub> BM/BCP/Ag		-----	-----	390	22	8
ITO/PTAA/MAPbI <sub>x</sub> Cl <sub>3-x</sub> /PC <sub>61</sub> BM/BCP/Ag				180		
FTO/TiO <sub>2</sub> /MAPbI <sub>x</sub> Cl <sub>3-x</sub> /P3HT/Au			3.8×10 <sup>16</sup>	660 240	18	9
FTO/TiO <sub>2</sub> /MAPbI <sub>3-x</sub> Cl <sub>x</sub> /spiro/Au	1.19	1.8×10 <sup>17</sup>				10
ITO/PTAA/ FAPbI <sub>3</sub> / PC <sub>61</sub> BM/BCP/Au	1.3	2.5×10 <sup>16</sup>			15	11
ITO/PTAA/ MAPbI <sub>3</sub> / PC <sub>61</sub> BM/BCP/Au	1.0	2.8×10 <sup>16</sup>			24	
FTO/SnO <sub>2</sub> /C <sub>60</sub> -SAM/ MA <sub>0.7</sub> FA <sub>0.3</sub> PbI <sub>3</sub> /spiro-OMeTAD/Au				172	32.5	12
ITO/PEDOT:PSS/MA0 <sub>.7</sub> FA <sub>0.3</sub> PbI <sub>3</sub> /C <sub>60</sub> /BCP/Ag				19 363	26	
ITO/PTAA/MA0 <sub>.7</sub> FA <sub>0.3</sub> PbI <sub>3</sub> /C <sub>60</sub> /BCP /Ag				371	37	
ITO/PEDOT:PSS/MAPbI <sub>3</sub> /PCBM /Ca	1.0	1.2×10 <sup>16</sup>				13
ITO/PEDOT:PSS/MAPbI <sub>3</sub> /PCBM /Cr <sub>2</sub> O <sub>3</sub> /Cr	0.79	1.3×10 <sup>16</sup>				

## Supplementary References

1. Daming Zheng, Tao Zhu, and Thierry Pauporté, Using Monovalent- to Trivalent-Cation Hybrid Perovskites for Producing High-Efficiency Solar Cells: Electrical Response, Impedance, and Stability. *ACS Appl. Energy Mater.* 2020, **3**, 11, 10349–10361.
2. S. Ravishankar, S. Gharibzadeh, C. Roldán-Carmona, G. Grancini, Y. Lee, M. Ralaiarisoa, A. M. Asiri, N. Koch, J. Bisquert and M. K. Nazeeruddin, Joule, 2018, **2**, 788–798.
3. M. T. Khan, M Salado, A. Almohammed, S Kazim, S Ahmad, *Adv. Mater. Interfaces* 2019, **6**, 1901193.
4. D. Prochowicz, P. Yadav, M. Saliba, M. Saski, S. M. Zakeeruddin, J. Lewiński, and M. Grätzel, *ACS Appl. Mater. Interfaces*, 2017, **9**, 28418–28425.
5. O. Almora, I. Zarazua, E. Mas-Marza, I. M.-Sero, J. Bisquert, and G. G.-Belmonte, *J. Phys. Chem. Lett.* 2015, **6**, 1645–1652.
6. Shao, Y., Xiao, Z., Bi, C. et al. Origin and elimination of photocurrent hysteresis by fullerene passivation in  $\text{CH}_3\text{NH}_3\text{PbI}_3$  planar heterojunction solar cells. *Nat Commun.*, 2014, **5**, 5784.
7. Dharmadasa, I.M., et al., Perovskite solar cells: a deep analysis using current–voltage and capacitance–voltage techniques. *Journal of Materials Science: Materials in Electronics*, 2018, **30**, 1227-1235.
8. M. T. Khan, P. Huang, A. Almohammed, S. Kazim, S. Ahmad, Mechanistic, *iScience*, 2021, **24**, 102024.
9. M. Samiee, S. Konduri, B. Ganapathy, R. Kottokkaran, H. A. Abbas, A. Kitahara, P. Joshi, L. Zhang, M. Noack, and V. Dalal, *Appl. Phys. Lett.* 2014, **105**, 153502.
10. Guerrero, A., et al., Electrical field profile and doping in planar lead halide perovskite solar cells. *Applied Physics Letters*, 2014, **13**, 105.
11. M. Fischer, K. Tvingstedt, A. Baumann, and V. Dyakonov, *ACS Appl. Energy Mater.* 2018, **1**, 5129–5134.
12. A. Awani, Z. Song, C. Chen, C. Li, C. Wang, M. A. Razooqi, L. Chen, X. Wang, R. J. Ellingson, J. V. Li, and Y. Yan, Joule, 2020, **4**, 644-657.
13. Guerrero, A., et al., *Interfacial Degradation of Planar Lead Halide Perovskite Solar Cells*. *ACS Nano*, 2016, **10**, 218.