

Supplementary information

Maintaining pronounced proton transportation of solid oxides prepared with a sintering additive

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Table S1. Calcination and sintering regimes for BaCeO₃- and BaZrO₃-based ceramics prepared by solid-state synthesis technique: T_{calc} – calcination temperature, T_{sint} – sintering temperature, τ – holding time, ρ – relative density.

Materials	T _{calc} , °C/ τ, h	T _{sint} , °C/ τ, h	ρ, %	Refs
Doped BaCeO₃				
Ba(Ce _{0.95} Ti _{0.05}) _{1-y} Y _y O _{3-δ}	1200/24	1500/24	>90	19
BaCe _{0.9} Y _{0.1} O _{3-δ}	1300/10	1550/10	92	20
BaCe _{0.8} Eu _{0.2} O _{3-δ}	900/2	1600/10	97	21
BaCe _{0.8} Ho _{0.2} O _{3-δ}	1250/10	1650/10	92	22
BaCe _{1-x} Gd _x O _{3-δ}	1250/10	1650/10	96	23
BaCe _{0.75} Y _{0.25} O _{3-δ}	1500/10	1675/10	>90	24
Doped BaZrO₃				
BaZr _{0.8} Y _{0.2} O _{3-δ}	1150/5	1450/5	57	25
BaZr _{0.9} Y _{0.1} O _{3-δ}	-	1600/3	80	26
BaZr _{0.85} Y _{0.15} O _{3-δ}	1300/2	1670/24	84	27
BaZr _{0.8} Y _{0.2} O _{3-δ}	-	1700/-	92	28
BaZr _{0.9} Y _{0.1} O _{3-δ}	1250/10	1715/30	96	29
BaZr _{0.9} Y _{0.1} O _{3-δ}	1200/-	1850/1	>99	26
BaZr _{0.9} Y _{0.1} O _{3-δ}	1400/10	2200/-	98	30

Table S2. Crystallographic data used for EBSD data treatment.

Oxide	a, Å	b, Å	c, Å	α, °	β, °	γ, °	Sp. gr.
BaSnO ₃	4.13	4.13	4.13	90.00	90.00	90.00	221
CuO	4.27	4.27	4.27	90.00	90.00	90.00	224

Table S3. EBSD data treatment results.

Oxide	Phase composition, %	MAD ^a	Standard deviation of MAD	Minimal value of MAD	Maximal value of MAD
BaSnO ₃	96.98	0.87	0.16	0.39	1.80
CuO	2.07	1.36	0.29	0.50	1.96
Null solutions ^b	0.96	-	-	-	-

^a MAD (Mean Angle Deviation) is the deviation from theoretical Kikuchi band position.

^b Null solutions include null solutions in pores of the ceramic sample.

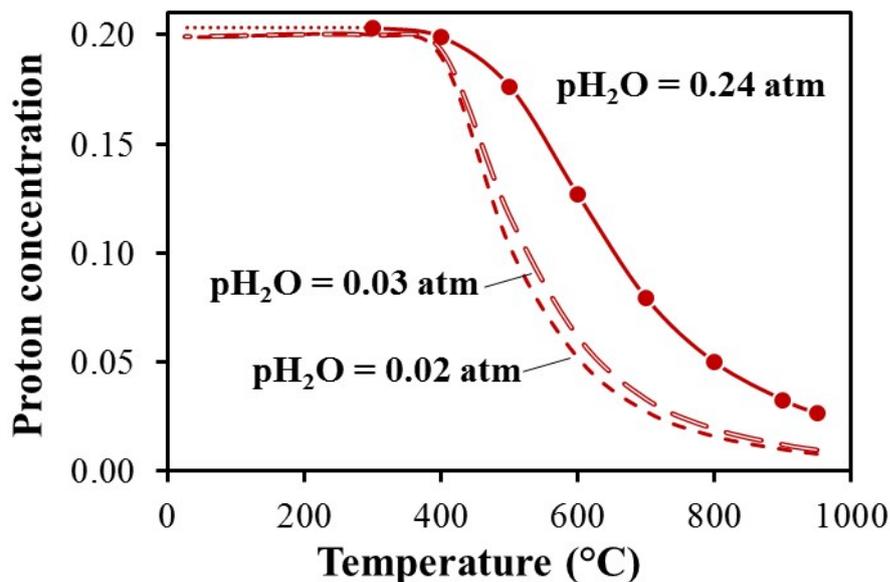


Figure S1.* Comparative analysis of proton concentrations for $\text{BaSn}_{0.8}\text{Sc}_{0.2}\text{O}_{3-\delta} + 0.5 \text{ wt.}\% \text{ CuO}$ in a wet Ar atmosphere: $p_{\text{H}_2\text{O}} = 0.24 \text{ atm}$ (experimental data); $p_{\text{H}_2\text{O}} = 0.02$ and 0.03 atm (recalculated data).

* To compare our results with the literature data, we recalculated the proton concentrations at lower $p_{\text{H}_2\text{O}}$, taking thermodynamic parameters (standard enthalpy and standard entropy) and defects relations (mass action law, electroneutrality conditions) into account. These recalculated (continuous) curves are presented in Fig. S1 based on the starting points of experimental data at $p_{\text{H}_2\text{O}} = 0.24$ (red circle symbols) and determined thermodynamic parameters. As can be seen, the lower $p_{\text{H}_2\text{O}}$ values do not change the character of hydration curves, leading to their displacement towards lower temperatures only. The latter means that a saturation level (the same for a given material) is always reached despite of $p_{\text{H}_2\text{O}}$ used.

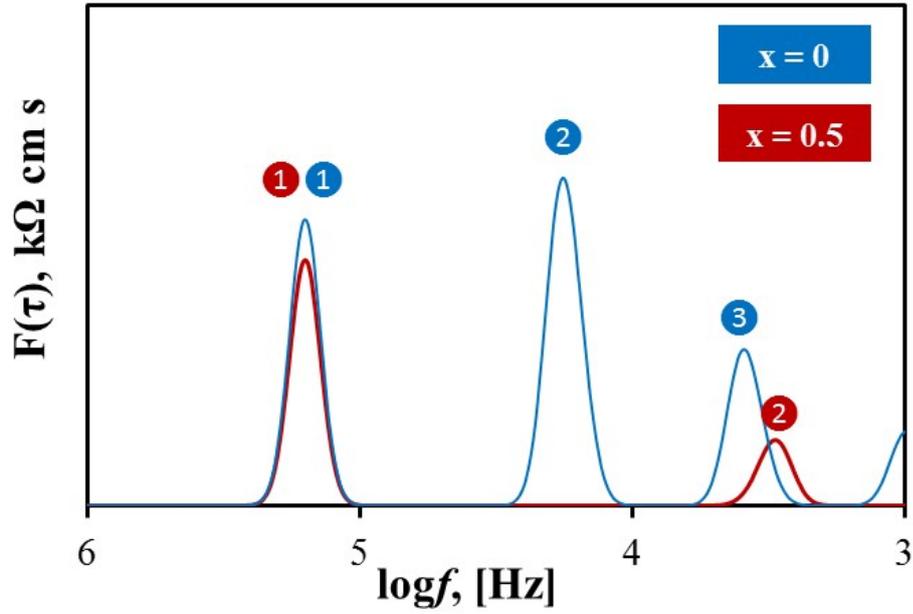


Figure S2. Comparative analysis of the DRT spectra obtained for the $\text{BaSn}_{0.8}\text{Sc}_{0.2}\text{O}_{3-\delta} + x$ wt.% CuO ceramic samples at 350 °C: **1** is the grain process, **2** is the grain boundary process, **3** is the part of the electrode process.

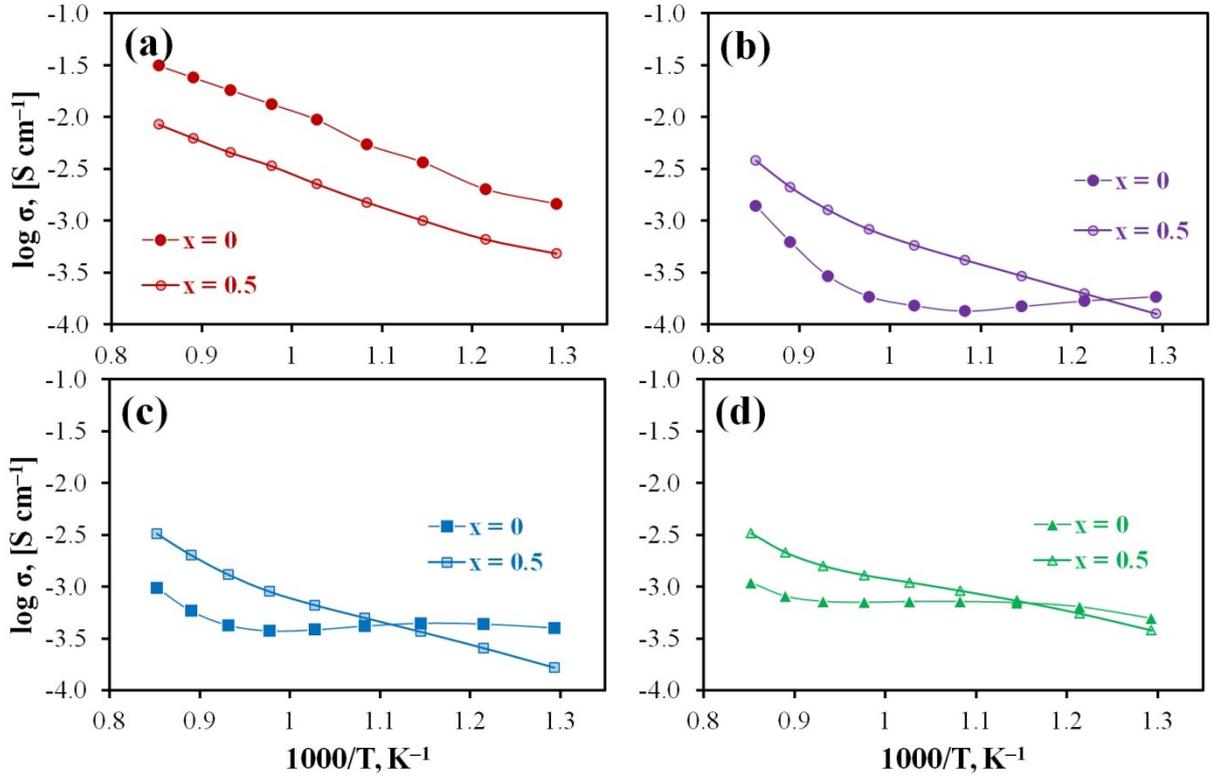


Figure S3. Comparative analysis of transport properties for $\text{BaSn}_{0.8}\text{Sc}_{0.2}\text{O}_{3-\delta} + x$ wt.% CuO: 3 vol.% $\text{H}_2\text{O}/\text{air}$ (a), 3 vol.% $\text{H}_2\text{O}/\text{H}_2$ (b), 10 vol.% $\text{H}_2\text{O}/\text{H}_2$ (c) and 30 vol.% $\text{H}_2\text{O}/\text{H}_2$ (d).