Supplementary Information

Ultrathin oxysulfide semiconductors from liquid metal: a wet chemical approach

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Figure S1. XPS spectra of S2*p* taken from samples treated at different temperatures in 5 hours. No signal was detected in the S2*p* regions of In₂O₃ as-synthesized, blank sample, and the sample treated at room temperature. The S2*p* peaks of S²⁻ are located at ~162 eV.¹



Figure S2. XPS spectra of S2*p* taken from samples treated at 150°C after different reaction times. The S2*p* peaks of S²⁻ are located at ~162 eV, while the broaden peak at 169.1 eV detected in the sample after 8 hours is associated with sulfite/sulfate compounds (SO₄²⁻).^{1, 2}



Figure S3. A representative 2D $In_2O_{3-x}S_x$ sheet printed on a SiO₂/Si substrate, approaching centimetre scale. It can be observed that the 2D sheet covers a homogeneous area of several millimetres with some inevitable folded regions at the edge and where the molten droplet was placed.



Figure S4. Statistical distribution of 2D $In_2O_{3-x}S_x$ sheets thickness. The mean and standard deviation (SD) were found to be 2.41 nm and 0.34 nm, respectively. The difference between measurements is due to AFM tip-surface interactions, noises from environment, and substrate effects.³



Figure S5. SAED pattern and HRTEM image of 2D In_2O_3 directly exfoliated from molten indium metal. The diffraction spot in red circle and a lattice spacing of 0.32 nm can be indexed to (130) plane of *c*-In₂O₃.⁴ The crystal structure of cubic 2D In₂O₃ derived from liquid metal has been reported in the previous study.⁵



Figure S6. Crystal structure of *c*-In₂O₃ with the unit cell thickness of 10.12 Å.⁴



Figure S7. (a) XPS survey spectrum of 2D $In_2O_{3-x}S_x$ nanosheets. Si2*p* and Si2*s* peaks arising from the silicon substrate have been detected at 103.2 eV and 155.5 eV respectively. (b) High resolution XPS spectrum of Na 1*s* for 2D $In_2O_{3-x}S_x$ with no peaks being detected, indicating the absence of surface absorbed Na₂S.



Figure S8. XPS spectra of 2D In₂O₃ nanosheets. (a) XPS scanning for In 3*d*: the doublet signal at 444.3 eV and 451.77 eV is associated with In $3d_{5/2}$ and In $3d_{3/2}$, respectively, no signals of indium metal have been recorded. (b) XPS scanning for O 1*s*: the dominant peak at 532.3 eV is related to Si-O-Si bonds in the substrate. The peak at 529.7 eV is attributed to O²⁻ ions in indium oxide lattice. The peak at 531.2 eV may be assigned to C=O bonds originating from surface contamination as well as oxygen atoms near defect sites. ⁶⁻¹¹ (c) No peaks were detected in the S 2*p* region.



Figure S9. SEM-EDX spectrum of a 2D $In_2O_{3-x}S_x$ nanosheet prepared on 300 nm SiO₂/Si wafer with corresponding maps for the elements Si, In, O, and S. The analysis was conducted at a folded edge region that was thicker and hence provided better signal. The map sum spectrum reveals a In:S ratio of 3:1, with the signal being close to the detection limit. The excess oxygen is due to surface SiO₂ from the Si substrate. Because of the limitations of SEM-EDX, XPS elemental analysis is better suited and should be referred to when determining the composition of the synthesised nanosheets.



Figure S10. UV-vis spectra of 2D In_2O_3 and 2D $In_2O_{3-x}S_x$ (x=0.41) indicates the shift in light absorption from 340.1 nm to 376.7 nm.



Figure S11. UV-vis Tauc plot of 2D In₂O₃ sample reveals a direct bandgap of 3.68 eV.



Figure S12. The transfer (I_{ds} - V_{gs}) curve with 1V source drain bias (V_{ds}) shown in a semi-logarithmic scale. The dashed black lines indicate the change in the position of the charge neutrality point. Red arrows indicate the back-gate sweep direction. The hysteresis can be assigned to the trapping and de-trapping of charge carriers by adsorbed water molecules or by trap states on the dielectric substrate.^{12, 13}



Figure S13. The photocurrent response of 2D $In_2O_{3-x}S_x$ based device as a function of time under (a) different applied powers of the incident light and (b) different wavelengths. The V_{ds} was set at 0.5 V.



Figure S14. Responsivity and detectivity values of 2D $In_2O_{3-x}S_x$ based device at different power intensities, with the wavelength of 285 nm and 0.5 V bias voltage.



Figure S15. Transfer characteristic I_{ds} - V_{gs} of 2D In_2O_3 based device with V_{ds} being varied from 0 to 1 V.



Figure S16. The photocurrent response of 2D In_2O_3 based photodetector under 285 nm and 365 nm UV exposure at P = 2 mW cm⁻² and V_{ds} = 0.5 V. The device is more sensitive to shorter wavelengths and features a persistent photocurrent for several hours.



Figure S17. Elemental composition derived from XPS results of (a) 2D bismuth oxysulfide and (b) 2D tin oxysulfide.



Figure S18. Optical image of an as-grown 2D In_2O_3 nanosheet on silicon substrate with a folded region being observed at the edge of the 2D sheet.

Table S1. Screening the reaction parameters for the transformation of $2D In_2O_3$ into $2D In_2O_{3-x}S_x$ (calculated from XPS data).

Sample	Temp.	Time	S content	X
	(°C)	(h)	(At%)	
In ₂ O ₃ as-synthesized *	-	-	0	0
In ₂ O ₃ blank sample **	150	5	0	0
$In_2O_{3-x}S_x$	RT	5	<1	-
$In_2O_{3-x}S_x$	80	5	4.2	0.21
$In_2O_{3-x}S_x$	120	5	4.6	0.23
$In_2O_{3-x}S_x^{***}$	150	5	8.3±0.5	0.41 ± 0.02
In ₂ O _{3-x} S _x	200	5	7.8	0.39
In ₂ O _{3-x} S _x	150	1	<1	-
$In_2O_{3-x}S_x$	150	2	1.8	0.09
$In_2O_{3-x}S_x$	150	8	6.4	0.32

* 2D In₂O₃ was analysed after squeeze printing process
** 2D In₂O₃ was treated at optimized conditions, but without sulfur sources
*** Experiments have been repeated 4 times for standard error calculation *Note:* x is calculated from the relation x=2/(In:S ratio) with In:S ratio being obtained
from XPS data of 2D In₂O_{3-x}S_x. Sulfur content is calculated from the relation
%S=100x/5

Thickness Method Ref. Materials Mobility Ion/Ioff $(cm^2V^{-1}s^{-1})$ (nm)44 * vdW Exfoliation and 0.6×10^{2} $In_2O_{3-x}S_x$ 2.4 This work wet chemical process 20.4±6.3 ** 5.5±1.5 ** In_2O_3 2.0 vdW Exfoliation 4×10^{2} This work vdW Exfoliation 10^{5} In_2O_3 1.5 4 15 3.98 10^{8} In₂O₃ 12-20 Inkjet-printing 16 SnO vdW Exfoliation 0.7 20 0.4 17 10^{6} 50 Solution process and 0.23 SnO₂ annealing 18 5 10^{5} ZnO 2.8 ALD 19 IGZO 10 Solution process and 9.1 10^{6} annealing 1 vdW Exfoliation and 10^{4} In_2S_3 3.7 56 CVD 20 10^{4} InSe 12 Mechanical 0.1 exfoliation 21 10^{6} InSe 41 Solvent exfoliation 19 22 2 vdW Exfoliation and 3.5 10^{2} α -Ga₂S₃ CVD 23 9 CVD 1.1 10⁵ γ -Ga₂S₃ 24 vdW Exfoliation and 1.5 0.2 150 GaS CVD 25 10^{5} 1.1 0.6 GaSe Mechanical exfoliation 26 CVD 12.24 10^{6} MoS₂ 3 27 10^{3} MoS_2 0.8 CVD 17 28 Mechanical 10^{6} MoSe₂ 3-80 50 exfoliation 29 10^{6} MoTe₂ 1.6 Mechanical 10 exfoliation 30 WS_2 0.7 CVD 0.91 10^{6}

Table S2. Comparison of FETs based on 2D $In_2O_{3-x}S_x$ nanosheets with some reported 2D oxides and chalcogenides over the past few years.

*	Highest value
**	Average value with standard error

Table S3. Comparison of 2D $In_2O_{3-x}S_x$ nanosheets with other recently reported 2D nanosheets and commercial materials for photodetector performance. NM: not mentioned.

Materials	Wavelength	Thickness	R	D^*	EQE	Response time	Ref.
	(nm)	(nm)	$(A W^{-1})$	(Jones)	(%)	(ms)	
$In_2O_{3-x}S_x$	285	2.4	3.4×10^{3}	2.18×10^{13}	1.47×10^{6}	$t_{rise} = 42 \times 10^3$	This
						$t_{fall} = 87 \times 10^3$	work
SnO-In ₂ O ₃	280	5.5	10 ³	5×10 ⁹	NM	≤1	5
Bi ₂ O ₃	365	0.75	4×10^{2}	1.1×10^{13}	NM	0.07	31
Graphene-	254	2	1.48	2.24×10^{12}	7.27×10^{2}	3.1×10 ⁵	32
β -Ga ₂ O ₃							
γ -Ga ₂ S ₃	350	9	61.3	1.52×10^{10}	2.17×10^{4}	10-15	23
GaS	254	2.3	4.2	10 ¹³	2×10^{3}	<30	33
WS ₂	365	25.2	53.3	1.22×10^{11}	NM	NM	34
MoS ₂	405	1.3	6.3×10 ⁻⁵	4.2×10^{8}	NM	20	35
InSe	254	30	5.6×10 ⁴	2×10 ¹³	NM	5	36
SnSe	370	9	5.5	6×10 ¹⁰	1.83×10^{3}	NM	37
In ₂ Se ₃	300	3.9	3.9×10^2	2.26×10^{12}	16.3×10^4	18	38
Commercial	400-1100	NM	0.5	3×10 ¹²	NM	10-6	39
Si							
Commercial	150-550	NM	0.1	2×10 ¹³	NM	1.1×10 ⁻⁴	39
GaP							

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