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Supporting Information

A hybrid ZIF-8/ZIF-62 glass membrane for gas separation

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1. Experimental procedures

1.1 Materials

All chemicals were commercially available without further purification. $Zn(NO_3)_2 \cdot 6H_2O$ (99.99%, Sigma-Aldrich), imidazole (Im, 99.5%, Sigma-Aldrich), benzimidazole (Bim, 99%, Sigma-Aldrich), 2-methylimidazole (Hmim, 99%, Sigma-Aldrich), N,N-dimethylformamide (DMF, 99.5%, Sinopharm Chemical Reagent Co., Ltd), methanol (99.5%, Sinopharm Chemical Reagent Co., Ltd). Porous α -Al₂O₃ disks (18 mm in diameter, 1.0 mm in thickness, and 70 nm particles on the top layer) were purchased from Fraunhofer Institute IKTS, Germany.

1.2 Synthesis of ZIF-62 powder

A mixture of $Zn(NO_3)_2 \cdot 6H_2O(0.1 \text{ M})$, Im (1.35 M), and Bim (0.15 M) was dissolved in DMF and stirred for 15 min to obtain a homogeneous solution.¹ Then the solution was transferred to Teflonlined stainless-steel autoclaves and heated in an oven at 130 °C for 96 h. The product was collected by centrifugation, washed with clean DMF thoroughly and dried at 100 °C for 24h.

1.3 Synthesis of ZIF-8 powder

ZIF-8 nanoparticles were synthesized according to our previously reported method.² A methanol solution (60 mL) of $Zn(NO_3)_2 \cdot 6H_2O$ (0.879 g) was quickly added to another methanol solution (60 mL) of Hmim (1.776 g), and the mixture solution was stirred at room temperature for 1 h. The product was collected by centrifugation, washed with clean methanol thoroughly and dried at 100 °C for 24h.

1.4 Synthesis of (ZIF-8)_x(ZIF-62)_{1-x} hybrid crystals and glasses

A series of hybrid crystals with different doping amounts of ZIF-8 had been prepared, which were named (ZIF-8)_x(ZIF-62)_{1-x}, where x was the mass percentage of ZIF-8. Specifically, a mixture of ZIF-8 and ZIF-62 at a certain mass ratio was dispersed in deionized water and ball-milled with a 7 mm stainless steel ball at 25 Hz for 5 min. The obtained hybrid sample was obtained by centrifugation and dried at 80 °C.

Subsequently, the hybrid ZIF-8/ZIF-62 crystal was heated in a tube furnace with argon atmosphere protection, at a rate of 10°C/min to 500°C and kept for 10 min, and finally cooled naturally to room temperature.³ The obtained hybrid glasses were named $a_g[(ZIF-8)_x(ZIF-62)_{1-x}]$.

1.5 Synthesis of ZIF-8/ZIF-62 hybrid membrane via post-synthetic ligand exchange approach

Post-synthetic ligand exchange approach was employed for preparing the mixed-phase ZIF-8/ZIF-62 polycrystalline membrane. First, the ZIF-62 polycrystalline membrane was prepared by in situ growth method at room temperature. A mixture of $Zn(NO_3)_2 \cdot 6H_2O$, imidazole and benzimidazole with molar ratio of 1:13.5:1.5 was dissolved in DMF and stirred for 15 min. The precursor solution was then transferred to a glass container with α -Al₂O₃ support in advance, reacted at room temperature for 12h for fabricating ZIF-62 polycrystalline membrane. To prepare ZIF-8/ZIF-62 hybrid membrane, the ZIF-62 membrane was subsequently immersed in the DMF solution containing Hmim (8 M) ligand of ZIF-8 to react at 50 °C for 2 h. Finally, the resultant ZIF-8/ZIF-62 polycrystalline membrane was heated at 500 °C for 30 min in a tube furnace filled with Ar atmosphere.

1.6 Synthesis of ZIF-8/ZIF-62 hybrid membrane via in-situ solvothermal method

A DMF solution containing Zn(NO₃)₂·6H₂O, Im, Bim and Hmim was prepared according to the optimal formula, i.e., Zn:L: Hmim: DMF molar ratio of 1:5:8:208, where L=im+bim and im: bim is 5:1. The precursor solution was transferred to a Teflon-lined stainless-steel autoclave in which the α -Al₂O₃ support was placed horizontally in advance, and then heated at 50 °C for 2 h. The obtained ZIF-8/ZIF-62 polycrystalline membrane was then heated at 500 °C for 30 min in a tube furnace filled with Ar atmosphere to achieve the corresponding hybrid glass membrane.

We conducted controlled experiments by using single ligand of ZIF-8 or ZIF-62. All operations are consistent with the above-mentioned ZIF-8/ZIF-62 membrane preparation, except for changing the solution to Zn:Hmim:DMF molar ratio of 1:8:208 or Zn:L:DMF (L=im+bim, im:bim=5:1) molar ratio of 1:5:208, respectively.

1.7 Characterizations

Powder X-ray diffraction (PXRD) patterns were carried out on a Bruker D8 ADVANCE X-ray diffractometer (accelerating voltage: 40 kV, current: 40 mA) using a Cu Kα radiation source with a 20 range between 5° and 40°. Thermogravimetric analysis (TGA) was performed on NETZSCH STA 2500 at a heating rate of 10 °C/min from 50 to 700 °C under nitrogen atmosphere. Differential scanning calorimeter (DSC) curve was obtained on NETZSCH DSC 214 instrument. Flourier transformed infrared spectroscopy (FTIR) was carried out on a VERTEX 70 (Bruker). A Nova Nano SEM 450 scanning electron microscope (SEM) instrument was used to characterize the sample

morphology at an accelerating voltage of 5 kV. The gas adsorption isotherms were measured on a Micromeritics 3Flex Surface Characterization Analyzer. Prior to gas measurement, all the samples were degassed at 80 °C for 48 h under dynamic vacuum. Proton NMR (¹H NMR) spectroscopy was performed on Magnet System 500/54 Ascend. The sample was dissolved in a mixture of DMSO-D6 (500 μ L) and DCI (36.5 %)/D₂O (100 μ L).

1.8 Gas permeation experiments

The gas separation performances of glass membranes were measured by the Wicke-Kallenbach method. Single gas or equimolar gas mixture was used as feed gas, and helium was used as sweep gas. A gas chromatography (Agilent 7890B) is used to measure the concentration of the gas on the permeate side.

The gas permeance was determined by Eq. (1):

$$P_i = \frac{N_i}{A \cdot \triangle P_i}$$
(Eq. 1)

where P_i (mol m⁻² s⁻¹ Pa⁻¹) is the gas permeance, N_i (mol s⁻¹) is the molar flow rate of component i, Δp_i (Pa) is the transmembrane pressure difference, and A (m²) represents the effective membrane area.

The ideal selectivity was determined by Eq. (2):

$$S_{ij} = \frac{P_i}{P_j}$$
(Eq. 2)

where P_i and P_j represent the permeance of component i and component j, respectively.

1.9 Calculation of the gas adsorption isotherms

All the C_3H_6 and C_3H_8 isotherms were fitted using Dual-site Langmuir (DSL) model (Eq. 3):

$$q = q_1 \frac{b_1 p}{1 + b_1 p} + q_2 \frac{b_2 p}{1 + b_2 p}$$
(Eq. 3)

where q_1 and q_2 are the adsorbed amount per mass of adsorbent (mmol g⁻¹) for each site, p is the pressure of the bulk gas at equilibrium (kPa), b_1 and b_2 are the affinity coefficients of each site (kPa⁻¹).

The IAST selectivity S_{ads} was determined by Eq. (4):

$$S_{ads} = \frac{q_i \times p_j}{q_j \times p_i}$$
(Eq. 4)

where q_i and q_j are the adsorbed amount per mass of adsorbent of component i and j, p_i and p_j are the pressure of the bulk gas at equilibrium of component i and j in the bulk.

2. Supplementary Figures and Tables



Fig. S1 SEM images of (a, b) ZIF-62 and (c, d) ZIF-8.



Fig. S2 PXRD patterns of $(ZIF-8)_x(ZIF-62)_{1-x}$ samples.



Fig. S3 (a) TG and (b) DSC curves of (ZIF-8)_x(ZIF-62)_{1-x} samples.



Fig. S4 FTIR spectra of $(ZIF-8)_x(ZIF-62)_{1-x}$ samples.



Fig. S5 C₃H₈ adsorption isotherms of $a_g[(ZIF-8)_x(ZIF-62)_{1-x}]$ samples at 273K.



Fig. S6 (a) Schematic diagram of the preparation of hybrid ZIF-8/ZIF-62 glass membrane. (b) XRD patterns of membranes obtained from different treatment processes, the ZIF-62 polycrystalline membrane (purple), the ZIF-8/ZIF-62 hybrid membrane via post-synthetic ligand exchange (blue) and a_T (ZIF-8/ZIF-62) hybrid membrane after heat-treatment process (green), which a_T MOF means temperature-induced amorphization.



Fig. S7 SEM images of (a, b) the ZIF-62 polycrystalline membrane, (c, d) the ZIF-8/ZIF-62 hybrid membrane synthesized via post-synthetic ligand exchange approach, and (e, f) the a_T (ZIF-8/ZIF-62) hybrid membrane after heat-treatment process.



Fig. S8 Schematic diagram of the preparation of a_g (ZIF-8/ZIF-62) hybrid membrane, which a_g MOF means amorphization upon melt-quenching.



Fig. S9 XRD pattern of the membranes obtained by using single ligand of ZIF-8 or ZIF-62.



Fig. S10 SEM images of the membrane via in-situ solvothermal process at Zn:Hmim:DMF=1:8:208. Top view and cross-sectional SEM images of (a, b) the membrane, and (c, d) after heat-treatment process.



Fig. S11 SEM images of the membrane via in-situ solvothermal process at Zn:L:DMF (L=im+bim)=1:5:208. Top view and cross-sectional SEM images of (a, b) the resultant membrane and (c, d) after heat-treatment process.



Fig. S12 ¹H NMR spectra of polycrystalline and glassy ZIF-8/ZIF-62 hybrid membranes.



Fig. S13 Long-term stability of the hybrid glass membrane for \mbox{CO}_2/\mbox{N}_2 separation.

Table S1 The gas separation performances of hybrid ZIF-8/ZIF-62 glass membranes for equimolar CO_2/CH_4 and C_3H_6/C_3H_8 mixtures.

Membran e	CO ₂ permeance (×10 ⁻⁹ mol m ⁻² s ⁻¹ Pa ⁻¹)	CO ₂ /CH ₄ separation factor	C ₃ H ₆ permeance (×10 ⁻⁹ mol m ⁻² s ⁻¹ Pa ⁻ ¹)	C ₃ H ₆ /C ₃ H ₈ separatio n factor
M1	6.5	29.9	0.09	18.7
M2	11.7	25.2	0.13	12.9
M3	35.9	19	0.67	10

3. References

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