Electronic Supplementary Information

Ultrahigh mechanical flexibility induced superior piezoelectricity of InSeBr-type 2D Janus materials

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Table S1. The optimized lattice constants of bulk InSeBr-type materials calculated by different vdW correction methods, including D2 method of Grimme (DFT-D2),¹ zero damping D3 method of Grimme (DFT-D3),² D3 method of Grimme with Becke-Jonson damping (DFT-D3+BJ),³ Tkatchenko-Scheffler method (DFT-TS).⁴ And the corresponding experiment data are also listed for comparison as well. DFT-D2 method can effectively reproduce the experiment data.

Material	DFT-D2	DFT-D3	DFT-D3+BJ	DFT-TS	Exp. ^{5, 6}
InSCl	a=3.76 Å,	a=3.82 Å,	a=3.80 Å,	a=3.84 Å,	a=3.72 Å,
	c=17.91 Å	c=17.81 Å	c=17.70 Å	c=17.98 Å	c=17.95 Å
InSBr	a=3.81 Å,	a=3.86 Å,	a=3.85 Å,	a=3.89 Å,	a=3.83 Å,
	c=18.51 Å	c=18.53 Å	c=18.46 Å	c=18.82 Å	c=18.75 Å
InSeCl	a=3.88 Å,	a=3.94 Å,	a=3.92 Å,	a=3.97 Å,	a=3.85 Å,
	c=18.18 Å	c=18.12 Å	c=18.09 Å	c=18.38 Å	c=18.44 Å
InSeBr	a=3.92 Å,	a=3.98 Å,	a=3.96 Å,	a=4.01 Å,	a=3.92 Å,
	c=18.94 Å	c=18.92 Å	c=18.84 Å	c=19.25 Å	c=19.01 Å



Fig. S1. Phonon spectra of (a) InSCl, (b) InSBr, and (c) InSeCl monolayers.



Fig. S2. Evolutions of atomic structures and potential energies of Janus (a) InSCl, (b) InSBr, and (c) InSeCl monolayers during the AIMD simulations at 300K for 10 ps. The red line indicates the average total energy during the simulations. For InSCl and InSBr monolayers, the initial equilibrium nanostructures are well maintained and the total energy has small fluctuation during the AIMD simulation at 300 K, suggesting that their good thermal stability. For InSeCl monolayer, the obvious

broken bonds and reduction of total energy indicate that it is thermally unstable.



Fig. S3. Electronic band structures and partial density of states (PDOS) of Janus (a) InSCl, (b) InSBr, and (c) InSeCl monolayers calculated by HSE06 functional. The red arrow indicates the positions of the VBM and CBM, and the Fermi level is set at zero.



Fig. S4. The calculated stress-strain relations of Janus (a) InSCl and (b) InSBr monolayers under tensile deformation along armchair (*x*-) and zigzag (*y*-) directions.

Material	Piezoelectric strain coefficient					
2D monolayer systems: d_{11} (pm/V)						
InSCl	149~154					
InSBr	51~53					
InSeBr	2361~3224					
SnSe	251					
SbAs	243					
BiInTe ₃	362					
MoS ₂	3.73					
Janus GaInS ₂	8.33					
Janus SbTeI	12.95					
3D perovskite-based systems: d_{33} (pm/V)						
Pb(Zr, Ti)O ₃ (PZT)	360~900					
PbTiO ₃ (PT)	1000~2000					
K _{0.5} Na _{0.5} NbO ₃ (KNN)	355~1030					
Bi _{0.5} Na _{0.5} TiO ₃ -Bi _{0.5} K _{0.5} TiO ₃ (BNKT)	250~1400					
Bi _{0.5} Na _{0.5} TiO ₃ (BNT)	249~1100					
$\boxed{ Ba(Zr_{0.2}Ti_{0.8})O_3-x(Ba_{0.7}Ca_{0.3})TiO_3 (BZT-BCT) }$	400~1310					
(Ba _{0.94} Ca _{0.06})-(Ti _{0.95} Zr _{0.05})O ₃ (BCTZ)	2027					

Table S2. The comparison of piezoelectric strain coefficients between InSeBr-type monolayers and
other reported 2D monolayers and 3D perovskite-based materials.

Table S3. The in-plane elastic stiffness coefficients C_{11} and C_{12} of the 2D Janus InSeBr-type monolayers. The superscripts of *el* and *ion* represent electronic and ionic contributions, respectively. $C_{11} = C_{11}^{el} + C_{11}^{ion}$ and $C_{12} = C_{12}^{el} + C_{12}^{ion}$

Material	<i>C</i> ^{<i>el</i>} ₁₁	<i>C</i> ^{<i>el</i>} ₁₂	C ^{ion} ₁₁	C ^{ion} ₁₂	<i>C</i> ₁₁	<i>C</i> ₁₂
InSCl	62.20	25.57	-27.98	1.98	34.22	27.55
InSBr	60.57	24.32	-21.24	-1.11	39.33	23.21
InSeCl	53.78	21.89	-40.42	17.94	13.36	39.83
InSeBr	52.65	20.78	-25.71	5.75	26.94	26.53

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