

Supplementary information

Temperature-induced Polymorphism of a Benzothiophene Derivative: Reversibility and Impact on the Thin Film Morphology

Shunya Yan,^a Alba Cazorla,^a Adara Babuji,^a Eduardo Solano,^b

Christian Ruzié,^c Yves H. Geert,^{c,d} Carmen Ocal,^{*a} Esther Barrena^{*a}

^a Institut de Ciència de Materials de Barcelona (ICMAB-CSIC), 08193 Bellaterra, Barcelona, Spain

^b ALBA synchrotron, C/ de la Llum 2-26. Cerdanyola del Vallès, Barcelona, 08290, Spain

^c Laboratoire de Chimie des Polymères, Faculté des Sciences, Université Libre de Bruxelles (ULB), Boulevard du Triomphe, CP 206/01, 1050 Bruxelles, Belgium

^d International Solvay Institutes of Physics and Chemistry, Université Libre de Bruxelles (ULB), Boulevard du Triomphe, CP 231, 1050 Bruxelles, Belgium

Email: e.barrena@csic.es

Cell parameters calculation

In a monoclinic system, such as the ones observed for C₈O-BTBT-OC₈ thin-film, *c* parameter can be directly determined from the out-of-plane spacing and the angle between *c* and *a*-axis (β), as described by equation S1:

$$c = d_{001} / \sin\beta \quad (S1)$$

In general, for crystal structures with the (001) plane parallel to the substrate surface, in-plane and out-of-plane components of the reciprocal vector *q* can be written as equations S2 and S3, respectively:

$$q_{xy}^2 = q_x^2 + q_y^2 = h^2 z_a^2 + k^2 z_b^2 - 2hk z_a z_b \cos \gamma \quad (S2)$$

$$q_z = h a^* \cos \beta^* + k b^* \cos \alpha^* + l c^* \quad (S3)$$

where $z_a = 2\pi / (a \sin \gamma)$ and $z_b = 2\pi / (b \sin \gamma)$, and a^* , b^* , c^* , α^* , β^* refer to reciprocal lattice parameters. Thus, from in-plane *q*-values (q_{xy}) the unit cell parameters *a*, *b*, and γ can be determined, while out-of-plane *q*-values (q_z) lead to parameters *c*, α , and β .¹

According to equation S2, the lattice parameter *b* can be extracted from the in-plane position of the (020) peak. If γ is assumed to be 90°, *b* parameter is given by equation S4:

$$b = \frac{4\pi}{q_{xy(020)}} \quad (S4)$$

Once the *b* parameter is calculated, the lattice parameter *a* can be extracted from the in-plane position of the (-1-20) peak, following the equation S5:

$$a = \frac{2\pi}{\sqrt{q_{xy(-1-20)}^2 - 4 \left(\frac{2\pi}{b}\right)^2}} \quad (S5)$$

Additionally, since we observe a third diffraction rod, we can verify the calculated unit cell parameters by comparing the expected position for (-1-10) with the observed position. The expected position for (-1-10) is given by equation S6:

$$q_{xy(-1-10)} = \sqrt{4\pi^2 \left(\frac{1}{a^2 \sin^2 \beta} + \frac{1}{b^2} \right)} \quad (S6)$$

For each lattice parameter we have considered the uncertainty given by the error propagation in the equation, considering the uncertainty of the peak position, i.e., the standard deviation given by the Gaussian fit for each peak.

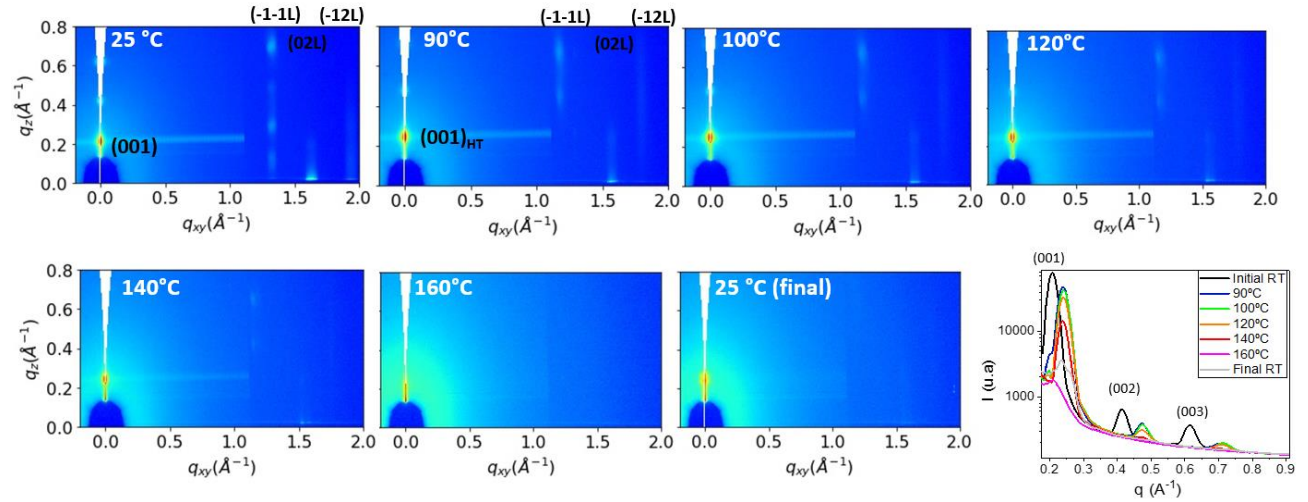
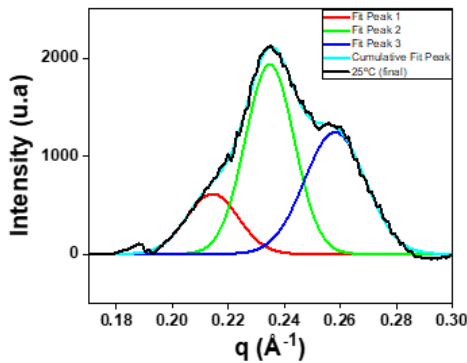


Figure S11. 2D GIWAXS patterns of a C₈O-BTBT-OC₈ thin film (incidence angle of 0.13°) acquired at each temperature. The horizontal line appearing $q_z \approx 0.24 \text{ \AA}^{-1}$ is a blooming effect from the CCD camera employed. Line cuts along the out-of-plane direction are plotted.



Model	Gauss		
Equation	$y=y_0 + (A/(w*\sqrt{\pi/2})) * \exp(-2*((x-xc)/w)^2)$		
y0	3.60497 ± 5.09011	3.60497 ± 5.09011	3.60497 ± 5.09011
xc	0.21431 ± 0.00142	0.23476 ± 3.13516E-4	0.25808 ± 6.0576E-4
w	0.01849 ± 0.00159	0.01749 ± 8.51026E-4	0.02175 ± 7.8747E-4
A	14.19056 ± 2.22215	42.35017 ± 3.45375	33.84385 ± 1.80722
Reduced Chi-Sqr	1659.64057		
R-Square (COD)	0.99658		

Figure S12. Multiple peak Gaussian fit of the (001) peak from C₈O-BTBT-OC₈ films at 25 °C after post-annealing at 160 °C (grey spectrum in Figure 1b after background subtraction). The table corresponds to the results of the fits with Gaussian functions. The three peaks are centered at 0.214 \AA^{-1} , 0.235 \AA^{-1} and 0.258 \AA^{-1} .

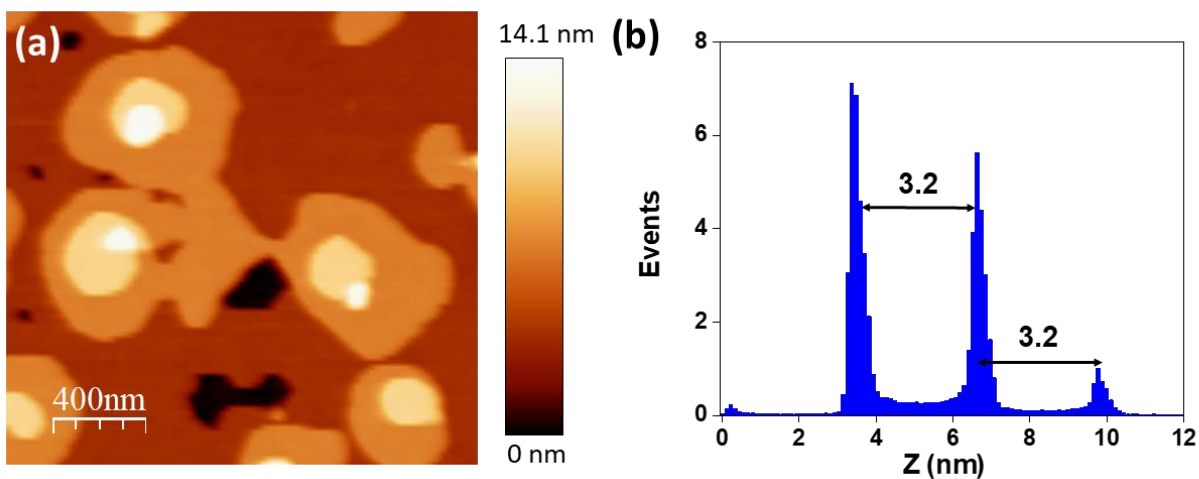


Figure S13. Topographic AFM image of the C_8O -BTBT- OC_8 film (nominal thickness of 14 nm) grown at RT (a). (b) Height distribution of the image in (a).

Figure S13a is the topography of a thin film of C_8O -BTBT- OC_8 with a nominal thickness of 14 nm, grown at 25 °C (room temperature). The film consists of 4 complete layers (each with single layer height) plus one nearly complete layer (large brown layer in the image) and islands with two or three layers. The height estimated from the histogram (Figure S13b) indicates that the molecules within the film are nearly perpendicular with respect to the silicon substrate, as corresponds to the SIP structure.

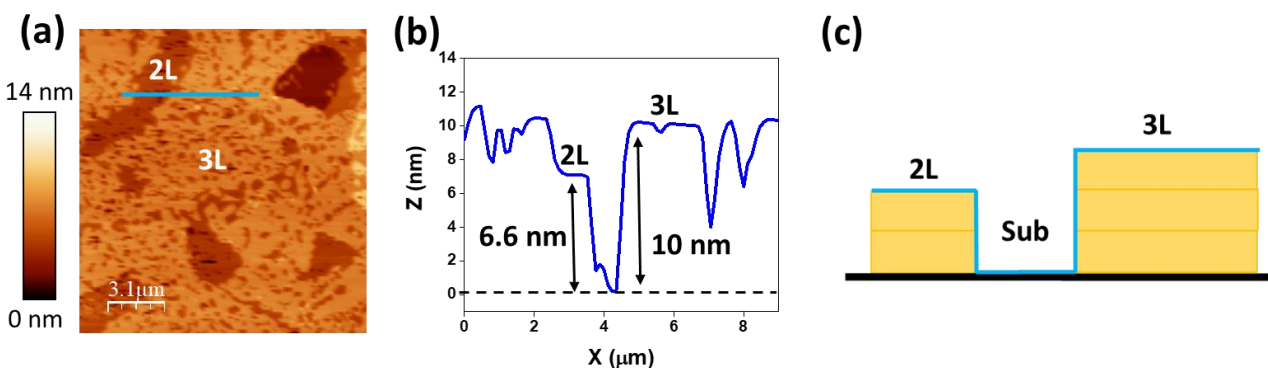


Figure S14. (a) Topographic AFM image of a C_8O -BTBT- OC_8 film (nominal thickness of 14 nm) grown at RT and cooled down to RT after post-annealing at 130 °C, (b) height profile corresponding to the blue segment in (a). The sketch in (c) illustrates schematically the height profile correspondence between height and number of layers indicated in (b).

As it can be seen from the topographic profile (blue line) in Figure S14b, reaching the substrate, the film consists of regions 2 and 3 layers thick.

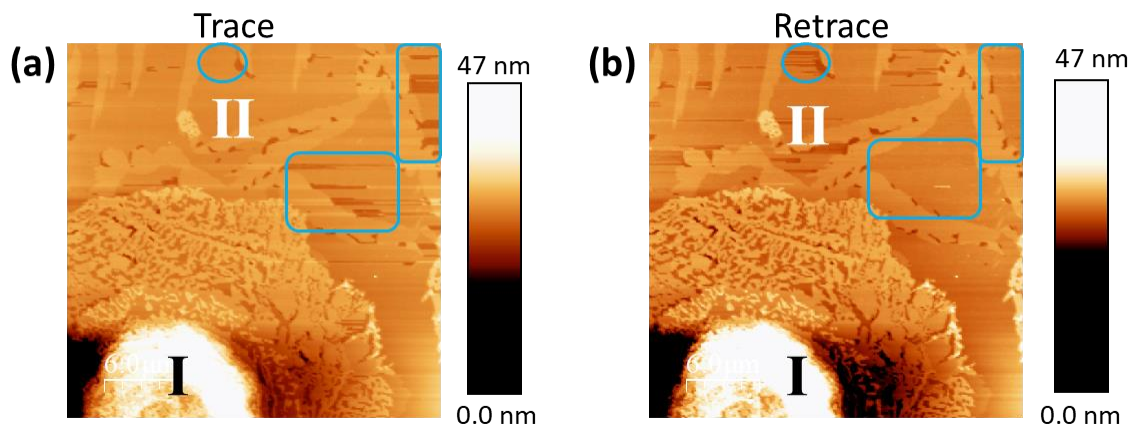


Figure S15. Trace (a) and retrace (b) topographic AFM images for a C_8O -BTBT- OC_8 film (nominal thickness of 14 nm) grown at RT and cooled down to RT after post-annealing at 150 °C. As it can be seen at the areas marked in blue, the lower layer is easily penetrated by the tip during scanning. The lateral size of the image is 30 μm x 30 μm .

In Figure S15 we show the trace and retrace images of the sample post-annealed at 150 °C. In region I (see detailed explanation in the main manuscript), the flat film of 2.5 nm thickness is easily damaged by tip scanning. In addition, some holes appear between the HTP and SIP structures arising from the different molecular density of these phases.

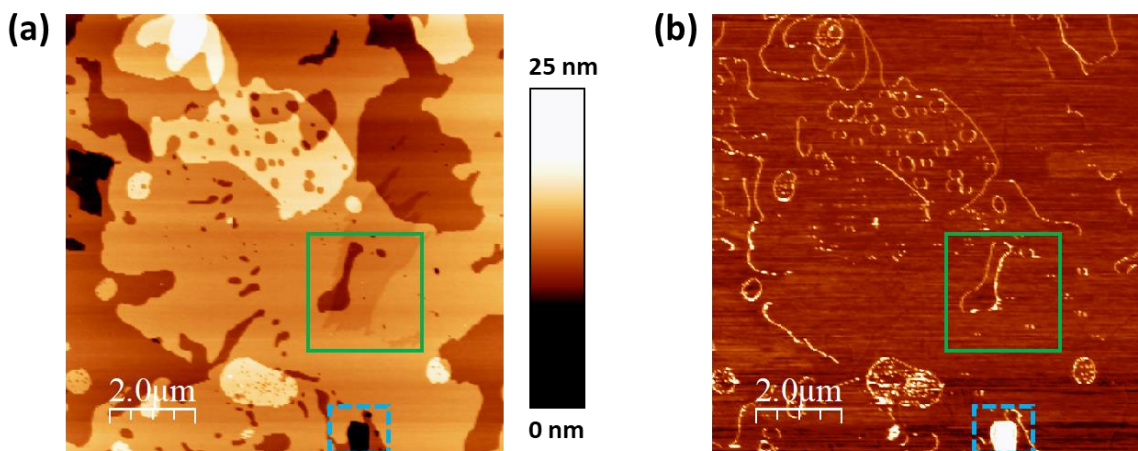


Figure S16. Topographic AFM image (a) and simultaneously obtained lateral force image (b) for a C_8O -BTBT- OC_8 film (nominal thickness of 14 nm) grown at 130 °C. The area marked by green and dashed blue squares correspond to Fig.6c and a hole reaching down the substrate, respectively.

Figure S16 is a larger image of the region imaged in Figure 6c of the main manuscript. It is clearly seen that the low 0.6 nm steps do not show any contrast in the lateral force (LF) image. Some other regions with similar steps can be identified, this fact demonstrates that, upon cooling at RT after the post-annealing treatment, some regions remain in the HTP structure while other regions convert to the most stable SIP at RT.

- [1] J. Simbrunner, C. Simbrunner, B. Schrode, C. Röthel, N. Bedoya-Martinez, I. Salzmann and R. Resel, Indexing of grazing-incidence X-ray diffraction patterns: The case of fibre-textured thin films, *Acta Crystallogr. Sect. A Found. Adv.*, 2018, **74**, 373–387.