#### Electronic supplementary information for

# Surface energy and surface stability of cesium tin halide perovskites:

## A theoretical investigation

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#### **Formation energy**

The formation energies ( $E_f$ , eV) of SnX<sub>2</sub>, CsX and CsSnX<sub>3</sub> (X=Cl, Br, I) with respect to element substances are calculated according to the follow formulas:

$$E_{f(SnX_2)} = E_{SnX_2} - E_{Sn} - 2E_X$$
(1)

$$E_{f(CSX)} = E_{CSX} - E_{CS} - E_X \tag{2}$$

$$E_{f(CsSnX_3)} = E_{CsSnX_3} - E_{Cs} - E_{Sn} - 3E_X$$
(3)

where  $E_{SnX_2}$ ,  $E_{CsX}$ ,  $E_{CsSnX_3}$  are the total energies of bulk SnX<sub>2</sub>, CsX, CsSnX<sub>3</sub> respectively.  $E_{Cs}$ ,  $E_{Sn}$ ,  $E_X$  are the energies (eV per atom) of element substances, i.e., Cs metal, Sn metal and X<sub>2</sub> gas molecule. One exception is I<sub>2</sub> for which a solid structure is used. The  $E_f$  of these bulks define the CsX-rich and SnX<sub>2</sub>-rich conditions at which we calculate the  $E_{surf}$  values for (100)-CsX and (100)-SnX<sub>2</sub> surfaces (Table 2, Table S2 and Table S3). CsSnX<sub>3</sub> bulk is thermodynamically stable at the chemical potential region between CsX-rich and SnX<sub>2</sub>-rich condition:  $-4.30 \le \Delta \mu_{Cs} + \Delta \mu_{Cl} \le -4.00 \ eV$  for CsSnCl<sub>3</sub>;  $-4.04 \le \Delta \mu_{Cs} + \Delta \mu_{Br} \le -3.79 \ eV$  for CsSnBr<sub>3</sub>; and  $-3.21 \le \Delta \mu_{Cs} + \Delta \mu_I \le 3.09 \ eV$  for CsSnI<sub>3</sub>. These conditions define the red lines in Fig. 2 of main text. Binary phase CsX or SnX<sub>2</sub> may appear beyond these regions.

#### Conditions to calculate $E_{surf}$ as a function of $\Delta \mu x$

In order to show  $E_{surf}$  as a function of  $\Delta\mu_X$ , we calculate  $E_{surf}$  at specific chemical potential conditions of Cs. We assume Sn-rich condition (i.e.,  $\Delta\mu_{Sn} = 0$  eV) where Sn source is in equilibrium with CsSnX<sub>3</sub>. For Sn-perovskites, excess Sn may be reached by adding Sn sources such as SnF<sub>2</sub> additive.<sup>1, 2</sup> Nakamura et al. showed that Sn (0) nanoparticles produced from the reaction of a dihydropyrizine derivative with SnF<sub>2</sub> (II) additive can avoid Sn (IV) impurities and improve the efficiency of tin-based solar cells.<sup>2</sup> Previous works also reported the existence of metal Pb nanoparticles in CsPbBr<sub>3</sub> system.<sup>3, 4</sup> With  $\Delta\mu_{Sn} = 0$  eV, we obtain  $\Delta\mu_{Cs} + 3\Delta\mu_{Cl} = -7.46$  eV for CsSnCl<sub>3</sub> ( $\Delta\mu_{Cs} + 3\Delta\mu_{Br} = -6.80 \text{ eV}$  for CsSnBr<sub>3</sub> and  $\Delta\mu_{Cs} + 3\Delta\mu_{I} = -4.67$  eV for CsSnI<sub>3</sub>). Combining Sn-rich condition with CsCl-rich and SnCl<sub>2</sub>-rich conditions of stable CsSnCl<sub>3</sub>, we can obtain two typical conditions for Cs:  $\Delta\mu_{Cs} = -2.72 \text{ eV}$  and  $\Delta\mu_{Cs} = -2.7$ 

-2.26 eV for CsSnCl<sub>3</sub>. Similarly,  $\Delta\mu_{Cs} = -2.66 \text{ eV}$  and  $\Delta\mu_{Cs} = -2.29 \text{ eV}$  for CsSnBr<sub>3</sub> and  $\Delta\mu_{Cs} = -2.48 \text{ eV}$  and  $\Delta\mu_{Cs} = -2.30 \text{ eV}$  for CsSnI<sub>3</sub> can be obtained. The  $E_{\text{surf}}$  as a function of  $\Delta\mu_X$  are calculated at these conditions (Figure 3).

#### Calculating monolayer exfoliation energy

We calculate the monolayer exfoliation energy (i.e., the energy required in peeling off one-unit-cell CsSnX<sub>3</sub> from CsSnX<sub>3</sub> surface) using slab method. <sup>5</sup> The calculation models are displayed in Fig. S2. The monolayer exfoliation energy is calculated as the difference in total energies of configurations between Fig. S2(b) and Fig. S2(a). We check the convergence of monolayer exfoliation energy with the thickness of slab by using a CsSnCl<sub>3</sub> slab with eleven layers. The calculated monolayer exfoliation energy is 0.330 J/m<sup>2</sup>, the same as the result from nine-layers model. This indicates the convergence of the quantity. Note that the slab in Fig. S2 is terminated with CsX-layer. We also calculated monolayer exfoliation energy by using SnX<sub>2</sub>-terminated nine-layers slab. The calculated results are 0.328, 0.307 and 0.280 J/m<sup>2</sup> for CsSnCl<sub>3</sub>, CsSnBr<sub>3</sub> and CsSnI<sub>3</sub>, respectively. These values are very close to those (0.330, 0.307 and 0.281 J/m<sup>2</sup> in order) obtained from CsX-terminated slabs, providing consistent data for exfoliation energy regardless of the termination of slab.

Table. S1 Formation energies (E)	f) of bulk SnX <sub>2</sub> , CsX	and CsSnX <sub>3</sub> with respect to the			
element substances. The structures of tin halides (SnCl <sub>2</sub> , SnBr <sub>2</sub> and SnI <sub>2</sub> ) are taken from					
ref. 6-8 The $E_{\rm f}$ are also calculated with spin-orbital coupling ( $E_{\rm f}$ -SOC), listed in the					
third column.					
Substances	$E_{\rm f}({\rm eV})$	$E_{\rm f}$ -SOC (eV)			

Substances	$E_{\rm f}({ m eV})$	$E_{\rm f}$ -SOC (eV)
SnCl <sub>2</sub>	-3.16	-3.14
$\mathrm{SnBr}_2$	-2.76	-2.73
$SnI_2$	-1.46	-1.40
CsCl	-4.00	-3.99
CsBr	-3.79	-3.79
CsI	-3.09	-3.05
CsSnCl <sub>3</sub>	-7.46	-7.43
CsSnBr <sub>3</sub>	-6.80	-6.77
CsSnI <sub>3</sub>	-4.67	-4.58

Cutoff; mesh		350 eV;	350 eV; 8×8×8		500 eV; 12×12×12	
CsSnX <sub>3</sub>	$\Delta \mu_{\rm Cs} + \Delta \mu_{\rm X}({\rm eV})$	CsX	SnX <sub>2</sub>	CsX	$SnX_2$	
C <sub>2</sub> C <sub>2</sub> C <sub>1</sub>	-4.00	0.069	0.135	0.070	0.134	
CsSnCl <sub>3</sub>	-4.30	0.148	0.055	0.146	0.057	
C <sub>a</sub> S <sub>m</sub> D <sub>m</sub>	-3.79	0.059	0.162	0.058	0.162	
CsSnBr <sub>3</sub>	-4.04	0.119	0.101	0.119	0.102	
C C I	-3.09	0.043	0.156	0.045	0.158	
CsSnI <sub>3</sub>	-3.21	0.068	0.130	0.069	0.135	

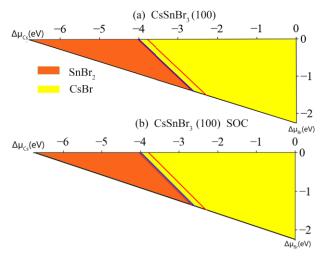
**Table S2** Surface energies ( $E_{surf}$ , J/m<sup>2</sup>) of relaxed (100)-CsX and (100)-SnX<sub>2</sub> surfaces calculated with a cutoff of 500 eV and a mesh of 12×12×12 at the CsX-rich (e.g.,  $\Delta\mu_{Cs}$  +  $\Delta\mu_{Cl}$ = -4.00 eV for CsSnCl<sub>3</sub>) and SnX<sub>2</sub>-rich conditions ( $\Delta\mu_{Cs} + \Delta\mu_{Cl}$ = -4.30 eV).

**Table S3** Surface energies ( $E_{surf}$ , J/m<sup>2</sup>) of relaxed (100)-CsBr and (100)-SnBr<sub>2</sub> surfaces of CsSnBr<sub>3</sub> calculated without SOC and with SOC at the CsBr-rich (e.g.,  $\Delta\mu_{Cs} + \Delta\mu_{Br} = -3.79 \text{ eV}$ ) and SnBr<sub>2</sub>-rich conditions ( $\Delta\mu_{Cs} + \Delta\mu_{Br} = -4.04 \text{ eV}$ ).

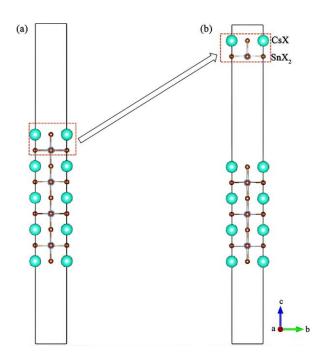
	) 2	(	P C5 P D1	/	
		Without SOC		With SOC	
-	$\Delta \mu_{\rm Cs} + \Delta \mu_{\rm Br}(eV)$	CsBr	SnBr <sub>2</sub>	CsBr	SnBr <sub>2</sub>
	-3.79	0.059	0.162	0.062	0.160
_	-4.04	0.119	0.101	0.123	0.100

**Table S4** Cleavage energies ( $E_{cl}$ , J/m<sup>2</sup>) of CsSnBr<sub>3</sub> and CsPbBr<sub>3</sub> cutting through its (100), (110), and (111) surfaces. The values of CsPbBr<sub>3</sub> are taken from ref. 9.

CsMBr <sub>3</sub>	(100)	(110) polar	(110) non- polar	(111) polar
CsSnBr <sub>3</sub>	0.355	1.193	0.652	0.964
CsPbBr <sub>3</sub>	0.344	1.407	0.658	1.166



**Fig. S1** Calculated phase diagrams of CsSnBr<sub>3</sub> (100) surfaces: (a) without SOC; (b) SOC.



**Fig. S2** Calculation models of the monolayer exfoliation energy through (100) plane of  $CsSnX_3$ : (a) a slab with nine atomic layers (4.5-unit cells); (b) the topmost unit cell of the slab being separated. The monolayer exfoliation energy is calculated as the difference in total energies between the configuration (b) and (a). The separated model in (b) remains its split style without further relaxation. The vacuum thicknesses are larger than 15 Å.

### References

- R. Lin, K. Xiao, Z. Qin, Q. Han, C. Zhang, M. Wei, M. I. Saidaminov, Y. Gao, J. Xu, M. Xiao, A. Li, J. Zhu, E. H. Sargent and H. Tan, *Nat. Energy*, 2019, 4, 864-873.
- T. Nakamura, S. Yakumaru, M. A. Truong, K. Kim, J. Liu, S. Hu, K. Otsuka, R. Hashimoto, R. Murdey, T. Sasamori, H. D. Kim, H. Ohkita, T. Handa, Y. Kanemitsu and A. Wakamiya, *Nat. Commun.*, 2020, 11, 3008.
- 3. T. Udayabhaskararao, M. Kazes, L. Houben, H. Lin and D. Oron, *Chem. Mater.*, 2017, **29**, 1302-1308.
- 4. M. Zhang, H. Li, Q. Jing, Z. Lu and P. Wang, *Crystals*, 2017, **8**, 2.
- 5. J. H. Jung, C. H. Park and J. Ihm, *Nano Lett*, 2018, **18**, 2759-2765.
- 6. J. M. van den Berg, *Acta Crystallogr.*, 1961, 14, 1002-1003.
- 7. P. Ravindran, A. Delin, R. Ahuja, B. Johansson, S. Auluck, J. M. Wills and O. Eriksson, *Phys. Rev. B*, 1997, **56**, 6851-6861.
- 8. P. Eckold, W. Hügel, R. E. Dinnebier and R. Niewa, Z. Anorg. Allg. *Chem.*, 2015, **641**, 1467-1472.
- 9. Y. Yang, C. Hou and T. X. Liang, *Phys. Chem. Chem. Phys.*, 2021, 23, 7145-7152.