Supplementary Information:

Efficient and Stable Formamidinium-Caesium Perovskite Solar Cells and Modules from Lead Acetate-Based Precursor

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Fig. S1. Solution ¹H-NMR spectrum of the $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films made from FA-Ac-FACs. The films were dissolved in DMSO-d₆ for the measurement. In addition to the peaks from FA at 7.85 ppm, we found an additional peak at 9.31 ppm that can be assigned to *sym*-triazine.



Fig. S2. XRD patterns of the FACs perovskite films made from MA-Ac-FACs (red) and A-Ac-FACs (black). Peaks marked with * and # are assigned to FTO and *N*-methyl formamidinium iodide (MFAI), respectively.



Fig. S3 Solution ¹H NMR results of dissolved $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films made from MA-Ac-FACs (blue) and A-Ac-FACs (red) and reference $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ film made via the PbI₂ route (black). The films were directly dissolved in DMSO-d₆ and spiked with HI for the measurement.



Fig. S4. Solution ¹H-NMR spectrum of the $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films made from $PbAc_2$ route when using MA⁺ as volatile cation. The films were dissolved in DMSO-d₆ and spiked with HI for the measurement.



Fig. S5. ¹H-¹H COSY NMR spectrum of $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films made from PbAc₂ route using MA⁺ as volatile cation. The films were dissolved in DMSO-d₆ for the measurement. The left part shows the correlation of the H^{CH} with H^{NH} protons while the right part shows coupling between H^{CH} and H^{Me}. The films were directly dissolved in DMSO-d₆ and spiked with HI for the measurement.



Fig. S6. Mass spectrum results of FAPbl₃ films made from PbAc₂ route using MA⁺ as volatile cation. The films were dissolved in methanol for the measurement. The top section contains the anions, the m/z peak at 126.8 belongs to iodide. The bottom section contains cations. (m/z 32.0: MA, m/z 45.0: FA, m/z 59.0: MFA, m/z 73: DMFA).





Fig. S7. Proposed side reaction pathway in the precursor solution of FA-Ac-FACs based on Wang et al. and Grundamm and Kreutzberger. (1, 2)



Fig. S8. Proposed side reaction pathway in the precursor solution of MA-Ac-FACs based on Wang et al. (1)



Fig. S9. Solution ¹H NMR results of the perovskite precursor solution. Green: MA-Ac-FACs, Blue: FAMA perovskite Pbl₂ route, Red: A-Ac-FACs, Black: FA perovskite Pbl₂ route. The solutions were directly prepared in DMSO-d₆ for the measurement.



Fig. S10. Solution ¹³C-NMR spectrum of MA-Ac-FACs precursor solution. The solutions were prepared in DMSO-d₆ for the measurement.



Fig. S11. Superimposed ¹H-NMR spectra of MA-Ac-FACs precursor solution as a function of aging time. The solutions were prepared in DMSO-d₆ for the measurements.



Fig. S12. Proposed reaction pathway in the precursor solution of (A-Ac-FACs).



Fig. S13. Photograph of a blade-coated perovskite thin film on a 10 cm × 10 cm substrate.



Fig. S14. XRD patterns of $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ prepared from PbI₂ route and PbAc₂ route, with and without HPA addition (11.25 mol% HPA with respect to Pb).



Fig. S15. The photovoltaic performance of PSCs prepared from Pbl_2 route with and without HPA addition. (11.25 mol% HPA with respect to Pb). Statistic device parameters are obtained from the reverse J-V scan.



Fig. S16. Characterization of $FA_{0.83}Cs_{0.17}Pb(I_xBr_{(1-x)})_3$ films made from $PbAc_2$ route with different Br composition. (A) XRD patterns. (B) UV-vis absorption and PL spectra. (C) SEM (Scale Bar: 1µm).



Fig. S17. $FA_{0.83}Cs_{0.17}Pb(I_{0.8}Br_{0.2})_3$ device photovoltaic performance. (A) *J-V* characteristics of champion $FA_{0.83}Cs_{0.17}Pb(I_{0.8}Br_{0.2})_3$ device. (PCE: 17.8% (reverse) 18.1% (forward)) (B) Stabilized power output (SPO) of champion $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ device.



Fig. S18. Solution ¹H NMR spectra of $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films made via PbAc₂ route after coating (black), annealed at 120 °C for 20 min (red) and 40 min (blue), respectively. The films were dissolved in DMSO-d₆ for the measurement.



Fig. S19. Two-dimensional grazing-incidence wide-angle X-ray scattering (GIWAXS) patterns of the reference and target $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ films. Perovskite thin films were thermally aged at 130°C in N₂ environment.



Fig. S20. Pole figure of Pbl₂ (001) peak in fresh perovskite films integrated azimuthally along $q \sim 0.9$ Å⁻¹.



Fig. S21. FLIM: Carrier lifetime histogram.



Fig. S22. Cross sectional SEM of perovskite thin films. (Scale Bar: 500nm)



Fig. S23. Atomic force microscopy (AFM) images. (A and B) $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ film made from PbI₂ route (A) and PbAc₂ route (B). Scan area: 5 µm × 5 µm. (C and D) $FA_{0.83}Cs_{0.17}Pb(I_{0.9}Br_{0.1})_3$ film made from PbI₂ route (C) (RMS roughness: 28.42nm) and PbAc₂ route (D) (RMS roughness: 19.34nm). Scan area: 30 µm × 30 µm.



Fig. S24. EQE and the integrated J_{SC} of the champion device.



Fig. S25. The distribution of the PCEs of 16 sub-cells on a 10cm*10cm substrate.



Fig. S26. The sketch of the top view of module and metal mask.



Fig. S27. Pole figure of Pbl₂ (001) peak in fresh and thermally aged perovskite films integrated azimuthally along $q \sim 0.9$ Å⁻¹.



Fig. S28. Photographs of perovskite thin films (thermally aged at 130 °C in nitrogen environment under dark condition).



Fig. S29. Long-term device operational stability test. The devices were kept at room temperature without active cooling in a N_2 environment. PTAA was used to replace spiro-OMeTAD in the normal structured (*p*-*i*-*n*) device.

Lead Source	Perovskite composition	PCE (%)	Reference
Pb(SCN) ₂	MAPb(SCN) ₂ I	8.3	(3)
Pb(SCN) ₂	MAPbl ₃	7.6	(4)
Pb(SCN) ₂	MAPb(SCN) _x I _{3-x}	15.1	(5)
Pb(NO ₃) ₂	MAPbl ₃	12.6	(6)
Pb(NO ₃) ₂	MAPbl ₃	8.4	(4)
Pb(acac) ₂	MAPbl ₃	1.5	(7)
Pb(HCOO) ₂	MAPbl ₃	20.3	(8)
Pb(Ac) ₂	MAPbl ₃	10.8	(7)
Pb(Ac) ₂	MAPbl ₃	12.5	(9)
Pb(Ac) ₂	MAPbl ₃	15.2	(10)
Pb(Ac) ₂	MAPbl ₃	16.2	(11)
Pb(Ac) ₂	MAPbl ₃	18.0	(12)
Pb(Ac) ₂	MAPbl _{3-x} Br _x	18.7	(13)
Pb(Ac) ₂	MAPbl _{3-x} Br _x	18.3	(14)
Pb(Ac) ₂	MAPbl ₃	11.6	(15)
Pb(Ac) ₂	MAPbl _{3-x} Br _x	16.3	(16)
Pb(Ac) ₂	MAPbl ₃	18.5	(17)
Pb(Ac) ₂	CsPbl₃	12.9	(18)
Pb(Ac) ₂	FA _{0.83} Cs _{0.17} Pb(I _{0.9} Br _{0.1}) ₃	21.0	This work

Table S1. Summary of the PCE of PSCs made from non-halide lead source-based precursor solution.

Table S2. Summary of the processing parameters for P1, P2, P3 laser cutting

	Scanning speed (mm/s)	Power (W)	Repeat frequency (kHz)
P1	500	80	20
P2	1000	30	80
P3	1000	30	80

EXPERIMENTAL PROCEDURES

Materials

Unless otherwise specified, all materials were purchased from either Alfa Aesar or Sigma-Aldrich without any further purification. Formaminidium iodide (FAI), formamidinium bromide (FABr), methylammonium iodide (MAI) were purchased from Greatcell Solar Ltd. Poly(triaryl amine) (PTAA) (Mn<6000 and 15000</p>
Mn<30000) and *iso*-butylammonium bromide (*i*-BABr) were purchased from Xi'an Polymer Light Co., Ltd. Lead acetate (PbAc₂) was purchased from TCI. Spiro-OMeTAD and FK209 Co(III) TFSI Salt were purchased from Luminescence Technology Corp. 4-isopropyl-4'-methyldiphenyliodonium Tetrakis(pentafluorophenyl) borate (TPFB) was purchased from TCI.

Preparation of precursor solutions

PbAc₂ based precursor solutions: The precursor solutions were prepared in N₂ glovebox by mixing stoichiometric amount of FABr, FAI, CsI and PbAc₂ in anhydrous *N*,*N*-dimethylformamide (DMF) with further addition of FAI, MAI and NH₄I at a molar ratio of 2:1 (with respect to Pb²⁺) as volatile organic salts for FA-Ac-FACs, MA-Ac-FACs and A-Ac-FACs, respectively. The precursor solution was stirred for 5 min, then three equimolar amounts (relative to Pb²⁺) of deionized water were added to help dissolve PbAc₂ and CsI. The precursor solution was further stirred for 10 min until fully dissolved. Finally, hypophosphorous acid (HPA) solution was added, and the molar ratio of HPA/Pb²⁺ was 11.25%. (19)

 PbI_2 based precursor solutions: The 1.45 M precursor solution was prepared in a glovebox by mixing stoichiometric amounts of FAI, CsI, PbI_2 and $PbBr_2$ in a solvent mixture (4:1) of DMF and dimethyl sulfoxide (DMSO). The precursor solution was stirred for 1hour before use.

Device fabrication

FTO glass was cleaned and sonicated with detergent, deionized water, acetone and isopropanol for 15 min each. Then, it was treated by UV-plasma for 15 min before use. The ETL SnO₂ was deposited onto FTO glass by a chemical bath deposition method (CBD). 12 mM of SnCl₂·2H₂O with 5 g urea was dissolved in 100 μ L of thioglycolic acid, 5 mL of HCl (37%) and 400 mL deionized water to make the precursor solution. The as-cleaned FTO substrate was dipped into the diluted solution (2 mM) for 2.5 hour at 90 °C. Then it was sonicated in deionized water for 5 min followed by the annealing at 180 °C for 1 hour. Then the substrate was treated by UV-plasma for 15 min for the deposition of perovskite. For Pbl₂ route, 1.45 M perovskite precursor solution was spin-coated and then annealed at 100 °C for 60mins, according to previous report. (20) For PbAc₂ route, 1.1M of A-Ac-FACs solution was spin-coated at 3000rpm of 30s with a ramping speed of 1000rpm/s. For the blade coating, 0.8

M of A-Ac-FACs solution was blade coated at 1.67 mm/s with N₂ knife blowing at room temperature in an ambient air environment. The coating gap is 150 µm, and the gas pressure was controlled at ~0.04 MPa. The as-coated film was then annealed at 120 °C for 40 min. After annealing, for the fabrication of PSCs, large area-substrate (10 cm × 10 cm) coated with perovskite was divided into 16 pieces with each piece having 2.5 × 2.5 cm² size. Then 2.5 mg/mL i-BABr in IPA was dynamically spun on top of the perovskite at 5000 rpm for 30 s followed by the annealing at 100 °C for 5 min. For HTL, spiro-OMeTAD (73 mg, 30 µL 4-tertbutylpridine, 18 µL LiTFSI (520 mg/mL in acetonitrile) and 29 µL FK209 (300 mg/mL in acetonitrile) in 1 mL chlorobenzene) was spin-coated at 3000 rpm for 30 s. For the operational stability test, 10mg/mL of PTAA in CB with (10wt% of TPFB) was used to replace spiro-OMeTAD following a previous literature. (21) Finally, an 80 nm gold layer was deposited by thermal evaporation to complete the device. For the thermal stability test, the inverted device architecture was used. PTAA (2 mg/mL in CB) was spin-coated at 2000 rpm for 20 s, and annealed at 150 °C for 10 min as the HTL. Then an Al₂O₃ nanoparticles layer (1:150 v/v ratio Al₂O₃ NPs/IPA) was dynamically spun on PTAA at 4000 rpm for 30 s. A subsequent anneal at 100 °C for 5 min was performed. The deposition of perovskite used the same procedure described before. PC₆₁BM (20 mg/mL in CB) was dynamically spun onto the perovskite layer at a speed of 2000 rpm for 20 s. The samples were then annealed at 100 °C for 5 min. An interlayer of BCP (0.5 mg/mL in IPA) was spin-coated at 5000rpm for 30 s. Finally, an 80 nm gold layer was thermally evaporated.

Module Fabrication

The module was designed to have 6 subcells connected in series with P1, P2 and P3 interconnections. The FTO glass was first etched by nanosecond laser to from P1 lines. After the deposition of perovskite and spiro-OMeTAD, the P2 lines were formed by laser cutting. Finally, after the Au evaporation, the module was re-etched to form P3 lines. The gap between P1, P2 and P3 are about 200µm. The detailed processing parameter is shown in Table S3.

Material characterization

¹H-NMR, ¹H-¹H COSY, ¹³C NMR spectra were recorded on a Bruker AVIII 400MHz spectrometer. For perovskite thin film samples, 4 perovskite films were dissolved in DMSO-d₆ for the measurement. Small amount of HI (1 μ L of 57% w/w HI in 750 μ L of DMSO-d₆) was added to get a better quality ¹H-NMR spectra. (22)

The external quantum efficiency (EQE) of the PSCs were measured by Fourier transform photocurrent spectroscopy based on a Bruker Vertex 80v Fourier transform spectrometer. A tungsten-halogen lamp was used as the light source and the intensity was calibrated against

a reference silicon photodiode. The wavelength interval we used for the measurement is 10nm.

The GIWAXS experiments were performed at the SAXS/WAXS beamline at the Australian Synchrotron. (23) The beam energy was fixed at 15 keV. The scattering patterns were collected using an in-vacuum Dectris Pilatus 2M detector with a total exposure time of 3 s. The entire beam path from source to sample to detector was placed under vacuum to minimise diffuse air scattering hence improving the signal-to-noise ratio of the scattering patterns. An incident angle of 0.4° which is higher than the critical angle was used to probe the entire depth of the perovskite film. The sample-to-detector distance was calibrated to be \sim 66 cm using a silver behenate standard. Data reduction and analysis was performed using an altered version of NIKA analysis package implemented in Igor. (24)

Uv-vis spectra were collected with a Perkin Elmer Lambda 1050 spectrometer.

The PL emission spectra and time-resolved confocal PL mapping were measured with a Picoquant Microtime 200 instrument with 485nm laser excitation.

TRPL spectra were recorded using a luminescence spectrometer (Edinburgh instruments, FLSP920) with a pulsed diode laser as the excitation source (466 nm). PL was collected using a grating monochromator, and the excitation and PL detection was from the perovskite side. Spectra are corrected for the wavelength-dependence for the detector.

SEM images were taken using a FEI Nova NanoSEM 450 FEG scanning electron microscope with a beam accelerate voltage at 5kV.

AFM images were collected with a Dimension iCon AFM in tapping mode in the air. For the thermal stability test, the encapsulated devices were placed on a 65°C hot plate in the air.

The X-ray diffraction (XRD) data was collected by using a Bruker D8 Advance diffractometer in Brag-Brentano geometry. A Cu- $k\alpha$ ($\lambda = 0.15406$ nm) radiation was used with a generator voltage at 40kV and current at 40mA. Diffraction data was collected by a Lyxeye XE position sensitive detector in 1D mode. As the sample film was prepared with a thickness that the Xray beam is able to penetrate through, this is evident with the substrate diffraction peaks can always been visible. A dynamic divergence slit size was used to maintain a constant beam illumination size at 9 mm. In-situ XRD data was collected with sample thermally aged under 130 °C in dried N₂ environment in an MTC wide range variable temperature chamber. The XRD data was analysed using a Rietveld refinement software TOPAS Academic version 7. Pawley refinement was used to refine the perovskites phase with a cubic crystalline structure with a space group of *Pm-3m*. The microstrain and crystallite size were extracted from the refinement results with calculated errors. The environmental contribution for in-situ XRD pattern was measured and has been modelled in the refinement.

Device characterization

The photovoltaic performance was measured under 1-sun illumination (100 mW/cm⁻²) using an Oriel solar simulator equipped with a xenon lamp and an AM 1.5G filter. The light intensity was calibrated by a standard silicon reference solar cell (Oriel, VLSI standards). The *J-V* characteristics were performed with forward and reverse scan between -0.1 to 1.2V with a voltage step of 10mV and settling time of 100ms (100mV/s) by a Keithley 2400 source meter. All devices were tested using a black metal mask with an aperture area of 0.16cm² and 10cm² for small area devices and solar modules, respectively.

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