

## Supporting Information

### **Low-Temperature Metal-Catalyzed Synthesis of Metal Oxide Nanoparticles Encapsulated and Nitrogen-doped Carbon Nanotubes from Carbon Nitride as Anodic Materials of High-Performance Lithium-Ion Batteries**

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## Experimental Section

### Preparation of Samples.

**The synthesis of C<sub>3</sub>N<sub>4</sub>:** 10 g melamine was pressed into a pellet under a pressure of 20 MPa. The pellet was transferred to a semiclosed quartz boat and calcined at 600 °C for 0.5 h in N<sub>2</sub> atmosphere. Then the obtained yellow powder was collected as the carbon and nitrogen precursor for the preparation of metal@NCNT.

**The in situ synthesis of Co@NCNT and Ni@NCNT arrays:** 2 g Co(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O or Ni(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O and 5 g C<sub>3</sub>N<sub>4</sub> were added to 5 ml aqueous solution under tempestuously stirring to form a yellow slurry. Then, the colloidal compounds were milled continually to form a yellow paste, following the evaporation of water. The obtained yellow paste was dried at 80 °C for 24 h and manually ground into powder. The powder was transferred to a semiclosed quartz boat and heated at 350 °C for 1 h at a heating rate of 2 °C min<sup>-1</sup> in a tubular furnace under N<sub>2</sub> flow, and the temperature was increased to 700 °C at 2 °C min<sup>-1</sup> and kept at 700 °C for 3 h, followed by cooling to room temperature naturally.

**The in situ synthesis of Co<sub>3</sub>O<sub>4</sub>@NCNT and NiO@NCNT arrays:** The obtained Co@NCNT and Ni@NCNT powder was calcined in a muffle furnace in air at 400 °C for 0.5 h with a heating rate of 10 °C min<sup>-1</sup>, and the final product was obtained.

**The synthesis of Co<sub>3</sub>O<sub>4</sub>/CB:** 2 g Co(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O and 0.5 g carbon black was dispersed in 3 ml deionized water and ground by mortar and pestle to form a slurry. Then the slurry was dried at 80 °C in oven overnight. The obtained powder was transferred to a quartz boat and covered by a quartz cap. Then heat treated at 400 °C for 1 h in air to form Co<sub>3</sub>O<sub>4</sub>/CB.

### Material Characterization

The crystalline structure of the products was characterized by X-ray diffraction (XRD) on a Rigaku SmartLab9 powder diffractometer equipped with Cu  $K_\alpha$  radiation ( $\lambda = 1.541 \text{ \AA}$ ). Thermogravimetric analysis (TGA) was performed on a TA SDT Q600 analyser in air with a heating rate of  $10 \text{ K min}^{-1}$ . The morphology of the products was observed by field emission scanning electron microscopy (FE-SEM) on a ZEISS SUPRA 55 microscope and transmission electron microscopy (TEM) on a Tecnai G2 F20 microscope. X-ray photoelectron spectroscopy (XPS) was conducted using a Kratos Axis Ultra DLD (delay line detector) spectrometer equipped with a monochromatic Al  $K_\alpha$  X-ray source (1486.6 eV). Raman spectra were collected on a Renishaw-1000 spectrometer by exciting a 514.5 nm Ar ion laser.  $\text{N}_2$  adsorption-desorption isotherms were recorded at 77 K on a Quantachrome NOVA 2000e sorption analyzer. Optical absorption spectroscopy was performed in the 300-700 nm range in 1 nm steps on an ultraviolet-visible-near-infrared, double beam spectrophotometer (America PerkinElmer Lambda 950).

### **Electrochemical Characterization**

$\text{Co}_3\text{O}_4@\text{NCNT}$ ,  $\text{NiO}@\text{NCNT}$  or  $\text{Co}_3\text{O}_4/\text{CB}$  and polyvinylidene fluoride (PVDF) binder in a weight ratio of 9 : 1 were mixed in N-methylpyrrolidone (NMP) and stirred for 24 h to make a slurry. The slurry was then spread on a Cu foil (13 mm in diameter, 0.3 mm in thickness) with a surface density of  $1.0 \text{ mg cm}^{-2}$  and dried at  $120 \text{ }^\circ\text{C}$  for 24 h to fabricate the working electrodes in vacuum. Lithium foil was used as both the reference electrode and the counter electrode (13 mm in diameter, 0.5 mm in thickness). 1.0 M  $\text{LiPF}_6$  in a 1:1 (v/v) mixture of ethylene carbonate (EC) and diethyl carbonate (DEC) was employed as the electrolyte. Celgard 2300 membrane (25  $\mu\text{m}$ -thick polyethylene) was adopted as a separator. The assembly of CR2032-type coin cells was conducted in a high-purity Ar filled glovebox. Galvanostatic cycling was performed between 0.01 and 3 V vs  $\text{Li}^+/\text{Li}$  at various C rates on a Land Battery Tester (Wuhan, China), where 1 C corresponds to  $1000 \text{ mA g}^{-1}$ . Cyclic voltammetry (CV) was conducted between 0.01 and 3 V at  $0.1 \text{ mV s}^{-1}$  using a CHI660E electrochemical

workstation. Electrochemical impedance spectroscopy (EIS) was performed on the same electrochemical system over the frequency range from 100 kHz to 100 mHz with a perturbation voltage of 5 mV. All of the electrochemical measurements were performed at 25 °C in an ambient atmosphere.

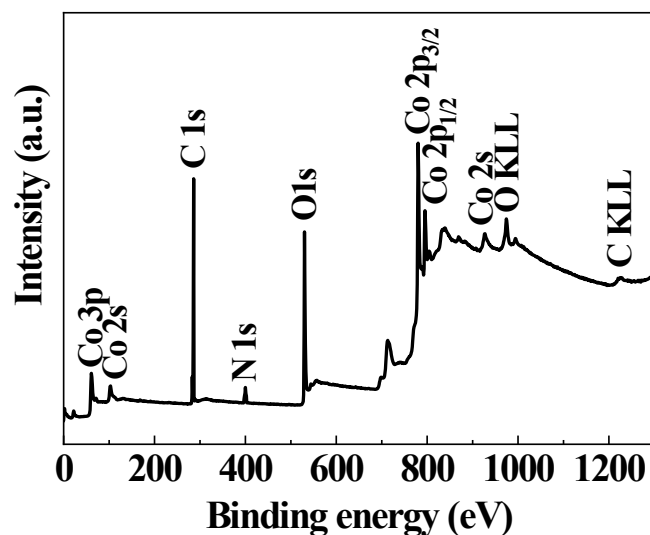


Fig. S1 XPS survey spectrum of  $\text{Co}_3\text{O}_4@\text{NCNT}$  hybrid.

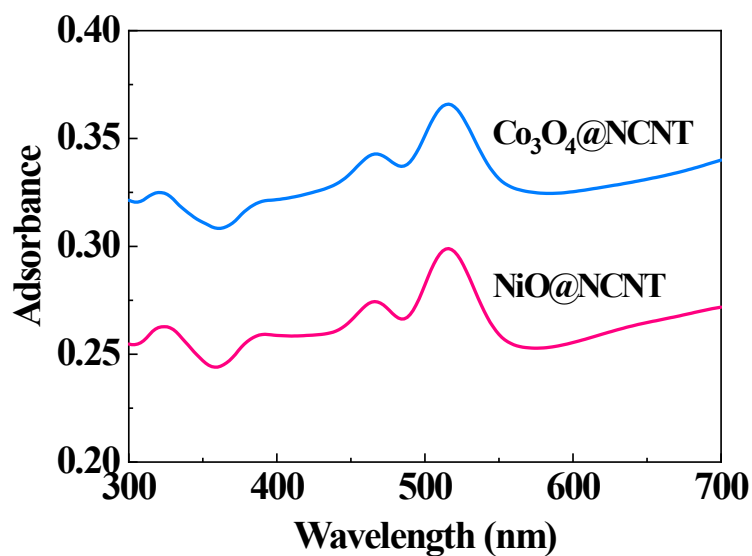


Fig. S2 Absorption spectrum of  $\text{Co}_3\text{O}_4@\text{NCNT}$  and  $\text{NiO}@\text{NCNT}$  hybrids.

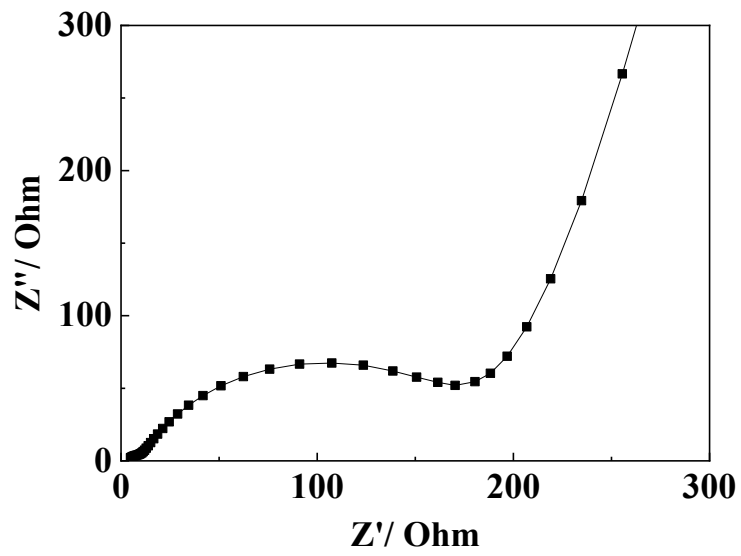


Fig. S3 Nyquist plot of Co<sub>3</sub>O<sub>4</sub>/CB.

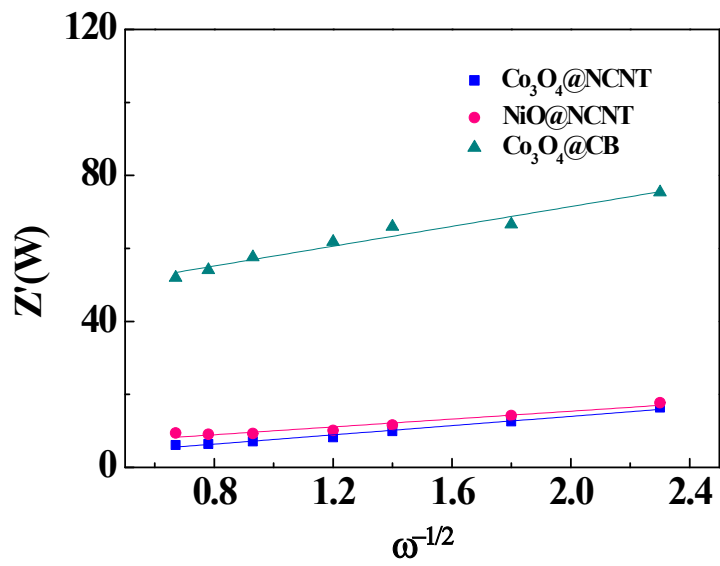


Fig. S4 The relationship between  $Z'$  and  $\omega^{-1/2}$  in the low-frequency region for LiFePO<sub>4</sub>@C@CNT anodes.

To further confirm the effect of one-dimensional nanostructure on the electrolyte ions diffusion, the measurement of electrolyte ions diffusion coefficients ( $D_{Li^+}$ ) based on eq. 1 are very necessary.<sup>1</sup>

$$D = \frac{R^2 T^2}{2A^2 n^4 F^4 C^2 \sigma^2} \quad (1)$$

where  $D$  is the diffusion coefficient of  $Li^+$ ,  $R$  is the gas constant,  $T$  is the absolute temperature,  $A$  is the surface area of the anode ( $0.15 \text{ m}^2$ ),  $n$  is the number of electrons involved in the redox reaction of  $Co^{3+}/Co^{2+}/Co$  at the electrode,  $F$  is the Faraday constant,  $C$  is the concentration of  $Li^+$  ( $7.69 \times 10^{-3} \text{ mol cm}^{-3}$ ),  $\sigma$  is the Warburg factor. According to the eq. 2, the  $\sigma$  varies negatively with  $D$ , suggesting that the lower  $\sigma$  corresponds to the higher-rate ions diffusion in the electrode. The  $\sigma$  can be directly derived from the slope of straight line in the Nyquist plots based on eq. 2.<sup>2</sup> Fig. S4 presents the relationship between  $Z'$  and reciprocal square root

$$Z' = R_D + R_L + \sigma \omega^{-1/2} \quad (2)$$

of angular frequency ( $\omega^{-1/2}$ ) in the low-frequency region, in which the slope of straight line represents  $\sigma$ . The  $\sigma$  firstly slightly decreases from  $Co_3O_4@NCNT$  (6.3) to  $NiO@NCNT$  (5.4) and then largely increases to  $Co_3O_4@CB$  (13.6), demonstrating the  $D_{Li^+}$  of  $Co_3O_4@NCNT$  and  $NiO@NCNT$  was effectively enhanced in the one-dimensional nanostructure.

1. B. Wang, Z. Zhang, Y. Ning, X. Li, T. Ruan, F. Wang, D. Wang and Y. Zhou, *Energy Technology*, 2020, **8**, 2000171.
2. D. Saikia, J. R. Deka, C.-J. Chou, C.-H. Lin, Y.-C. Yang and H.-M. Kao, *ACS Applied Energy Materials*, 2019, **2**, 1121-1133.