

Supporting Information

Two-in-one template-assisted construction of hollow phosphide nanotubes for electrochemical energy storage

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Experimental section

Chemicals

The chemical reagents used in the experiments were all analytically pure and not further purified. Ammonium molybdate tetrahydrate ($(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}\cdot 4\text{H}_2\text{O}$), Nitric acid (HNO_3 , 65%), Hydrochloric acid (HCl , 38%), Nickel sulphate hexahydrate ($\text{NiSO}_4\cdot 6\text{H}_2\text{O}$), Ferric chloride hexahydrate ($\text{FeCl}_3\cdot 6\text{H}_2\text{O}$), Ammonium persulfate ($(\text{NH}_4)_2\text{S}_2\text{O}_8$), Polyvinyl alcohol, and Potassium persulfate ($\text{K}_2\text{S}_2\text{O}_8$) are purchased from Sinopharm Chemical Reagent Co., Ltd. Aniline solution ($\text{C}_6\text{H}_7\text{N}$), Ammonia solution ($\text{NH}_3\cdot \text{H}_2\text{O}$), and Sodium hypophosphite monohydrate ($\text{NaH}_2\text{PO}_2\cdot \text{H}_2\text{O}$) were purchased from Shanghai Macklin Biochemical Co., Ltd.

Synthesis of MoO_3 nanorods

MoO_3 nanorods was prepared according to previous works with slight modifications [S1]. The detailed process is as follows: 1.4 g of $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}\cdot 4\text{H}_2\text{O}$ was stirred and sonicated in 6.66 mL of 65% HNO_3 and 33.33 mL of deionized water. The mixture was then transferred into a 70 mL Teflon-lined stainless-steel autoclave and heated to 180°C at room temperature for 20 h. When the reaction was finished, the precipitate was collected by centrifugation and washed several times with deionized water and ethanol. Finally, the cleaned precipitate was dried in a vacuum oven at 80°C for 12 h.

Synthesis of PANI@ MoO_3 nanorods

PANI@ MoO_3 nanorods were prepared by *in-situ* polymerization at room temperature. The process is as follows: the prepared 0.10 g MoO_3 nanorods were added

to 10 mL of deionized water and stirred ultrasonically for 20 min for uniform dispersion. The dispersed milky white solution was added drop by drop to a 10 mL (0.1 mol L⁻¹) HCl solution containing 0.30 mL of aniline monomer and stirred continuously for 20 min to allow complete dispersion. Subsequently, 10 mL (1.0 mol L⁻¹) HCl solution containing 0.18 g (NH₄)₂S₂O₈ was added dropwise to the aforementioned dispersion solution with continuous stirring and the pH was adjusted to 1-2 by adding HCl (1.0 mol L⁻¹). With continuous stirring, the milky white dispersion gradually changed to a dark green solution, as displayed in **Figure S2(a-c)**. The reaction time was extended to 4 h after the solution turned dark green. After the reaction, the dark green precipitate was collected by centrifugation, and repeated with deionized water and ethanol. Finally, the precipitate was dried in a vacuum oven at 80°C for 12 h.

Synthesis of Ni(OH)₂@PANI hollow nanotubes

Ni(OH)₂@PANI hollow nanotubes were prepared by CBD at room temperature. The process is as follows: the prepared PANI@MoO₃ nanorods 0.10 g were dispersed in 25 mL NiSO₄·6H₂O solution (1.0 mol L⁻¹) and 20 mL K₂S₂O₈ solution (0.125 mol L⁻¹) and sonicated for 20 min. After ultrasonic dispersion, 5 mL NH₃·H₂O was added under continuous stirring, and the reaction was continued for 2 h. After the reaction, the precipitated product was collected by centrifugation and washed several times with deionized water and ethanol. The treated precipitated product was dried in a vacuum oven at 80°C for 12 h, and the final powder sample was collected and denoted as Ni(OH)₂@PANI-1M (1M indicates the concentration of 1.0 mol L⁻¹ of NiSO₄·6H₂O). Similarly, the amount of other reagents and reaction conditions remained unchanged,

and only the concentration of $\text{NiSO}_4 \cdot 6\text{H}_2\text{O}$ was changed to 0.5 and 1.5 mol L⁻¹, denoted as $\text{Ni}(\text{OH})_2@\text{PANI-0.5M}$ and $\text{Ni}(\text{OH})_2@\text{PANI-1.5M}$, respectively. $\text{Ni}(\text{OH})_2$ hollow nanotubes were prepared similarly, the MoO_3 nanorods were used as templates and obtained without the addition of aniline solution in the preparation process. Finally, the preparation of PNAI hollow nanotubes was obtained by sonicating the prepared $\text{PANI}@\text{MoO}_3$ nanorods in 20 mL of deionized water under continuous stirring, followed by the addition of 5 mL of ammonia solution for 2 h, without the addition of other reagents.

Synthesis of $\text{Ni}_2\text{P}@\text{PANI}$ hollow nanotubes

The hollow $\text{Ni}_2\text{P}@\text{PANI}$ nanotubes (denoted as PNP) were finally prepared by a low-temperature phosphorization reaction in a tube furnace under a nitrogen atmosphere. The process is as follows: $\text{Ni}(\text{OH})_2@\text{PANI-1M}$ hollow nanotubes of 0.10 g were placed in a ceramic crucible with a cover at one end, and 0.22 g of $\text{NaH}_2\text{PO}_2 \cdot \text{H}_2\text{O}$ was placed at the other end of the crucible and covered with a lid. Next, the end of the crucible containing $\text{NaH}_2\text{PO}_2 \cdot \text{H}_2\text{O}$ was placed on the upstream side of the tube furnace. The sample was then heated to 300°C for 2 h at room temperature in the tube furnace with a gas collection device at a rate of 3°C/min under nitrogen protection (*notably, the low-temperature phosphorization reaction in this experiment should be covered with a lid in the ceramic crucible, otherwise the product cannot be obtained. Furthermore, the phosphorization process produces the toxic and explosive gas PH_3 , which requires additional gas treatment and sealing devices to ensure safety during the experiment*). After the reaction was completed, the final product was

collected after cooling to room temperature and marked as PNP-2.2 (2.2 indicates a 2.2:1 mass ratio of phosphorus source to precursor material). Similarly, the amounts of other reagents and reaction conditions were maintained constant, and the mass ratios of phosphorus source and precursors were changed to 1.2:1 and 3.2:1 and marked as PNP-1.2 and PNP-3.2, respectively. Finally, the blank control Ni₂P hollow nanotubes were prepared in the similar manner as the aforementioned Ni(OH)₂ hollow nanotubes as precursors, and the products were obtained under the same conditions of phosphatization.

Synthesis of PANI derived nitrogen-doped carbon nanotubes

PANI derived nitrogen-doped carbon nanotubes (denoted as PNCT) were prepared by removing the MoO₃ template from the PANI@MoO₃ nanorods obtained by *in-situ* polymerization at room temperature as described previously, followed by a high-temperature carbonization reaction in the tube furnace under nitrogen protection. The specific process is as follows: the prepared PANI@MoO₃ nanorods 1.0 g were ultrasonically dispersed in 60 mL of deionized water containing 20 mL of NH₃·H₂O and the reaction was continuously stirred for 1 h to remove MoO₃. The precipitate was collected by centrifugation and washed several times with deionized water and ethanol, followed by drying at 80°C for 12 h in a vacuum oven. Finally, an amount of the dried sample was placed in a ceramic crucible in the tube furnace under nitrogen protection at a rate of 2°C/min at room temperature to 600°C for 2 h. At the end of the reaction, the samples were finally collected.

Synthesis of Fe₂O₃ modified PANI-derived nitrogen-doped carbon nanotubes

The PNCT obtained from the above preparation was used as a precursor to prepare Fe₂O₃ modified PANI derived nitrogen-doped carbon nanotubes (denoted as Fe₂O₃@PNCT) through hydrothermal and subsequent calcination treatment processes. The process is as follows: 0.135 g of FeCl₃·6H₂O and 0.07 g of Na₂SO₄ were ultrasonically dissolved in 50 mL of deionized water with continuous stirring to form a homogeneous solution. Next, 0.015 g of PNCT was added to this mixture and stirred continuously for 20 min to obtain a well-dispersed black suspension. The suspension was then transferred to a 100 mL Teflon-lined stainless-steel autoclave and heated to 160°C at room temperature for 6 h. After the reaction, the sample was washed several times by centrifugation with deionized water and ethanol, and the sample was dried in a vacuum oven at 80°C for 12 h. Finally, several dried samples were placed in a ceramic crucible protected by nitrogen in a tube furnace at a rate of 3°C/min at room temperature to 400°C for 2 h. After the reaction, the brownish red powder samples were collected.

Assembly of quasi-solid-state asymmetric supercapacitors

To demonstrate the electrochemical performance and practical applications of the samples, a quasi-solid state asymmetric supercapacitor was assembled using Ni₂P@PANI hollow nanotubes as the positive electrode, Fe₂O₃@PNCT as the negative electrode, and PVA/KOH gel as the electrolyte. In particular, the preparation of PVA/KOH gel was prepared according to a previous report with slight modifications [2, 25]. The specific process is as follows: 2 g of polyvinyl alcohol (PVA) was dissolved in 40 mL of deionized water and stirred at 130°C until PVA was completely dissolved to form a clear and transparent solution. The solution was cooled to room temperature

with continuous stirring. Next, 10 mL of deionized water containing 2 g of KOH was slowly added to the PVA solution with continuous stirring to form the PVA/KOH gel electrolyte. Subsequently, the positive and negative electrodes were immersed into the PVA/KOH gel for 5 min, and the immersion was repeated three times so that the positive and negative electrode materials were completely coated with gel. Finally, a quasi-solid asymmetric supercapacitor with a sandwich (positive electrode/gel electrolyte/negative electrode) structure can be assembled. In asymmetric SCs, the two electrodes exhibit specific capacitances, masses and potential windows. In order to construct asymmetric SCs, a charge balance relationship, $Q^+ = Q^-$, should be followed between the positive and negative electrodes, where the charge stored by the positive and negative electrodes depends on the specific capacity (C), the potential range of the discharge process (ΔV) and the mass of the electrodes (m), following the below equation [24, 48]:

$$Q^+ = Q^- \quad \#(1)$$

$$Q = C \times \Delta V \times m \quad \#(2)$$

$$C^+ \times \Delta V^+ \times m^+ = C^- \times \Delta V^- \times m^- \quad \#(3)$$

Thus, calculations based on GCD curves at 2 A g^{-1} showed that the $\text{Ni}_2\text{P@PANI}$ electrode specific capacity was about 892 C g^{-1} (**Figure 3(c)**), and the specific capacity of $\text{Fe}_2\text{O}_3\text{@PNCT}$ was 328 C g^{-1} (the potential window is 1.0 V , equivalent to 328 F g^{-1} , 91 mA h g^{-1} , in **Figure S23(c)** at 2 A g^{-1}). In this work, the loading mass of active materials was $3.5 \pm 0.5 \text{ mg}$ for $\text{Ni}_2\text{P@PANI-2.2}$ (more details about the mass loading of active material see **Table S4**). From this, it can

be calculated that the load ratio of the positive and negative electrode was approximately 0.61, then the negative electrode Fe₂O₃@PNCT mass loading range was about 6.0 mg and the whole device mass loading was about 10.0 mg.

Material characterization

The morphologies and structures of the samples were characterized by scanning electron microscopy (FE-SEM, JEOL JSM 6700F) and transmission electron microscopy (FEI Tecnai G2 S-Twin F20 D573). The crystal structure of the materials was analyzed by X-ray diffraction (XRD, D/max 2550V/PC, CuK α radiation), and surface elemental analysis. Oxidation states were characterized by X-ray photoelectron spectroscopy (XPS, ESCALAB 250, Mg K α radiation) and energy dispersive X-ray spectroscopy (EDS, FEI Helios Nanolab 600i), and the specific surface area and pore size distribution results were obtained by specific surface area analyzer (ASAP 2420). The functional groups of the samples were characterized by Fourier transform infrared spectroscopy (Bruker FTIR IFS-66V/S, and pure KBr for spectral backgrounds). The samples were prepared by grinding the resulting powder and KBr in a mortar at a mass ratio of 1:10, followed by pressing at 10 MPa at room temperature to obtain the test samples. Raman spectroscopy was measured in air on a confocal Raman microscope (RM 1000, Renishaw) using a 532 nm laser. All spectral peaks were calibrated against the value of 520.7 cm⁻¹ of a silicon wafer.

Electrochemical measurements

Electrochemical measurements were performed on an electrochemical

workstation (CHI760e, CH Instruments Inc., Shanghai) in 2.0 mol L⁻¹ KOH. A standard three-electrode system was used, with Ni₂P@PANI or Fe₂O₃@PNCT as the working electrode, Pt foil as the counter electrode, Hg/HgO electrode as the reference electrode. The asymmetric device was investigated using a two-electrode testing cell. The Fe₂O₃@PNCT negative was prepared by grinding a mixture of active material, acetylene black, and PTFE in a mass ratio of 8:1:1, and pressed onto a nickel foam (2 cm × 2 cm) under 3.0 MPa (the positive electrode was prepared in the same way). The average mass loading of Fe₂O₃@PNCT was 4.0 ± 0.5 mg (more details about the mass loading of active material see **Table S4**). Cyclic voltammetry (CV) curves were performed at different scan rates in the potential range of 0.0-0.7 V and -1.0-0.0 V. Galvanostatic charge/discharge (GCD) curves were performed at different current densities in the voltage range 0.0-0.6 V and -1.0-0.0 V. Electrochemical impedance spectroscopy (EIS) was performed at an AC voltage of 5.0 mV amplitude applied in the frequency range 0.01-100 kHz.

The mass specific capacity (C_m), area specific capacity (C_s), energy density (E), and power density (P) of the electrodes and devices are calculated by the following equations [3, 48]:

$$C_{m(Cg^{-1})} = \frac{i \times \Delta t}{m} \#(4)$$

$$C_{m(Ahg^{-1})} = \frac{i \times \Delta t}{3600 \times m} \#(5)$$

$$E = \frac{1}{2} \times C \times \Delta V^2 \#(6)$$

$$P = \frac{E}{\Delta t} \#(7)$$

where C_m (C g⁻¹) is the mass specific capacity, i (A) is the charge/discharge current, Δt (s) is the discharge time, ΔV (V) is the discharge potential window, and m (g) is the electrode active mass loading.

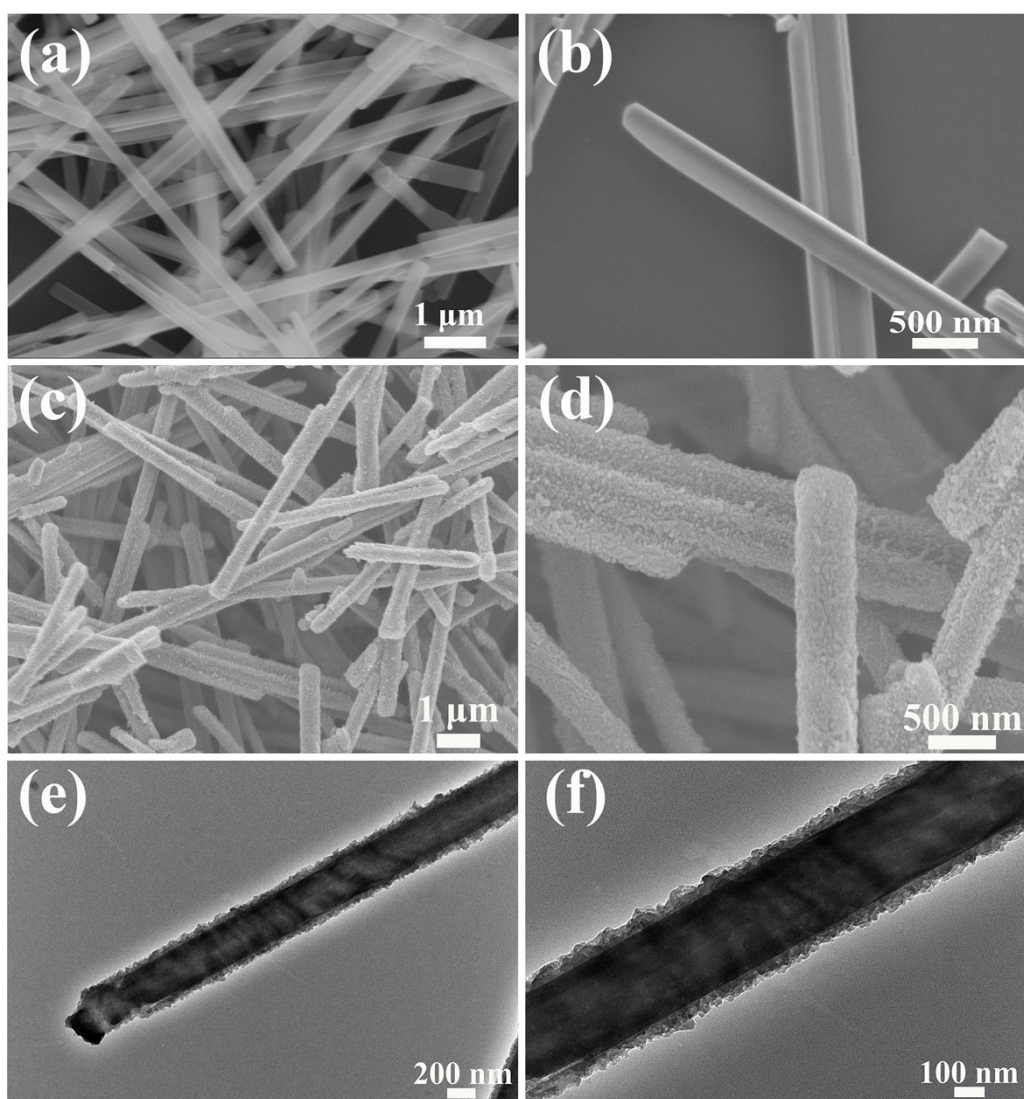


Figure S1: (a, b) SEM images of the MoO₃ template, (c, d) PANI @ MoO₃, and (e, f) the TEM of PANI @ MoO₃.

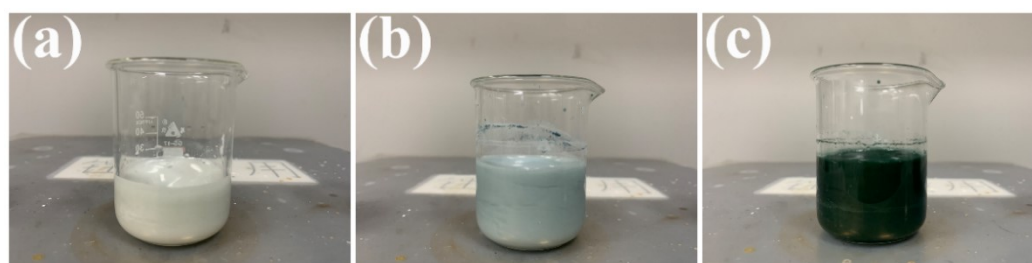


Figure S2: Digital photo image of PANI@MoO₃ sample: (a) 0 minute, (b) 30 minutes, and (c) 4 hours.

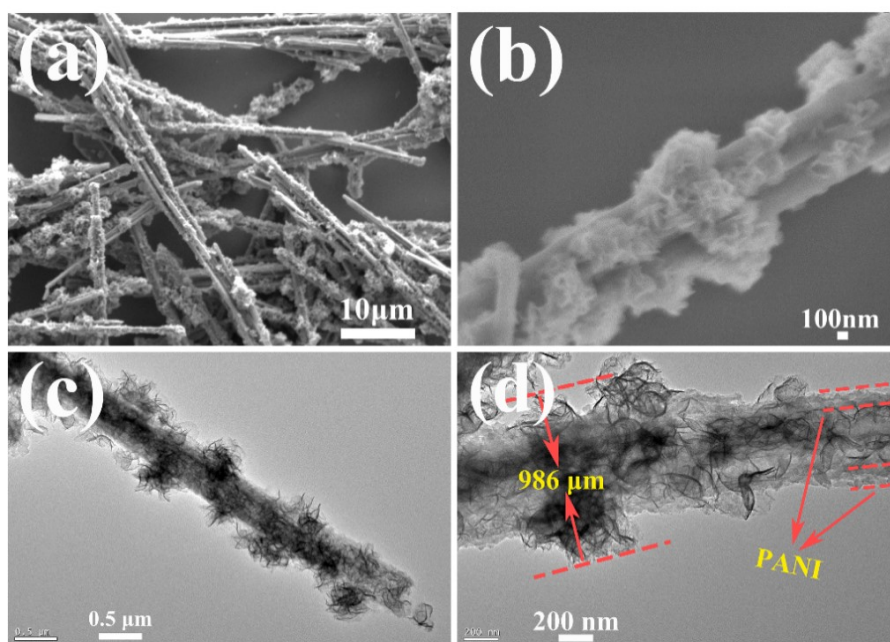


Figure S3: SEM and TEM images of the Ni(OH)₂@PANI hollow nanotubes obtained via the CBD reactions under 0.5 mol L⁻¹ NiSO₄

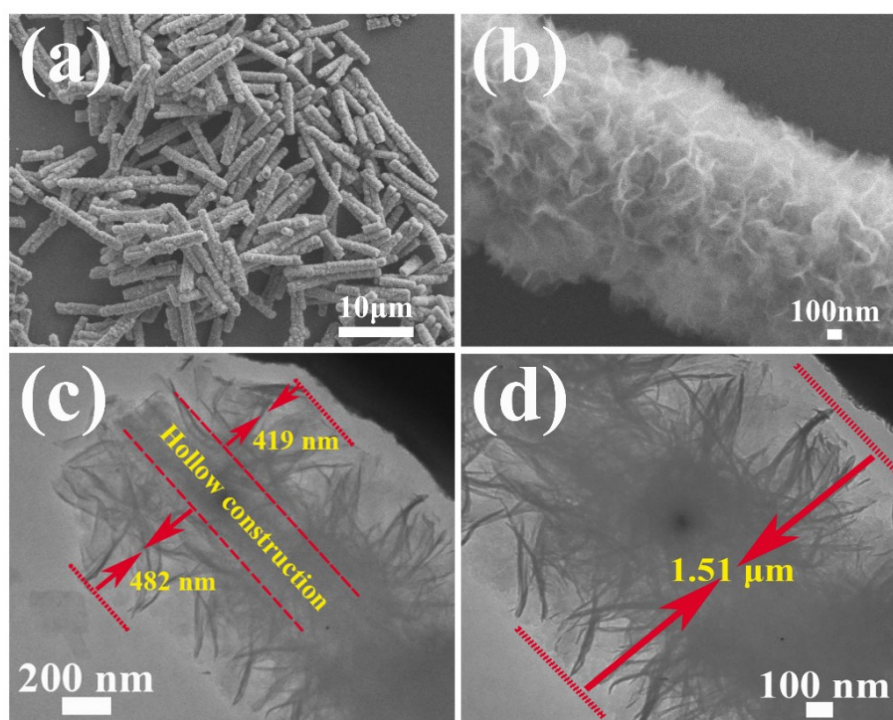


Figure S4: SEM and TEM images of the Ni(OH)₂@PANI hollow nanotubes obtained via the CBD reactions under 1.0 mol L⁻¹ NiSO₄

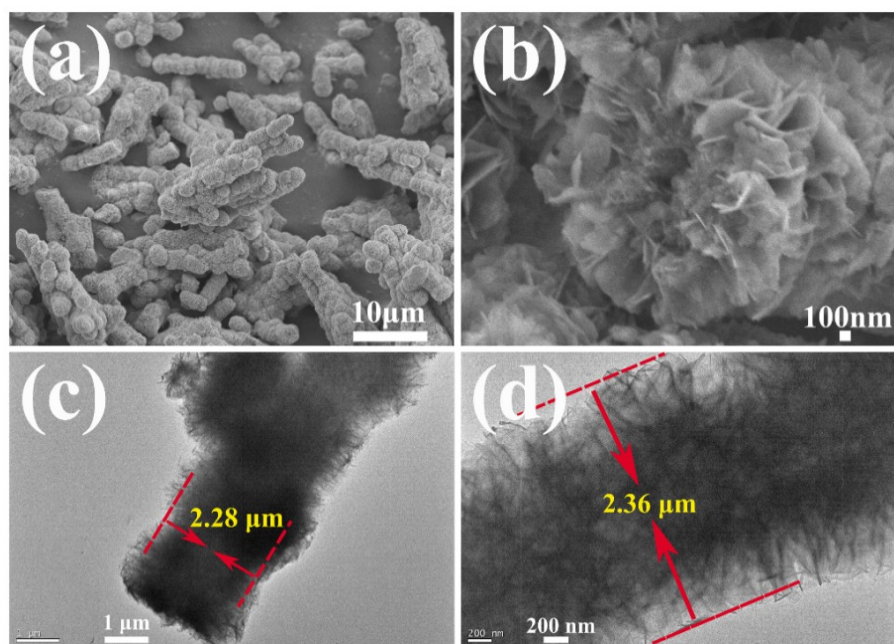


Figure S5: SEM and TEM images of the Ni(OH)₂@PANI hollow nanotubes obtained via the CBD reactions under 1.5 mol L⁻¹ NiSO₄

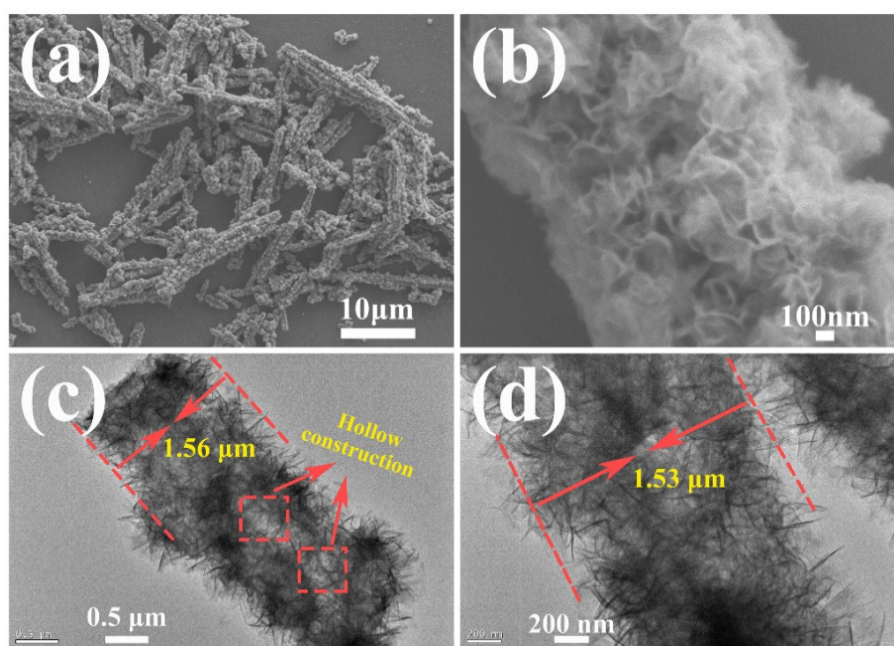


Figure S6: SEM and TEM images of the blank Ni(OH)₂ hollow nanotubes obtained via the CBD reactions under 1.0 mol L⁻¹ NiSO₄ without PANI layer.

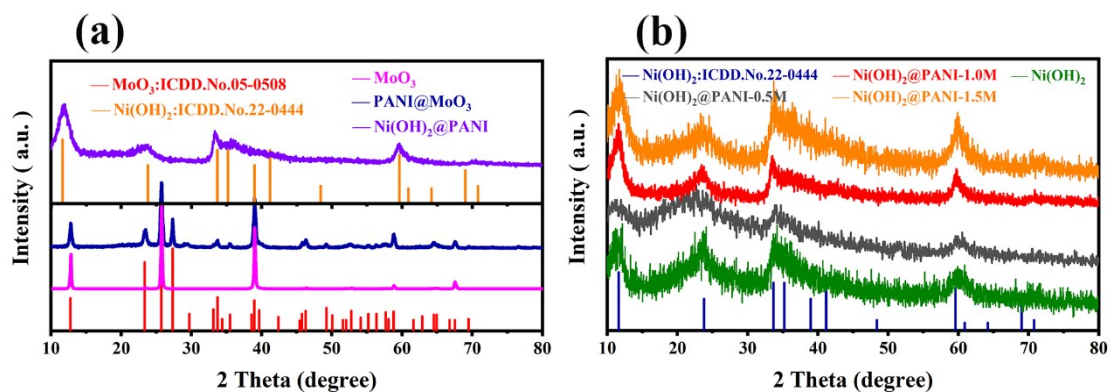


Figure S7: (a) XRD patterns of the MoO₃, PANI@MoO₃, and Ni(OH)₂@PANI precursors, (b) XRD patterns of the Ni(OH)₂@PANI-x precursors obtained via the CBD reactions at different NiSO₄ concentrations

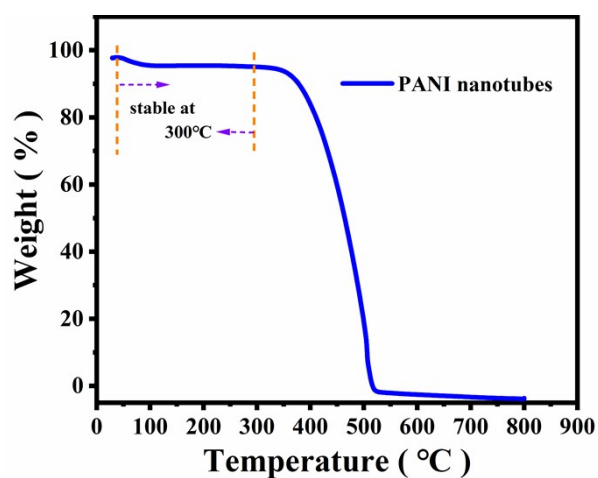


Figure S8: The TGA curves of PANI hollow nanotubes obtained through the removal of MoO₃ template.

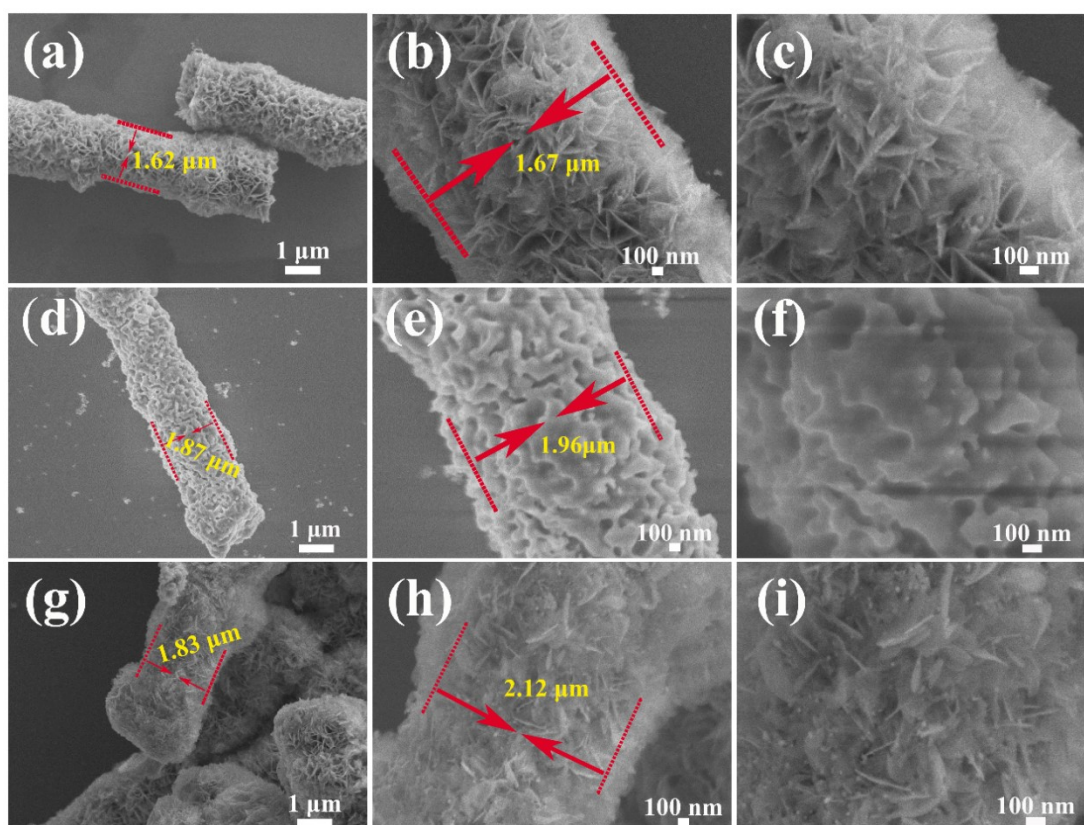


Figure S9: Morphological characterization of different ratios of phosphate samples and the blank samples: (a-c) SEM images of the Ni₂P@PANI-1.2 hollow nanotubes, (d-f) SEM images of the Ni₂P@PANI-3.2 hollow nanotubes, and (g-i) SEM images of the blank Ni₂P hollow nanotubes samples.

At a lower phosphorization ratio of 1.2:1, **Figure S9(a, b)**, due to the low phosphating mass ratio, the reaction rate was slower and thus beneficial to maintain the whole morphological stability, the sample still retained the morphology of the precursor without structural damage. In the high magnification SEM images in **Figure S9(c)**, the smooth surface of Ni(OH)₂ nanosheets in PNP-1.2 was transformed into a 'spot-like' particle shape and became rough overall. When the mass ratio was further increased to 3.2:1, as

shown in **Figure S9(d-f)**. The phosphatization at a high mass ratio no longer has uniformly dispersed rough porous nanosheets although the highly ordered nanotube morphology of the precursor was still retained. The excessive reaction destroyed the porous and interconnected Ni₂P sheets, causing the sheets to accumulate on the surface of PANI nanotubes and form a bulk, which no longer has a perfect morphology. For the blank Ni₂P sample, when the phosphatization was carried out with a mass ratio of 2.2:1, the morphology was shown in **Figure S9(g-i)**, and the surface of the nanosheets still formed a rough particle-like morphology with uniform dispersion. Since it did not contain the stable support of PANI nanotubes, the large-scale collapse and fragmentation of the nanosheets with poor morphological retention can be seen in the high magnification images in **Figure S9(h, i)** after the phosphatization process.

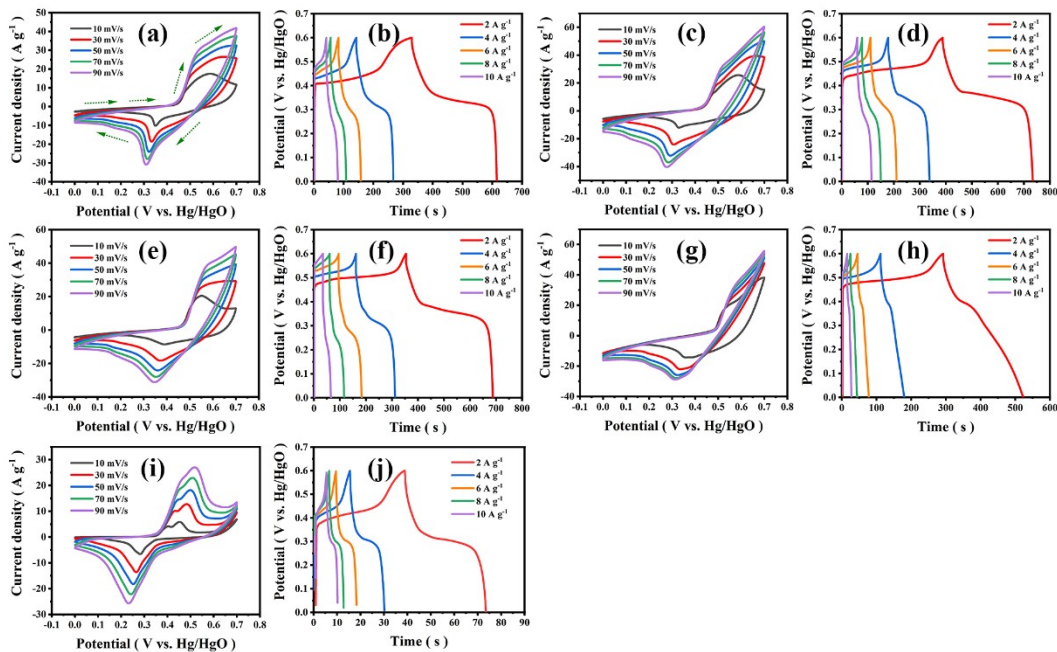


Figure S10 : Electrochemical characterization of the CV and GCD curves of Ni(OH)₂@PANI-x and blank samples prepared at different NiSO₄ concentrations: (a, b) 0.5M, (c, d) 1.0M, (e, f) 1.5M, (g, h) the blank Ni(OH)₂ hollow nanotubes, and (i, j) the blank PANI hollow nanotubes samples.

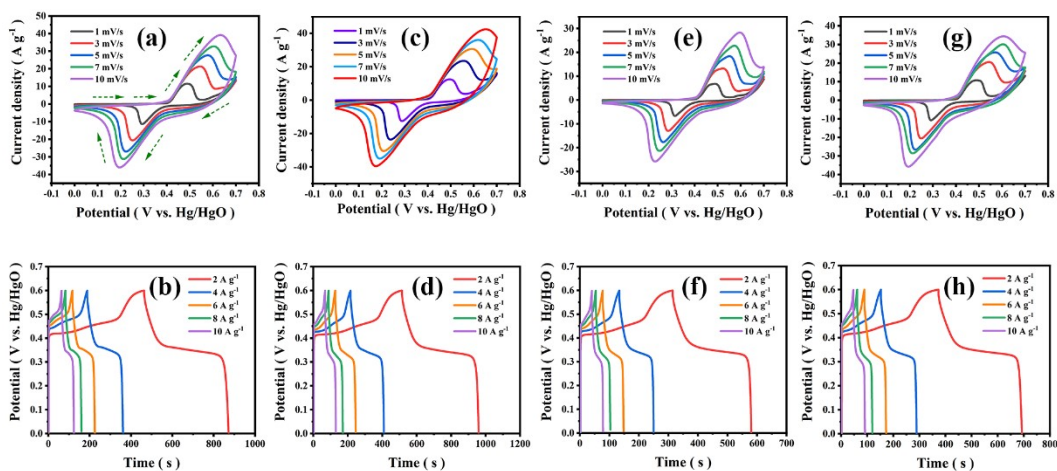


Figure S11: Electrochemical characterization of the CV and GCD curves of Ni₂P@PANI-x and the blank samples prepared at different phosphorization ratios: (a, b) 1:1.2, (c, d) 1:2.2, (e, f) 1:3.2, and (g, h) the blank Ni₂P hollow nanotubes samples.

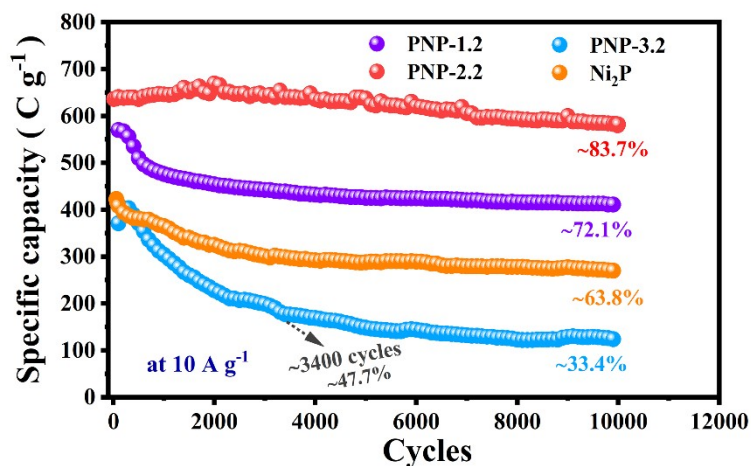


Figure S12: Cycling stability of PNP-x samples measured at a current density of 10 A

g^{-1} .

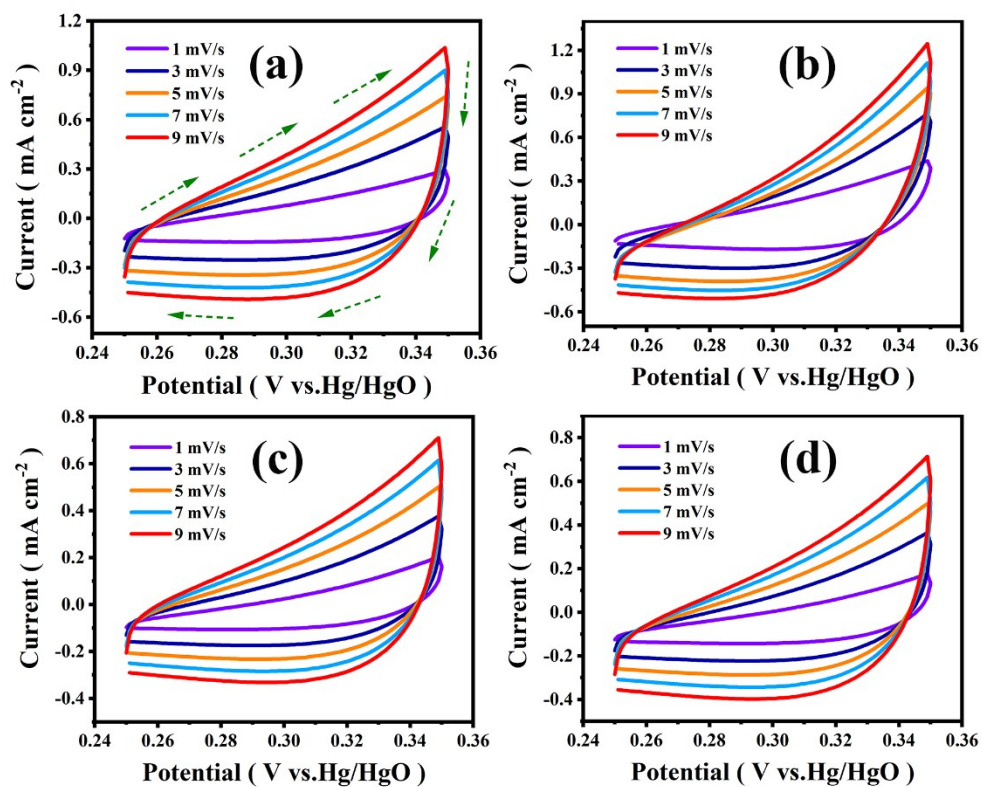


Figure S13: CV curves recorded at different scan rates between 0.25 and 0.35 V for (a) $\text{Ni}_2\text{P@PANI-1.2}$, (b) $\text{Ni}_2\text{P@PANI-2.2}$, (c) $\text{Ni}_2\text{P@PANI-3.2}$, and (d) the blank Ni_2P samples.

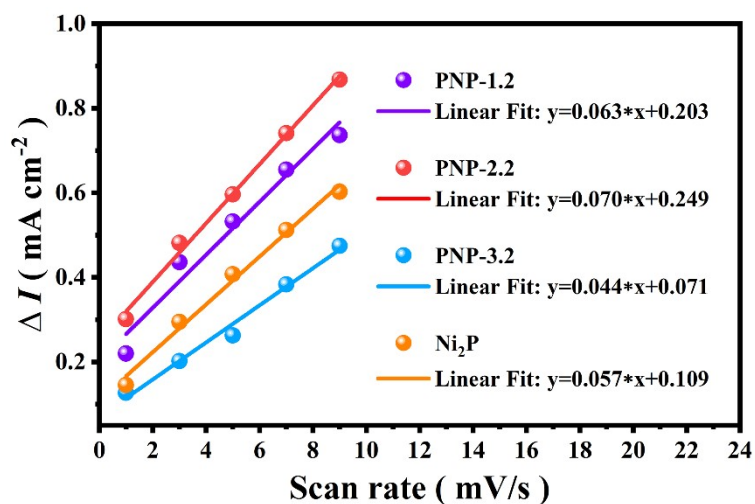


Figure S14: ΔI measured at 0.30 V vs Hg/HgO plotted as a function of scan rate.

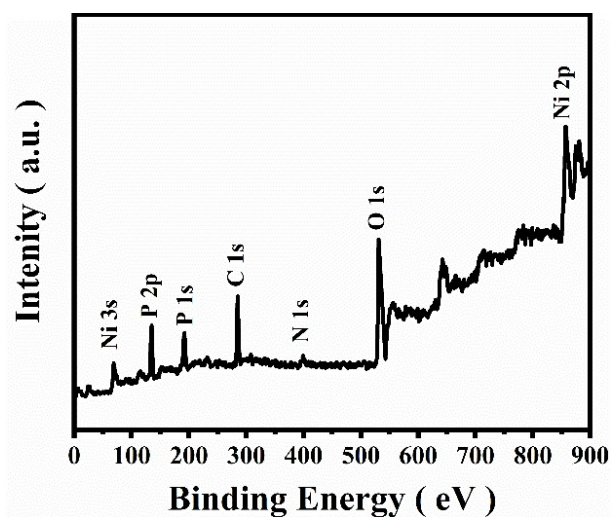


Figure S15: XPS survey spectrum of Ni₂P@PANI hollow nanotubes samples.

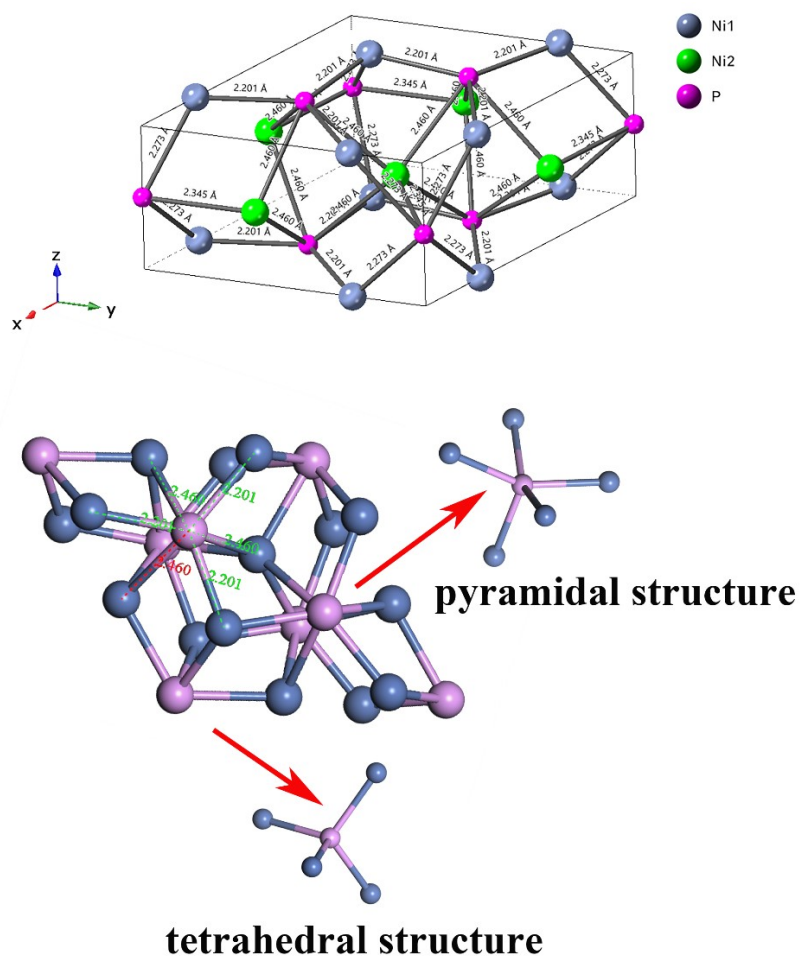


Figure S16: Ni_2P crystal structure unit and its composition.

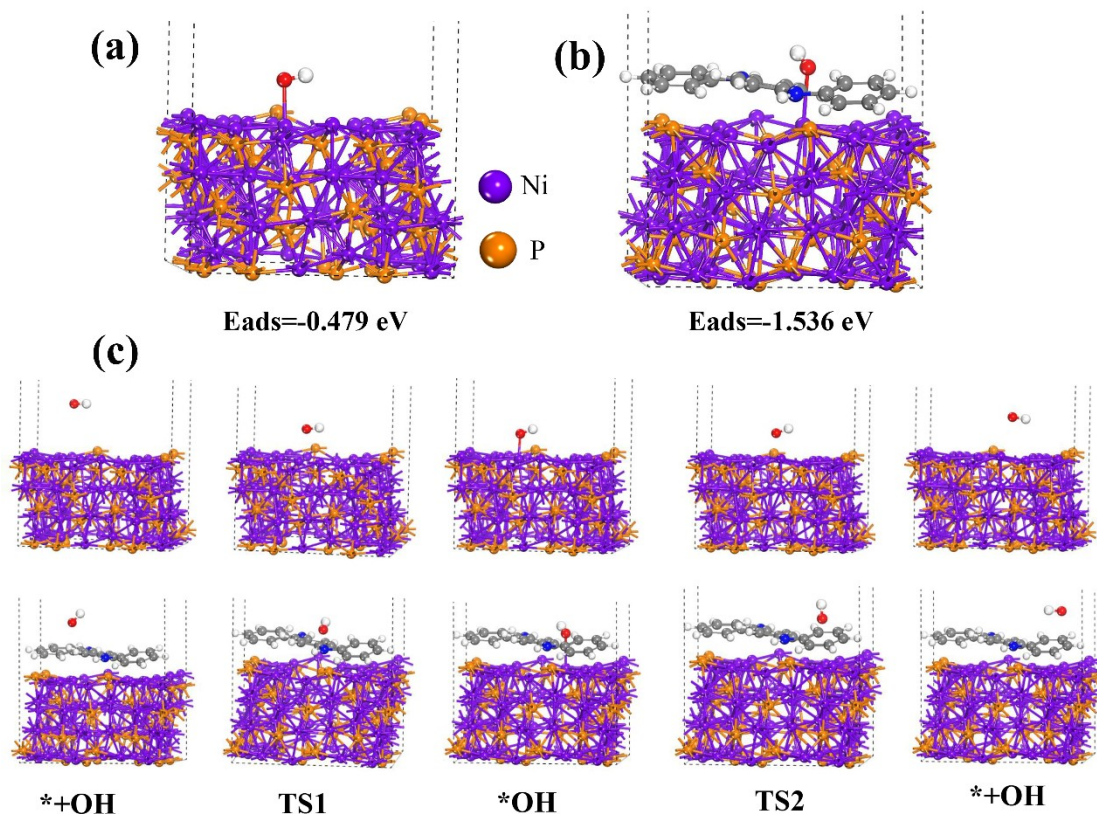


Figure S17: DFT calculations based on the transition state energy of adsorbed hydroxyl (ΔG_{OH^-}): (a) Energy barrier required for stable adsorption of OH^- on blank Ni_2P samples, (b) Energy barrier required for stable adsorption of OH^- on $Ni_2P@PANI$ samples, and (c) the structure of OH^- adsorption and desorption in the structure of the blank Ni_2P and $PANI@Ni_2P$ samples.

Computational methods

We had employed the Vienna Ab Initio Package (VASP) [S2, S3] to perform all the density functional theory (DFT) calculations within the generalized gradient approximation (GGA) using the PBE formulation [S4]. We had chosen the projected augmented wave (PAW) potentials to describe the ionic cores and taken valence electrons into account using a plane wave basis set with a kinetic energy cutoff of 400 eV [S5, S6]. Partial occupancies of the Kohn-Sham orbitals were allowed using the

Gaussian smearing method and a width of 0.05 eV. The electronic energy was considered self-consistent when the energy change was smaller than 10^{-4} eV. A geometry optimization was considered convergent when the force change was smaller than 0.05 eV/Å. The vacuum spacing perpendicular to the plane of the structure is 18 Å. The Brillouin zone integral uses the surfaces structures of $1 \times 1 \times 1$ monkhorst pack K point sampling. Finally, the adsorption energies (E_{ads}) are calculated as $E_{ads} = E_{ad/sub} - E_{ad} - E_{sub}$, where $E_{ad/sub}$, E_{ad} and E_{sub} are the optimized adsorbate/substrate system, the adsorbate in the structure and the clean substrate respectively. The free energy is calculated as follows:

$$G = E + ZPE - TS \quad (8)$$

where G , E , ZPE and TS are the free energy, total energy from DFT calculations, zero-point energy and entropic contributions, respectively. In our calculation, the U had been employed to calculate the electronic properties, and the 3.68 eV for 3d had been used in our systems.

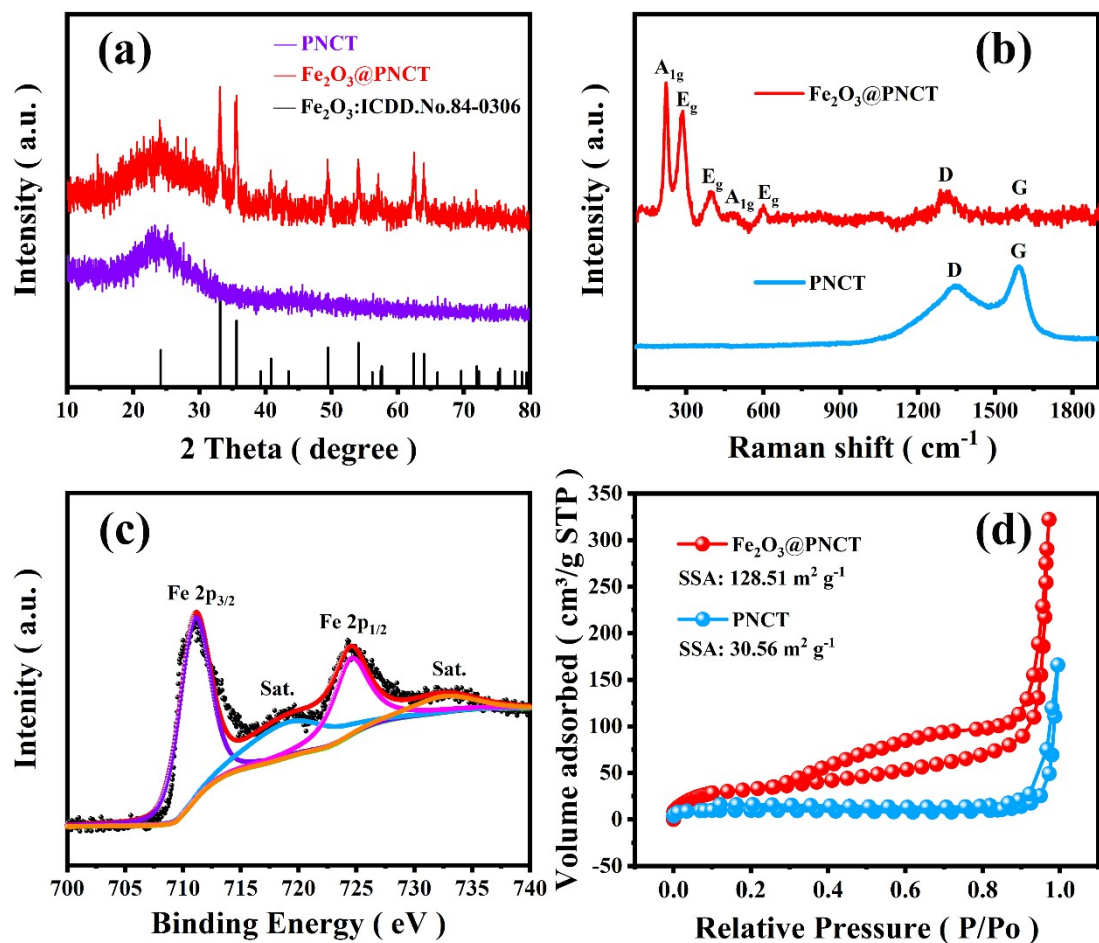


Figure S18: Structural characterization of $\text{Fe}_2\text{O}_3@\text{PNCT}$ samples: (a) XRD patterns, (b) Raman spectrum (powder samples tested at room temperature with a 532 nm laser and calibrated against the value of 520.7 cm^{-1} of a silicon wafer), (c) Fe 2p XPS spectrum, and (d) N_2 adsorption-desorption isotherms.

The phase composition of the product was characterized by XRD, as shown in **Figure S18(a)**. The XRD spectra of $\text{Fe}_2\text{O}_3@\text{PNCT}$ (red spectrum) and PNCT precursor (purple spectrum) both had two relatively broad diffraction peaks at 2θ of $\sim 23.9^\circ$ and $\sim 43.8^\circ$, which were assigned to the graphitic structure of sp^2 hybridization after pyrolytic carbonization. The strong diffraction peaks at the 2θ of $24.2, 33.2, 35.6, 40.8, 49.5, 54.1, 57.4, 62.4, 64.0, 71.9,$ and 75.5° matched

well with the characteristic diffraction peaks of the rhombohedral phase Fe_2O_3 , respectively (ICDD No. 84-0306) [49]. The Raman spectroscopy further characterized the sample structural information, as shown in **Figure S18(b)**. The spectral peaks at ~ 214.3 and 479.2 cm^{-1} correspond to Fe-O symmetric stretching vibrations (A_{1g} mode), while the peaks near ~ 277.6 , 391.2 , and 593.8 cm^{-1} correspond to Fe-O symmetric bending vibrations of Fe_2O_3 (E_g mode), and the peaks at ~ 1307.8 and 1608.3 cm^{-1} correspond to the D and G bands of carbon [S37]. In addition, the chemical composition and valence of the surface of $\text{Fe}_2\text{O}_3@$ PNCT samples were analyzed by XPS. The survey spectrum of the samples showed the presence of Fe, C, N and O (**Figure S19**). In the high-resolution Fe 2p spectrum, **Figure S18(c)**, the two peaks at ~ 724.5 and 711.2 eV correspond to the spin-orbit split electrons of Fe $2p_{1/2}$ and Fe $2p_{3/2}$ in Fe_2O_3 [8, S37]. Finally, nitrogen adsorption-desorption tests were conducted for PNCT and $\text{Fe}_2\text{O}_3@$ PNCT to characterize the pore structure and specific surface area, and the results were shown in **Figure S18(d)** and **Figure S20**. The $\text{Fe}_2\text{O}_3@$ PNCT had a high specific surface area of $128.51 \text{ m}^2 \text{ g}^{-1}$ and an average pore size distribution of 17.65 nm , which was much larger than that of PNCT at $30.56 \text{ m}^2 \text{ g}^{-1}$ and 8.47 nm . A large specific surface area of $\text{Fe}_2\text{O}_3@$ PNCT with a rational pore size distribution ensures a fast electron and ion transport, which will contribute to the electrochemical performance of the electrode.

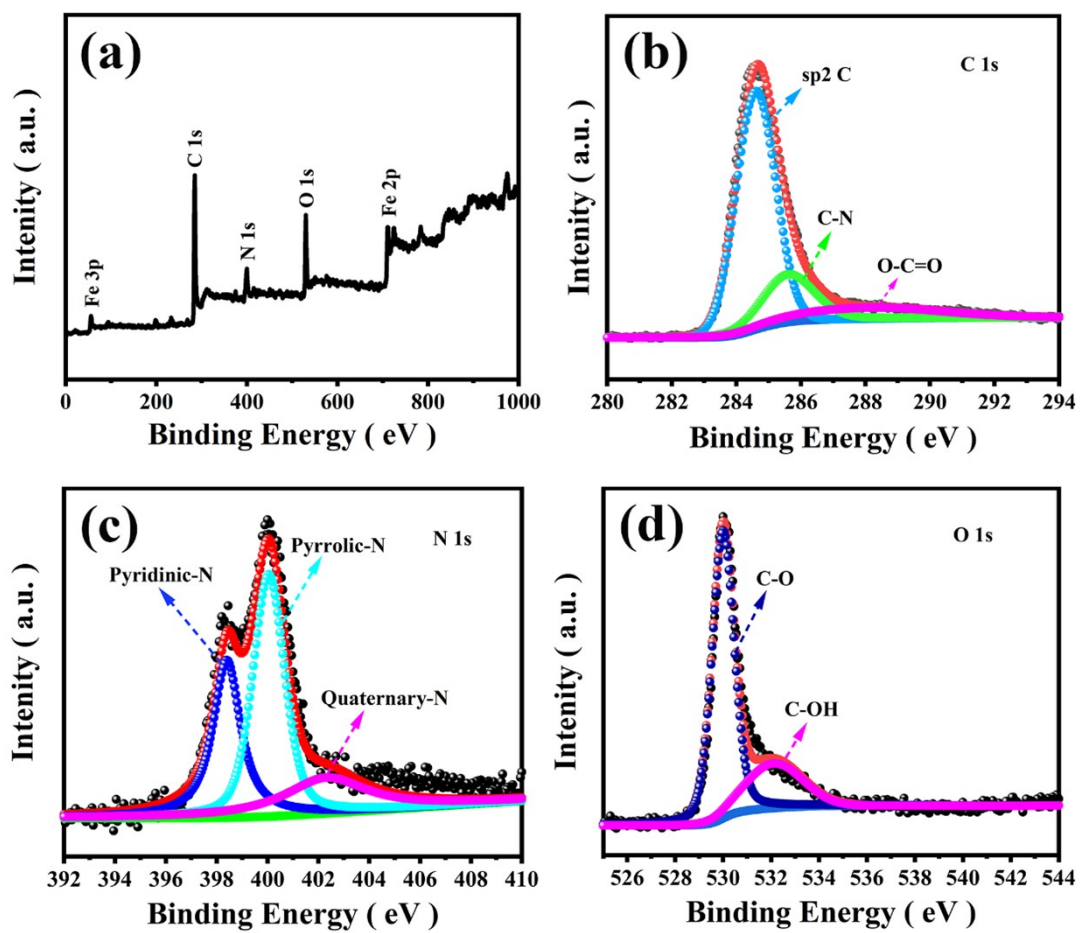


Figure S19: XPS spectrum of $\text{Fe}_2\text{O}_3@\text{PNCT}$: (a) XPS survey of $\text{Fe}_2\text{O}_3@\text{PNCT}$, (b) C 1s, (c) N 1s, and (d) O 1s.

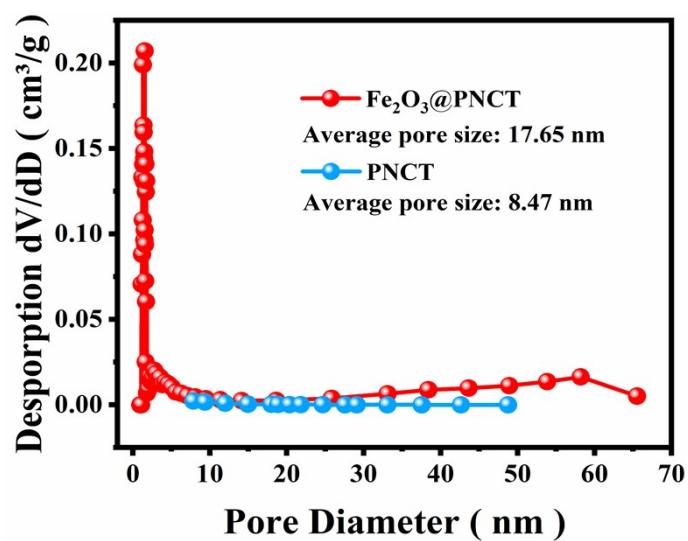


Figure S20: Pore-size distribution curve of Fe₂O₃@PNCT and PNCT samples obtained from the BJH method.

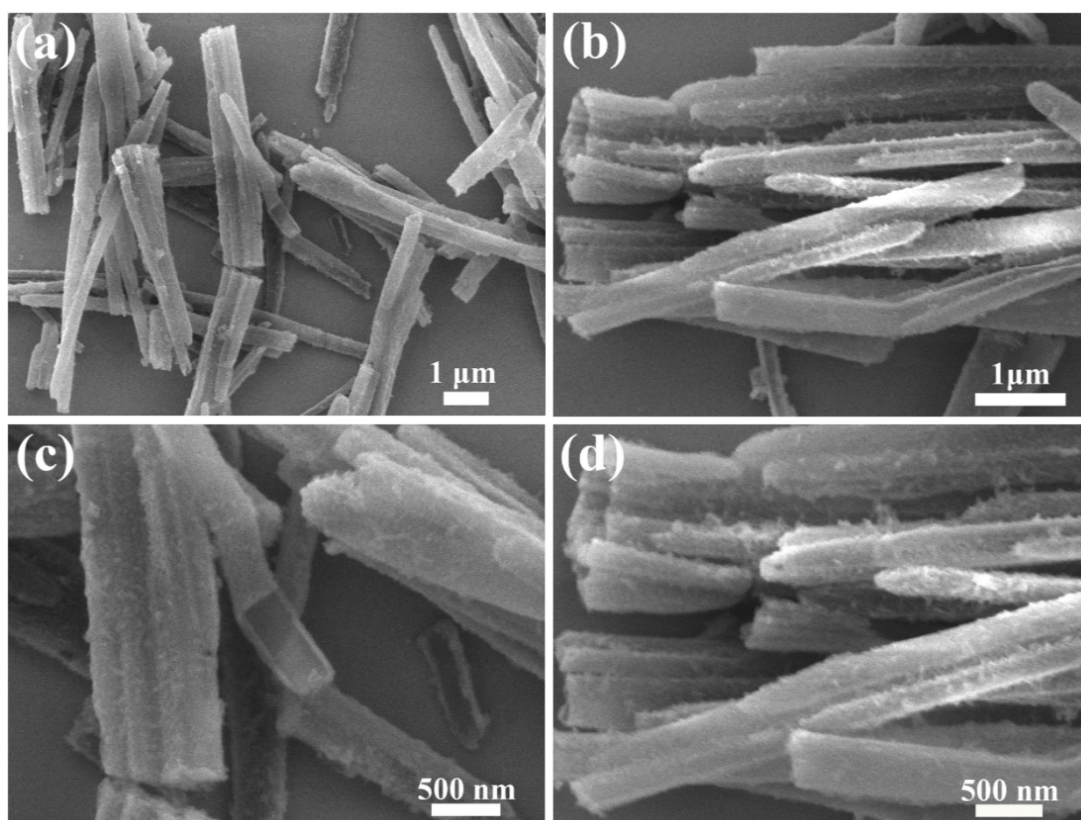


Figure S21: The morphology of PANI nanotubes obtained after the removal of MoO₃ template.

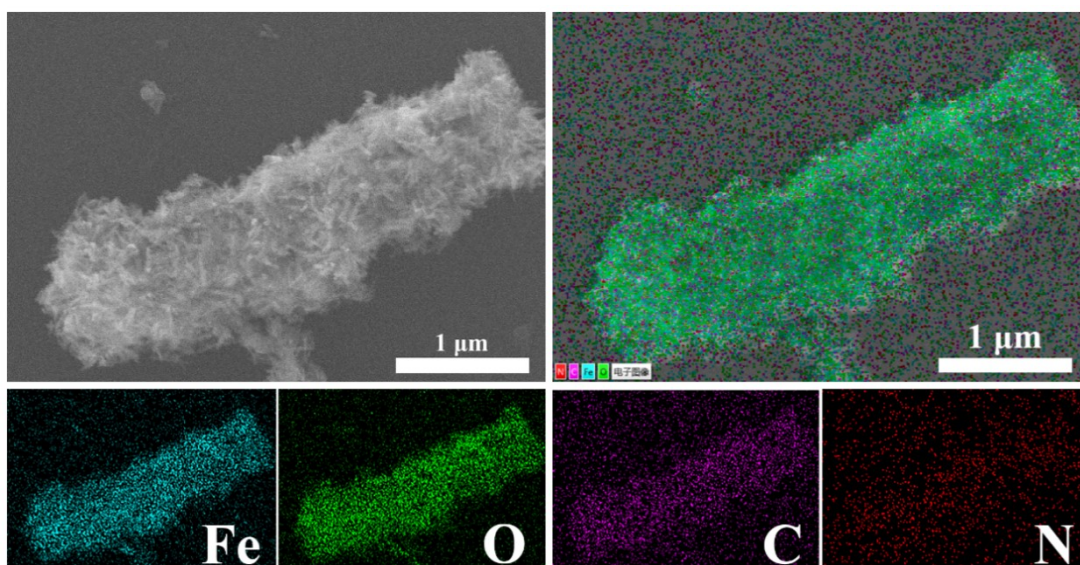


Figure S22: SEM of $\text{Fe}_2\text{O}_3@\text{PNCT}$ and the corresponding elemental mappings of different elements.

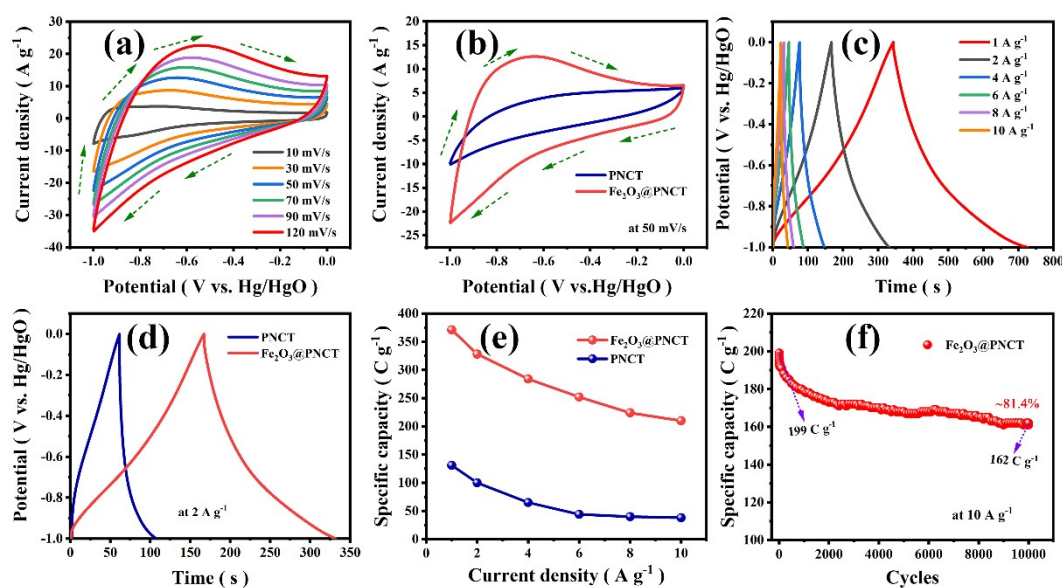


Figure S23: Electrochemical performances of negative electrode: (a) CV curves of the $\text{Fe}_2\text{O}_3@\text{PNCT}$ electrode at various scan rates, (b) comparative CV curves of $\text{Fe}_2\text{O}_3@\text{PNCT}$ at 50 mV s^{-1} , (c) GCD curves of $\text{Fe}_2\text{O}_3@\text{PNCT}$ at different current densities in the range of $1\text{--}10 \text{ A g}^{-1}$, (d) comparative GCD curves of PNCT and $\text{Fe}_2\text{O}_3@\text{PNCT}$ samples at 2 A g^{-1} , (e) GCD curves of PNCT and $\text{Fe}_2\text{O}_3@\text{PNCT}$ samples as a function of discharge current density, and (f) cycling stability of $\text{Fe}_2\text{O}_3@\text{PNCT}$ measured at a current density of 10 A g^{-1} .

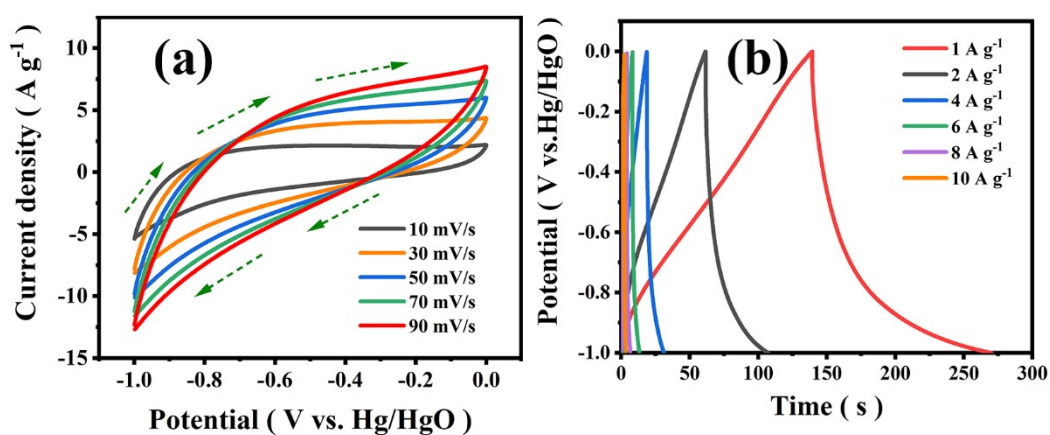


Figure S24: Electrochemical performances of the blank PNCT electrodes: (a) CV curves at various scan rates, and (b) GCD curves at different current densities in the range of 1-10 A g⁻¹.

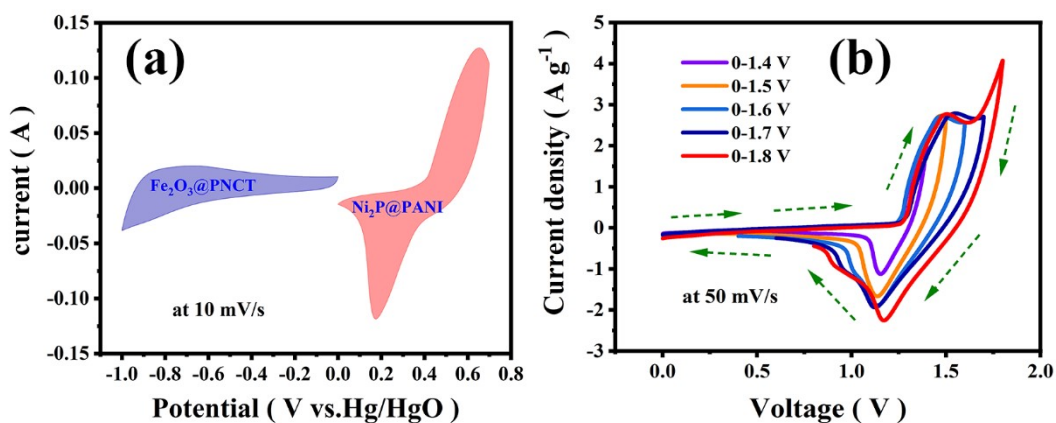


Figure S25: (a) CV curves of Ni₂P@PANI positive electrode and Fe₂O₃@PNCT negative electrode in a three-electrode system at 10 mV s⁻¹ after charge balancing, (b) CV curves of the device measured in different voltage ranges.

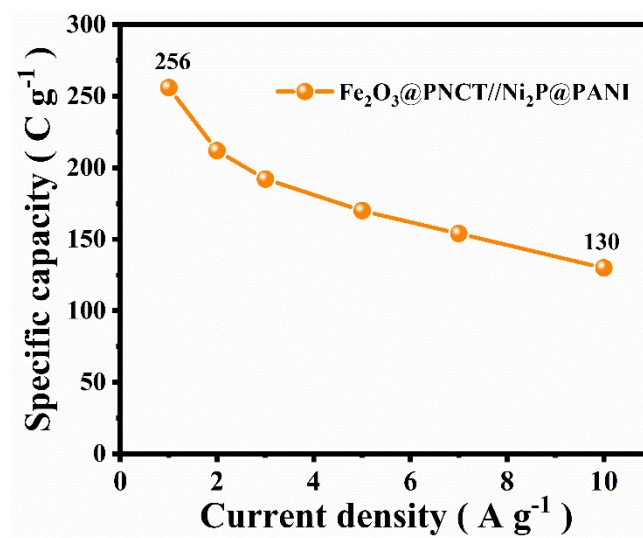


Figure S26: Rate capability of Fe₂O₃@PNCT//Ni₂P@PANI QSSC devices.

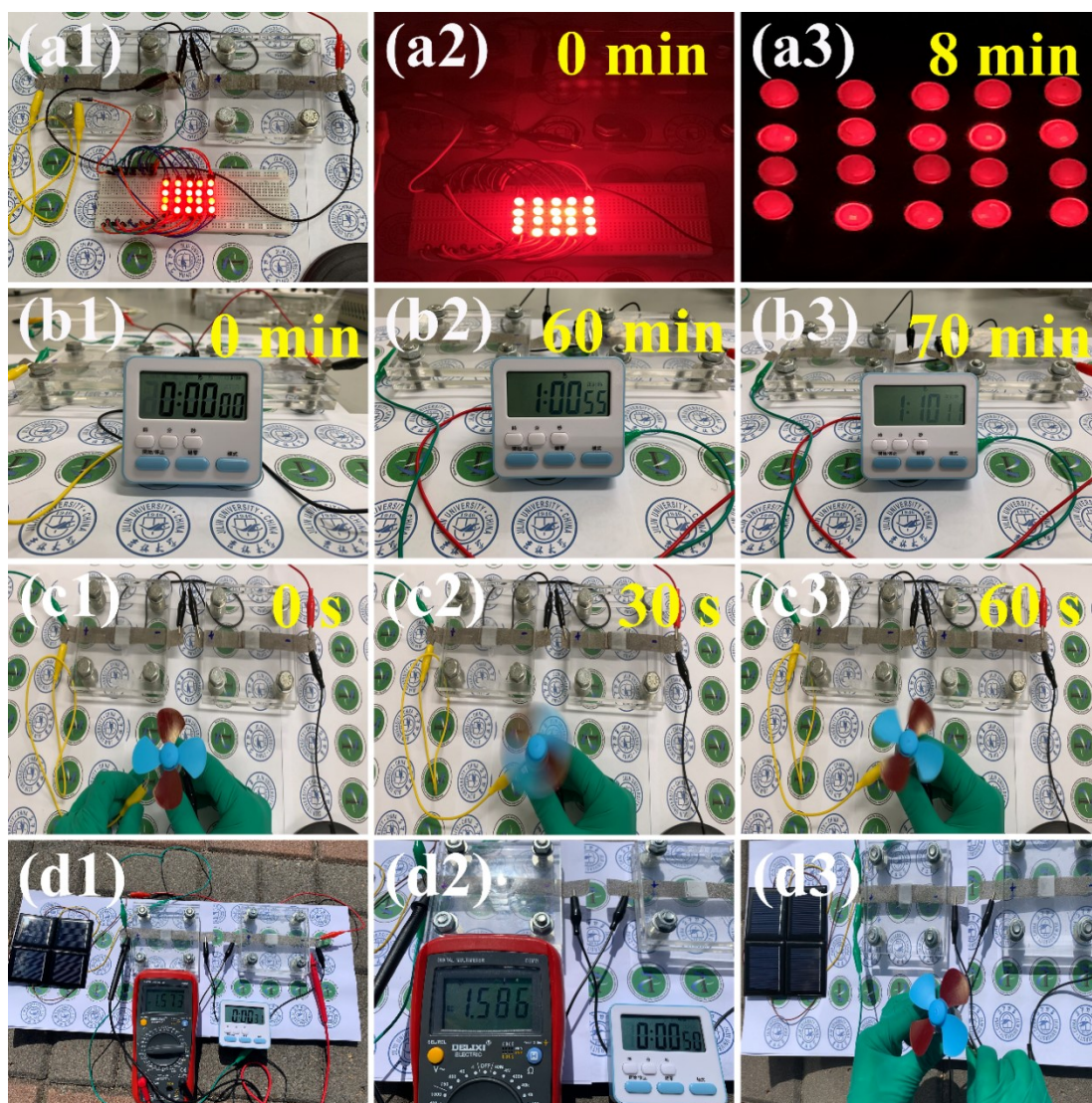


Figure S27: The digital photographs show the practical application of the device: (a1-a3) two devices in series could stably light a light source consisting of 20 red LEDs, (b1-b3) power the alarm clock, (c1-c3) drive a small fan, and (d1-d3) the energy storage and supply system built in combination with commercial solar panels.

Table S1: Comparisons of Ni₂P@PANI electrodes and QSSC devices performance with previous work.

Electrode	Capacity (Current density)	Cell (Config)	Cycles (devices)	Retention (devices)	ΔV (V)	ED (Wh kg ⁻¹)	Ref.
Ni₂P@PANI hollow nanotube	866 C g⁻¹ 2.0 A g⁻¹	2E Fe₂O₃@ PNCT	10,000	79.8%	1.70	60.60	This work
Ni ₂ P NS/NF	1398 C g ⁻¹ 2.5 A g ⁻¹	2E AC	5000	91.3%	1.60	26.00	S7
Ni-P	448 C g ⁻¹ 2.0 A g ⁻¹	2E AC	1000	84.5%	1.60	29.20	S8
Ni ₂ P/Ni/C	521 C g ⁻¹ 2.0 A g ⁻¹	2E AC	5000	99.0%	1.40	32.02	S9
Ni ₂ P or Ni ₅ P ₄	337 C g ⁻¹ 1.0 A g ⁻¹	2E Fe ₂ O ₃	1000	96.0%	1.60	35.50	S10
Co ₂ P nanorod	270 C g ⁻¹ 2.0 A g ⁻¹	2E Graphene	6000	97.0%	1.50	24.00	S11
Cu ₃ P nanorod	1400 C g ⁻¹ 2.0 A g ⁻¹	2E AC	15,000	65.0%	1.50	76.85	S12
Ni ₁₂ P ₅ nanowires	707 C g ⁻¹ 1.0 A g ⁻¹	2E AC	100	~55.0%	1.60	108.20	S13
NiO-GNS/PANI	744 C g ⁻¹ 2.0 A g ⁻¹	2E Sym.	1000	86.0%	0.60	---	S14
Co _x Ni _{1-x} P/CNF	1405 C g ⁻¹ 5.0 A g ⁻¹	2E AC	10,000	72.0%	1.40	32.20	S15
Ni _x P _y nanosheets	1272 C g ⁻¹ 2.0 A g ⁻¹	2E AC	5000	84.6%	1.50	67.20	S16
Ni-P@NiCo-LDH	1449 C g ⁻¹ 2.5 A g ⁻¹	2E AC	5000	70.0%	1.60	35.10	S17
CoP@NiCoP nanowire	956 C g ⁻¹ 2.0 A g ⁻¹	2E Fe ₂ O ₃	1000	73.3%	1.75	37.16	S18
2D nickel phosphate	176 C g ⁻¹ 2.0 A g ⁻¹	2E AC	5000	100.0%	1.45	50.00	S19
FeNiP@CoNi-LDH	975 C g ⁻¹ 2.0 A g ⁻¹	2E PC	20,000	73.9%	1.60	87.30	S20
NiP@CoAl-LDH	556 C g ⁻¹ 1.7 A g ⁻¹	2E AC	4000	95.5%	1.60	37.18	S21
NiCoP/ NiCo-OH	550 C g ⁻¹ 1.0 A g ⁻¹	2E PC	1000	92.0%	1.60	34.00	S22
Ni-PANI	315 C g ⁻¹ 2.0 A g ⁻¹	3E	5000	~75.0%	---	---	S23

NiCoP@NiCo-LDH	728 C g ⁻¹ 2.0 A g ⁻¹	2E AC	10,000	97.0%	1.70	57.00	S24
NiCoP@NF nanosheet	780 C g ⁻¹ 2.0 A g ⁻¹	2E AC	500	67.2%	1.20	27.00	S25

Note: **3E** represents three-electrode systems for single-electrode, **2E** represents two-electrode systems for devices, **AC** represents the activated carbon, **PC** represents the porous carbon, **Sym.** represents the symmetrical devices, and --- represents not mentioned. All values in **Table S1** were recalculated using the standard set of **equation (4)** (as reported in the experimental part) for normalization and more meaningful comparisons.

Table S2: Parameters of the proposed equivalent circuit model.

Electrode	R_s (Ω)	R_{ct} (Ω)	Z_w (Ω)
Ni ₂ P@PANI-1.2	0.78	7.85	7.18
Ni ₂ P@PANI-2.2	0.76	5.50	6.67
Ni ₂ P@PANI-3.2	2.00	16.05	12.09
Ni ₂ P	0.79	13.19	8.28

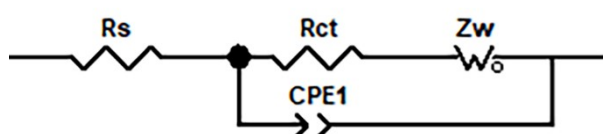


Figure S28: Equivalent Simulation Circuit.

Table S3: Comparisons of Fe₂O₃@PNCT electrodes performance with previous work.

Electrode	Capacity (Current density)	Electrolyte	Cycles (3E)	Retention (3E)	ΔV (V)	Ref.
Fe ₂ O ₃ @PNCT	371 C g ⁻¹ 1 A g ⁻¹	2 M KOH	10,000	81.4%	1.0	This work
FeOCl	278 C g ⁻¹ 0.8 A g ⁻¹	1 M Na ₂ SO ₄	10,000	75.0%	1.0	S26
g-C ₃ N ₄ /α-Fe ₂ O ₃	260 F g ⁻¹ 0.5 A g ⁻¹	2.5 M Li ₂ SO ₄	1000	100.0%	1.0	S27
SiC@Fe ₂ O ₃	721 F g ⁻¹ 2.0 A g ⁻¹	2 M KOH	---	---	1.2	S28
α-Fe ₂ O ₃ /C	345 F g ⁻¹ 1.8 A g ⁻¹	1 M Na ₂ SO ₄	4000	91.8%	1.0	S29
N doped graphene/Fe ₂ O ₃	260 F g ⁻¹ 2.0 A g ⁻¹	1 M Na ₂ SO ₄	1000	82.5%	1.2	S30
Fe ₂ O ₃ /N-CNT	264 F g ⁻¹ 1.0 A g ⁻¹	2 M KOH	10,000	84.0%	1.0	S31
α-Fe ₂ O ₃ /rGO	270 F g ⁻¹ 1.0 A g ⁻¹	1 M Na ₂ SO ₄	5000	123.0%	1.5	S32
Fe ₇ S ₈ /α-FeOOH	520 F g ⁻¹ 1.0 A g ⁻¹	1 M KOH	5000	90.6%	1.1	S33
PANI@Fe ₃ O ₄ @C F	245 F g ⁻¹ 0.5 A g ⁻¹	1 M H ₂ SO ₄	10,000	82.4%	0.65	S34
FeS ₂ nano- ellipsoids	515 C g ⁻¹ 1.0 A g ⁻¹	2 M KOH	5000	88%	1.1	S35
NS-Fe ₃ O ₄ @N-PC	866 F g ⁻¹ 1.0 A g ⁻¹	1 M KOH	5000	78.2%	0.4	S36

Note: 3E represents three-electrode systems for single-electrode and the values of ΔV in **Table S3** are all obtained from the discharge voltage range in the GCD curves in the cited reference.

Table S4: The mass loading of active material in three-electrode system for all samples.

Electrode	M_{total} (mg cm⁻²)	Percentage (%)	M_{active} (mg cm⁻²)	Average
Ni(OH) ₂ @PANI-0.5M	4.56		3.65	
	4.71	8:1:1	3.77	3.5 ± 0.5
	4.12		3.29	
Ni(OH) ₂ @PANI-1.0M	5.11		4.08	
	5.63	8:1:1	4.50	4.0 ± 0.5
	4.43		3.54	
Ni(OH) ₂ @PANI-1.5M	4.27		3.42	
	4.61	8:1:1	3.69	3.4 ± 0.5
	4.83		3.86	
Ni(OH) ₂	5.03		4.02	
	4.08	8:1:1	3.26	3.5 ± 0.5
	4.52		3.62	
Ni ₂ P@PANI-1.2	3.92		3.14	
	4.64	8:1:1	3.71	3.6 ± 0.5
	5.22		4.18	
Ni ₂ P@PANI-2.2	4.39		3.51	
	5.10	8:1:1	4.08	3.5 ± 0.5
	4.13		3.30	
Ni ₂ P@PANI-3.2	5.20		4.16	
	6.35	8:1:1	5.08	4.5 ± 0.5
	5.82		4.66	
Ni ₂ P	3.63		2.90	
	3.52	8:1:1	2.82	2.8 ± 0.5
	4.10		3.28	
Fe ₂ O ₃ @PNCT	4.44		3.55	
	5.03	8:1:1	4.02	4.0 ± 0.5
	5.64		4.51	
PNCT	4.66		3.72	
	4.25	8:1:1	3.40	3.5 ± 0.5
	4.90		3.92	

Note: **M_{total}** represents the total mass of the electrode material (2 cm × 2 cm), including the PTFE and acetylene black, **Percentage** represents the active material, acetylene

black, and PTFE in a mass ratio of 8:1:1, M_{active} represents the active material available on the actual electrode ($2 \text{ cm} \times 2 \text{ cm}$), and the **Average** represents the mass loading range of the active material on the electrode (Since there is no guarantee that the operation is exactly the same during each electrode preparation, the mass loading deviation of each sample cannot be avoided. As such, an average specific capacity is more appropriate in this paper).

Table S5: The average specific capacity of the electrode under optimal phosphatization conditions.

Electrode	M_{total} (mg cm^{-2})	Percentage (%)	M_{active} (mg cm^{-2})	Capacity (C g^{-1})	Average e (C g^{-1})
	4.39	8:1:1	3.51	892	
Ni₂P@PANI-2.2	5.10	8:1:1	4.08	753	~866
	4.13	8:1:1	3.30	952	

Note: The electrode test was performed at 2 A g^{-1} .

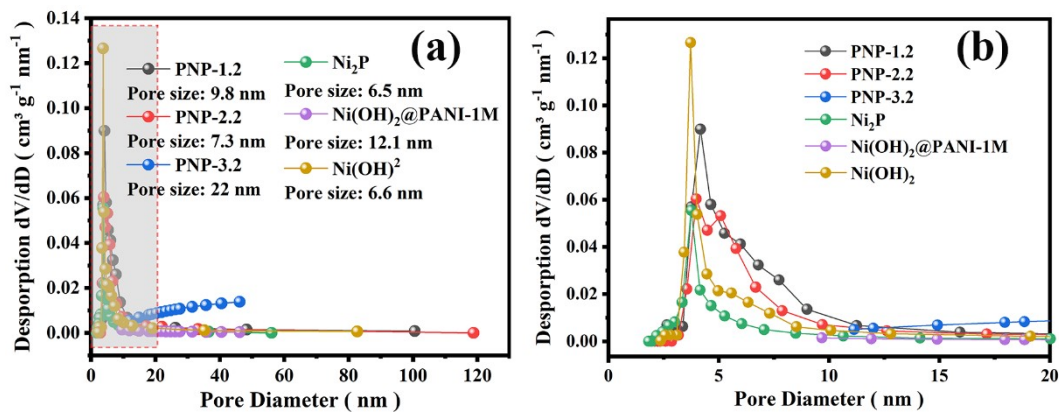


Figure S29: Pore-size distribution curve of PNP-x and precursor samples obtained from the BJH method (note that the data are average pore sizes).

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