

## Experimental Section

### Synthesis of La-VS<sub>2-x</sub>

3 mmol Na<sub>3</sub>VO<sub>4</sub>·12H<sub>2</sub>O and 15 mmol C<sub>2</sub>H<sub>5</sub>NS were dissolved in 40 mL of deionized water, and then 0.05 mM La<sub>2</sub>(CO<sub>3</sub>)<sub>3</sub>·8H<sub>2</sub>O was added under stirring for 10 min. Afterwards, a piece of carbon cloth (CC, 1 cm × 2 cm) was transferred to a Teflon-lined stainless-steel autoclave and kept at 170 °C for 24 h. The obtained La-VS<sub>2-x</sub> grown on CC was washed with deionized water and ethanol several times. For comparison, pristine VS<sub>2</sub> grown on CC was prepared by the same procedure without addition of La<sub>2</sub>(CO<sub>3</sub>)<sub>3</sub>·8H<sub>2</sub>O.

### Electrochemical measurements

Electrochemical measurements were carried out on a CHI-660E electrochemical workstation using a standard three-electrode system. The Ag/AgCl, graphite rod, and catalyst coated on CC (1 × 1 cm<sup>2</sup>) were used as reference electrode, counter electrode, and working electrode, respectively. All potentials were referenced to reversible hydrogen electrode (RHE) according to the equation:  $E$  (V vs. RHE) =  $E$  (V vs. Ag/AgCl) + 0.198 V + 0.059 × pH. The NO<sub>3</sub>RR measurements were carried out in Ar-saturated 0.5 M Na<sub>2</sub>SO<sub>4</sub> + 0.1 M KNO<sub>3</sub> electrolyte using an H-type two-compartment electrochemical cell separated by a Nafion 211 membrane. The Nafion membrane was pretreated by heating it in 5% H<sub>2</sub>O<sub>2</sub> aqueous solution at 80 °C for 1 h and then in deionized water at 80 °C for another 1 h. After each chronoamperometry test for 1 h, the produced NH<sub>3</sub> and other possible by-products (NO<sub>2</sub><sup>-</sup> and N<sub>2</sub>H<sub>4</sub>) were analyzed by various colorimetric methods using UV-vis absorbance spectrophotometer (MAPADA P5), while the gas products (H<sub>2</sub>, N<sub>2</sub>) were analyzed by gas chromatography (Shimadzu GC2010). The detailed determination procedures are given in our previous publication[1].

### Calculations of NH<sub>3</sub> yield rate and NH<sub>3</sub>-Faradaic efficiency

$$\text{NH}_3 \text{ yield rate } (\mu\text{g h}^{-1} \text{ mg}_{\text{cat}}^{-1}) = \frac{c_{\text{NH}_3} \times V}{t \times A} \quad (1)$$

$$\text{NH}_3\text{-Faradaic efficiency (\%)} = \frac{8 \times F \times c_{\text{NH}_3} \times V}{17 \times Q} \times 100\% \quad (2)$$

where  $c_{\text{NH}_3}$  ( $\mu\text{g mL}^{-1}$ ) is the measured  $\text{NH}_3$  concentration,  $V$  (mL) is the volume of the electrolyte,  $t$  (h) is the reduction time,  $A$  ( $\text{cm}^2$ ) is the surface area of CC ( $1 \times 1 \text{ cm}^2$ ),  $F$  ( $96500 \text{ C mol}^{-1}$ ) is the Faraday constant,  $Q$  (C) is the quantity of applied electricity.

### Characterizations

Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) were carried out on a Tecnai G<sup>2</sup> F20 microscope. X-ray diffraction (XRD) pattern was conducted on a Rigaku D/max 2400 diffractometer. Scanning electron microscopy (SEM) was carried out on a ZEISS GeminiSEM-500 microscope. Electron paramagnetic resonance (EPR) measurements were conducted on a Bruker ESP-300 spectrometer.

### Calculation details

Spin-polarized DFT calculations were performed using a Cambridge sequential total energy package (CASTEP). The exchange-correlation interactions were modeled by using Perdew–Burke–Ernzerhof (PBE) generalized gradient approximation (GGA) functional. The description of van der Waals interactions was based on a DFT-D correction method. During the geometry optimization, a cutoff energy of 450 eV was used and the convergence tolerance was set to be  $1.0 \times 10^{-5}$  eV for energy and 0.02 eV  $\text{\AA}^{-1}$  for force. The Brillouin zone was sampled by  $3 \times 3 \times 1$  Monkhorst–Pack k-point mesh. The  $\text{VS}_2$  (001) was modeled by a  $4 \times 4$  supercell, and a vacuum region of 15  $\text{\AA}$  was used to separate adjacent slabs. The adsorption energy ( $\Delta E$ ) is calculated as[2]

$$\Delta E = E_{\text{ads/slab}} - E_{\text{ads}} - E_{\text{slab}} \quad (3)$$

where  $E_{\text{ads/slab}}$ ,  $E_{\text{ads}}$  and  $E_{\text{slab}}$  are the total energies for adsorbed species on slab, adsorbed species and isolated slab, respectively.

The computational hydrogen electrode (CHE) model was adopted to calculate the Gibbs free energy change ( $\Delta G$ ) for each elementary step as follows:

$$\Delta G = \Delta E + \Delta \text{ZPE} - T\Delta S \quad (4)$$

where  $\Delta E$  is the adsorption energy,  $\Delta \text{ZPE}$  is the zero-point energy difference and  $T\Delta S$  is the entropy difference between the gas phase and adsorbed state. The entropies of

free gases were acquired from the NIST database. Given that it is difficult to directly calculate the energy of charged  $\text{NO}_3^-$ , the adsorption free energy of  $\text{NO}_3^-$  ( $\Delta G(*\text{NO}_3)$ ) was calculated with assistance of the gaseous  $\text{HNO}_3$  as follows[3]

$$\Delta G(*\text{NO}_3) = G(*\text{NO}_3) - G(*) - [G(\text{HNO}_3) - 0.5 \times G(\text{H}_2)] + \Delta G_{\text{correct}} \quad (5)$$

where  $G(*)$  and  $G(*\text{NO}_3)$  are the Gibbs free energies of the bare catalyst and that with the adsorbed  $\text{NO}_3^-$ , respectively.  $G(\text{HNO}_3)$  and  $G(\text{H}_2)$  represent the Gibbs free energies of  $\text{HNO}_3$  and  $\text{H}_2$  molecule, respectively.

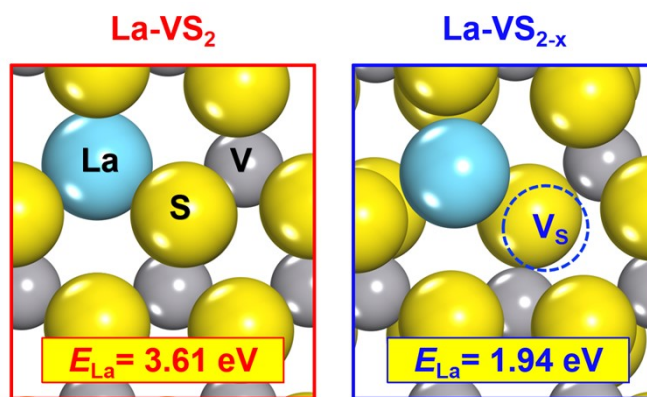


Fig. S1. La-dopant formation energy ( $E_{\text{La}}$ ) of  $\text{V}_\text{S}$ -free  $\text{La-VS}_2$  and  $\text{V}_\text{S}$ -containing  $\text{La-VS}_{2-x}$ .

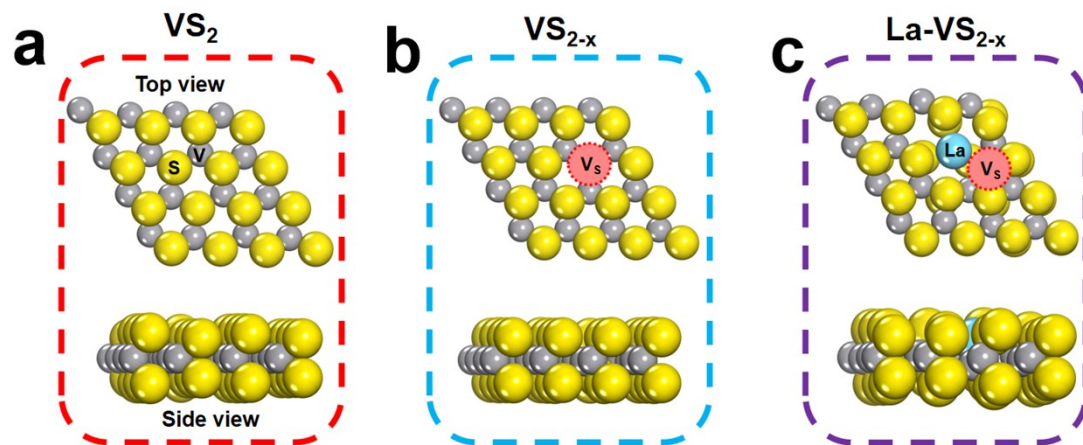


Fig. S2. Atomic structures of  $VS_2$ ,  $VS_{2-x}$  and  $La-VS_{2-x}$ .

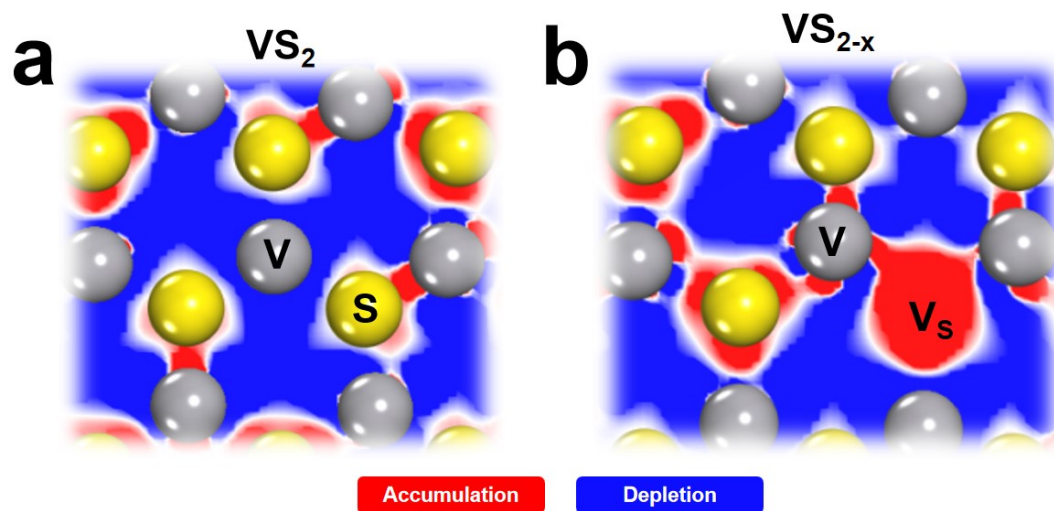


Fig. S3. Electron contour maps of (a)  $VS_2$  and (b)  $VS_{2-x}$ .

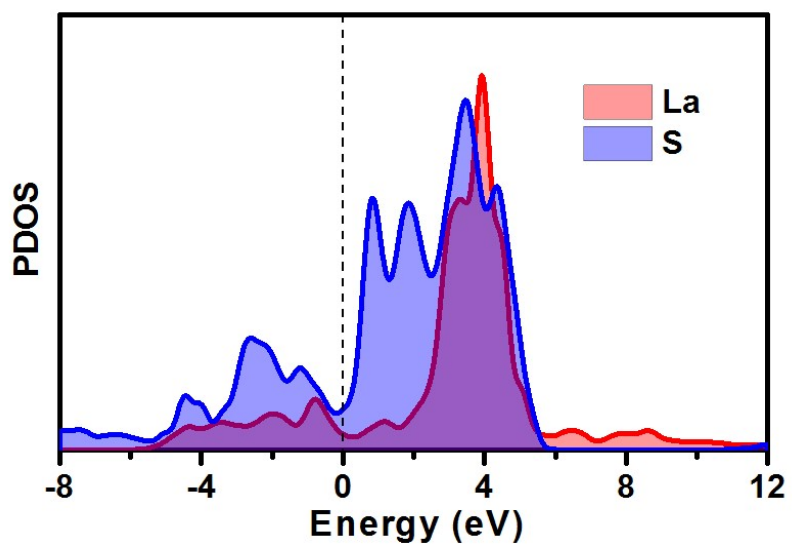


Fig. S4. PDOS plot of La/S orbital hybridization in La-VS<sub>2-x</sub>.

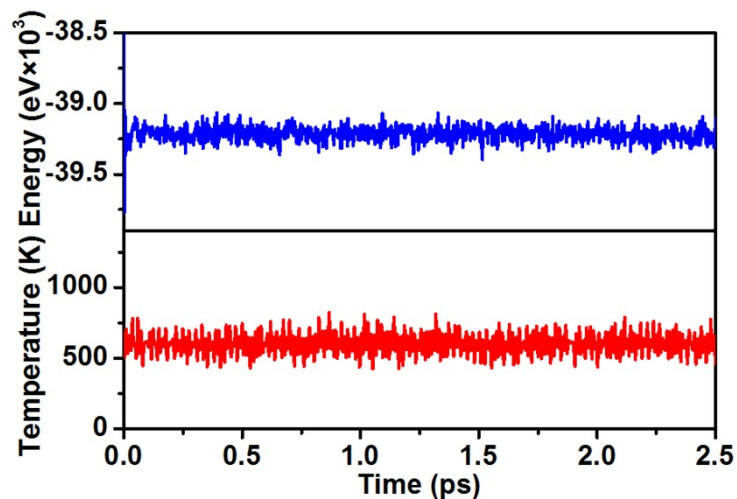


Fig. S5. Variations of energy and temperature during the AIMD simulation for assessing the thermodynamic stability of La- $\text{VS}_{2-x}$  (inset: geometric structure of La- $\text{VS}_{2-x}$  after AIMD simulation).

It is seen that both energy and temperature keep the equilibrium states at 600 K, confirming the high thermodynamic stability of La- $\text{VS}_{2-x}$ .



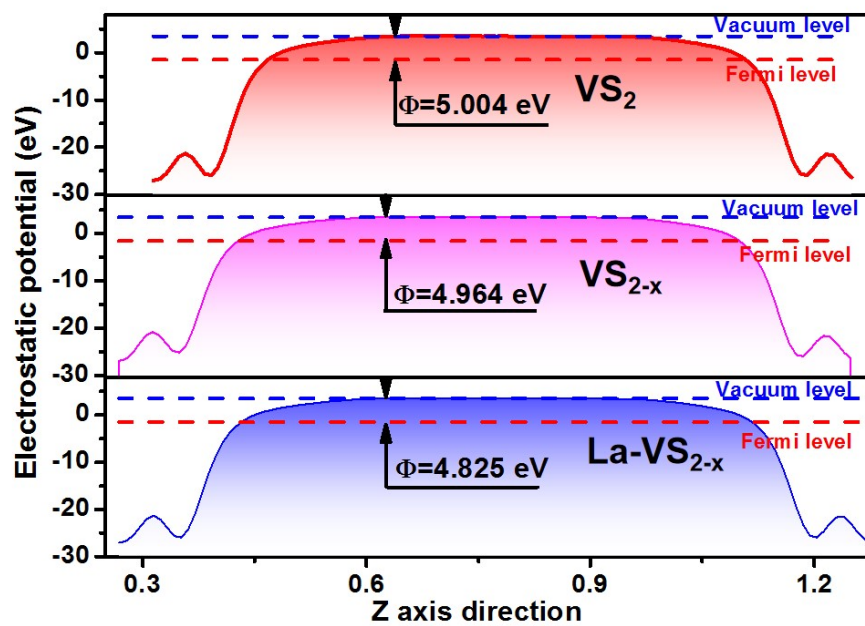


Fig. S6. Average potential profiles along c-axis direction for calculating the work functions ( $\Phi$ ) of VS<sub>2</sub>, VS<sub>2-x</sub> and La-VS<sub>2-x</sub>.

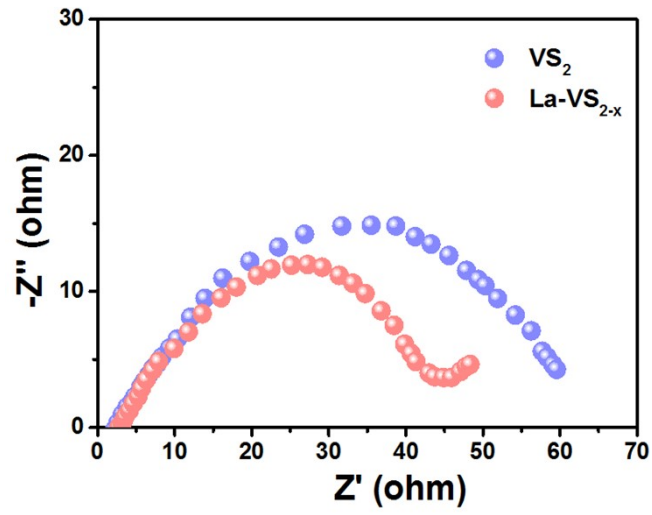


Fig. S7. Electrochemical impedance spectra of  $\text{VS}_2$  and  $\text{La-VS}_{2-x}$ .

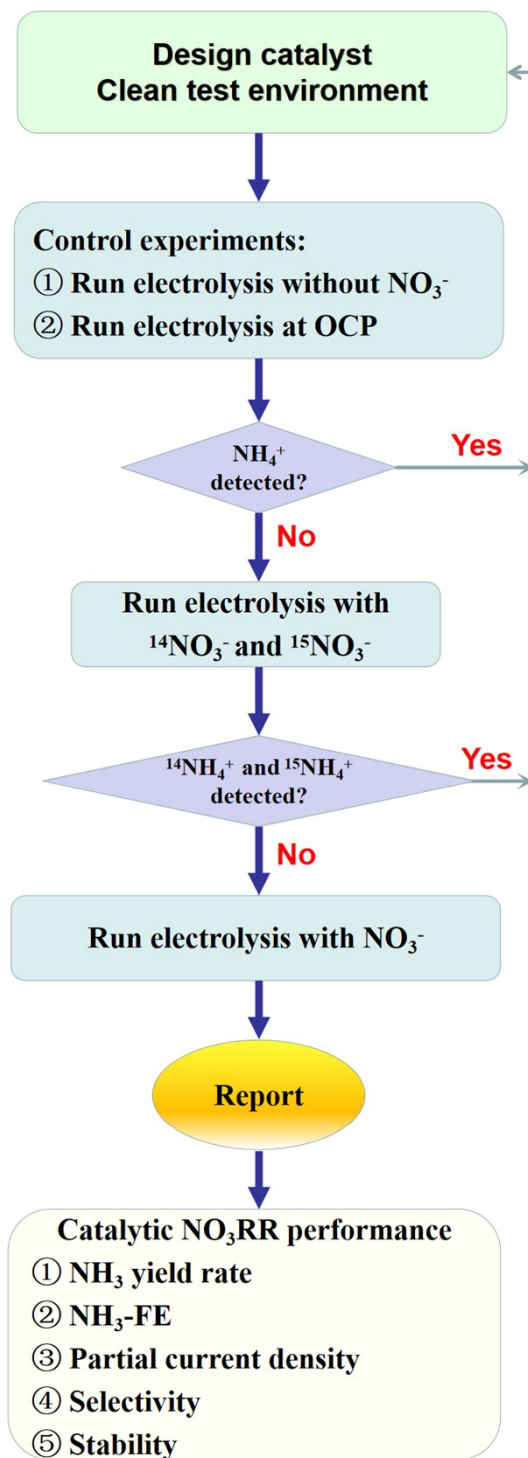


Fig. S8. Flow chart of the electrocatalytic NO<sub>3</sub>RR measurement procedure.

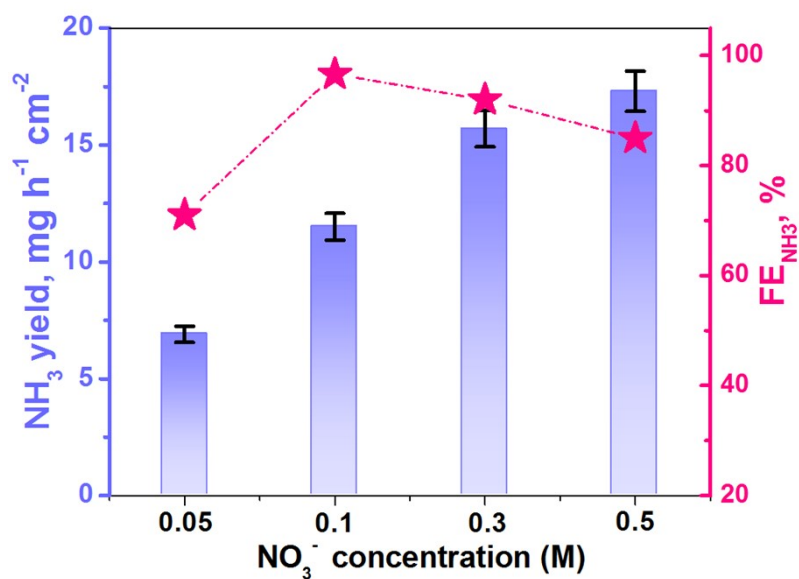


Fig. S9. NH<sub>3</sub> yield rates and FE<sub>NH<sub>3</sub></sub> of La-VS<sub>2-x</sub> at different nitrate concentrations.

We investigate the influence of starting nitrate concentration on the NO<sub>3</sub>RR performance of La-VS<sub>2-x</sub>. It is seen in Fig. S9 that NH<sub>3</sub> yield rate is monotonously increased with increasing nitrate concentration, while FE<sub>NH<sub>3</sub></sub> peaks at 0.1 M. As FE<sub>NH<sub>3</sub></sub> is commonly regarded as a more important indicator than NH<sub>3</sub> yield rate for NH<sub>3</sub> electrosynthesis, we take 0.1 M as the optimum nitrate concentration for our NO<sub>3</sub>RR measurements. The reduced FE<sub>NH<sub>3</sub></sub> beyond 0.1 M can be due presumably to the adverse effects of expedited competing HER at lower nitrate concentration (0.05 M) and plentiful NO<sub>3</sub><sup>-</sup> occupying the active sites at higher nitrate concentration (0.3~0.5 M)[4].

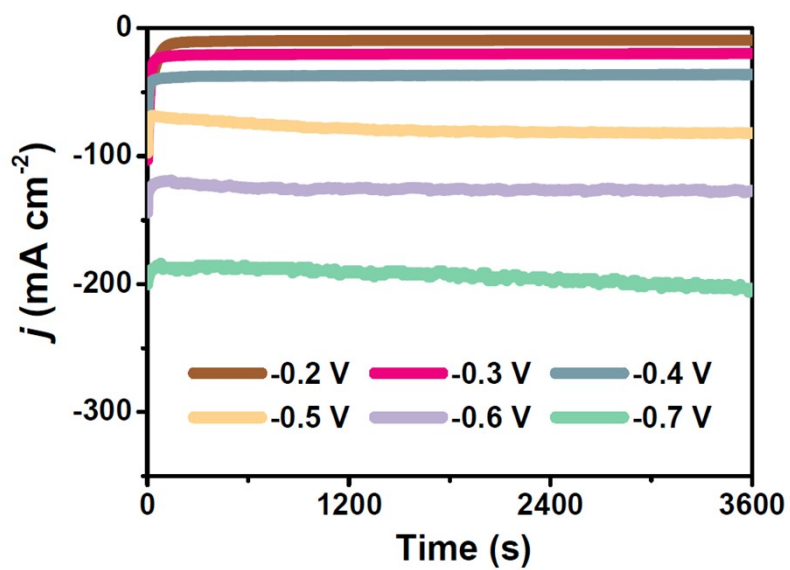


Fig. S10. Potential-dependent chronoamperometry curves of La- $\text{VS}_{2-x}$  after 1 h of  $\text{NO}_3\text{RR}$  electrolysis.

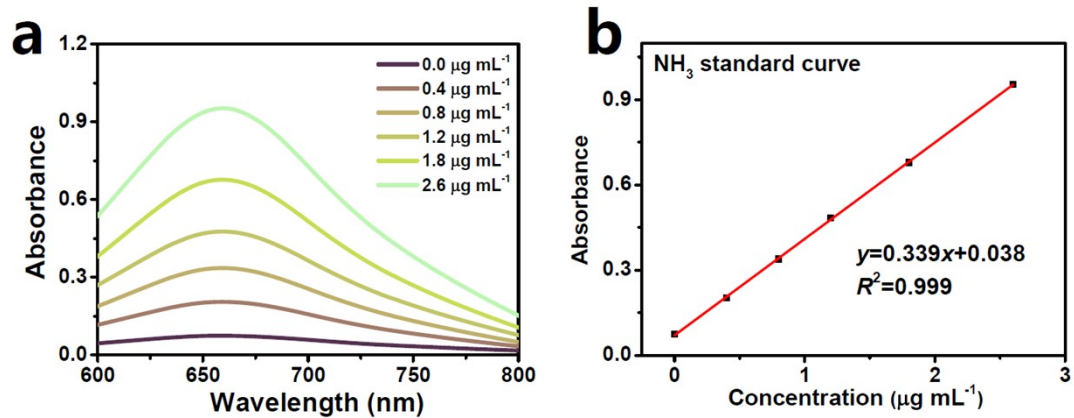


Fig. S11. (a) UV-vis absorption spectra of  $\text{NH}_4\text{Cl}$  assays after incubated for 2 h at ambient conditions. (b) Calibration curve used for the calculation of  $\text{NH}_3$  concentrations.

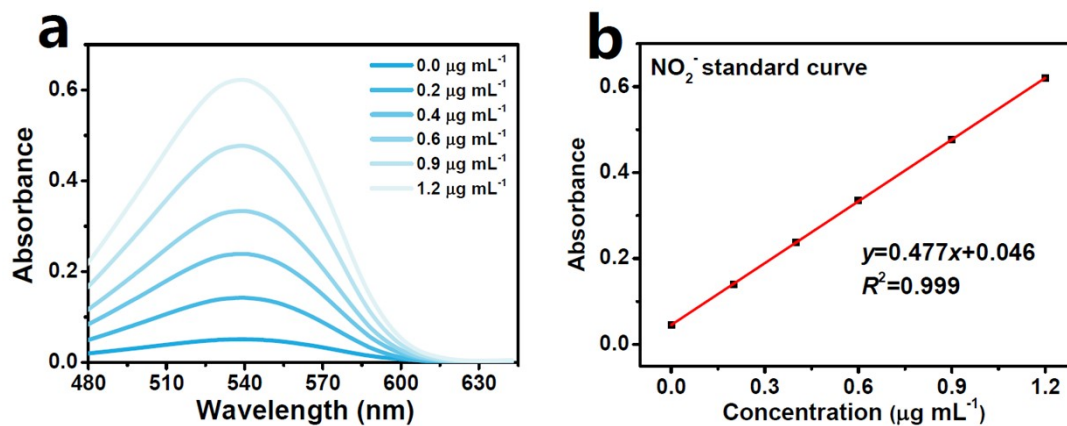


Fig. S12. (a) UV-vis absorption spectra of  $\text{KNO}_2$  assays after incubated for 20 min at ambient conditions. (b) Calibration curve used for calculation of  $\text{NO}_2^-$  concentrations.

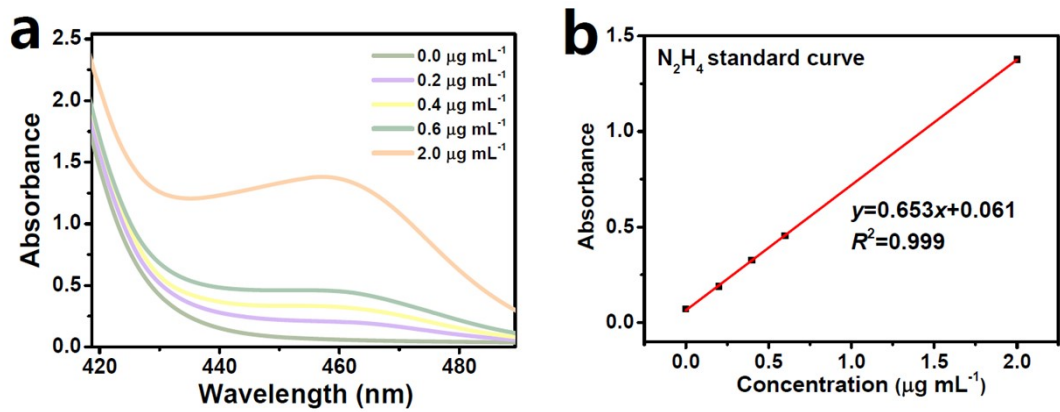


Fig. S13. (a) UV-vis absorption spectra of  $N_2H_4$  assays after incubated for 20 min at ambient conditions. (b) Calibration curve used for calculation of  $N_2H_4$  concentrations.



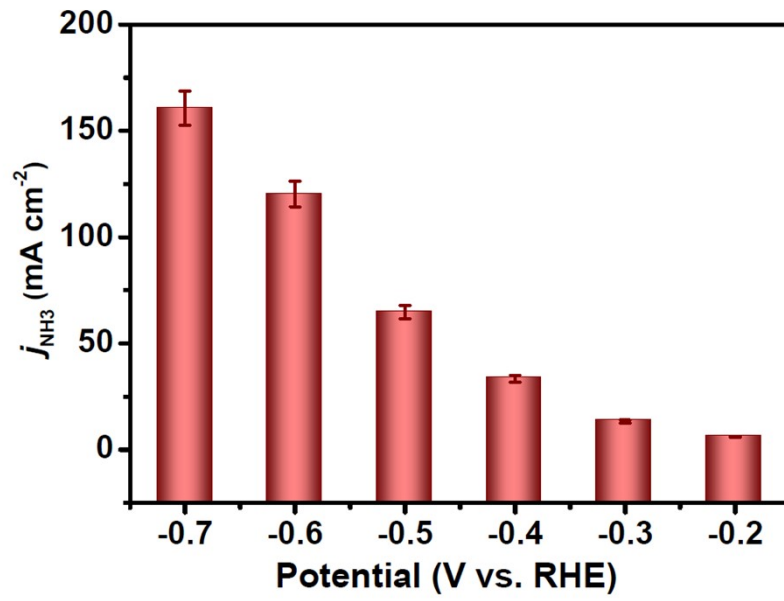


Fig. S14. Calculated NH<sub>3</sub> partial current densities of La-VS<sub>2-x</sub> at various potentials.

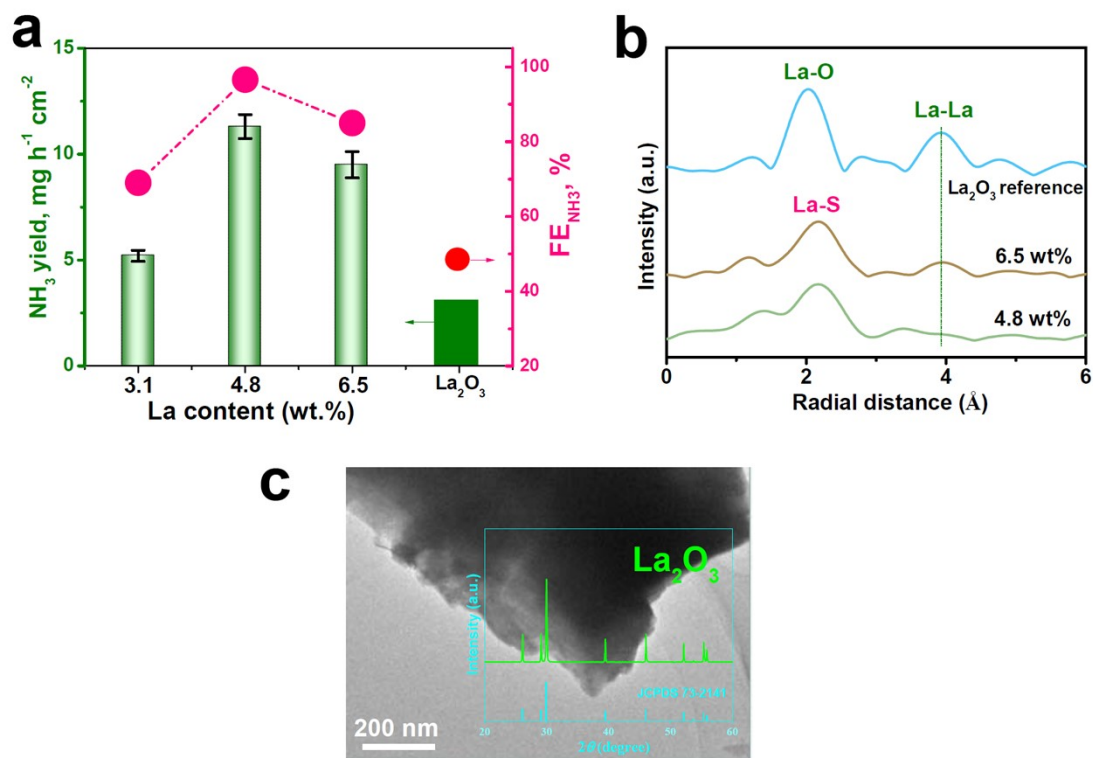


Fig. S15. (a)  $\text{NO}_3\text{RR}$  performances of  $\text{La-VS}_{2-x}$  with different La contents (3.1-6.5 wt%) at -0.6 V. (b) EXAFS spectra of 4.8 and 6.5 wt%  $\text{La-VS}_{2-x}$  and  $\text{La}_2\text{O}_3$  reference. (c) TEM image and XRD pattern (inset) of  $\text{La}_2\text{O}_3$  nanoplates, which was prepared by a reported method[5].

It is seen in Fig. S15a that the  $\text{NO}_3\text{RR}$  performance of  $\text{La-VS}_{2-x}$  peaks at 4.8 wt%, whereas the lower (3.1 wt%) and higher (6.5 wt%) La contents lead to the distinct performance degradation. The poor performance of 3.1 wt%  $\text{La-VS}_{2-x}$  can be explained by the insufficient single-atom La active sites, while the reduced performance of 6.5 wt%  $\text{La-VS}_{2-x}$  can be attributed possibly to the formation of  $\text{La}_2\text{O}_3$  species due to the excessive La addition, as evidenced by the EXAFS spectra (Fig. S15b) where 6.5 wt%  $\text{La-VS}_{2-x}$  shows the characteristic La-La bond of  $\text{La}_2\text{O}_3$  reference. To verify this, we prepare  $\text{La}_2\text{O}_3$  nanoplates (Fig. S15c) which exhibits a very poor  $\text{NO}_3\text{RR}$  property (Fig. S15a). Hence, the presence of  $\text{NO}_3\text{RR}$ -unfavored  $\text{La}_2\text{O}_3$  species on 6.5 wt%  $\text{La-VS}_{2-x}$  is the main cause for its decreased performance. Therefore, the abundant atomically dispersed La-dopants involved in 4.8 wt%  $\text{La-VS}_{2-x}$  contribute to the highest  $\text{NO}_3\text{RR}$  performance.

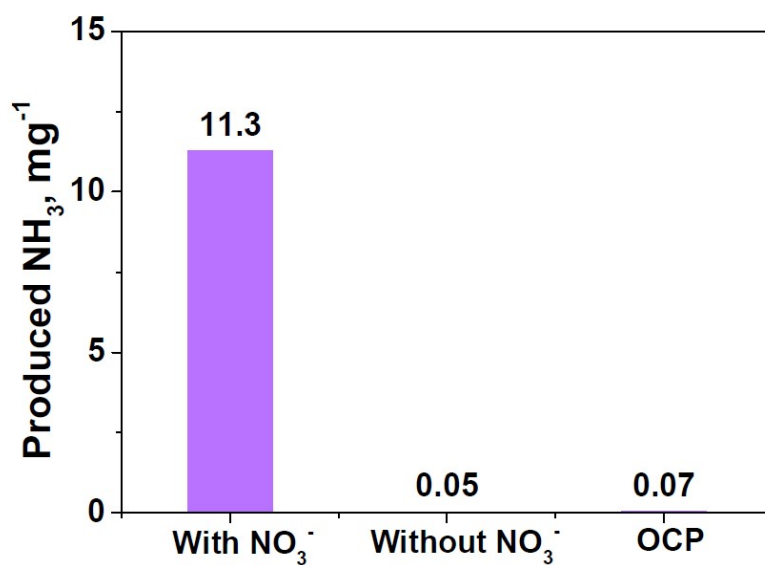


Fig. S16. Amounts of produced NH<sub>3</sub> under different conditions: electrolysis in NO<sub>3</sub><sup>-</sup>-containing electrolyte at -0.6 V; electrolysis in NO<sub>3</sub><sup>-</sup>-free electrolyte at -0.6; electrolysis in NO<sub>3</sub><sup>-</sup>-containing electrolyte at open-circuit potential (OCP).

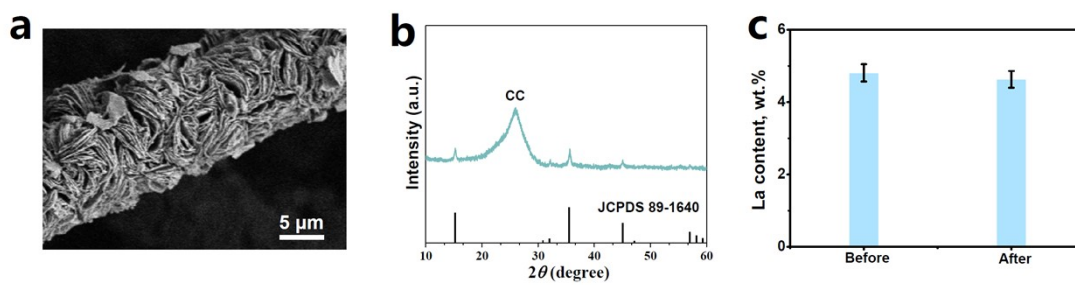


Fig. S17. (a) SEM image, (b) XRD pattern and (c) La content of La- $\text{VS}_{2-x}$  after stability tests.

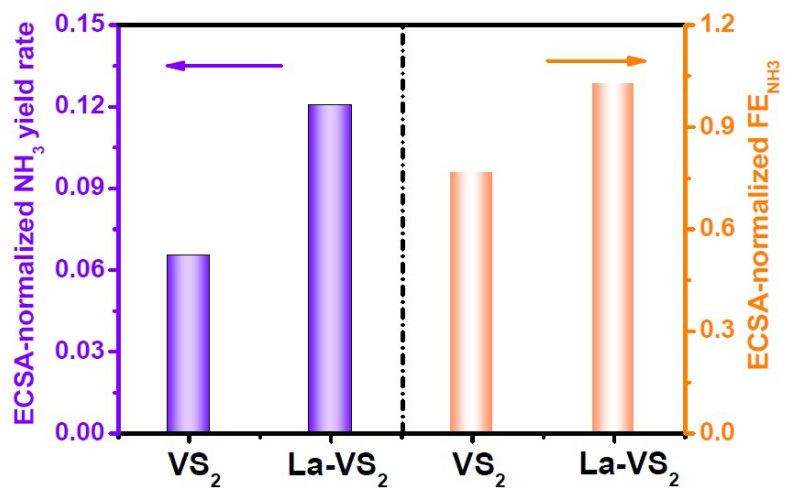


Fig. S18. ECSA-normalized NH<sub>3</sub> yield rates and FE<sub>NH3</sub> of VS<sub>2</sub> and La-VS<sub>2-x</sub> at -0.6 V.

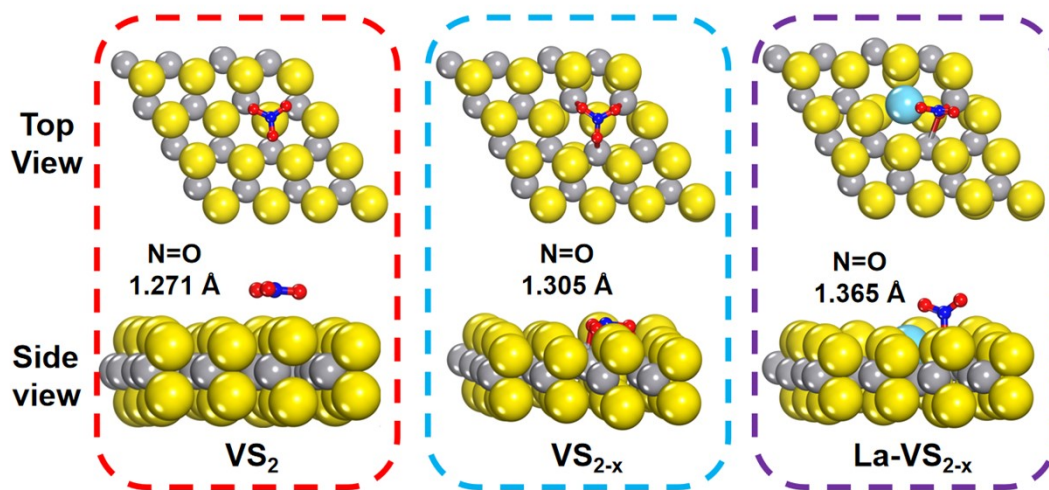


Fig. S19. Optimized atomic configurations of  $\text{NO}_3^-$  adsorption on  $\text{VS}_2$ ,  $\text{VS}_{2-x}$  and  $\text{La-VS}_{2-x}$ .

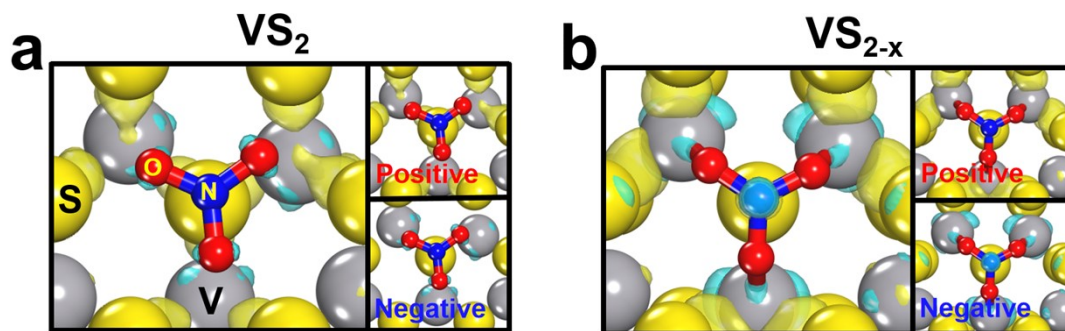


Fig. S20. Differential charge density maps of  $\text{NO}_3^-$  adsorption on (a)  $\text{VS}_2$  and (b)  $\text{VS}_{2-x}$  (yellow: accumulation, cyan: depletion).

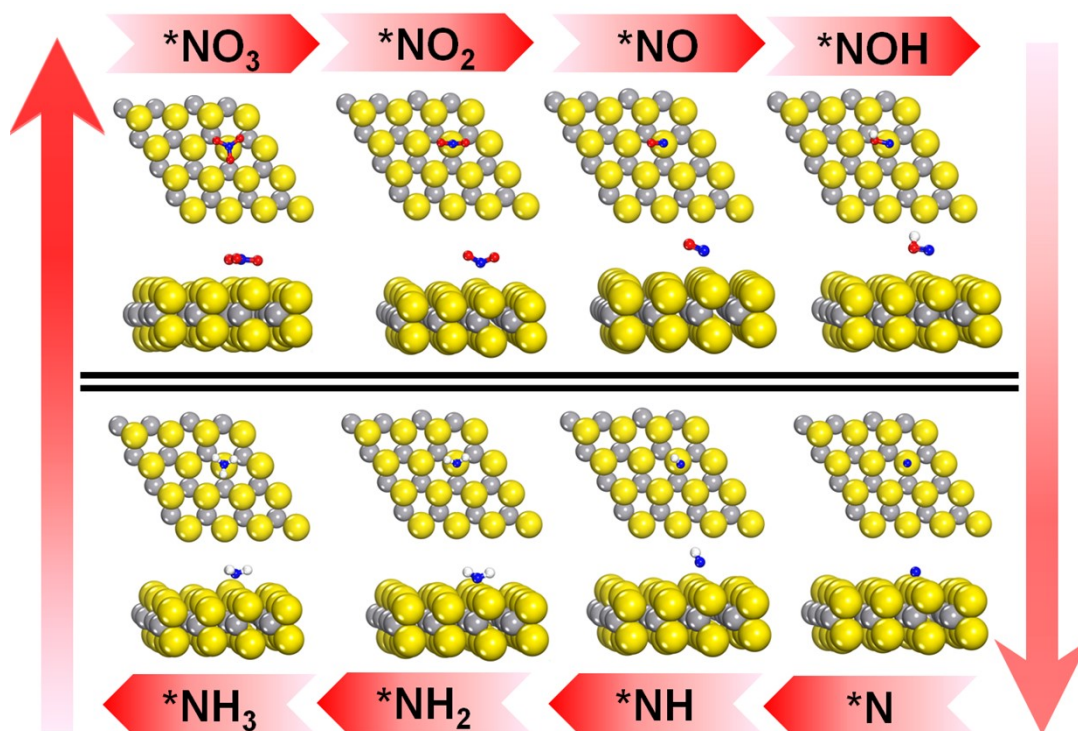


Fig. S21. Optimized atomic configurations of NO<sub>3</sub>RR intermediates on VS<sub>2</sub>.



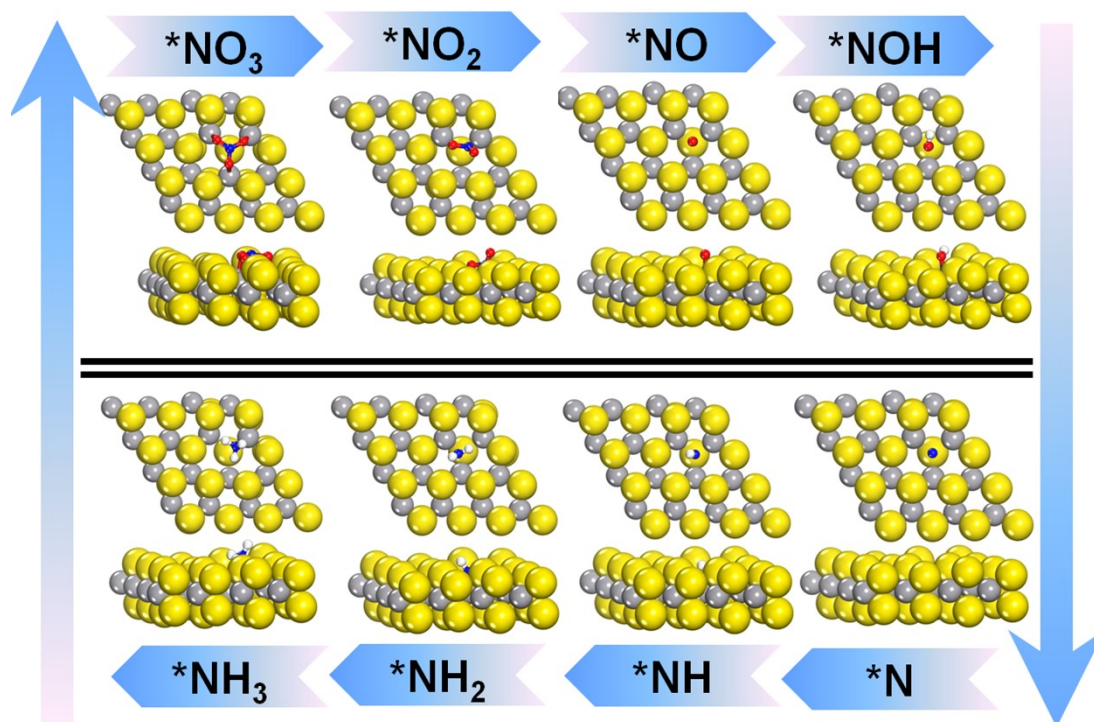


Fig. S22. Optimized atomic configurations of NO<sub>3</sub>RR intermediates on VS<sub>2-x</sub>.

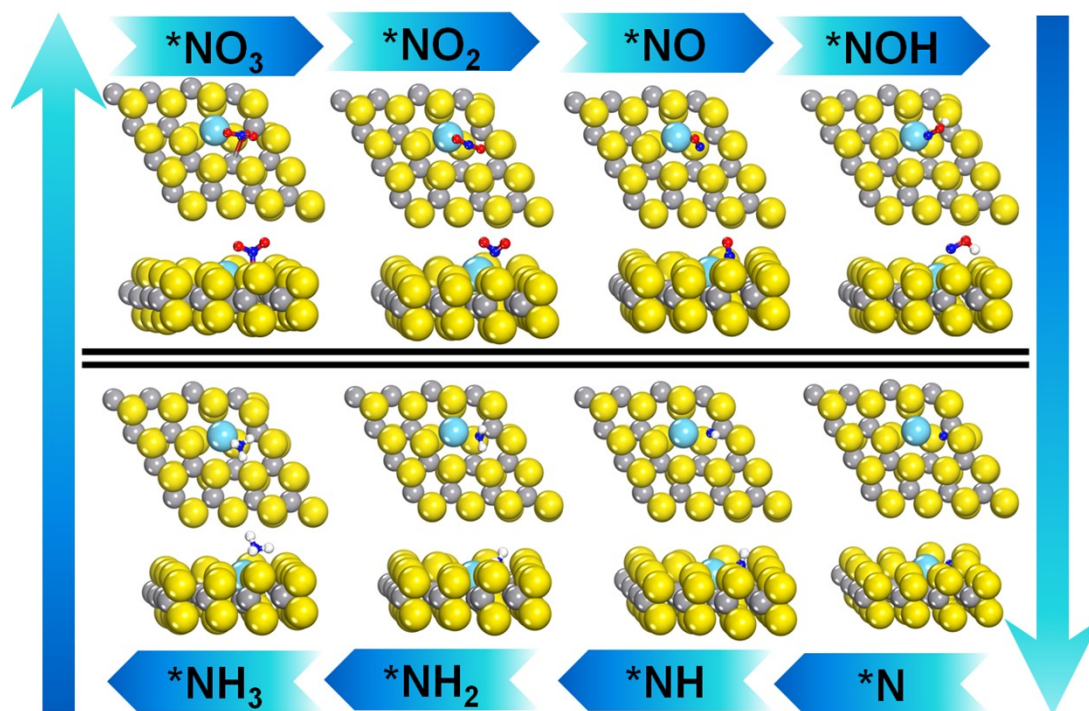


Fig. S23. Optimized atomic configurations of NO<sub>3</sub>RR intermediates on La-VS<sub>2-x</sub>.

Table S1. Structural parameters extracted from the La L<sub>3</sub>-edge EXAFS fitting.

Sample	Shell	CN	R (Å)	$\sigma^2$ (10 <sup>-3</sup> Å)	$\Delta E_0$ (eV)	R factor (%)
La-VS <sub>2-x</sub>	La-S	5.2	2.47	8.5	-3.7	0.012

CN is the coordination number; R is interatomic distance;  $\sigma^2$  is Debye-Waller factor;  $\Delta E_0$  is edge-energy shift; R factor is used to value the goodness of the fitting.

Table S2. Comparison of the optimum NH<sub>3</sub> yield and FE<sub>NH<sub>3</sub></sub> for the recently reported state-of-the-art NO<sub>3</sub>RR electrocatalysts at ambient conditions.

Catalyst	Electrolyte	NH <sub>3</sub> yield rate (mg h <sup>-1</sup> cm <sup>-2</sup> )	FE <sub>NH<sub>3</sub></sub> (%)	Potential (V vs RHE)	Ref.
Fe <sub>3</sub> O <sub>4</sub> /SS	0.1 M NaOH (0.1 M NaNO <sub>3</sub> )	10.15	91.5	-0.5	[6]
Co <sub>2</sub> AlO <sub>4</sub> /CC	0.1 M PBS (0.1 M NO <sub>3</sub> <sup>-</sup> )	7.9	92.6	-0.7	[7]
ZnCo <sub>2</sub> O <sub>4</sub> NSA/CC	0.1 M NaOH (0.1 M NaNO <sub>3</sub> )	10.79	98.33	-0.6	[8]
PdCoO/NF	0.5 M K <sub>2</sub> SO <sub>4</sub> (200 mg/LNO <sub>3</sub> <sup>-</sup> )	3.47	88.6	-1.3	[9]
Poly-Cu <sub>14</sub> cba	0.5 M K <sub>2</sub> SO <sub>4</sub> (250 ppm NO <sub>3</sub> <sup>-</sup> )	2.84	90	-0.15	[10]
Pd nanocrystalline	0.1 M Na <sub>2</sub> SO <sub>4</sub> (0.1 M NO <sub>3</sub> <sup>-</sup> )	9.32	79.91	-0.7	[11]
a-RuO <sub>2</sub>	0.5 M Na <sub>2</sub> SO <sub>4</sub> (200 ppm NO <sub>3</sub> <sup>-</sup> )	1.97	97.46	-0.35	[12]
Rh@Cu	0.1 M Na <sub>2</sub> SO <sub>4</sub> 0.1 M KNO <sub>3</sub>	13.57	93	-0.2	[13]
Cu-PTCDA	1 M PBS (500 ppm KNO <sub>3</sub> )	0.44	85.9	-0.4	[14]
CoO@NCNT /GP	0.1 M NaOH 0.1 M NaNO <sub>3</sub>	9.04	93.8	-0.6	[15]
TiO <sub>2</sub> NTs/CuO <sub>x</sub>	0.5 M Na <sub>2</sub> SO <sub>4</sub> 100 ppm KNO <sub>3</sub>	1.24	92.23	-0.75	[16]
Co-P/TP	0.2 M Na <sub>2</sub> SO <sub>4</sub> (200 ppm NO <sub>3</sub> <sup>-</sup> )	0.42	93.6 ± 3.3	-0.3	[17]
Cu <sub>3</sub> P NA/CF	0.1 M PBS (0.1 M NaNO <sub>2</sub> )	1.63	91.2 ± 2.5	-0.5	[18]
La-VS <sub>2-x</sub>	0.1 M Na <sub>2</sub> SO <sub>4</sub> 0.1 M NaNO <sub>3</sub>	11.3	96.6	-0.6	This work

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