Supporting Information

Dynamics of a two-dimensional active polymer chain with a rotation-restricted active head

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1. Orientational autocorrelation function

The orientational autocorrelation function C(t) of the orientation of self-propulsion force, $\mathbf{n}(t)$, is defined as $C(t) = \langle \mathbf{n}(t_0) \cdot \mathbf{n}(t + t_0) \rangle$. The evolution of C(t) with time t is presented in Figure S1. We find that C(t) roughly decays exponentially as $C(t) = e^{-t/\tau_r}$. Here the persistence time τ_r is estimated mostly from the decay of C(t) in the region 0.05 $\langle C(t) \langle 0.5 \rangle$ where C(t) shows the best exponential decay.



Fig. S1. Semi-logarithm plot of the evolution of the orientational autocorrelation function of the self-propulsion force, C(t), for the active polymer with a rotation-restricted head ABP. Polymer length N = 64 and self-propulsion force $f_s = 30$.

2. Elongation of the active polymer chain with a rotation-restricted head ABP

Figure S2 presents the dependence of the mean square radius of gyration $\langle R_G^2 \rangle$ on the rotational friction coefficient η_r for polymer chains of length N = 64 under selfpropulsion force $f_s = 7$ and 30. We can see that $\langle R_G^2 \rangle$ of the active polymer under selfpropulsion is always larger than that of a corresponding passive polymer ($f_s = 0$) with $\langle R^2 \rangle_0 = 69 \pm 2$ for N = 64.

Figure S3 presents the dependence of mean square end-to-end distance $\langle R^2 \rangle$ and $\langle R_G^2 \rangle$ on self-propulsion force f_s . $\langle R^2 \rangle$ and $\langle R_G^2 \rangle$ increase gradually with increasing f_s .

Figure S4 presents the dependence of $\langle R^2 \rangle$ and $\langle R_G^2 \rangle$ on N for the active polymer with $f_s = 30$ and the corresponding passive polymer ($f_s = 0$). For the passive polymer, we have $\langle R^2 \rangle \propto N^{1.5}$ and $\langle R_G^2 \rangle \propto N^{1.5}$. The exponent 1.5 is consistent with the known result for two-dimensional self-avoiding polymers. While for the active polymer with $f_s = 30$, we find the scaling exponent is close to 2 for relatively short chains, indicating a rod-like conformation under self-propulsion of the head ABP. We find the data deviate gradually from the scaling relations $\langle R^2 \rangle \propto N^2$ and $\langle R_G^2 \rangle \propto N^2$ with the increase of N. And $\langle R^2 \rangle$ and $\langle R_G^2 \rangle$ of long RRAP tend to approach that of the passive polymer. The reason is that the effect of the self-propulsion of the head ABP is limited, i.e., the RRAP becomes flexible with increasing N.



Fig. S2. Log-log plot of the mean square radius of gyration $\langle R_G^2 \rangle$ versus the rotational friction coefficients η_r for the active polymer of length N = 64 under self-propulsion force $f_s = 7$ and 30. The dashed line with $\langle R_G^2 \rangle_0 = 69 \pm 2$ denotes $\langle R_G^2 \rangle$ of the corresponding passive polymer chain of N = 64.



Fig. S3. Plot of mean square end-to-end distance $\langle R^2 \rangle$ and mean square radius of gyration $\langle R_G^2 \rangle$ versus self-propulsion force f_s for the active polymer. Polymer length N = 64 and rotational friction coefficient $\eta_r = 1$.



Fig. S4. Log-log plot of mean square end-to-end distance $\langle R^2 \rangle$ (a) and mean square radius of gyration $\langle R_G^2 \rangle$ (b) versus polymer length *N* for the active polymer (red squares) with $f_s = 30$ and the corresponding passive polymer (blue stars). The rotational friction coefficient is $\eta_r = 1$. The red solid lines have a slope of 2 and the blue dashed lines have a slope of 1.5.

3. Diffusion property

The mean squared displacement of the two-dimensional active polymer at long timescales can be expressed as $\langle \Delta r^2(t) \rangle = 4D_{\text{eff}}t$. The diffusion of the active polymer can be attributed to the active force of ABP and thermal noise. Thus we have $D_{\text{eff}} = D_{\text{P}} + D_{\text{T}}$ with D_{P} and D_{T} are the coefficients of propulsive and thermal diffusion, respectively. For the present active polymer model, we have $D_{\text{P}} \propto f_s^2 \tau_r / N^2 \eta_t^2$ and $D_{\text{T}} \propto k_{\text{B}}T/N\eta_t$. It is interesting to estimate the crossover value of $f_{\text{s,c}}$ at which $D_{\text{P}} = D_{\text{T}}$. We thus have a simple relation $f_{\text{s,c}}^2 \propto k_{\text{B}}T\eta_t N/\tau_r$.



Fig. S5. Plot of the crossover value of $f_{\rm s,c}$ versus polymer length N for the active polymer with a rotation-restricted head ABP (with M) and that with a freely rotating head ABP (M = 0). Solid lines show the relation $f_{\rm s,c} \propto N^{0.5}$.

We have checked the dependence of $f_{s,c}$ on N for $\eta_t = 1$ and $\eta_r = 1$. The simulation results are presented in Figure S5 for the active polymer with a rotation-restricted head ABP (with M) and that with a freely rotating head ABP (M = 0). We find $f_{s,c}$ is small for the active polymer with a rotation-restricted head ABP, in agreement with that the restriction of rotation of the head ABP will enhance the diffusion. It is interesting to see that $f_{s,c} \propto N^{0.5}$ still holds for the active polymer with a rotation-restricted head ABP. The reason may be that, for the active polymer with M, the variation of $f_{s,c}$ is very small, so that τ_r can be regarded as a constant independent of f_s and N. The influence of N on τ_r is rather weak at small f_s since the deformation of polymer is small at small f_s .