

Supplementary Information

Structural, Dynamic, and Diffusion Properties of Li₆(PS₄)SCl Superionic Conductor from Molecular Dynamics Simulations

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Section S1: DFT calculation setup

We used VASP (Vienna *Ab initio* Simulation Package, v5.4.4) for all density functional theory (DFT) computations on the ground state (0 K).^{1,2} The Projector-augmented-wave (PAW) potentials with valence configurations of $1s^22s^1$ for Li, $2s^22p^3$ for P, $2s^22p^4$ for S, and $2s^22p^5$ for Cl was applied to describe the valence electrons.^{3,4} The Perdew, Burke, and Ernzerhof (PBE) flavor of the generalized gradient approximation (GGA) functional was applied to describe the exchange-correlation potentials of the constituting elements.^{5,6} In addition, we included the Grimme D3 vdW correction (PBE-D3) with the Becke-Johnson parameters. The accuracy of the electronic calculations was within 1 μ eV. A single point energy calculation was performed to test the convergence of cut-off energy and k -points for input lattices. We concluded from the k -mesh calculations that a k -spacing of 0.2 \AA^{-1} was sufficient for the required accuracy of 1 meV/atom when the cutoff energy was set to 500 eV. Ionic relaxations were performed until the Hellmann-Feynman force on each atom reached an order of 10 meV/ \AA .^{7,8}

Section S2: DDEC6 calculated charges and charge assignment on Li and other species

DDEC6 method⁹ was employed in the literature to accurately predict NAC (net atomic charges) on charged species. Here we used the same method to calculate the partial charge on all the ionic species in the system (Table S3). The calculated charge was later scaled to with charge on Li to accurately predict the room temperature Li-ion diffusivity obtained from experiments. It was observed with charge of +0.4 and +0.5 on Li-ion the ionic diffusivity is respectively higher or lower than experimental diffusivity at room temperature but a charge of +0.45 gives more accurate prediction.

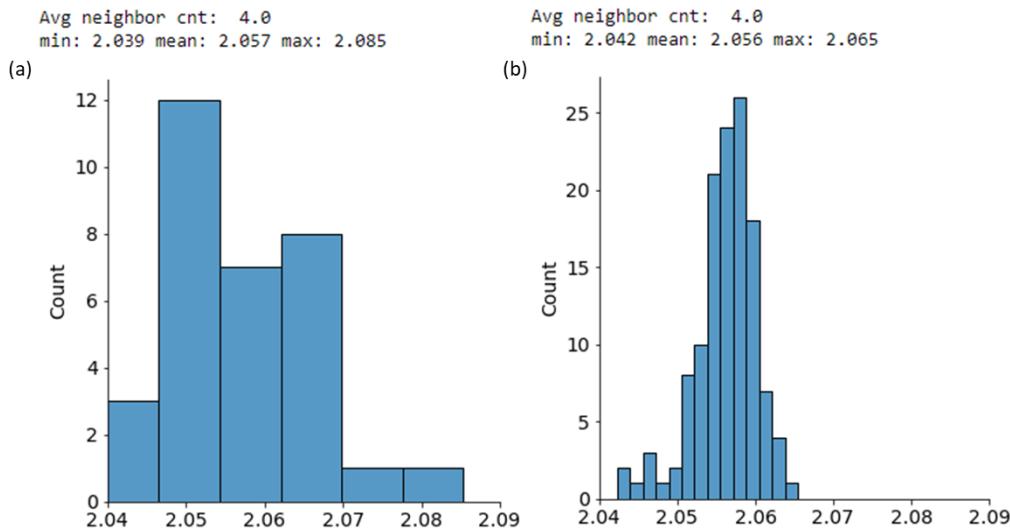


Figure S1. (a) DFT predicted P–S bond length distribution from energy minimized structure (b) LAMMPS MD predicted P–S bond length distribution from the UFF structural minimization.

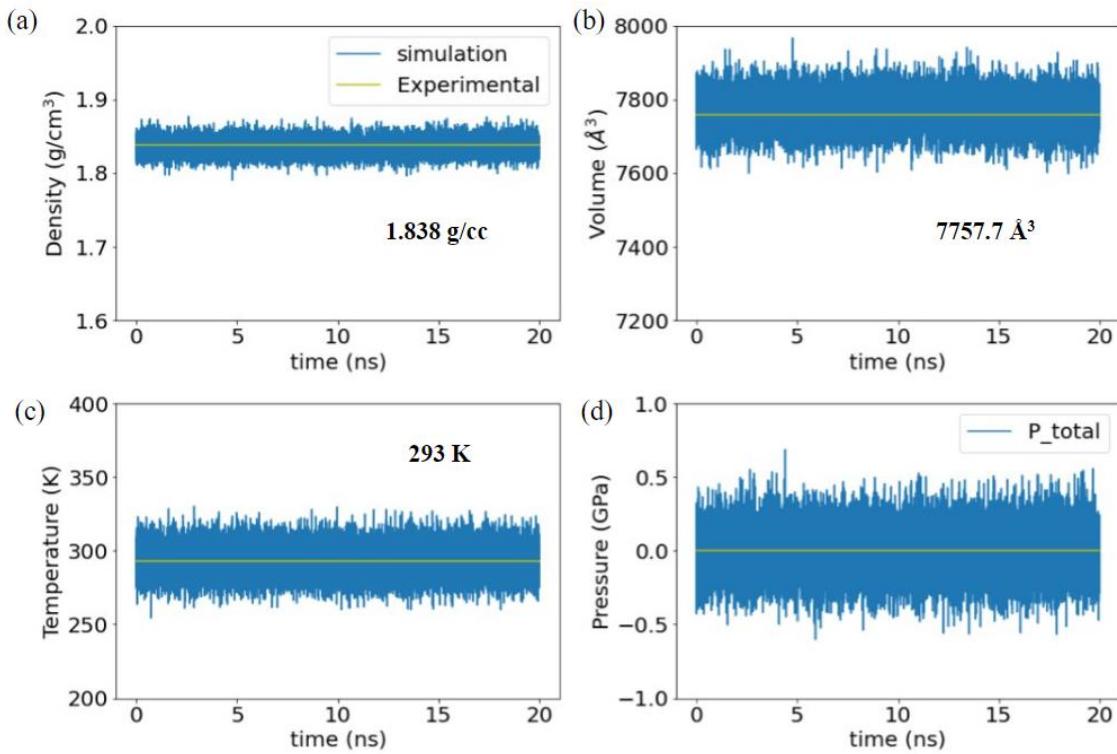


Figure S2. (a) Density, (b) volume, (c) temperature, and (d) pressure fluctuations in the 20 ns NPT simulation on $\text{Li}_6(\text{PS}_4)\text{SCl}$ at 293 K. The yellow line shows the experimental target.

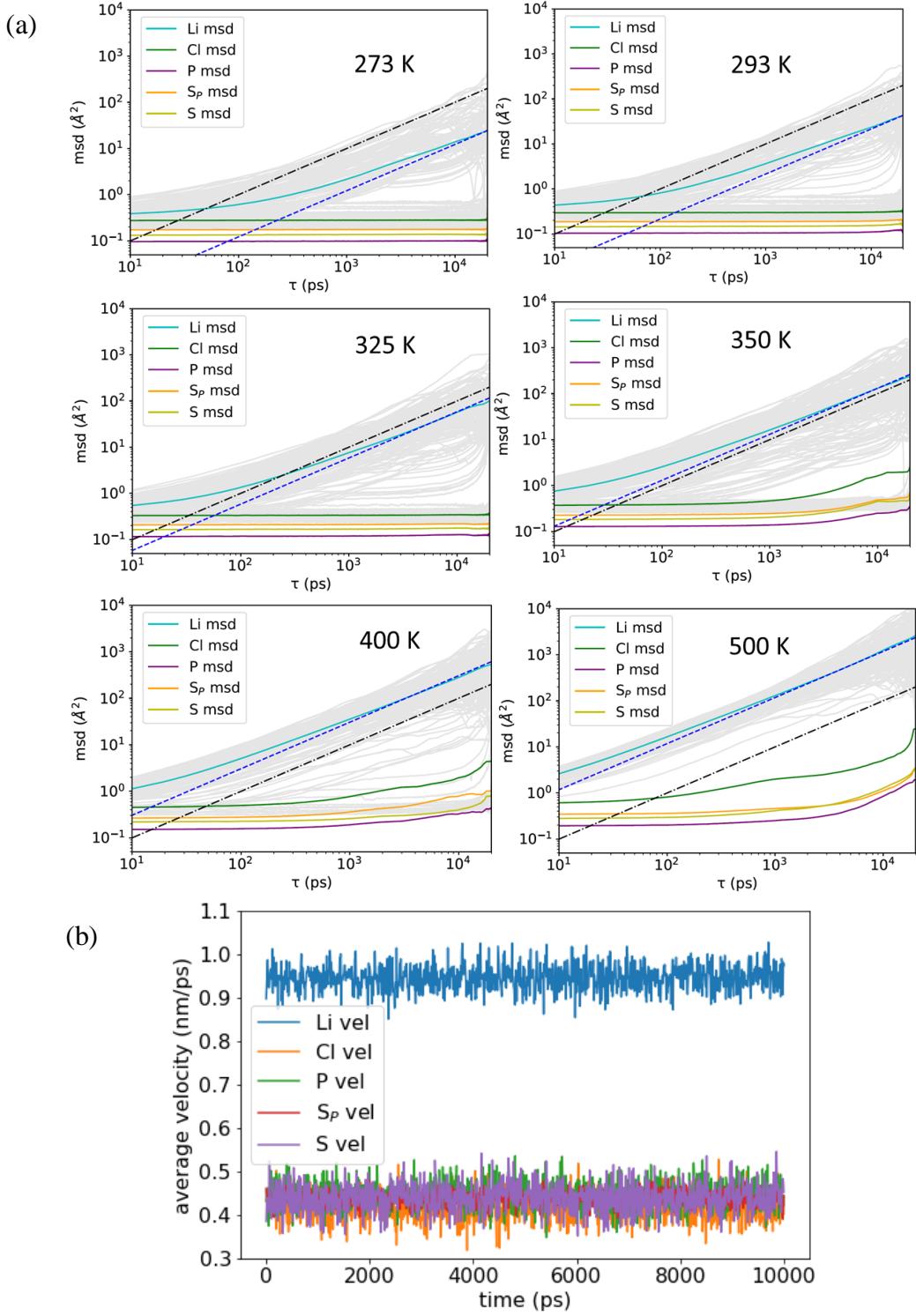


Figure S3. (a) MSD of the different species calculated from the 20 ns MD simulations on $\text{Li}_6(\text{PS}_4)\text{Cl}$ at 273, 293, 325, 350, 400, and 500 K. The dotted blue lines have a slope of 1 represented by the dash-dotted black lines. (b) Results from application of a constant electric field of 0.1 V/nm. Average velocity of each ion type at room temperature (293 K) *vs.* time.

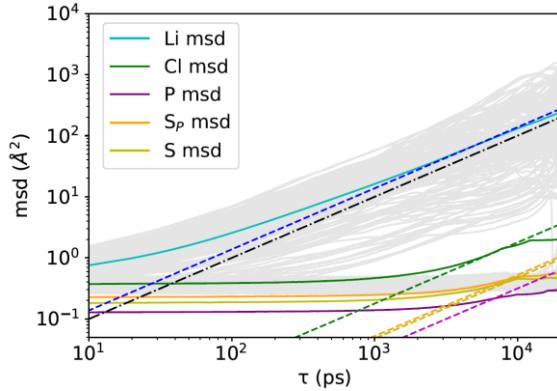


Figure S4. Mean square displacements (MSD) of the different species calculated from the 20 ns MD simulation on $\text{Li}_6(\text{PS}_4)\text{SCl}$ at 350 K. The dotted lines have a slope of 1 represented by the dash-dotted black line. The Li-ion diffusivity ($2.1 \times 10^{-7} \text{ cm}^2/\text{s}$) is by 2-3 orders of magnitude higher than that of the anions ($\text{S} - 8.0 \times 10^{-10} \text{ cm}^2/\text{s}$, $\text{Cl} - 3.0 \times 10^{-9} \text{ cm}^2/\text{s}$, $\text{P} - 5.2 \times 10^{-10} \text{ cm}^2/\text{s}$, and $\text{S}_\text{P} - 8.7 \times 10^{-10} \text{ cm}^2/\text{s}$). A transference number for the Li-ions was calculated as 0.98.

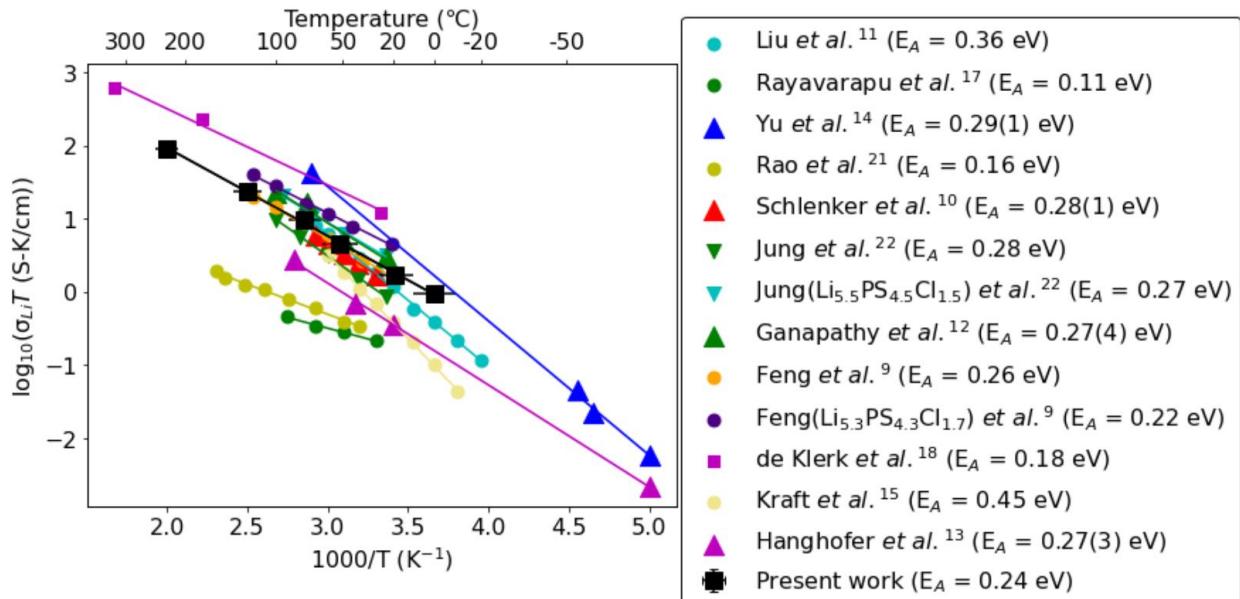


Figure S5. The log plots of the Li-ion conductivity multiplied by temperature (σT) vs. temperature reported in literature. Note that our calculations lead to a comparable slope with $E_a = 0.24$ eV. Filled circles represent the experimental EIS data. Filled triangles represent the experimental NMR data. Filled squares represent computationally predicted data. Computationally predicted activation energy should be compared with the activation energy from NMR measurement due to no impact of grain boundaries and electrode contacts on the bulk conductivity measurement and minimal variation, $0.27(4) - 0.29(1)$ eV.

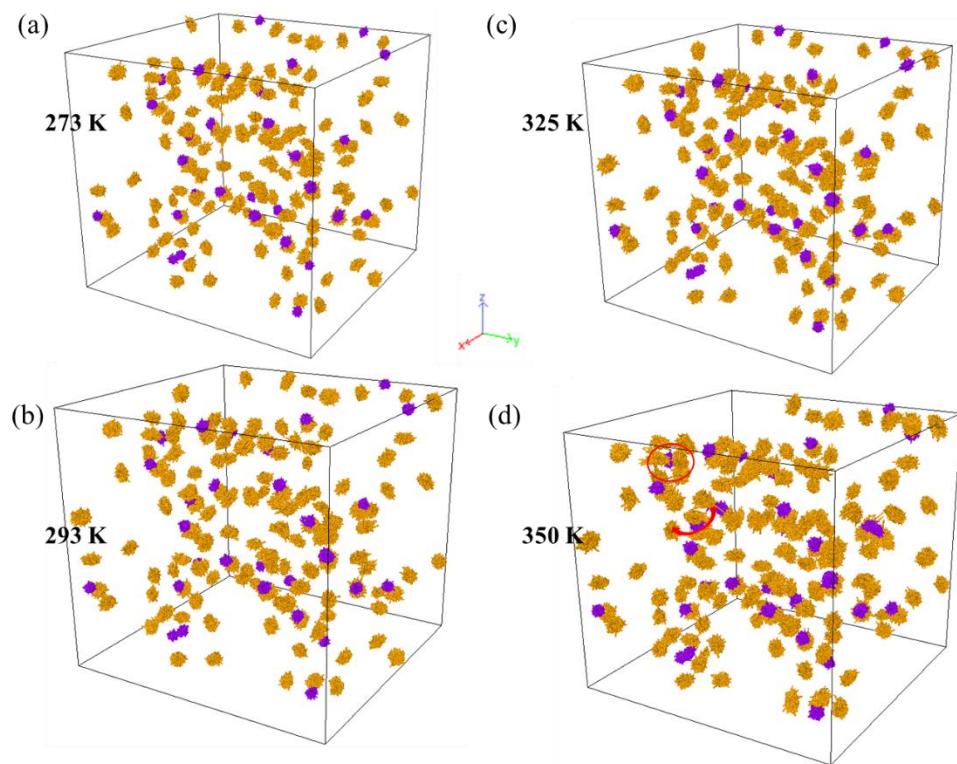


Figure S6. Trajectories of P (purple) and S (orange) atoms of PS₄ tetrahedra indicating rotation of the PS₄ units at 350 K (d). No rotation of PS₄ units were observed at (a) 273 K, (b) 293 K, and (c) 325 K for 20 ns of simulations.

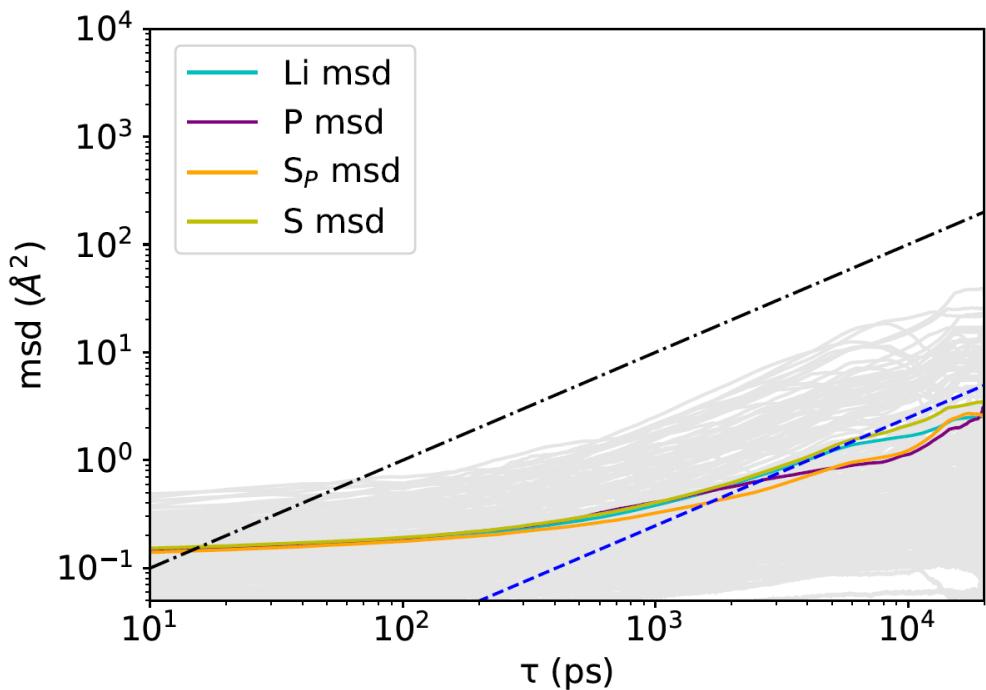


Figure S7. Ion diffusions in $\text{Li}_7(\text{PS}_4)\text{S}_2$: mean square displacement (MSD) of the species *vs.* time at room temperature (293 K). The dotted blue line has a slope of 1 leading to Fick's law of Δr^2 proportional to time. This resulted in the Li-ion diffusivity of $4.12 \times 10^{-9} \text{ cm}^2/\text{s}$ and Li-ion conductivity of $7.56 \times 10^{-4} \text{ S/cm}$ which are ~ 10 times lower than in $\text{Li}_6(\text{PS}_4)\text{SCl}$.

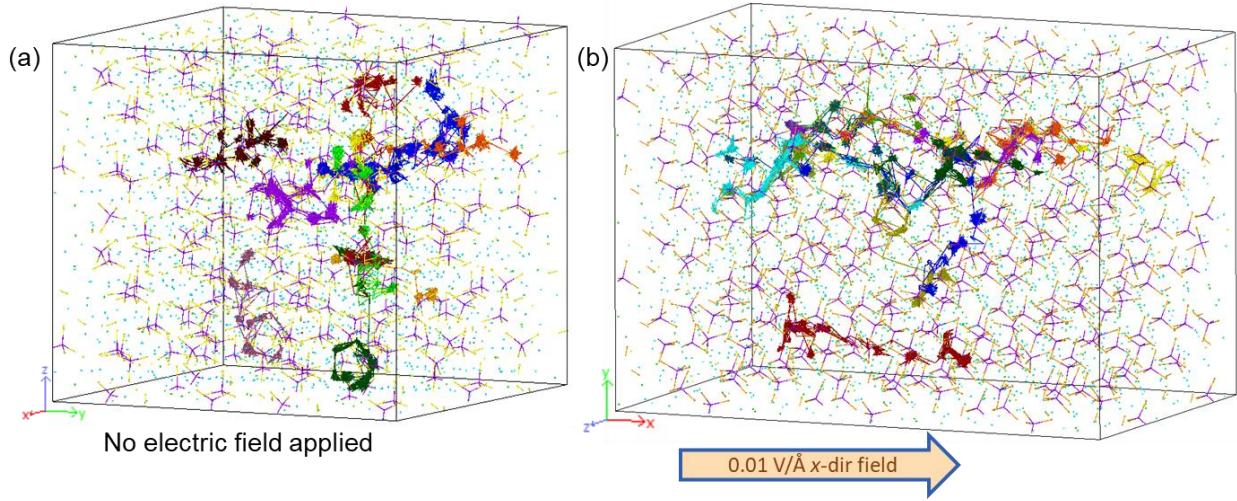


Figure S8. (a) Trajectories of 10 fastest Li-ions without electric field at 293 K. They randomly move along different directions in a $2 \times 2 \times 2$ supercell ($39.45 \text{ \AA} \times 40.91 \text{ \AA} \times 38.09 \text{ \AA}$). (b) Trajectories of 10 fastest Li-ions under electric field of 0.01 V/\AA along the x -direction. They show preferable motion along the x -direction in a $3 \times 2 \times 2$ supercell ($59.18 \text{ \AA} \times 40.91 \text{ \AA} \times 38.09 \text{ \AA}$).

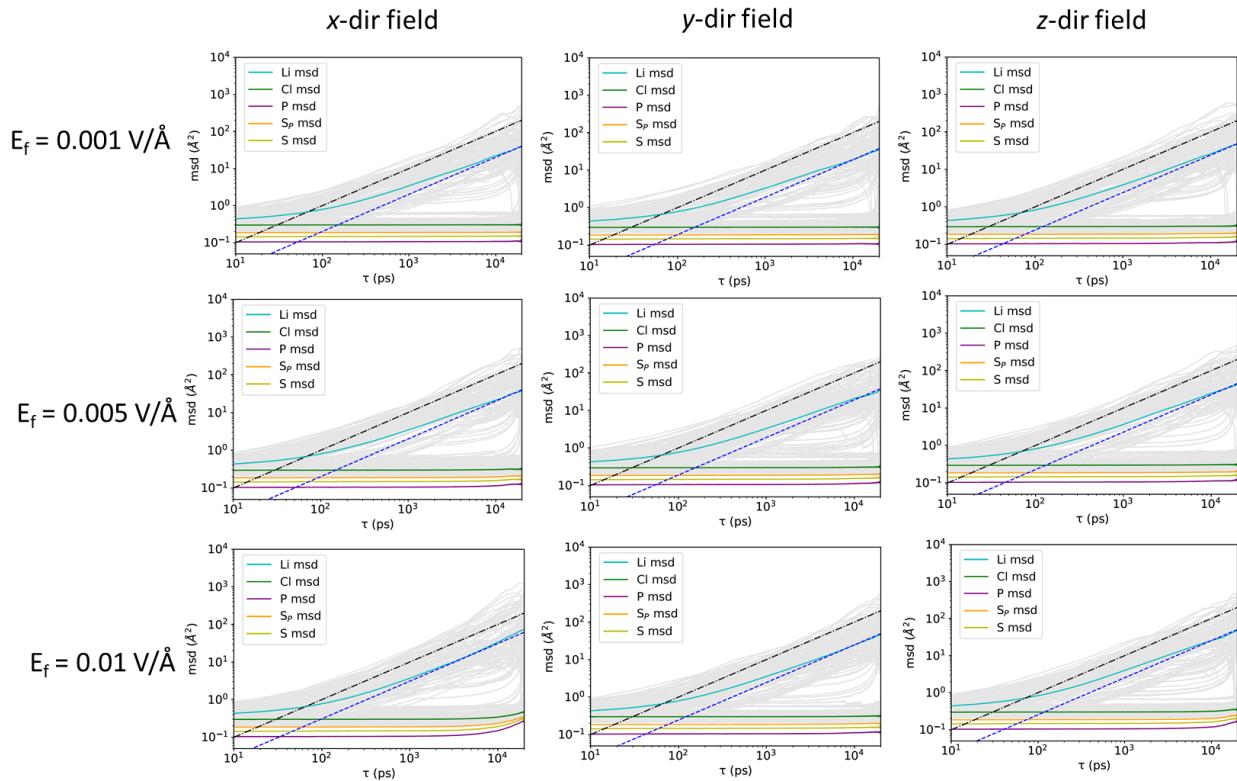
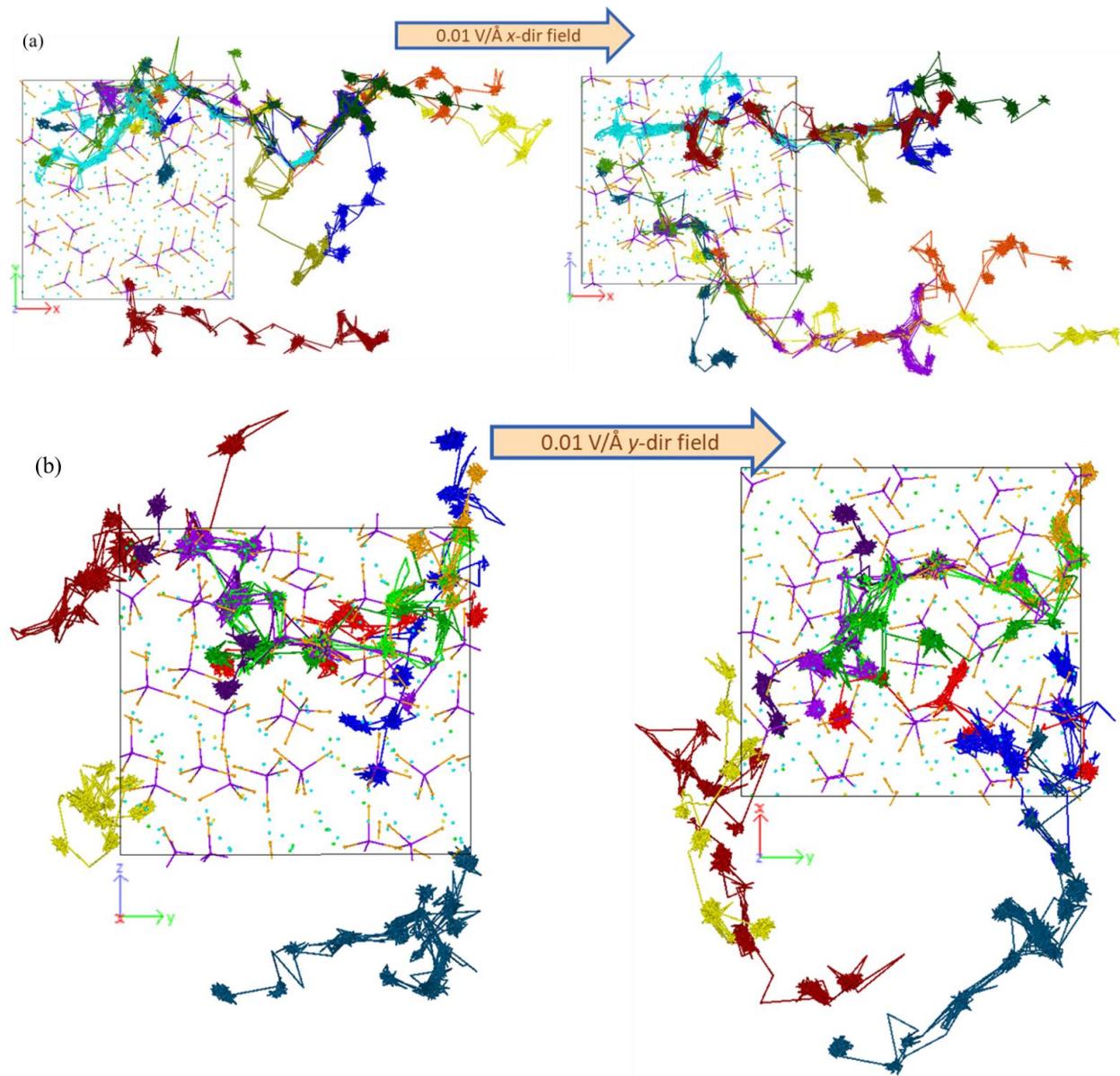


Figure S9. Trajectories of the species in $\text{Li}_6(\text{PS}_4)\text{Cl}$ under different electric field strengths at room temperature (293 K). The dotted blue lines have a slope of 1 represented by the dash-dotted black lines.



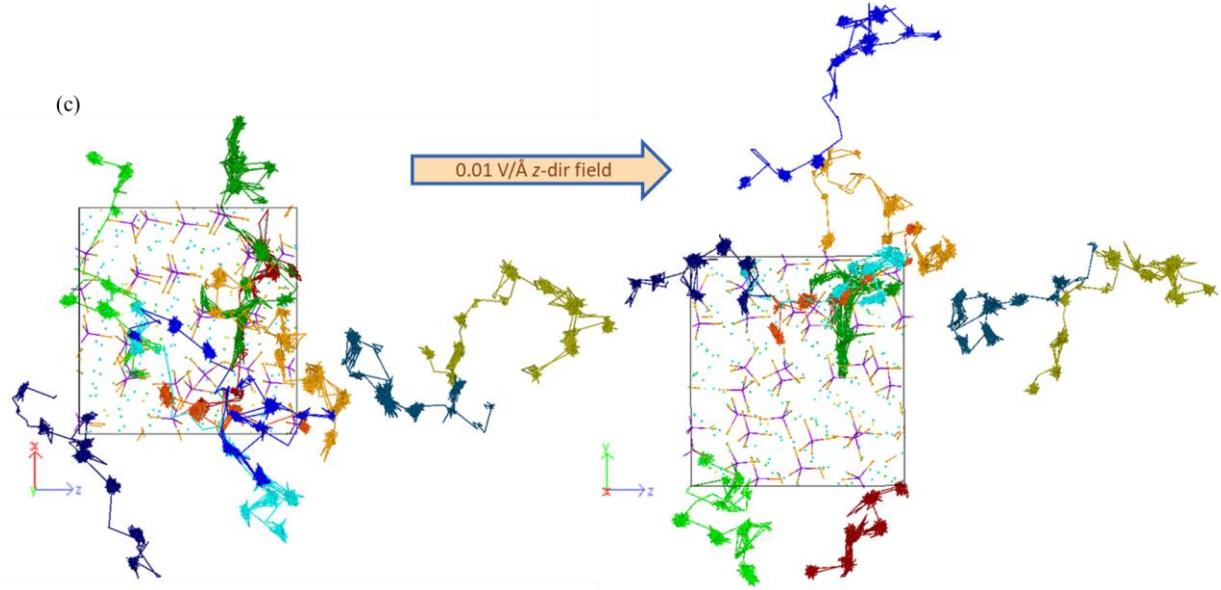


Figure S10. Trajectories of 10 fastest Li-ions over 20 ns under the 0.01 V/Å electric field applied along the (a) x, (b) y, and (c) z directions. Some Li-ions preferentially diffuse in (101) planes under the strong 0.01 V/Å electric field. When an electric field was applied along the y and z directions, similar preferential trajectories were observed along the 45° direction to the applied field.

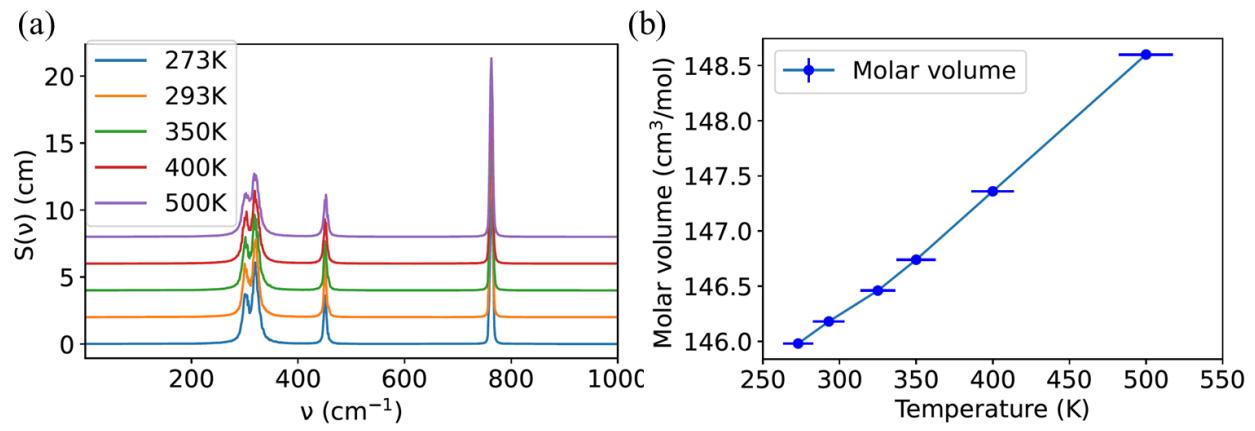


Figure S11. (a) Vibrational DoS contribution from the PS_4 tetrahedra, and (b) the molar volume predicted from one phase dynamics calculations. The same nature of the peaks at high temperatures ensures the stability of the P–S bonds with temperature. Molar volume changes linearly with temperature in a single phase. Our findings are in agreement with the change in the thermodynamic properties with temperature.

Table S1. Summary of reported experimental measurements: Li-ion conductivity and activation energy in $\text{Li}_6(\text{PS}_4)\text{SCl}$ at 298 K.

Sample Preparation method*	Measurement Technique‡	Li-ionic conductivity (mS/cm)	Activation energy (eV)
BMA 550 °C/ 12 hr (pellet)	EIS	7.1	0.257(7)
BMA 460 °C/ 12 hr ($\text{Li}_{5.3}\text{PS}_{4.3}\text{Cl}_{1.7}$) (pellet)	EIS	17.3	0.218(6)
Feng <i>et al.</i> ⁹ BMA 480 °C/ 12 hr ($\text{Li}_{5.5}\text{PS}_{4.5}\text{Cl}_{1.5}$) (pellet)	EIS	16.1	0.221(8)
AIMD	MSD (Einstein relation)	45	0.182
Schlenker <i>et al.</i> ¹⁰ commercial (powder)	NMR	3.9	0.28(1)
Liu <i>et al.</i> ¹¹ BMA 550 °C/ 4 hr (pellet)	EIS	6.11	0.30
BMA 550 °C/ 4 hr (powder)	EIS	3.50	0.33
Ganapathy <i>et al.</i> ¹² BMA 550 °C/ 10 hr (powder)	NMR	10.3	0.27(4)
Hanghofer <i>et al.</i> ¹³ BMA 550 °C/ 7 days (pellets)	NMR	2.46	0.273(5)
SSM 550 °C/ 10 hr (powder)	NMR	4.96	0.29(1)
BMA 550 °C (powder)	NMR	4.73	0.19(9)
Yu <i>et al.</i> ¹⁴ SSM 550 °C/ 10 hr (powder)	EIS	5.99	0.33(8)
BMA 550 °C (powder)	EIS	3.25	0.35(2)
Kraft <i>et al.</i> ¹⁵ Hand ground + annealed 550 °C/ 14 days (powder)	EIS	1.3	0.45(2)
BMA 550 °C/ 5hr (powder)	NMR	20	0.29(1)
Yu <i>et al.</i> ¹⁶ AIMD	Transition rate	NA	0.10(1)

	BMA 550 °C/ 5hr (pellet)	EIS	0.74	0.11
Rayavarapu <i>et al.</i> ¹⁷	BM (not annealed)	EIS	0.033	0.38
de Klerk <i>et al.</i> ¹⁸	AIMD (50% Cl at site 4c)	MSD (Einstein relation)	40	0.17 – 0.23

*BMA - Ball milled + annealed

*SSM - Solid state method

*AIMD - *Ab initio* molecular dynamics

[‡]EIS - Electrochemical impedance spectroscopy

[‡]NMR - Nuclear magnetic resonance

[‡]MSD - Mean square displacement of Li-trajectories

Table S2. UFF parameters used in our LAMMPS calculations.

Pair Coefficients # lj/charmm/coul/charmm

	D_I	X_I
Li	0.025	2.13992
Cl	0.227	3.16474
P	0.305	3.62066
S _P	0.274	3.52288
S	0.274	3.59478

Bond Coefficients # harmonic

	K	r_0
P–S	266.873	2.06543

Angle Coefficients # cosine/squared

	K	θ_0
S–P–S	179.88	109.471

Table S3. Atomic charge comparison from LAMMPS scaled charges vs. DFT calculated DDEC6 charges along with diffusivity at room temperature (293 K).^{9,20}

Species	DDEC6 charge (DFT-D3)	QE _q charges (LAMMPS UFF)	Scaled charge of 0.40 on Li	Scaled charge of 0.45 on Li	Scaled charge of 0.50 on Li
Li	+0.78	+0.31	+0.40	+0.45	+0.50
Cl	-0.80	-0.41	-0.41	-0.46	-0.51
PS ₄	-2.26	-1.20	-1.18	-1.32	-1.46
S	-1.61	-0.27	-0.81	-0.92	-1.03
D (cm ² /s)		4.59×10^{-7}	9.8×10^{-8}	3.72×10^{-8}	9.4×10^{-9}

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